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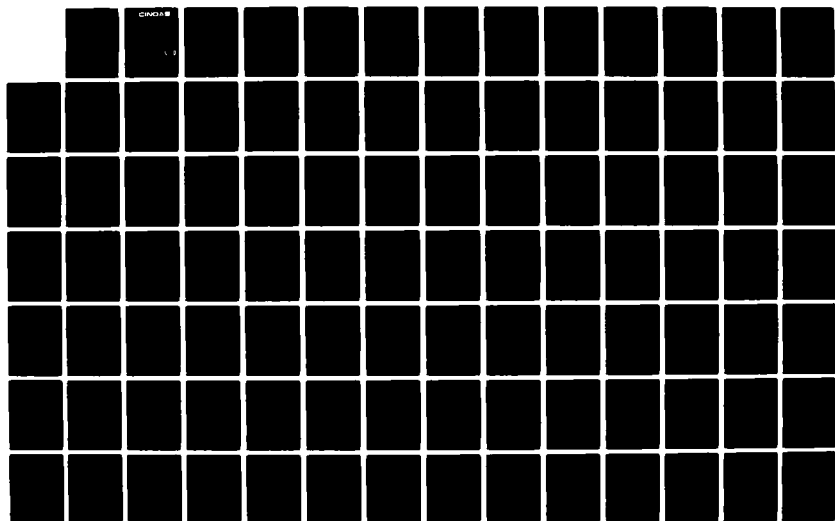
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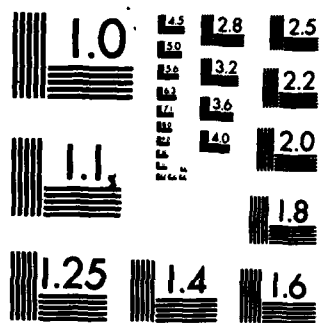
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## ELECTRICAL RESISTIVITY OF TEN SELECTED BINARY ALLOY SYSTEMS

By

C. Y. Ho, M. W. Ackerman, K. Y. Wu, T. N. Havill, R. H. Bogaard,  
R. A. Matula, S. G. Oh, and H. M. James

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## PREFACE

This technical report was prepared by the Center for Information and Numerical Data Analysis and Synthesis (CINDAS), Purdue University, West Lafayette, Indiana, under the auspices of the Office of Standard Reference Data of the National Bureau of Standards (NBS), Department of Commerce, Washington, D.C. It represents the most exhaustive review and critical evaluation of the recorded world knowledge on the electrical resistivity of ten selected binary alloy systems, and is a continuation of a similar work on the thermal conductivity of the same ten binary alloy systems already published. The recommended self-consistent electrical resistivity values presented in this report cover the full ranges of composition and temperature for most of the alloy systems and go far beyond the limited experimental data, which are often conflicting and uncertain in many cases. Thus, new knowledge has been generated in this process of data analysis and synthesis.

This report serves multiple purposes. It provides engineering and design data for virtually all compositions of the ten binary alloy systems for industrial applications at all temperatures. It provides reliable data against which theoreticians can test their theories. Furthermore, the knowledge of the electrical resistivity of binary alloy systems is essential for the study and estimation of the electrical resistivity of ternary and more complex engineering alloys.

Although this report is primarily the result of financial support and interest of the NBS Office of Standard Reference Data, the extensive documentary activity essential to this work was supported by the Defense Logistics Agency of the Department of Defense. Throughout the course of this work, Dr. P. G. Klemens, who is a Visiting Research Professor at CINDAS and Professor of Physics at the University of Connecticut, has given the staff of this project valuable technical guidance and advice; his contributions are hereby gratefully acknowledged. Thanks are also due Dr. H. J. White, Jr., of the NBS Office of Standard Reference Data for his sympathetic understanding and help in many ways.

Y. S. TOULOUKIAN

Director of CINDAS  
Distinguished Atkins Professor of  
Engineering  
Purdue University

## ABSTRACT

This work compiles, reviews, and discusses the available data and information on the electrical resistivity of ten selected binary alloy systems and presents the recommended values resulting from critical evaluation, correlation, analysis, and synthesis of the available data and information. The ten binary alloy systems selected are the systems of aluminum-copper, aluminum-magnesium, copper-gold, copper-nickel, copper-palladium, copper-zinc, gold-palladium, gold-silver, iron-nickel, and silver-palladium. The recommended values for each of the ten binary alloy systems except three (aluminum-copper, aluminum-magnesium, and copper-zinc) are given for 27 compositions: 0 (pure element), 0.5, 1, 3, 5, 10(5)95, 97, 99, 99.5, and 100% (pure element). For aluminum-copper, aluminum-magnesium, and copper-zinc alloy systems, the recommended values are given for 26, 12, and 11 compositions, respectively. For most of the alloy systems the recommended values cover the temperature range from 1 K to the solidus temperature of the alloys or to about 1200 K. For most of the nine elements constituting the alloy systems, the recommended values cover the temperature range from 1 K to above the melting point into the molten state. The estimated uncertainties in most of the recommended values are about  $\pm 3\%$  to  $\pm 5\%$ .

Key words: Alloy systems; alloys; conductivity; critical evaluation; data analysis; data compilation; data synthesis; electrical conductivity; electrical resistivity; metals; recommended values; resistivity.

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## NOMENCLATURE

$a, b$	Constants in eq (17)
$A, B$	Constants in eq (20)
$c$	Impurity concentration; mole fraction; composition
$\delta_c$	Composition error
$C$	Constant in eqs (3) to (6)
$e$	Electron charge; base of natural logarithm
$\hbar$	Planck constant divided by $2\pi$
$k$	Boltzmann constant
$\delta l$	Dimensional error
$L$	Length of specimen at $T$
$L_0$	Length of specimen at $T_0$
$\Delta L$	$\Delta L = L - L_0$
$m$	Electron mass
$M$	Atomic weight; net magnetization
$N$	Electron density
$p$	Number of positive holes per atom
$T$	Temperature
$T_0$	Reference temperature
$T_t$	Test temperature
$U$	Quantity defined by eq (15)
$v$	Electron velocity
$V_\alpha, V_\beta$	Volume fractions of $\alpha$ and $\beta$ phases
$z$	$z = \hbar\omega/kT$
$\delta Z$	Difference in valence
$\alpha, \beta, \gamma$	Positive quantities in eq (10)
$\Delta$	Deviation from the Matthiessen's rule; deviation of $p_1(c, T)$ of an alloy from its interpolated value $\bar{p}_1(c, T)$

$\delta\Delta$	Change in $\Delta$
$\epsilon_F$	Fermi energy
$\theta_D$	Debye temperature
$\theta_R$	Characteristic temperature for intrinsic electrical resistivity
$\kappa$	Constant
$\mu$	$\mu = \rho_{od}/\rho_{os}$
$\nu$	$\nu = \rho_{id}/\rho_{is}$
$\rho$	Electrical resistivity
$\rho_0$	Residual electrical resistivity
$\rho_{ob}, \rho_{ib}$	Residual and intrinsic electrical resistivities of the belly electrons (those associated with the spherical portions of the Fermi surface)
$\rho_{od}$	Residual electrical resistivity of the d-bands
$\rho_{on}, \rho_{in}$	Residual and intrinsic electrical resistivities of the neck electrons (those associated with the portions of the Fermi surface which approach or make contact with the zone boundary)
$\rho_{os}$	Residual electrical resistivity of the s-bands
$\delta\rho_0$	Increase of residual electrical resistivity
$\rho_e$	Electrical resistivity due to electron-electron scattering
$\rho_i(T)$	Intrinsic electrical resistivity of a pure metal
$\rho_i(c,T)$	Temperature-dependent part of electrical resistivity of an alloy or impure metal
$\rho_{id}$	Intrinsic electrical resistivity of the d-bands
$\rho_{is}$	Intrinsic electrical resistivity of the s-bands
$\rho_{sd}$	Electrical resistivity due to s-d transitions
$\rho_{ss}$	Electrical resistivity due to s-s scattering
$\delta\rho_{vac.}$	Increase in electrical resistivity due to vacancies
$\delta\rho_{int.}$	Increase in electrical resistivity due to interstitials
$\phi$	Potential energy
$\delta\phi$	Potential energy perturbation

$\psi$	Wave function
$\omega$	Phonon angular frequency

## 1. INTRODUCTION

The principal objective of this study was to critically evaluate, correlate, analyze, and synthesize all the available data and information on the electrical resistivity of ten selected binary alloy systems and to generate recommended values over the widest practicable ranges of temperature and alloy composition for each of the alloy systems. This study is a continuation of a similar work on the thermal conductivity of the same ten binary alloy systems [1].<sup>1</sup>

The ten binary alloy systems selected are the systems of aluminum-copper, aluminum-magnesium, copper-gold, copper-nickel, copper-palladium, copper-zinc, gold-palladium, gold-silver, iron-nickel, and silver-palladium. These systems include all the three different kinds of binary alloy systems: nontransition-metal and nontransition-metal systems (aluminum-copper, aluminum-magnesium, copper-gold, copper-zinc, and gold-silver), nontransition-metal and transition-metal systems (copper-nickel, copper-palladium, gold-palladium, and silver-palladium), and a transition-metal and transition-metal system (iron-nickel). Most of these alloy systems are among those for which the largest amounts of experimental data are available. However, it will become evident later that even for these alloy systems serious gaps still exist in the electrical resistivity data, as concerns dependence on both composition and temperature, and that some of the available experimental data sets show large uncertainties or wide divergences.

The resulting electrical resistivity values presented in this work are designated as recommended or provisional values depending upon the level of confidence placed on the values and, hence, upon the uncertainty in the values assigned. The uncertainty in the recommended electrical resistivity values is less than or equal to  $\pm 5\%$  and that in the provisional values is greater than  $\pm 5\%$ . It will be noted that most of the resulting values are designated as recommended values and the uncertainty in the values is generally of the order of  $\pm 3\%$  to  $\pm 5\%$ .

The recommended electrical resistivity values are based on the room-temperature dimensions of the alloys, as thermal expansion corrections have not been made. This is due to the fact that the available experimental data on the electrical resistivity of the alloys, upon which the recommendations are based,

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<sup>1</sup> Numbers in brackets indicate literature references listed in Section 6.

are not corrected for thermal expansion, and that thermal expansion values for the respective alloys are not available for such corrections to be made. If the values of thermal expansion,  $\Delta L(T)/L_0$ , for the alloys are available, the electrical resistivity values corrected for thermal expansion,  $\rho_{\text{corrected}}$ , can easily be calculated from the given values uncorrected for thermal expansion,  $\rho_{\text{uncorrected}}$ , by the following relation:

$$\rho_{\text{corrected}}(T) = \left( 1 + \frac{\Delta L(T)}{L_0} \right) \rho_{\text{uncorrected}}(T)$$

where  $\Delta L = L - L_0$ , and  $L$  and  $L_0$  are the lengths of the specimen at any temperature  $T$  and at a reference temperature  $T_0$ , respectively. The thermal expansion correction amounts roughly to about -0.2% to -0.7% at very low temperatures, zero at room temperature, about 0.3% to 0.7% at 500 K, and about 2% near the melting point of the alloy.

The recommended (or provisional) values generated are for alloys which are not ordered and have not been cold-worked severely; the values would be lower for ordered alloys and higher for cold-worked alloys at low temperatures.

The general background information on this work is given in Section 2, which includes a brief summary of the theory of the electrical resistivity of metals and alloys, a short description of the procedure and method used for the data evaluation, correlation, analysis, and synthesis and the generation of recommended values, and a detailed explanation of the specifics and conventions used in the presentation of the data and information.

The experimental data and information and the recommended (or provisional) values for the electrical resistivity of the ten binary alloy systems are presented in Section 3. In the discussion of the electrical resistivity of each alloy system, individual pieces of available data and information are reviewed, details of data analysis and synthesis are given, the considerations involved in arriving at the final assessment and recommendation are discussed, the recommended values and the experimental data are compared, and the uncertainties in the recommended values are stated. For each of the alloy systems except three (aluminum-copper, aluminum-magnesium, and copper-zinc), the recommended values are given for 27 compositions: 0 (pure element), 0.5, 1, 3, 5, 10(5)95, 97, 99, 99.5, and 100% (pure element). For aluminum-copper alloy system, recommended values are presented for 26 compositions, without that

containing 80% copper. For aluminum-magnesium alloy system, recommended values are generated only for 12 compositions, lacking those containing 15 to 85% magnesium. Recommended values are given for only 11 compositions of the copper-zinc alloy system, and no values are for those with 35 to 99.5% zinc. For most of the compositions the recommended values cover the temperature range from 1 K to near the solidus temperature (melting starting point). The recommended values have been smoothed simultaneously over both temperature and composition dependences.

The last three sections are for acknowledgments, appendices, and references. There are three appendices given. The first appendix presents, as an example, a detailed account of analysis and synthesis of the electrical resistivity of gold-silver alloy system. A logical organization of the methods for the measurement of electrical resistivity is given in the second appendix. The third appendix presents conversion factors for the units of electrical resistivity, which may be used to convert easily the electrical resistivity values in the SI units given in this work to values in any of the several other units listed.

## 2. GENERAL BACKGROUND

### 2.1. Theoretical Background

#### a. Matthiessen's Rule

It was found experimentally by Matthiessen [2,3] that the increase in the electrical resistivity of a metal due to the presence of a small amount of another metal in solid solution is independent of the temperature. According to this Matthiessen's rule, the total electrical resistivity of an impure metal may therefore be separated into two additive contributions and written in the form

$$\rho(c,T) = \rho_0(c) + \rho_1(T), \quad (1)$$

where  $\rho_0$  is the residual resistivity caused by the scattering of electrons by impurity atoms and lattice defects and is temperature-independent but dependent on the impurity concentration,  $c$ , and  $\rho_1$  is the temperature-dependent intrinsic resistivity arising from the scattering of electrons by lattice waves, or phonons.

The validity of the Matthiessen's rule has been assessed by Sondheimer [4], who concluded that the maximum deviation should not exceed one percent of  $\rho_0$  in the idealized case in which the impurities do not affect the phonon and electron spectra. However, significant deviations from Matthiessen's rule do occur, which will be discussed in Subsection e. Thus, in general the electrical resistivity of an impure metal is given by

$$\rho(c,T) = \rho_0(c) + \rho_1(T) + \Delta(c,T), \quad (2)$$

where  $\Delta$  is the deviation from the Matthiessen's rule.

In separating the electrical resistivity into its components, the temperature dependent part sometimes includes the electrical resistivity due to electron-electron scattering,  $\rho_e$ ; indeed, this is thought to be the dominant temperature-dependent term in transition metals at low temperatures.

In what follows, we will first discuss  $\rho_1$ ,  $\rho_e$ ,  $\rho_0$ , and  $\Delta$ , then consider the electrical resistivity of nontransition-metal alloys, and finally comment on aspects of the electrical resistivity of transition metals and their alloys. The theory of electrical resistivity of metals and alloys has been the subject of a number of reviews and has constituted a large portion of the material in several books [5-21].

### b. Intrinsic Electrical Resistivity

The intrinsic electrical resistivity which is due to scattering of electrons by phonons may be approximated by the Bloch-Grüneisen formula [22-24]:

$$\rho_i = \frac{C}{M\theta_R} \left( \frac{T}{\theta_R} \right)^5 \int_0^{\theta_R/T} \frac{z^5 e^z dz}{(e^z - 1)^2}, \quad (3)$$

where  $C$  is a constant characteristic of the metal and proportional to the square of the electron-phonon interaction constant,  $M$  is the atomic weight, and  $\theta_R$  is a characteristic temperature of the metal which characterizes its intrinsic electrical resistivity in the same way as the Debye temperature,  $\theta_D$ , characterizes its lattice specific heat. The dimensionless variable of integration  $z = \hbar\omega/kT$ , where  $\hbar$  is the Planck constant divided by  $2\pi$ ,  $\omega$  is the phonon angular frequency, and  $k$  is the Boltzmann constant. The derivation of eq. (3) is based on the simplifying assumptions that the Fermi surface is spherical, that the conduction electrons can be treated as free in the first approximation, that the spectrum of lattice vibrations is that of the Debye model, that the phonon distribution is essentially undisturbed by the scattering processes, and that electron-phonon Umklapp processes can be ignored. Consequently, it is perhaps most reasonable to expect the Bloch-Grüneisen formula to agree with experiment in the case of monovalent metals. Nevertheless, the intrinsic resistivity of many metals can be well represented by eq. (3) over a wide temperature range by a suitable choice of  $\theta_R$  and  $C$ , though no single value of  $\theta_R$  can fit the data at all temperatures.

At low temperatures ( $T \leq \theta_R/20$ ), eq. (3) reduced to

$$\rho_i = \frac{124.4C}{M\theta_R} \left( \frac{T}{\theta_R} \right)^5, \quad (4)$$

while at high temperatures ( $T > \theta_R$ ), to a good approximation, it reduces to

$$\rho_i \approx \frac{C}{4M\theta_R} \left( \frac{T}{\theta_R} \right). \quad (5)$$

Thus it agrees with the experimental facts that at low temperatures (but above about 20 K) the electrical resistivity of some metallic elements is proportional to  $T^5$ , and at high temperatures the resistivity of most metals increases approximately linearly with temperature.



The intrinsic resistivity of a metal at the characteristic temperature,  $T = \theta_R$  in eq. (3), is

$$\rho_1(\theta_R) = \frac{C}{M\theta_R} \int_0^1 \frac{z^5 e^z dz}{(e^z - 1)^2} = \frac{0.2366C}{M\theta_R}. \quad (6)$$

Combining eqs. (3) and (6) yields the reduced intrinsic resistivity equation

$$\frac{\rho_1(T)}{\rho_1(\theta_R)} = 4.226 \left( \frac{T}{\theta_R} \right)^5 \int_0^{\theta_R/T} \frac{z^5 e^z dz}{(e^z - 1)^2}, \quad (7)$$

which gives a single curve for all metals. The values of the reduced intrinsic resistivity from eq. (7) as a function of the reduced temperature,  $T/\theta_R$ , have been compared favorably with the experimental data for a number of metals [25,26].

#### c. Electrical Resistivity due to Electron-Electron Scattering

As in the case of the scattering of electrons by phonons, electron-electron collisions are of two types: normal processes, in which the total wave vector is conserved, and Umklapp processes in which the total wave vectors before and after the collision differ by a reciprocal lattice vector. On the other hand, unlike electron-phonon Umklapp processes, electron-electron Umklapp processes are not frozen out at low temperatures.

Normal processes, involving the collision between two s-band conduction electrons, do not contribute directly to the electrical resistivity because they do not change the total momentum and thus have no effect on the current. Normal processes involving the scattering of an s-band conduction electron by a non-conducting d-band electron do contribute to the electrical resistivity, and are thought to be the dominant temperature-dependent resistive processes in transition elements and their alloys at very low temperatures, since their resistivities show the  $T^2$  temperature dependence expected for electron-electron scattering rather than the  $T^5$  temperature dependence expected for the intrinsic resistivity. This temperature dependence of the electrical resistivity due to electron-electron scattering:

$$\rho_e \propto T^2 \quad (8)$$

comes about through the double application of the exclusion principle in the scattering processes; it applies to both the initial states and final states. For simplicity assume all the states with energies less than the Fermi energy,  $\epsilon_F$ , to be occupied and consider the scattering of electron 1, in an excited state of energy greater than  $\epsilon_F$  by  $\epsilon_1$ , colliding with electron 2 having energy less than  $\epsilon_F$  by  $\epsilon_2$ . Then conservation of energy requires that  $\epsilon_1$  be greater than  $\epsilon_2$ , since the final states must be empty and, therefore, must have energies greater than  $\epsilon_F$ . Thus, the orbital of electron 2 must lie within a shell of thickness  $\epsilon_1$  within the Fermi surface so that only a fraction,  $\epsilon_1/\epsilon_F$ , of the electrons can scatter electron 1. Further, the final states must also lie within a shell of thickness less than  $\epsilon_1$  outside the Fermi surface so that only a fraction,  $\epsilon_1/\epsilon_F$ , of the final states compatible with conservation of energy and momentum are allowed by the exclusion principle. The average energy of an occupied excited state measured from  $\epsilon_F$  is approximately  $kT$  so that  $\epsilon_1/\epsilon_F \approx kT/\epsilon_F$  and the effect of the double application of the exclusion principle is to reduce the collision cross section for electron-electron scattering by a factor  $(kT/\epsilon_F)^2$  below its value for a screened Coulomb interaction.

Umklapp processes between two conduction electrons do contribute to the electrical resistivity. Because these processes involve a reciprocal lattice vector, the wave functions of the electrons involved cannot be regarded as simple plane waves, but must be treated as true Bloch functions having the periodicity of the lattice. The result of this is to introduce into the expression for the resistivity the square of an interference factor. Apparently this factor is quite small, as the low temperature electrical resistivity of most ordinary metals does not show the  $T^2$  temperature dependence expected for such a resistive mechanism; however, a contribution proportional to  $T^2$  has been observed in the liquid-helium-temperature resistivity of indium and aluminum.

#### d. Residual Electrical Resistivity

The electrical resistivity of a metal or an alloy approaches a constant value, the residual electrical resistivity, as the temperature decreases toward absolute zero. Exceptions to this rule are the phenomena of superconductivity, which are not considered here. The residual resistivity is due to the scattering of electrons by foreign atoms and lattice defects such as vacancies, interstitials, dislocations, and stacking faults.

The most important contribution to the residual resistivity is the scattering of electrons by foreign atoms, which will be discussed in Subsection f in connection with the electrical resistivity of alloys.

Electrical resistivities due to vacancies and interstitials have been estimated theoretically, and studied experimentally by using neutron-irradiated samples. The increase in the resistivity due to vacancies,  $\delta\rho_{\text{vac.}}$ , and that due to interstitials,  $\delta\rho_{\text{int.}}$ , have been given for copper by Blatt [20] as follows:

$$\delta\rho_{\text{vac.}} = 1.0 - 1.5 \times 10^{-8} \Omega \text{ m/atomic percent}$$

$$\delta\rho_{\text{int.}} = 0.5 - 1.0 \times 10^{-8} \Omega \text{ m/atomic percent.}$$

Dislocations scatter electrons weakly; however, it is sometimes energetically favorable for a dislocation to dissociate into two partial dislocations separated by a ribbon of stacking faults which, particularly if it is wide, may be effective in scattering electrons. Thus, the resistivity due to dislocations is often significantly larger than predicted by theoretical calculations, which may be caused by the substantial contributions of stacking faults and the core of the dislocation line to the resistivity of plastically deformed metals.

Calculation of the resistivity due to stacking faults is difficult and requires a knowledge of the Fermi surface. However, Howie [27] was able to account for the resistivity of a deformed copper specimen in terms of a reasonable density of stacking faults.

#### e. Deviations from Matthiessen's Rule

For Matthiessen's rule to be exact, the scattering of electrons by imperfections and by phonons would have to be independent of each other, and this is only approximately true. Other causes for deviations from Matthiessen's rule are as follows:

- (1) Alloying may alter the band structure of the metal.
- (2) Alloying may change the phonon spectrum or perturb the phonon distribution in the steady state.
- (3) There may be parallel conduction by two bands of electrons.

The first two effects are to be expected in concentrated alloys, but two-band effects can cause significant deviations in multivalent elements and in very dilute alloys of monovalent elements.

The two-band effect was first discussed by Sondheimer and Wilson [28], who considered the case in which the s- and d-bands of electrons of a transition element conduct in parallel. They found that if  $\rho_0$  depends only on  $\rho_{0s}$  and  $\rho_{0d}$  of the s- and d-bands, respectively, and  $\rho_1$  depends only on  $\rho_{1s}$  and  $\rho_{1d}$ , then the deviation from Matthiessen's rule is given by

$$\Delta = \rho - \rho_0 - \rho_1 = \frac{\rho_{0s} \rho_{1s} (\mu - \nu)^2}{(1 + \mu)(1 + \nu)[\rho_{0s}(1 + \mu) + \rho_{1s}(1 + \nu)]}, \quad (9)$$

where  $\mu = \rho_{0d}/\rho_{0s}$  and  $\nu = \rho_{1d}/\rho_{1s}$ . Thus, there is a positive deviation unless these ratios are equal. At low temperatures  $\rho_{1s}$  is smaller than  $\rho_{0s}$  so that  $\Delta$  is approximately proportional to  $\rho_{1s}$  and thus to  $\rho_1$ , while at high temperatures  $\rho_{1s}$  is larger than  $\rho_{0s}$  so that  $\Delta$  is approximately proportional to  $\rho_{0s}$  and thus to  $\rho_0$ . It follows that  $\Delta$  increases from zero at zero temperature to a constant value proportional to  $\rho_0$  at high temperatures. When  $\rho_0$  and  $\rho_1$  are of the same order of magnitude, the deviation can be approximated by

$$\Delta \approx \frac{\alpha \rho_0 \rho_1}{\beta \rho_0 + \gamma \rho_1}, \quad (10)$$

where  $\alpha$ ,  $\beta$ , and  $\gamma$  are positive quantities of the order of unity [6, p. 312]. Taking  $\alpha$ ,  $\beta$ , and  $\gamma$  equal to unity it follows that the largest relative deviation,  $\Delta/(\rho_0 + \rho_1)$ , occurs when  $\rho_0$  and  $\rho_1$  are equal.

Dugdale and Basinski [29] explained the deviations from Matthiessen's rule observed in dilute copper alloys and silver alloys on the bases of an argument by Ziman [30] concerning the different anisotropies of relaxation times for phonon and impurity scattering. While their expression for  $\Delta$  is equivalent to that obtained by Sondheimer and Wilson, the interpretation and temperature dependence are quite different.

Ziman divided electrons into two groups: (1) the belly electrons associated with the spherical portions of the Fermi surface and (2) the neck electrons associated with the portions of the Fermi surface which approach or make contact with the zone boundary. The electron-phonon interactions responsible

for  $\rho_i$  are of two types: normal processes in which the sum of the wave vectors remain unchanged, and Umklapp processes in which the sum of the wave vectors changes by a reciprocal lattice vector. In the case of belly electrons, Umklapp processes can take place only with phonons having a minimum wave vector and minimum energy determined by the differences between the belly electron wave vector and the nearest wave vector in the next Brillouin zone. At low temperatures such phonons disappear and the Umklapp processes involving belly electrons are frozen out. On the other hand, the neck electrons have wave vectors arbitrarily near to the zone boundary and Umklapp processes can occur at both low and high temperatures. Thus, while the strength of the electron-phonon interactions may be about the same for the two groups at room temperature and above, at low temperatures they are much stronger on the neck than on the belly so that the ratio  $\rho_{ib}/\rho_{in}$  increases from a small value at low temperatures to a constant value at room temperature and above. Ziman considered impurity scattering in the cases of silver in gold and of copper in gold. In the first case the atomic volumes are very nearly equal and the effect of silver atoms on the conduction electrons is due only to differences in the effective potential in the ion core, consequently the belly electrons, which have wave functions that are large in the region of the core, will be strongly scattered, while the neck electrons, which have wave functions that are large only in the region between the cores, will be weakly scattered. In the second case the atomic volumes are very different, and the effect of the copper atoms on the conduction electrons is much like that of a charged impurity, a change in the potential extending out into the region between the ion cores, so that the neck electrons are also scattered strongly.

It follows from the above that on this model if the ratio  $\rho_{ib}/\rho_{in}$  is always smaller than the ratio  $\rho_{ob}/\rho_{on}$ , then  $\Delta$  rises from zero at zero temperature, goes through a maximum, and takes on a constant value at high temperatures; this is the behavior observed in the dilute Cu + Au and Ag + Au alloys investigated by Dugdale and Basinski.

The deviations from Matthiessen's rule in iron alloys [31] and nickel alloys [32] have been explained in terms of parallel conduction by the spin-up and spin-down electrons. At low temperatures there are few interactions between the electrons of opposite spin, so that an electron with its spin in a given direction emerging from an impurity collision will reach the next impurity with

its spin unchanged; thus an impurity resistivity can be defined for each group. If the impurity relaxation time is considerably longer for one group of electrons than for the other, the scattering of electrons of that group will contribute relatively little to the residual resistivity. At high temperatures, on the other hand, there may exist (presumably as a result of electron-electron collisions) a process of spin mixing in which the electrons change their spin directions a number of times between impurity collisions. There will then be no group of electrons that experience relatively small impurity scattering, and the high-temperature impurity resistivity will become considerably greater than the low-temperature residual resistivity.

Deviations from Matthiessen's rule have been comprehensively reviewed by Bass [33] and by Cimberle et al. [34].

#### f. Electrical Resistivity of Alloys

Alloying affects the electrical resistivity of the host metal drastically, since it causes lattice perturbations, modifies the lattice-vibrational spectrum and electronic band structure, changes the Fermi energy and electronic density of states, etc.

For the residual electrical resistivity of binary solid-solution alloys containing  $c$  mole fraction of element A and  $(1 - c)$  mole fraction of element B, Nordheim's rule [35] is

$$\rho_0 \propto c(1 - c). \quad (11)$$

This rule is applicable only to homogeneous random (not ordered) solid solutions of two metals which are not transition elements. It is apparent from eq. (11) that  $\rho_0$  is directly proportional to  $c$  if  $c \ll 1$ .

Nordheim's rule can be made plausible by the following argument. The average cellular potential for a binary alloy with mole fraction  $c$  of element A in element B is given by

$$\phi = c\phi_A + (1 - c)\phi_B \quad (12)$$

and the perturbations of this potential at the sites of the A and B atoms are

$$\delta\phi_A = \phi_A - \phi = (1 - c)(\phi_A - \phi_B) = (1 - c)\phi_{AB} \quad (13)$$

$$\delta\phi_B = \phi_B - \phi = c(\phi_B - \phi_A) = -c\phi_{AB}. \quad (14)$$

In the Born approximation, the scattering cross section per scattering center is proportional to the square of the matrix element of the perturbation. The probability per unit time that an electron will be scattered from state  $\psi_k$  into state  $\psi_{k'}$  by a single A atom is  $(1-c)^2 U^2$  where

$$U = \int \psi_{k'}^* \phi_{AB} \psi_k d^3r \quad (15)$$

and  $\psi_k$  and  $\psi_{k'}$  are the initial and final wave functions. The probability per unit time that an electron will be scattered by a single B atom is  $c^2 U^2$ .

Since the fractions of A atoms and B atoms are  $c$  and  $(1-c)$ , the total scattering probability per unit time, and therefore the residual resistivity, is of the form

$$\rho_0 \propto [c(1-c)^2 + (1-c)c^2]U^2 = c(1-c)U^2. \quad (16)$$

This rule is obeyed quite well by many binary alloy systems, but not by those in which either constituent is a transition element.

In dilute solid-solution alloys, Norbury [36] found that the increase of residual resistivity per atomic percent solute,  $\delta\rho_0$ , increases with the valence difference,  $\delta Z$ , between the solute and the solvent atoms. By analyzing thoroughly the data on this effect, Linde [37-39] concluded that  $\delta\rho_0$  is proportional to the square of  $\delta Z$ , i.e.,

$$\delta\rho_0 = a + b(\delta Z)^2, \quad (17)$$

where  $a$  and  $b$  are constants for a given solvent metal and a given row of the periodic table to which the solute elements belong. This is the Norbury-Linde's rule.

A rough theoretical justification for Norbury-Linde's rule was given by Mott [40], who calculated  $\delta\rho_0$  on the assumption that the electrons behave as if they were free and used a screened Coulomb potential to describe the perturbation, and used the Born approximation to calculate the scattering cross section. For a solute in the same row of the periodic table, in which case  $a$  is zero, he obtained

$$\delta\rho_0 = \frac{2\pi(\delta Z)^2 e^2}{100 \text{ mv}^2} \left[ \ln \left( 1 + \frac{4m^2 v^2}{q^2 \hbar^2} \right) - \left( 1 + \frac{q^2 \hbar^2}{4m^2 v^2} \right)^{-1} \right]. \quad (18)$$

Here  $e$ ,  $m$ , and  $v$  are the electronic charge, mass, and velocity,  $q$  is the screening parameter, and  $\hbar$  is the reduced Planck constant. The  $(\delta Z)^2$  dependence is obtained because in the Born approximation the scattering is proportional to the square of the matrix element of the perturbation. The Born approximation overestimates the scattering for this type of potential so that eq. (18) is only an approximation; however, fair agreement with experiment can be obtained by taking the screening parameter to be somewhat greater than the usual estimate. Better estimates of  $\delta\rho_0$  can be obtained by the more refined phase-shift calculations, but these do not yield the simple  $(\delta Z)^2$  dependence.

Norbury-Linde's rule is not obeyed when the alloying involves transition elements or polyvalent elements such as aluminum and magnesium.

Another factor influencing the residual resistivity of alloys is short range order which may either decrease it, as in the case of  $\alpha$  brass [41], or increase it as in the case of copper-gold alloys [42,43]. Short range order is increased by annealing and destroyed by mechanical deformation and by heating and quenching.

In the case of binary alloy systems whose alloys are two-phase ( $\alpha + \beta$ ) mixtures, the electrical resistivity of the alloys is equal to the sum of the product of the electrical resistivity of each phase and its volume fraction, i.e.,

$$\rho = \rho_\alpha V_\alpha + \rho_\beta V_\beta, \quad (19)$$

where  $\rho_\alpha$  and  $\rho_\beta$  are the electrical resistivities and  $V_\alpha$  and  $V_\beta$  are the volume fractions of  $\alpha$  and  $\beta$  phases, respectively. Equation (19) is derived by considering the alloy as a rod which consists of a large number of small fibers in parallel, with each fiber having a certain volume of  $\alpha$ -phase in series with a certain volume of  $\beta$ -phase.

#### g. Electrical Resistivity of Transition Elements and Their Alloys

In terms of electronic transport properties, the most important feature of transition elements, at least those having face-centered cubic crystal structure such as nickel and palladium, is the presence of a narrow d-band with a high density of states overlapping the conduction band at the Fermi energy level. As originally suggested by Mott [44], the role of this d-band



is to provide a large number of levels into which the s-electrons can be scattered and lost from the current, thus causing the high electrical resistivity of transition elements as compared with that of ordinary metals.

As palladium is added to silver, the resistivity first increases in the way observed for binary alloys of ordinary metals. At a certain composition, about 40 atomic percent palladium, the number of conduction electrons is no longer sufficient to fill the d-band holes and the resistivity begins to increase rapidly, rising to an asymmetric peak. Mott [44] provided a semi-quantitative explanation of the shape of the curve which may be summarized as follows. The resistivity was taken to be the sum of terms due to s-s and s-d scattering. The resistivity due to s-s scattering was taken to have the composition dependence given by Nordheim's rule while that due to s-d scattering was taken to be proportional to the density of vacant states in the d-band, which in turn was taken to vary in the same way as the paramagnetic susceptibility, that is, as  $(p - c)^2$  for  $c < p$ , and zero for  $c > p$  for atomic fraction  $c$  of silver in palladium, where  $p$  is the number of positive holes per atom in pure palladium. Also, while silver and palladium in the alloy were assumed to have a common s-band, the d-band was assumed to be split into two bands and the one associated with silver atoms was taken to be below that associated with palladium atoms and thus below the Fermi energy level. With these assumptions Mott obtained the following equation for the electrical resistivity of the alloys:

$$\rho = A(p - c)^2(1 - c)c^2 + B(1 - c)c, \quad (20)$$

where  $A$  and  $B$  are constants. If  $A$  and  $B$  are properly chosen, eq. (20) will result in a resistivity-composition curve similar to that observed experimentally for this alloy system.

Coles and Taylor [45] used a similar model to explain the shape of this resistivity-composition curve but derived the density of states of the d-band from the electronic specific heat. Later, Dugdale and Guénault [46] modified Mott's model to make it consistent with the findings of Vuillemin and Priestley [47] concerning the Fermi surface of palladium. With this model they were able to explain not only the shape of the resistivity-composition curve but also the low-temperature thermoelectric power of these alloys.

The temperature dependence of the electrical resistivity of nickel and iron shows an anomaly caused by their being ferromagnetic: while above the Curie temperature the resistivity has the usual concave-downward curvature, below the Curie temperature it is concave upward. This behavior can be explained on the basis of a band model for the d-electrons in terms of s-d scattering, and on the basis of a model in which the d-electrons are localized on their atoms in terms of spin-disorder scattering. Mott and Stevens [48] argued that the former model applied to nickel and the latter to iron.

In the band model the narrow d-band overlaps the s-band. Above the Curie temperature, vacant d-states of both spin directions are present in equal numbers. Below the Curie temperature, there is a greater number of vacant states with spin up in a given domain and the number increases with decreasing temperature, while the number with spin down decreases until at zero temperature all of the spin down states are filled. If only states with spin up are vacant, then a spin-down electron cannot make a spin-conserving transition into the d-band, so that its mean free path is greater than in the unmagnetized state. Assuming that the energy is proportional to the square of the wave vector so that the density of d-states at the Fermi level,  $N_d(\epsilon_F)$ , is proportional to the square root of the Fermi energy, noting that the Fermi wave number, is proportional to the cube root of the electron density, one obtains

$$N_{d_{\pm}}(\epsilon_F) \propto \left(1 \mp \frac{M}{M_0}\right)^{1/3}, \quad (21)$$

where  $M$  is the net magnetization, reaching the saturation value  $M_0$  at  $T = 0$  K, and  $\pm$  refers to spin up and spin down electrons. Since the reciprocal relaxation time, and thus the resistivity due to s-d transitions, is proportional to  $N_d(\epsilon_F)$ , one obtains

$$\rho_{\pm} = \rho_{ss} + \rho_{sd} \left(1 \mp \frac{M}{M_0}\right)^{1/3}. \quad (22)$$

In an unmagnetized specimen there are many domains in which the magnetization of the d-spin is in different directions so that, instead of having parallel conduction by spin-up and spin-down electrons, both types of electrons see an average resistivity:

$$\rho = \rho_{ss} + \frac{\rho_{sd}}{2} \left[ \left( 1 - \frac{M}{M_0} \right)^{1/3} + \left( 1 + \frac{M}{M_0} \right)^{1/3} \right]. \quad (23)$$

For small  $M/M_0$  this reduces to

$$\rho \approx \rho_{ss} + \rho_{sd} \left[ 1 - \frac{1}{9} \left( \frac{M}{M_0} \right)^2 \right], \quad (24)$$

which is in qualitative agreement with the experimental data for nickel.

If the d-electrons are localized on their atoms, then instead of s-d scattering, the theory proposes an s-d exchange interaction with an energy depending on the relative orientation of the spin of a moving s-electron to that of the fixed d-electron near which it is passing. If the spins of the d-electrons in a domain are lined up parallel, then the electrons would not be scattered, but as the temperature is increased, the d-spins within the domain become disordered, and there is an extra contribution to the resistivity which increases rapidly as the Curie temperature is approached and takes a constant value above the Curie temperature.

Since s-d scattering due to the electron-phonon interaction continues to increase with temperature above the Curie temperature, while the spin-disorder scattering takes on a constant value, the two mechanisms can be distinguished experimentally. Coles [49] predicts the former behavior for a Ni-Pd alloy and the latter behavior for an Fe-Rh alloy.

## 2.2. Data Evaluation and Generation of Recommended Values

The recommended electrical resistivity values were generated through critical evaluation, correlation, analysis, and synthesis of the available experimental data and information compiled from all sources. The procedure involved critical evaluation of the validity of available data and related information, judgment on the reliability and accuracy of the data, resolution, and reconciliation of disagreements in conflicting data, correlation of data in terms of various controlling parameters, curve fitting with theoretical or empirical equations, and synthesis of the often fragmentary data to generate a full range of coverage of internally consistent "best" values.

In the critical evaluation of the validity and reliability of a set of experimental electrical resistivity data, the temperature dependence of the data was examined, and any unusual dependence or anomaly was carefully investigated. The experimental technique was reviewed to see whether the actual boundary conditions in the measurement agreed with those assumed in the theory, and the estimation of inaccuracies by the authors was checked to ensure that all the possible sources of error were considered. The sources of error might include the inaccuracy in the measurement of specimen dimensions and of the distance between potential probes, uncertainty due to the effect of thermal expansion, inaccuracy in temperature measurement, inaccuracy due to poor sensitivity of measuring devices or circuits, uncertainty at high temperature due to specimen instability or specimen and/or thermocouple contamination, etc. These and other possible sources of error were carefully considered in critical evaluation of experimental data.

The uncertainty of a set of experimental data is caused, however, not only by the experimental error in the measurement, but also by the inadequacy of characterization of the material for which the data are reported. Therefore, in data analysis it should be kept in mind that the total difference between two sets of experimental data is the sum of the difference due to experimental error and the real difference due to sample variance. It was found in this and other studies that the chemical composition of a specimen reported by the author was often unreliable. This might partly be due to the fact that in many cases the stated composition of a specimen was that the author obtained from the company who supplied the specimen and it could at best represent only the

nominal composition; the actual composition varied from sample to sample. In other cases there was a strong tendency for only certain particular elements to be detected by a particular chemical analysis which could miss other important constituents. Furthermore, the chemical composition of a specimen might change when it was measured at high temperature.

In many cases research papers did not contain adequate information for performing a truly critical evaluation. In these cases, some other considerations were used for data evaluation. For instance, if several author's data agreed with one another and, more importantly, these were obtained by using different experimental methods, these data were judged to be reliable. However, if the different sets of data were obtained by means of the same experimental method, even though they all agreed, the reliability of the data was still subject to questioning, because they might all suffer from a common, but unknown, source of error. Secondly, if the same apparatus had been used for measurements of other materials and the other results were reliable, the result for the new material was judged also to be reliable. If the information given by the author was entirely inadequate to make any value judgment, the data assessment was subjective. At times, judgments were based upon factors and considerations such as the central purpose of his research, the motivation for his measurement, general knowledge of the experimenter, his past performance, the reputation of his laboratory, etc.

In the process of critical evaluation of experimental data outlined above, unreliable and erroneous data were noted and set aside. The remaining data were then used for data correlation and synthesis, and graphical smoothing was often used. In graphical smoothing of experimental data for a binary alloy system, cross-plotting from electrical resistivity data versus temperature to resistivity data versus composition and vice versa were made. Smooth curves were drawn which approximate the best fit to the resistivity data versus temperature, and points from the smoothed curves were used to construct resistivity-versus-composition curves for a convenient set of selected temperatures. In a resistivity-versus-composition graph, the family of isotherms was similar and any required smoothing of the data could be done more easily and with greater confidence than when working directly with the resistivity-versus-temperature curves. The points from the resulting smoothed curves were then used to construct resistivity-versus-temperature curves for selected compositions, and these curves

were further smoothed. In the graphical smoothing process it was extremely important that the alloy phase diagrams be constantly consulted and the phase boundaries between solid solutions and/or mechanical mixtures and the boundaries of magnetic transitions be kept in mind, so as to be aware of any possible discontinuity or sudden change of slope in the resistivity curves.

In graphical smoothing and synthesis of data for a binary alloy system, instead of cross-plotting the total electrical resistivity,  $\rho(c,T)$ , it was often better to work with the temperature-dependent part,  $\rho_1(c,T)$ , of the electrical resistivity, i.e.,

$$\rho_1(c,T) = \rho(c,T) - \rho_0(c). \quad (25)$$

This temperature-dependent part changes more slowly with the alloy composition,  $c$ , than does the total electrical resistivity, and its isotherms may form a family more convenient for cross-plotting. Furthermore, for a binary alloy system that forms a continuous series of solid solutions over the entire range of compositions, it has proved very useful to go further in this direction by working with the quantity

$$\Delta(c,T) = \rho_1(c,T) - \bar{\rho}_1(c,T), \quad (26)$$

where  $\bar{\rho}_1$  is the atomic-fraction-weighted average of the intrinsic resistivities,  $\rho_1^{(A)}$  and  $\rho_1^{(B)}$ , of the two pure metals A and B:

$$\bar{\rho}_1(c,T) = c\rho_1^{(A)}(T) + (1-c)\rho_1^{(B)}(T) \quad (27)$$

with  $c \equiv c_A$  of metal A.

Since the quantity  $\Delta$  is at most a few percent of  $\rho$ , irregularities and discrepancies in the values of  $\rho$  become very conspicuous in plots of  $\Delta$  or  $\Delta/T$ , and this increases the confidence with which one can reject some data as aberrant or unreliable. In the low-temperature region it is better to cross-plot  $\Delta$  versus  $T$  or  $c$ , while at high temperature cross-plotting  $\Delta/T$  versus  $T$  or  $c$  is preferred. Cross-plotting of  $\Delta$  is convenient up to 300 K and cross-plotting of  $\Delta/T$  down to 100 K; the range of overlap in  $T$  makes it useful to employ both types of plot in relating low-temperature data to high-temperature data. As an example to illustrate the application of this method and procedure, a detailed account of analysis and synthesis of the electrical resistivity data for the gold-silver alloy system is presented in Appendix 5.1.

Combining eqs (25) to (27) gives an expression for the electrical resistivity of a binary alloy:

$$\rho(c,T) = \rho_0(c) + c\rho_1^{(A)}(T) + (1-c)\rho_1^{(B)}(T) + \Delta(c,T). \quad (28)$$

This expression reduces to eq (2) for an impure metal for which  $c$  approaches zero.

### 2.3. Presentation of Data and Information

In this work, the term "binary alloy system" refers to the full range of composition of two alloying elements and is signified by a hyphen between the two elements, such as aluminum-copper alloy system. The term "binary alloys" refers to a group of binary alloys in which the first alloying element is predominant and is signified by a plus between the two elements, such as aluminum + copper alloys. In specifying the composition of an alloy, weight percent is denoted by % and atomic percent by At.% or at.%.

In each of the subsections in Section 3, electrical resistivity data and information for each alloy system are presented in the following order:

- (1) A discussion text,
- (2) A table of recommended values given as a function of both temperature and composition,
- (3) Two figures presenting recommended values as a function of temperature,
- (4) Two comparable figures presenting experimental data as a function of temperature,
- (5) Two tables giving measurement information on the experimental data presented in the two figures of item (4),
- (6) Two comparable tables tabulating experimental data of all the data sets presented in the two figures of item (4) and/or listed in the two tables of item (5),
- (7) A figure presenting both recommended values and experimental data as a function of composition,
- (8) A table giving measurement information on the experimental data presented in the figure of item (7), and
- (9) A comparable table tabulating experimental data of all the data sets presented in the figure of item (7) and/or listed in the table of item (8).

Thus, there are normally five figures and seven tables for each alloy system, excepting that some additional figures presenting other useful information are also given for a few particular alloy systems.

In the discussion text on the electrical resistivity of each alloy system, individual pieces of available data and information are reviewed, details of data analysis and synthesis are given, the considerations involved in arriving



at the final assessment and recommendation are discussed, the recommended values and the experimental data are compared, and the uncertainties of the recommended values are stated.

In the table of recommended values, those values which are provisional are indicated each by a double dagger ( $\ddagger$ ). The designation as recommended or provisional values depends upon the level of confidence placed on the values and, hence, upon the uncertainty in the values assigned. The uncertainty in the recommended electrical resistivity values is  $\pm 5\%$  or smaller and that in the provisional values is greater than  $\pm 5\%$ .

Values in a temperature range where no experimental data are available are indicated each by an asterisk (\*). However, it is important to note that in many cases whether experimental data are available or not in a particular temperature range has no bearing on the accuracy or uncertainty in the recommended values generated for that temperature range. It is because in many cases the available experimental data in a particular temperature range are those already rejected as erroneous or unreliable and therefore not used at all in the generation of recommended values for that temperature range, and in many other cases the recommended values for a particular temperature range were generated from accurate values in other ranges and therefore are accurate even though no experimental data are available in that particular temperature range. In other words, recommended values with asterisks do not necessarily have larger uncertainty than those without, and in many cases the opposite is true.

The recommended (or provisional) values in some of the tables are given with more significant figures than warranted, which is merely for tabular smoothness or for the convenience of internal comparison. Hence, the number of significant figures given in the table has no bearing on the degree of accuracy or uncertainty in the values; the uncertainty in the values is always explicitly stated.

For each of the ten binary alloy systems except three (aluminum-copper, aluminum-magnesium, and copper-zinc), the recommended (or provisional) values are given for 27 compositions: 0(pure element), 0.5, 1, 3, 5, 10(5)95, 97, 99, 99.5, and 100%(pure element). The alloy compositions in atomic percent corresponding to these in weight percent are also specified. For aluminum-copper alloy system, recommended values are presented for 26 compositions,

without that containing 80% copper. For aluminum-magnesium alloy system, recommended values are generated only for 12 compositions, lacking those containing 15 to 85% magnesium. Recommended values are given for only 11 compositions of the copper-zinc alloy system, and no values are for those with 35 to 99.5% zinc. For most of the compositions the recommended values cover the temperature range from 1 K to near the solidus temperature (melting starting point). The recommended values have been smoothed simultaneously over both temperature and composition dependences.

The recommended values for the elements are for well-annealed high-purity specimens of the respective elements; however, those values for temperatures below about 100 K are applicable only to the particular specimens having residual electrical resistivities as given at 1 K in the tables. The recommended values generated for the alloys are for those which are not ordered and have not been severely cold-worked or quenched; the electrical resistivity values would be lower for ordered alloys and higher for cold-worked alloys at low temperatures. Furthermore, the values generated are based on the room-temperature dimensions of the alloys, as thermal expansion corrections have not been made. This is due to the fact that the available experimental data on the electrical resistivity of the alloys, upon which the recommendations are based, are not corrected for thermal expansion, and that thermal expansion values for the respective alloys are not available for such corrections to be made. If the values of thermal expansion,  $\Delta L(T)/L_0$ , for the alloys are available, the electrical resistivity values corrected for thermal expansion,  $\rho_{\text{corrected}}$ , can easily be calculated from the given values uncorrected for thermal expansion,  $\rho_{\text{uncorrected}}$ , by the following relation:

$$\rho_{\text{corrected}}(T) = \left( 1 + \frac{\Delta L(T)}{L_0} \right) \rho_{\text{uncorrected}}(T), \quad (29)$$

where  $\Delta L = L - L_0$ , and  $L$  and  $L_0$  are the lengths of the specimen at any temperature  $T$  and at a reference temperature  $T_0$ , respectively. The thermal expansion correction amounts roughly to about -0.2% to -0.7% at very low temperatures, zero at room temperature, about 0.3% to 0.7% at 500 K, and about 2% near the melting point of the alloy.

In the figures presenting recommended (or provisional) electrical resistivity values as a function of temperature, continuous (solid) curves represent recommended values and long-dashed curves represent provisional values. The short-dashed

portion of any of the above two types of curves represents values in that temperature range where no experimental data are available. In a particular temperature range experimental data are considered to be available for a particular alloy composition for which recommended values have been generated if the available experimental data are for a composition closer to that particular composition than to the next composition for which recommended values have also been generated. If the available experimental data are for a composition in the middle of two compositions for which recommended values have been generated, experimental data are considered available for both compositions.

In the figures presenting experimental data, a data set consisting of a single data point is denoted by a number enclosed by a square, and a curve that connects a set of two or more data points is denoted by a ringed number. These data set numbers correspond to those listed in the accompanying tables providing measurement information and tabulating numerical data for each of the data sets. When several sets of data are too close together to be distinguishable, some of the data sets, though listed and tabulated in the tables, are omitted from the figure for the sake of clarity. The data set numbers of those data sets omitted from the figure are asterisked in both tables providing the measurement information and tabulating the experimental data. If only part of the data points of a data set are omitted from the figure, only those data points omitted are asterisked in the table tabulating the experimental data.

The tables providing the measurement information contain for each set of experimental data the following information: data set number, reference number, author(s), year of publication, experimental method used for the measurement, temperature range covered by the data, alloy name and specimen designation, alloy composition, specimen specification and characterization, and information on measurement conditions, which are contained in the original paper. The experimental methods used for the measurement of the electrical resistivity of alloys are indicated in the tables by the following code letters:

- A Direct-current potentiometer method
- B Direct-current bridge method
- C Alternating-current potentiometer method
- E Eddy current decay method

- P Van der Pauw method
- R Rotating magnetic field method
- V Voltmeter and ammeter direct reading method

Details of these and other methods for the measurement of electrical resistivity may be found in the literature references given in Appendix 5.2, which presents a complete scheme for the classification and organization of the methods.

The last column of the table on measurement information with heading "Composition, Specification, and Remarks" should contain the following information on the specimen and its measurement if such information is provided in the original source document:

- (1) Chemical composition,
- (2) Type of crystal and crystalline axis orientation,
- (3) Microstructure and inhomogeneity,
- (4) Specimen shape and dimensions,
- (5) Method and procedure of fabrication,
- (6) Manufacturer, supplier, and stock number,
- (7) Prior heat history and cold-work history,
- (8) Heat treatment, cold working, irradiative and other treatments,
- (9) Test environment such as measured in vacuum, or in nitrogen under pressure,
- (10) Relevant physical properties such as density, transition temperature, Curie temperature, etc., and
- (11) Whether the data were corrected for the thermal expansion of the specimen.

It will be noted, however, that in the majority of cases the authors did not report in their research papers all the necessary pertinent information.

In the tables tabulating the experimental data, all the original data reported in different units have been converted to have the same units: the SI units  $10^{-8} \Omega \text{m}$ . The recommended values generated are also given in the same units. Conversion factors for the units of electrical resistivity, which may be used to convert the electrical resistivity values in the SI units given in this work to values in other units, are given in Appendix 5.3.

It should be noted that in this work the measurement information and experimental data are given in the above two kinds of tables only for the

binary alloys, but not for the pure metals, even though the recommended values for the pure metals are given in the tables of recommended values. The experimental data and measurement information on the nine metallic elements which constitute the ten binary alloy systems together with details of their critical reviews, evaluations, and discussions are presented elsewhere [50-54].

In the figures presenting both recommended values and experimental data as a function of composition, for the sake of clarity recommended or provisional values are presented only for a selected few temperatures and are all represented as continuous (solid) curves. In these figures the alloy compositions are given in atomic percent, with weight percent indicated at the top of the figures.

### 3. ELECTRICAL RESISTIVITY OF BINARY ALLOY SYSTEMS

#### 3.1. Aluminum-Copper Alloy System

The aluminum-copper alloy system does not form a continuous series of solid solutions. The maximum solid solubility of copper in aluminum is 5.7% (2.50 at.%) at 821 K and the solubility decreases to 0.1-0.2% (0.04-0.08 at.%) at 523 K. The maximum solid solubility of aluminum in copper is 9.4% (19.6 at.%) in the range from about 650 to 838 K and the solubility decreases at higher and lower temperatures. Thus the region of solid solution is limited.

There are 154 sets of experimental electrical resistivity data available for this system. Eight of the 27 data sets for Al + Cu alloys listed in table 2, tabulated in table 3, and shown in figure 3 are merely single data points. Most of the data were measured between 250 K and 700 K for alloys containing no more than 15% Cu. Only one single point is available at 4.2 K. Of the 106 data sets for Cu + Al alloys listed in table 4, tabulated in table 5, and shown in figure 4, 36 sets are single data points measured at 4.2 K or at room temperature. Most of the data were measured between room temperature and 1000 K. Three of the 21 resistivity - composition data sets listed in table 6, tabulated in table 7, and shown in figure 5 are for specimens in liquid state.

For the Al + Cu alloys, the resistivity - composition curve at 273 K was first determined in figure 5 following mainly the data of Smith [55] (Al-Cu data set 4). For ordinary alloys, it is usually reasonable to assume that the deviation of the electrical resistivity from the Matthiessen's rule due to alloying is small at low temperatures, say, below 50 K. The resistivity - composition curve at 4.2 K is then drawn parallel to the curve for 273 K. Starting from the values at these two temperatures, the recommended curves for the Al + Cu alloys were drawn according to the temperature dependence of the data of Griffiths and Schofield [56] (Al + Cu data sets 1-5).

For the Cu + Al alloys containing 10% Al or less, the recommended values for the resistivity at 293 K were obtained by drawing a best smooth isotherm so as to agree to within  $\pm 4\%$  with the data of Gaudig and Warlimont [57] (Cu + Al data sets 86, 87), Panin et al. [58,59] (Cu + Al data sets 88, 90, 92, 94), Hibbard [60] (Cu + Al data set 48), Wechsler and Kernohan [61] (Cu + Al data set 43), Gulyaev and Trusova [62] (Al-Cu data set 16), Linde [63] (Cu + Al

data sets 44-46; Cu + Al data set 47 is 10% higher than the recommended value), and Smith and Palmer [64] (Cu + Al data sets 2, 3, 5, 6; Cu + Al data set 4 is 11% higher than the recommended values). For specimens containing more than 10% Al, the isotherm representing the recommended values for the resistivity at 293 K follows the trend of the room-temperature data of Pecijare and Jannsen [65,66] (Cu + Al data sets 33-42), Köster and Rothenbacher [67] (Cu + Al data sets 67-70), and Hishiyama [68] (Al-Cu data set 21). For these more concentrated alloys, it should be noted that some authors, notably Smith and Palmer [64] (Cu + Al data sets 8, 9), Sinha and Prasad [69] (Cu + Al data sets 52, 53), and Griffiths and Schofield [56] (Cu + Al data set 1), have reported resistivities as much as 30% greater than the recommended values. The isothermal resistivities suggested by these authors increase more rapidly as a function of increasing aluminum content than do the recommended values. The discrepancy is largely unexplained due to lack of information, other than to suggest a probable difference in specimen impurities and crystalline structures, which points to a need for detailed specimen characterization in any future work on the resistivity of these alloys. For specimens containing from 15 to 25% Al, the data indicate a maximum in the isothermal resistivity as a function of composition. However, there is insufficient evidence to indicate where the maximum is, and not enough information is available in this region to warrant the recommendation of values for the resistivity of the 20% Al alloy at 293 K or other temperatures.

The recommended values for the residual resistivities of Cu + Al alloys were generated by drawing a best smooth isotherm through the data of Weinberg [70] (Cu + Al data sets 65, 66), Chu and Lipschultz [71,72] (Cu + Al data sets 28, 55), Charlsey and Salter [73,74] (Cu + Al data sets 10-17, 19; Al-Cu data set 19), Kusonoki and Suzuki [75] (Cu + Al data set 20), Wechsler and Kernohan [61] (Cu + Al data set 43), Lindenfeld and Pennebaker [76] (Cu + Al data set 18), Kapoor et al. [77] (Cu + Al data sets 103-105), and Mitchell et al. [78] (Cu + Al data sets 22, 24). The great majority of those data lie within  $\pm 0.5 \times 10^{-8} \Omega \text{m}$  of the recommended isotherm. For alloys containing more than 10% Al, no measurements were made below 250 K except a single point for a 46% Al alloy at 4.2 K. Consequently no recommendations are made below 250 K for the more dense alloys. The recommended values for the resistivities at temperatures above room temperature were based primarily on the data of Smith and Palmer [64] (Cu + Al data

sets 2, 3, 5, 6), Gaudig and Warlimont [57] (Cu + Al data sets 86, 87), and Panin et al. [59] (Cu + Al data sets 92, 94) with some weight given to the data of Jannsen and Pecijare [65,66] (Cu + Al data sets 33-42) and Sinha and Prasad [69] (Cu + Al data sets 50-53).

The resulting recommended electrical resistivity values for Al, Cu, and for 24 Al-Cu binary alloys are presented in table 1 and shown in figures 1, 2, and 5. No values were generated for Cu + 20% Al alloy. The recommended values for Al and for Cu are for well-annealed high-purity specimens, but those values for temperatures below about 100 K are applicable only to Al and Cu having residual electrical resistivities as given at 1 K in table 1. The alloys for which the recommended values are generated are not ordered and have not been quenched or cold-worked severely. For most of the alloys, the recommended values cover a full range of temperature from 1 K to the solidus temperature of the alloy where melting starts. These values are not corrected for the thermal expansion of the material. The estimated uncertainties in the values for the various alloys and for different temperature ranges are explicitly stated in a footnote to table 1. Some of the values in table 1 are indicated as provisional because their uncertainties are greater than  $\pm 5\%$ .



TABLE 1. RECOMMENDED ELECTRICAL RESISTIVITY OF ALUMINUM-COPPER ALLOY SYSTEM†  
[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Al: 100.00% (100.00 At. %) Cu: 0.00% (0.00 At. %)			Al: 99.50% (99.79 At. %) Cu: 0.50% (0.21 At. %)			Al: 99.00% (99.57 At. %) Cu: 1.00% (0.43 At. %)			Al: 97.00% (98.70 At. %) Cu: 3.00% (1.30 At. %)			Al: 95.00% (97.81 At. %) Cu: 5.00% (2.19 At. %)			Al: 90.00% (95.49 At. %) Cu: 10.00% (4.51 At. %)		
T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$	
1	0.00100*		1	0.043*		1	0.089*		1	0.269*		1	0.453*		1	0.934*	
4	0.00100		4	0.043*		4	0.089		4	0.269*		4	0.453*		4	0.934*	
7	0.00100		7	0.043*		7	0.089*		7	0.269*		7	0.453*		7	0.934*	
10	0.00103		10	0.043*		10	0.089*		10	0.269*		10	0.453*		10	0.934*	
15	0.00117		15	0.043*		15	0.089		15	0.269*		15	0.453*		15	0.934*	
20	0.00168		20	0.044*		20	0.090		20	0.270*		20	0.454*		20	0.935*	
25	0.00296		25	0.046*		25	0.092		25	0.273*		25	0.456*		25	0.936*	
30	0.00564		30	0.050*		30	0.095		30	0.276*		30	0.459*		30	0.940*	
40	0.0189		40	0.063*		40	0.109		40	0.289*		40	0.473*		40	0.954*	
50	0.0480		50	0.092*		50	0.138		50	0.318*		50	0.502*		50	0.983*	
60	0.0960		60	0.139*		60	0.186		60	0.366*		60	0.550*		60	1.03*	
70	0.163		70	0.206*		70	0.252*		70	0.432		70	0.616		70	1.10*	
80	0.246		80	0.289*		80	0.335*		80	0.515		80	0.699		80	1.18*	
90	0.340		90	0.383*		90	0.429*		90	0.609		90	0.793		90	1.27	
100	0.443		100	0.485*		100	0.531*		100	0.711		100	0.895		100	1.36	
150	1.010		150	1.05*		150	1.10*		150	1.28		150	1.46		150	1.94	
200	1.593		200	1.63*		200	1.68*		200	1.86		200	2.04		200	2.51	
250	2.167		250	2.21*		250	2.25*		250	2.44		250	2.62		250	3.09	
273	2.429		273	2.48		273	2.51		273	2.70		273	2.88		273	3.36	
293	2.653		293	2.69		293	2.74		293	2.92		293	3.10		293	3.59	
300	2.731		300	2.72		300	2.82		300	3.00		300	3.18		300	3.67	
350	3.292		350	2.45*		350	3.38*		350	3.56		350	3.75		350	4.25	
400	3.854		400	3.24*		400	3.95*		400	4.14		400	4.33		400	4.86	
500	4.992		500	5.15*		500	5.10*		500	5.32		500	5.54		500	6.14	
600	6.155		600	6.32*		600	6.28*		600	6.55		600	6.81		600	7.48	
700	7.353		700	7.41		700	7.50		700	7.82		700	8.12		700	8.89	
800	8.622		800	8.72		800	8.79		800	9.12		800	9.46		800	10.4	
900	10.019		900	10.1		900	10.2		875	10.1		895	9.73		821	10.7	
933.53	10.516*		923	10.4		913	10.3										

† Uncertainties in the electrical resistivity values are as follows:

100.00 Al - 0.00 Cu:  $\pm 2\%$  below 80 K,  $\pm 1\%$  from 80 to 200 K,  $\pm 0.5\%$  above 200 K to 500 K, and  $\pm 2\%$  above 500 K.

99.50 Al - 0.50 Cu:  $\pm 5\%$  below 200 K,  $\pm 2\%$  from 200 to 600 K, and  $\pm 3\%$  above 600 K.

97.00 Al - 3.00 Cu:  $\pm 5\%$  below 200 K,  $\pm 2\%$  from 200 to 600 K, and  $\pm 3\%$  above 600 K.

95.00 Al - 5.00 Cu:  $\pm 5\%$  below 200 K,  $\pm 2\%$  from 200 to 600 K, and  $\pm 5\%$  above 600 K.

90.00 Al - 10.00 Cu:  $\pm 5\%$  below 200 K,  $\pm 3\%$  from 200 to 600 K, and  $\pm 5\%$  above 600 K.

\* In temperature ranges where no experimental data are available.

TABLE 1. RECOMMENDED ELECTRICAL RESISTIVITY OF ALUMINUM-COPPER ALLOY SYSTEM† (continued)

[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Al: 85.00% (93.03 At. %) Cu: 15.00% (6.97 At. %)			Al: 80.00% (90.40 At. %) Cu: 20.00% (9.60 At. %)			Al: 75.00% (87.60 At. %) Cu: 25.00% (12.40 At. %)			Al: 70.00% (84.60 At. %) Cu: 30.00% (15.40 At. %)			Al: 65.00% (81.39 At. %) Cu: 35.00% (18.61 At. %)			Al: 60.00% (77.94 At. %) Cu: 40.00% (22.06 At. %)		
T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$	
1	1.44*		1	1.90*	2.28*	1	2.59*	2.82*	1	2.82*	3.08*	1	2.82*	3.08*	1	2.82*	3.08*
4	1.44*		4	1.90*	2.28*	4	2.59*	2.82*	4	2.59*	2.82*	4	2.59*	2.82*	4	2.59*	2.82*
7	1.44*		7	1.90*	2.28*	7	2.59*	2.82*	7	2.59*	2.82*	7	2.59*	2.82*	7	2.59*	2.82*
10	1.44*		10	1.90*	2.28*	10	2.59*	2.82*	10	2.59*	2.82*	10	2.59*	2.82*	10	2.59*	2.82*
15	1.44*		15	1.90*	2.28*	15	2.59*	2.82*	15	2.59*	2.82*	15	2.59*	2.82*	15	2.59*	2.82*
20	1.44*		20	1.90*	2.28*	20	2.59*	2.82*	20	2.59*	2.82*	20	2.59*	2.82*	20	2.59*	2.82*
25	1.44*		25	1.90*	2.28*	25	2.59*	2.82*	25	2.59*	2.82*	25	2.59*	2.82*	25	2.59*	2.82*
30	1.45*		30	1.91*	2.29*	30	2.60*	2.83*	30	2.60*	2.83*	30	2.60*	2.83*	30	2.60*	2.83*
40	1.46*		40	1.92*	2.30*	40	2.61*	2.84*	40	2.61*	2.84*	40	2.61*	2.84*	40	2.61*	2.84*
50	1.49*		50	1.93*	2.33*	50	2.64*	2.87*	50	2.64*	2.87*	50	2.64*	2.87*	50	2.64*	2.87*
60	1.54*		60	2.00*	2.38*	60	2.69*	2.92*	60	2.69*	2.92*	60	2.69*	2.92*	60	2.69*	2.92*
70	1.60*		70	2.06*	2.45*	70	2.75*	2.99*	70	2.75*	2.99*	70	2.75*	2.99*	70	2.75*	2.99*
80	1.69*		80	2.15*	2.53*	80	2.83*	3.06*	80	2.83*	3.06*	80	2.83*	3.06*	80	2.83*	3.06*
90	1.78		90	2.24*	2.62*	90	2.92*	3.15*	90	2.92*	3.15*	90	2.92*	3.15*	90	2.92*	3.15*
100	1.88		100	2.34*	2.72*	100	3.02*	3.25*	100	3.02*	3.25*	100	3.02*	3.25*	100	3.02*	3.25*
150	2.45		150	2.91*	3.29*	150	3.56*	3.80*	150	3.56*	3.80*	150	3.56*	3.80*	150	3.56*	3.80*
200	3.03		200	3.49*	3.87*	200	4.15*	4.39*	200	4.15*	4.39*	200	4.15*	4.39*	200	4.15*	4.39*
250	3.61		250	4.06*	4.45*	250	4.74*	5.00*	250	4.74*	5.00*	250	4.74*	5.00*	250	4.74*	5.00*
273	3.87		273	4.33*	4.72*	273	5.03*	5.31*	273	5.03*	5.31*	273	5.03*	5.31*	273	5.03*	5.31*
293	4.10		293	4.58	4.98	293	5.31	5.61	293	5.31	5.61	293	5.31	5.61	293	5.31	5.61
300	4.19		300	4.67	5.07	300	5.41	5.71	300	5.41	5.71	300	5.41	5.71	300	5.41	5.71
350	4.79		350	5.31*	5.76*	350	6.16*	6.48*	350	6.16*	6.48*	350	6.16*	6.48*	350	6.16*	6.48*
400	5.42		400	5.99*	6.49*	400	6.94*	7.30*	400	6.94*	7.30*	400	6.94*	7.30*	400	6.94*	7.30*
500	6.76		500	7.42*	8.05*	500	8.62*	9.04*	500	8.62*	9.04*	500	8.62*	9.04*	500	8.62*	9.04*
600	8.19		600	8.95*	9.68*	600	10.4*	10.9*	600	10.4*	10.9*	600	10.4*	10.9*	600	10.4*	10.9*
700	9.69		700	10.6	11.4	700	12.2	12.8	700	12.2	12.8	700	12.2	12.8	700	12.2	12.8
800	11.3		800	12.3	13.3	800	14.2	15.0	800	14.2	15.0	800	14.2	15.0	800	14.2	15.0
821	11.7		821	12.7	13.6	821	14.6	15.4	821	14.6	15.4	821	14.6	15.4	821	14.6	15.4

† Uncertainties in the electrical resistivity values are as follows:

- 85.00 Al - 15.00 Cu:  $\pm 5\%$  below 200 K,  $\pm 3\%$  from 200 to 600 K, and  $\pm 5\%$  above 600 K.  
 80.00 Al - 20.00 Cu:  $\pm 5\%$  below 200 K,  $\pm 3\%$  from 200 to 600 K, and  $\pm 5\%$  above 600 K.  
 75.00 Al - 25.00 Cu:  $\pm 5\%$  below 200 K,  $\pm 3\%$  from 200 to 600 K, and  $\pm 5\%$  above 600 K.  
 70.00 Al - 30.00 Cu:  $\pm 5\%$  below 200 K,  $\pm 3\%$  from 200 to 600 K, and  $\pm 5\%$  above 600 K.  
 65.00 Al - 35.00 Cu:  $\pm 5\%$  below 200 K,  $\pm 3\%$  from 200 to 600 K, and  $\pm 5\%$  above 600 K.  
 60.00 Al - 40.00 Cu:  $\pm 5\%$  below 200 K,  $\pm 3\%$  from 200 to 600 K, and  $\pm 5\%$  above 600 K.

\* Provisional values.

\* In temperature range where no experimental data are available.

TABLE 1. RECOMMENDED ELECTRICAL RESISTIVITY OF ALUMINUM-COPPER ALLOY SYSTEM† (continued)  
[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Al: 55.00% (74.22 Al.%) Cu: 45.00% (25.78 Al.%)		Al: 50.00% (70.20 Al.%) Cu: 50.00% (29.80 Al.%)		Al: 45.00% (65.83 Al.%) Cu: 55.00% (34.17 Al.%)		Al: 40.00% (61.09 Al.%) Cu: 60.00% (38.91 Al.%)		Al: 35.00% (55.91 Al.%) Cu: 65.00% (44.09 Al.%)		Al: 30.00% (50.23 Al.%) Cu: 70.00% (49.77 Al.%)	
T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
1	3.33*	1	3.64*								
4	3.33*	4	3.64*								
7	3.33*	7	3.64*								
10	3.33*	10	3.64*								
15	3.33*	15	3.64*								
20	3.33*	20	3.64*								
25	3.33*	25	3.64*								
30	3.34*	30	3.65*								
40	3.35*	40	3.66*								
50	3.38*	50	3.69*								
60	3.43*	60	3.74*								
70	3.48*	70	3.79*								
80	3.54*	80	3.85*								
90	3.64*	90	3.92*								
100	3.77*	100	4.00*								
150	4.27*	150	4.52*								
200	4.85*	200	5.15*								
250	5.51*	250	5.86*	250	6.39*	250	7.14*	250	8.27*	250	10.5*
273	5.84*	273	6.22*	273	6.76*	273	7.57*	273	8.75*	273	11.2*
293	6.16	293	6.55	293	7.14	293	7.96	293	9.19	293	11.8
300	6.28	300	6.67	300	7.26	300	8.10	300	9.36	300	12.0
350	7.11*	350	7.55	350	8.21	350	9.12*	350	10.6*	350	13.5*
400	8.00*	400	8.52	400	9.23	400	10.2*	400	11.9*	400	15.2*
500	9.98*	500	10.6	500	11.5	500	12.7*	500	14.7*	500	18.7*
600	12.1*	600	12.8	600	13.8	600	15.3*	600	17.6*	600	22.5*
700	14.2*	700	15.2*	700	16.3*	700	18.0*	700	20.7*	700	26.5*
800	16.5*	800	17.6*	800	18.9*	800	20.9*	800	24.0*	800	30.8*
821	17.0*	821	18.6*	821	20.6*	821	22.9*	821	26.0*	821	33.6*

† Uncertainties in the electrical resistivity values are as follows:

- 55.00 Al - 45.00 Cu:  $\pm 7\%$  below 200 K,  $\pm 3\%$  from 200 to 600 K, and  $\pm 8\%$  above 600 K.
- 50.00 Al - 50.00 Cu:  $\pm 10\%$  below 200 K,  $\pm 3\%$  from 200 to 600 K, and  $\pm 8\%$  above 600 K.
- 45.00 Al - 55.00 Cu:  $\pm 5\%$  up to 600 K and  $\pm 8\%$  above 600 K.
- 40.00 Al - 60.00 Cu:  $\pm 5\%$  up to 600 K and  $\pm 10\%$  above 600 K.
- 35.00 Al - 65.00 Cu:  $\pm 5\%$  up to 600 K and  $\pm 10\%$  above 600 K.
- 30.00 Al - 70.00 Cu:  $\pm 5\%$  up to 600 K and  $\pm 10\%$  above 600 K.

\* Provisional value.

† In temperature range where no experimental data are available.

TABLE 1. RECOMMENDED ELECTRICAL RESISTIVITY OF ALUMINUM-COPPER ALLOY SYSTEM † (continued)

[Temperature, T, K; Electrica Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Al: 25.00% (43.98 At.%) Cu: 75.00% (56.02 At.%)			Al: 15.00% (29.36 At.%) Cu: 85.00% (70.64 At.%)			Al: 10.00% (20.77 At.%) Cu: 90.00% (79.23 At.%)			Al: 5.00% (11.03 At.%) Cu: 95.00% (88.97 At.%)			Al: 3.00% (6.79 At.%) Cu: 97.00% (93.21 At.%)			Al: 1.00% (2.32 At.%) Cu: 99.00% (97.68 At.%)		
T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$	
250	15.3*		1	8.10*		1	7.32*		1	6.30*		1	6.30*		1	2.92*	
273	16.3*		4	8.10*		4	7.32*		4	6.30*		4	6.30*		4	2.92*	
293	17.2*		7	8.10*		7	7.32*		7	6.30*		7	6.30*		7	2.92*	
300	17.6*		10	8.10*		10	7.32*		10	6.30*		10	6.30*		10	2.92*	
350	19.8*		15	8.11*		15	7.34*		15	6.30*		15	6.30*		15	2.92*	
400	22.2*		20	8.12*		20	7.36*		20	6.30*		20	6.30*		20	2.92*	
500	27.2*		25	8.15*		25	7.39*		25	6.31		25	6.31		25	2.93*	
600	32.6*		30	8.18*		30	7.42*		30	6.33		30	6.33		30	2.94*	
700	38.4*		40	8.22*		40	7.49*		40	6.37		40	6.37		40	2.97*	
800	44.6*		50	8.29*		50	7.55*		50	6.42		50	6.42		50	3.00*	
900	51.1*		60	8.36*		60	7.61*		60	6.48		60	6.48		60	3.04*	
939	53.8*		70	8.41*		70	7.69*		70	6.50		70	6.50		70	3.08*	
			80	8.52*		80	7.78*		80	6.58		80	6.58		80	3.13*	
			90	8.61*		90	7.83*		90	6.62		90	6.62		90	3.17*	
			100	8.71*		100	7.92*		100	6.68		100	6.68		100	3.22*	
			150	9.29*		150	8.36*		150	6.92		150	6.92		150	3.52*	
			200	9.99*		200	8.79*		200	7.27		200	7.27		200	3.59*	
			250	10.5*		250	9.22*		250	7.68		250	7.68		250	4.29	
			273	10.8*		273	9.43*		273	7.87		273	7.87		273	4.46*	
			293	11.0*	12.3*	293	9.61*		293	8.01		293	8.01		293	4.60	
			300	11.1*		300	9.68*		300	8.10		300	8.10		300	4.65	
			350	11.7*		350	10.2*		350	8.59		350	8.59		350	5.06	
			400	12.3*		400	10.7*		400	9.10		400	9.10		400	5.37	
			500	13.4*		500	11.7*		500	10.1		500	10.1		500	6.11*	
			600	14.6*		600	12.7*		600	11.0		600	11.0		600	6.59*	
			700	15.8*		700	13.7*		700	12.0		700	12.0		700	7.75*	
			800	16.9*		800	14.8*		800	13.9*		800	13.9*		800	8.67*	
			900	18.1*		900	15.9*		900	14.1*		900	14.1*		900	9.65*	
			1000	19.3*		1000	17.0*		1000	15.2*		1000	15.2*		1000	10.7*	
			1100	20.5*		1100	18.2*		1100	16.3*		1100	16.3*		1100	11.7*	
			1200	21.6*		1200	19.3*		1200	17.5*		1200	17.5*		1200	12.7*	
			1300	22.9*		1300	20.5*		1300	18.7*		1300	18.7*		1300	13.8*	
			1313	23.0*		1313	20.8*		1313	19.1*		1313	19.1*		1313	14.3*	

† Uncertainties in the electrical resistivity values are as follows:

- 25.00 Al - 75.00 Cu:  $\pm 8\%$  up to 600 K and  $\pm 15\%$  above 600 K.
- 15.00 Al - 85.00 Cu:  $\pm 12\%$ .
- 10.00 Al - 90.00 Cu:  $\pm 12\%$  up to 150 K and  $\pm 10\%$  above 150 K.
- 5.00 Al - 95.00 Cu:  $\pm 7\%$ .
- 3.00 Al - 97.00 Cu:  $\pm 5\%$  below 200 K,  $\pm 3\%$  from 200 to 800 K, and  $\pm 5\%$  above 800 K.
- 1.00 Al - 99.00 Cu:  $\pm 5\%$  below 200 K,  $\pm 3\%$  from 200 to 800 K, and  $\pm 5\%$  above 800 K.

\* Provisional value.

\* In temperature range where no experimental data are available.

TABLE 1. RECOMMENDED ELECTRICAL RESISTIVITY OF ALUMINUM-COPPER ALLOY SYSTEM † (continued)  
(Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ )

Al: 0.50% ( 1.17 At.%) Cu: 99.50% (98.83 At.%)		Al: 0.00% ( 0.00 At.%) Cu: 100.00% (100.00 At.%)							
T	$\rho$	T	$\rho$						
1	1.49	1	0.00200						
4	1.49	4	0.00200						
7	1.49	7	0.00200						
10	1.49	10	0.00202						
15	1.49	15	0.00218						
20	1.49	20	0.00250						
25	1.49	25	0.00450						
30	1.50	30	0.00830						
40	1.51	40	0.0240						
50	1.54	50	0.0520						
60	1.58	60	0.0974						
70	1.62	70	0.154						
80	1.67	80	0.216						
90	1.72	90	0.282						
100	1.77	100	0.349						
150	2.12	150	0.701						
200	2.49	200	1.048						
250	2.97	250	1.388						
273	3.04	273	1.544						
293	3.17	293	1.678						
300	3.21	300	1.725						
350	3.56	350	2.051						
400	3.90	400	2.398						
500	4.62	500	3.079						
600	5.37	600	3.771						
700	6.18	700	4.481						
800	7.02*	800	5.213						
900	7.90*	900	5.973						
1000	8.79*	1000	6.766						
1100	9.71*	1100	7.596						
1200	10.72	1200	8.470						
1300	11.72	1300	9.395						
1354	11.6*	1357.6	9.946(s)						
		1358	21.01(L)						
		1700	24.41						

† Uncertainties in the electrical resistivity values are as follows:

0.50 Al - 99.50 Cu:  $\pm 5\%$  below 200 K,  $\pm 3\%$  from 200 to 800 K, and  $\pm 5\%$  above 800 K.

0.00 Al - 100.00 Cu:  $\pm 3\%$  up to 100 K,  $\pm 1\%$  above 100 K to 250 K,  $\pm 0.5\%$  above 250 K to 350 K,  $\pm 1\%$  above 350 K to 500 K,  $\pm 4\%$  above 500 K to 1357.6 K, and  $\pm 5\%$  above 1357.6 K.

\* In temperature range where no experimental data are available.

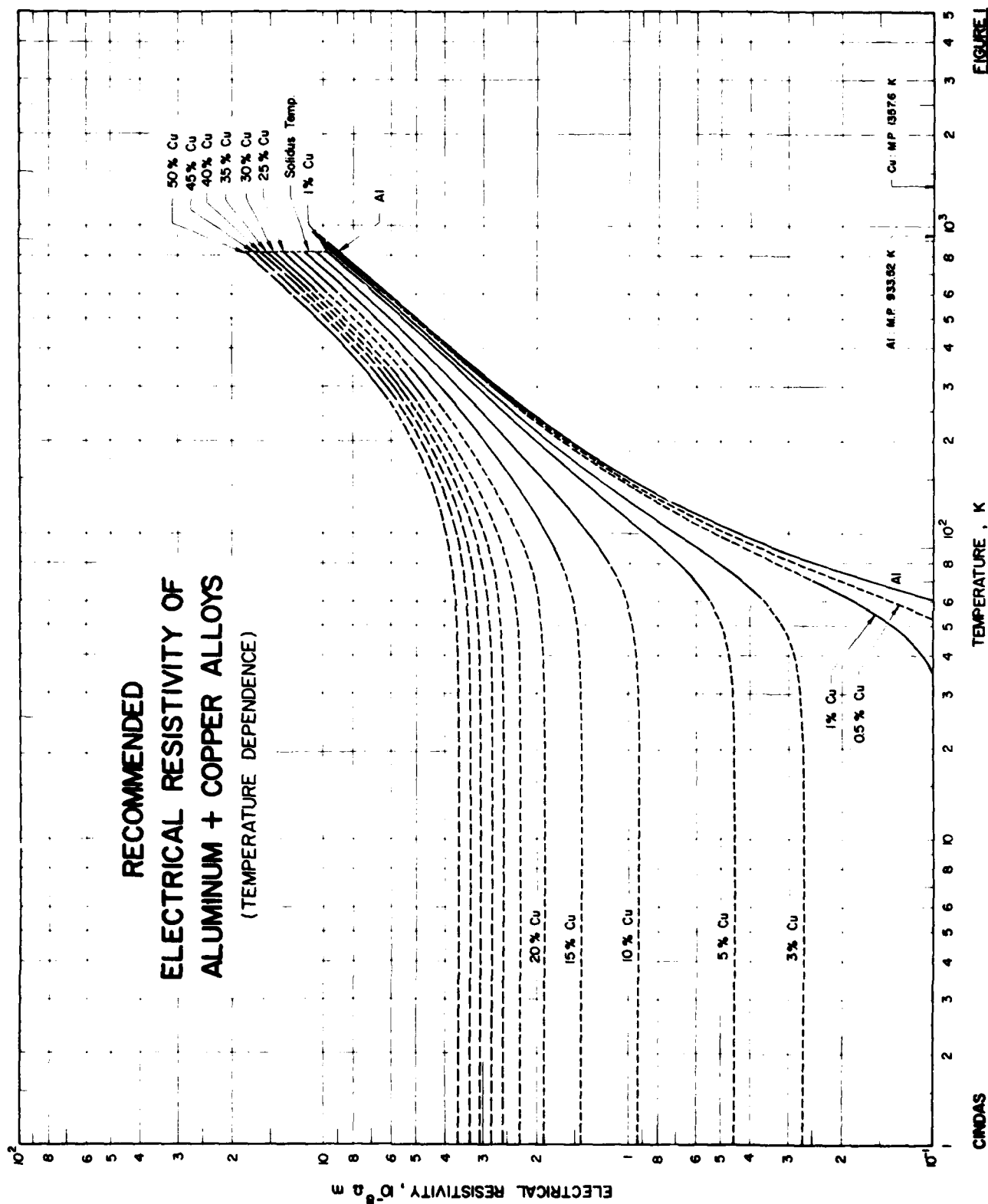


FIGURE 1

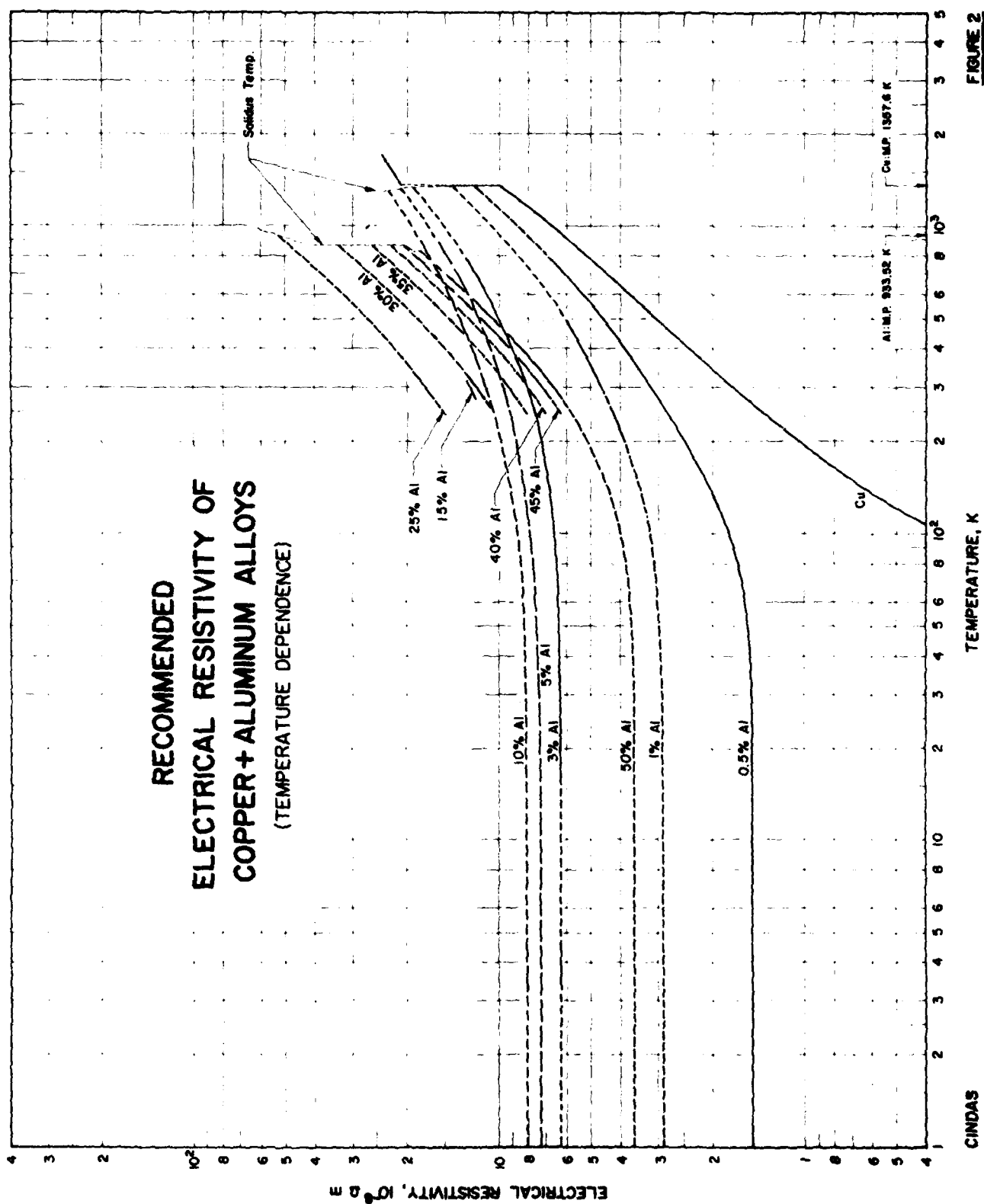


FIGURE 2

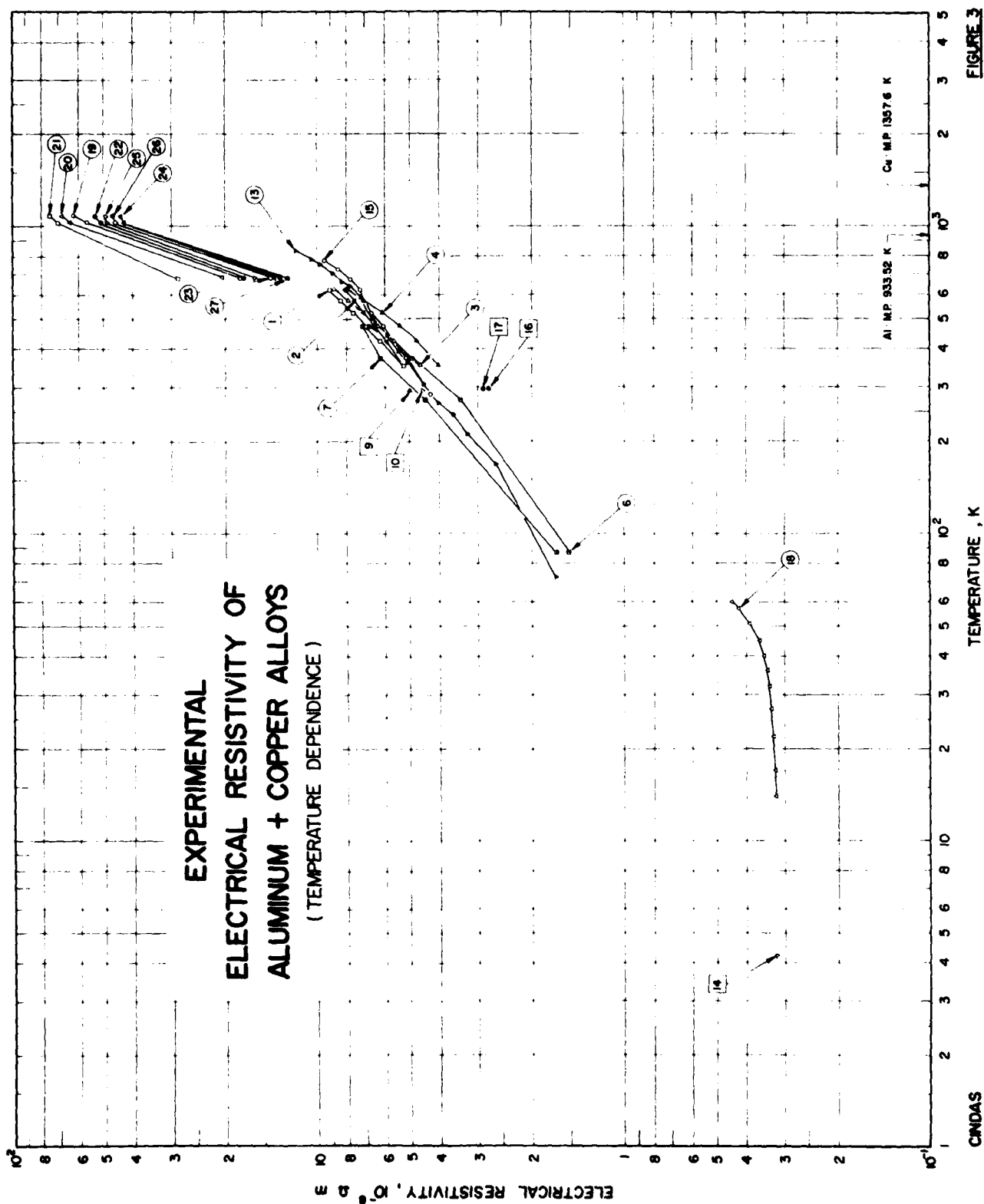


FIGURE 3

CNDS



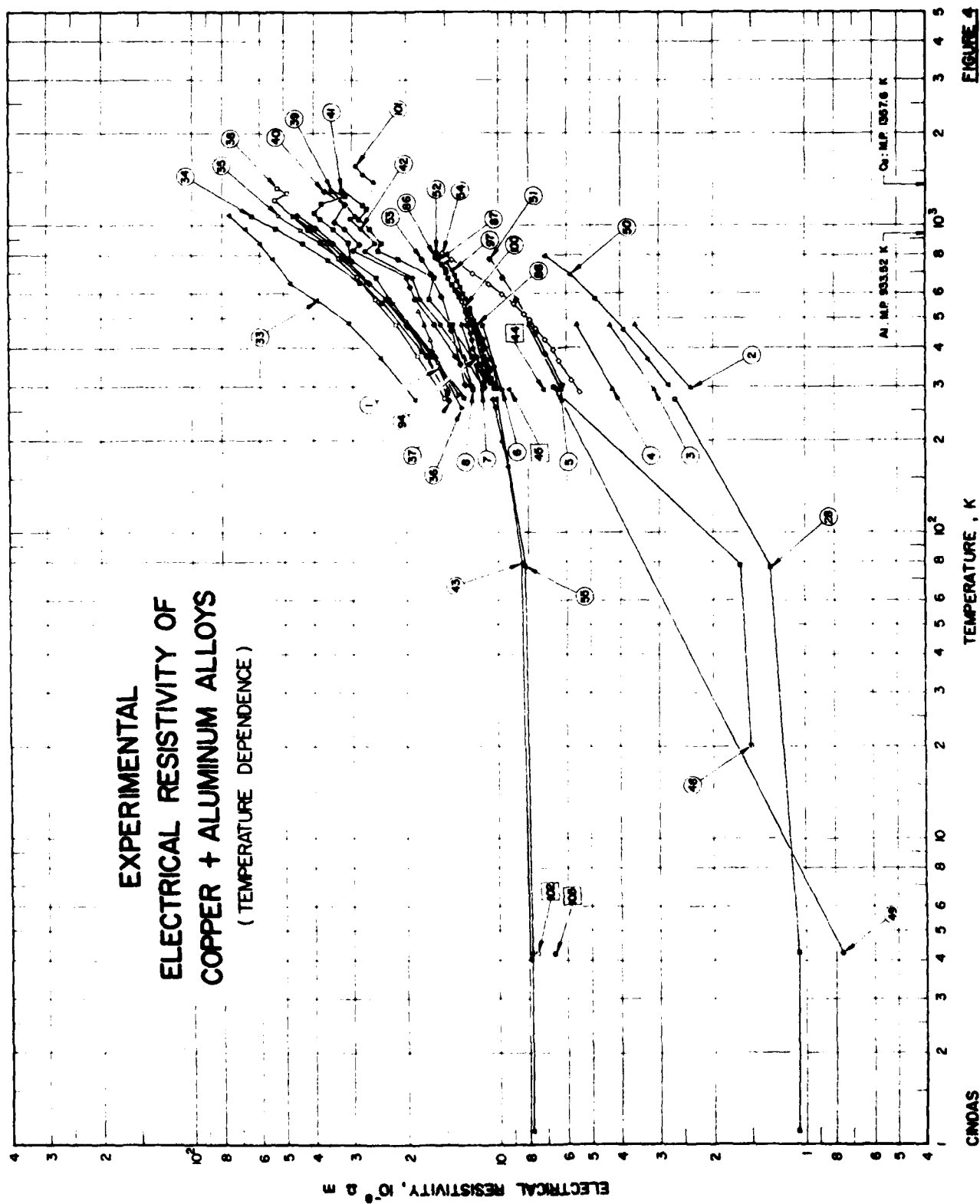


TABLE 2. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF ALUMINUM + COPPER ALLOYS (Temperature Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Al Cu	Composition (continued), Specifications, and Remarks
1	54	Griffiths, E. and Schofield, F. H.	1928	A	353-623	No. 655	86.0 14.0	Cast and annealed at 450 C for 1 hr.
2	56	Griffiths, E. and Schofield, F. H.	1928	A	353-623	No. 671	88.0 12.0	Similar to the above specimen.
3	56	Griffiths, E. and Schofield, F. H.	1928	A	353-623	No. 921	~88.0 ~12.0	Similar to the above specimen.
4	56	Griffiths, E. and Schofield, F. H.	1928	A	353-623	No. 2313	92.0 8.0	Similar to the above specimen.
5*	56	Griffiths, E. and Schofield, F. H.	1928	A	353-623	No. 2312	95.5 4.5	Similar to the above specimen.
6	79	Mauchien, W.	1931	V	87-473		92.0 8.0	Cast.
7	79	Mauchien, W.	1931	V	87-476		0 15.0	Cast.
8*	80	Grand, C. and Villey, J.	1927		373.2		88.0 12.0	Cast.
9	81	Elfein, M.	1937	B	293.2	I, 1	95 5	Cast from 98 to 99 pure Al bar (contamination: <1.0 Fe, <0.9 Si, and <0.1 Cu + Zn) and key alloy (50 Al and 50 Cu) at 750 C, and then cooled in air.
10	81	Elfein, M.	1937	B	293.2	I, 5	95 5	Similar to the above specimen except 99.5 pure Al notch bar (contamination: 0.28 Fe and 0.22 Si) used for the alloying.
11*	81	Elfein, M.	1937	B	293.2	I, 5A	95 5	Similar to the above specimen.
12*	81	Elfein, M.	1937	B	293.2	I, 5B	95 5	Similar to the above specimen.
13	82	Turnbull, D., Rosenbaum, H.S., and Treafie, H.N.	1960		72-839		3.9	Calculated composition (1.7 a/o Cu); 3 x 0.2 x 0.010 cm; prepared by cold swaging, drawing, and rolling; homogenized at 550 C and quenched to 223 K.
14	83	Caplan, A.D. and Rizzuto, C.	1971	A	4.2		1.01	Calculated composition (0.43 a/o Cu); 1.50 mm in diameter; prepared by quenching from the melt, drawing, annealing for several hr just below 660 C, and quenching in water.
15	94	Bollenrath, F. and Hank, V.	1948		285-775		5.08	0.15 Fe, 0.05 each of Si and Zn, 0.02 Ti, and traces Mg and Mn; cast; heated at 794 K for 6 hr and water quenched.
16	85	Robinson, A.T. and Dora, J.E.	1951	B	297.2		0.238	Calculated composition (0.101 a/o Cu); impurities: 0.003 Fe, 0.003 Si, 0.001 Mn, and 0.0006 Mg; sheet specimen 0.070 in. thick and 0.250 in. wide.
17	85	Robinson, A.T. and Dorn, J.E.	1951	B	297.2		0.547	Calculated composition (0.233 a/o Cu); impurities: 0.004 Fe, 0.003 Si, 0.001 Mn, and 0.0006 Mg; sheet specimen 0.070 in. thick and 0.250 in. wide.
18	86	Caplan, A.D. and Rizzuto, C.	1970	A	14-60		1.01	Calculated composition (0.43 a/o Cu); residual resistivity 0.320 $\mu\Omega$ cm.
19	87	Koro'kov, A.M. and Shashkov, D.P.	1962	R	673-1078		1.5	In solid and liquid states.

\* Not shown in figure.

TABLE 2. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF ALUMINUM + COPPER ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Al Cu	Composition (continued), Specifications, and Remarks
20	87	Korol'kov, A. M. and Shashkov, D. P.	1962	R	673-1075		2.7	In solid and liquid states.
21	87	Korol'kov, A. M. and Shashkov, D. P.	1962	R	673-1066		5.5	In solid and liquid states.
22	87	Korol'kov, A. M. and Shashkov, D. P.	1962	R	673-1053		11.5	In solid and liquid states.
23	87	Korol'kov, A. M. and Shashkov, D. P.	1962	R	673-1023		20.0	In solid and liquid states.
24	87	Korol'kov, A. M. and Shashkov, D. P.	1962	R	673-1023		26.8	In solid and liquid states.
25	87	Korol'kov, A. M. and Shashkov, D. P.	1962	R	673-1023		33.2	In solid and liquid states.
26	87	Korol'kov, A. M. and Shashkov, D. P.	1962	R	673-1023		37.9	In solid and liquid states.
27	87	Korol'kov, A. M. and Shashkov, D. P.	1962	R	673-1023		44.1	In solid and liquid states.

TABLE 3.  
EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF ALUMINUM + COPPER ALLOYS (Temperature Dependence)

[illegible]

\* Not shown in figure.

TABLE 4. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + ALUMINUM ALLOYS (Temperature Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Al	Composition (continued), Specifications, and Remarks
1	56	Griffiths, E. and Schofield, F.H.	1928	A	293-523	Aluminum bronze; 6	90.0 10.0	Chill-cast and annealed; $d\rho/dT = 0.00106 \mu\Omega \text{ cm K}^{-1}$ between 293 and 523 K.
2	64	Smith, C.S. and Palmer, E.W.	1935	B	293, 473	100	99.77 0.22	0.01 Fe; rolled, annealed, and cold drawn; heat-treated at 750 C for 2 hr.
3	64	Smith, C.S. and Palmer, E.W.	1935	B	293, 473	101	99.47 0.47	0.02 Fe; similar to the above specimen.
4	64	Smith, C.S. and Palmer, E.W.	1935	B	293, 473	76	99.20 0.71	0.09 Fe; similar to the above specimen.
5	64	Smith, C.S. and Palmer, E.W.	1935	B	293, 473	77	98.08 1.89	0.03 Fe; similar to the above specimen.
6	64	Smith, C.S. and Palmer, E.W.	1935	B	293, 473	45	95.25 4.61	0.14 Fe; similar to the above specimen.
7	64	Smith, C.S. and Palmer, E.W.	1935	B	293, 473	46	92.15 7.72	0.13 Fe; rolled, annealed, and cold drawn; heat-treated at 750 C for 3.5 hr, slowly cooled in furnace.
8	64	Smith, C.S. and Palmer, E.W.	1935	B	293, 473	102	90.56 9.37	0.07 Fe; similar to the above specimen.
9*	64	Smith, C.S. and Palmer, E.W.	1935	B	293, 473	130	87.76 12.15	0.09 Fe; similar to the above specimen.
10*	73	Salter, J.A.M. and Charsley, P.	1967	A	4.2	28	99.17 0.83	Calculated composition (1.93 a/o Al); single crystal; grown in a sealed graphite mold using the Bridgman technique.
11*	73	Salter, J.A.M. and Charsley, P.	1967	A	4.2	2	97.10 0.90	Calculated composition (2.69 a/o Al); polycrystalline; annealed at 750 C for 14 hr in vacuum; grain size 0.008 cm.
12*	73	Salter, J.A.M. and Charsley, P.	1967	A	4.2	2AR	99.17 0.83	Calculated composition (1.93 a/o Al); polycrystalline; grain size 0.0015 cm; 3 mm diameter x 12 cm long; prepared by International Research and Development Co.; cast, machined, swaged, and drawn; annealed at 750 C for 14 hr.
13*	73	Salter, J.A.M. and Charsley, P.	1967	A	4.2	8S	96.69 3.31	Calculated composition (7.45 a/o Al); single crystal; grown in a sealed graphite mold using the Bridgman technique.
14*	73	Salter, J.A.M. and Charsley, P.	1967	A	4.2	8	95.91 4.09	Calculated composition (9.12 a/o Al); polycrystalline; annealed in vacuum at 150 C for 14 hr; grain size 0.0083 cm.
15*	73	Salter, J.A.M. and Charsley, P.	1967	A	4.2	12	94.89 5.11	Calculated composition (11.25 a/o Al); similar to the above specimen.
16*	73	Salter, J.A.M. and Charsley, P.	1967	A	4.2	12(550)	94.72 5.28	Calculated composition (11.60 a/o Al); polycrystalline; grain size 0.0025 cm; 3 mm diameter x 12 cm long; prepared by International Research and Development Co.; cast, machined, swaged, and drawn; annealed in vacuum at 750 C for 14 hr.
17*	73	Salter, J.A.M. and Charsley, P.	1967	A	4.2	12(450)	94.72 5.28	Calculated composition (11.60 a/o Al); polycrystalline; annealed in vacuum at 450 C for 14 hr; grain size 0.0009 cm.
18*	76	Lindenfeld, P. and Pennebaker, W.C.	1962		4.2		99.363 0.617	Calculated composition; prepared from 99.999 pure Cu and 99.99% pure Al; materials melted, outgassed in vacuum, stirred for 0.5 h, then cast; annealed at 700 C for 22 hr.

\* Not shown in figure.

TABLE 4. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + ALUMINUM ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Al	Composition (continued), Specifications, and Remarks	
19*	74	Charaley, P. and Salter, J. A. M.	1965	A	4.2	128	94.89 5.11	Calculated composition (11.25 at/o Al); single crystal grown by Bridgman technique.	
20*	75	Kusunoki, M. and Suruki, H.	1969		4.2	Specimen No. 5	93.03 6.97	Calculated composition (15 at.% Al); single crystal; 2.546 cm <sup>2</sup> in cross-section and about 40 mm long; prepared from 99.999 pure Cu (Mitsubishi-Kinzoku Co. Ltd.) and 99.99 pure Al (Sumitomo-Kinzoku Co. Ltd.) by melting in a high purity graphite crucible by induction heating; grown in a splitting graphite mold by the Bridgman method using a seed crystal; annealed at 1000 C for 48 hr in a vacuum lower than 10 <sup>-5</sup> mm Hg; electrolytically polished in phosphoric acid - ethyl alcohol; dislocation density 5.8 x 10 <sup>10</sup> cm <sup>-2</sup> .	
21*	76	Mitchell, M. A., Klemens, P. G., and Reynolds, C. A.	1971	A	4.2	A		4.5	Obtained from Materials Research Corp., Orangeburg, N. Y.; prepared from 99.999 pure Al and Cu by vacuum induction melting, then machining and swaging to 0.125 in. in diameter; cold-worked in liquid nitrogen, then kept at 293 K for 3 hr.
22*	76	Mitchell, M. A., et al.	1971	A	4.2	B		4.5	Similar to the above specimen A except annealed at 1193 K for 48 hr after cold-work.
23*	76	Mitchell, M. A., et al.	1971	A	4.2	C		4.5	Similar to the above specimen A except annealed at 1123 K for 28 hr after cold-work, then given 9.5% torsional strain at 293 K, reannealed for 12 hr to 48 hr consecutively at various temperatures.
24*	76	Mitchell, M. A., et al.	1971	A	4.2	D		4.5	Same composition, supplier, and fabrication method as the above specimen A except swaged to 3/16 in. in diameter; annealed at 1205 K for 48 hr.
25*	76	Mitchell, M. A., et al.	1971	A	4.2	E		4.5	Similar to the above specimen D except given, after annealing, 9.33% tensile strain at 77 K with maximum stress 28.5 kg mm <sup>-2</sup> and strain rate 0.0093 s <sup>-1</sup> , then reannealed for 12 hr to 48 hr consecutively at various temperatures.
26*	76	Mitchell, M. A., et al.	1971	A	4.2	F		4.5	Similar to the above specimen E except annealed at 1202 K for 48 hr, then given 8.13% tensile strain at 77 K with maximum stress 29 kg mm <sup>-2</sup> and strain rate 0.0081 s <sup>-1</sup> , reannealed for 0.5 to 97 hr consecutively at various temperatures.
27*	76	Mitchell, M. A., et al.	1971	A	4.2	G		4.5	Similar to the above specimen F except given, after annealing, 9.26% tensile strain at 77 K with maximum stress 25.1 kg mm <sup>-2</sup> and strain rate 0.004 s <sup>-1</sup> , reannealed for 0.5 to 48 hr consecutively at various temperatures.
28	71, 72	Chu, T. K. and Lipschultz, F. P.	1972	A	1.1-273	C1		0.43	Supplied by American Anacosta Brass Co.; 0.5 in. diameter x 8 in. long with central 5 in. machined to 0.25 in. in diameter; annealed at 1273 K for 48 hr.
29*	71, 72	Chu, T. K. and Lipschultz, F. P.	1972	A	1.1-273	C2			The above specimen fatigued for 500 cycles with maximum load 6.4 kg mm <sup>-2</sup> .
30*	71, 72	Chu, T. K. and Lipschultz, F. P.	1972	A	1.1-273	C3			The above specimen fatigued for 10 <sup>4</sup> cycles with maximum load 6.4 kg mm <sup>-2</sup> .
31*	71, 72	Chu, T. K. and Lipschultz, F. P.	1972	A	1.1-273	C5			Similar to the above specimen C1 except given a 5% plastic deform under uniaxial stress.

\* Not shown in figure.

TABLE 4. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + ALUMINUM ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Al	Composition (continued), Specifications, and Remarks
32 *	71, 72	Chu, T.K. and Lipschultz, F.P.	1972	A	4.2-273	C6		The above specimen fatigued for $10^5$ cycles with maximum load $6.4 \text{ kg mm}^{-2}$ .
33	65	Pectjare, O. and Janssen, S.	1957		270-1080		18.30	In $\gamma_2$ phase.
34	65	Pectjare, O. and Janssen, S.	1957		285-1075		19.30	Similar to the above specimen.
35	65	Pectjare, O. and Janssen, S.	1957		280-1080		17.05	Similar to the above specimen.
36	65	Pectjare, O. and Janssen, S.	1957		255-1082		17.40	Similar to the above specimen.
37	65	Pectjare, O. and Janssen, S.	1957		270-1072		16.41	Similar to the above specimen.
38	66	Janssen, S. and Pectjare, O.	1957		275-1315		16.4	No details reported.
39	66	Janssen, S. and Pectjare, O.	1957		275-1300		15.15	No details reported.
40	66	Janssen, S. and Pectjare, O.	1957		300-1295		14.9	No details reported.
41	66	Janssen, S. and Pectjare, O.	1957		318-1300		14.5	No details reported.
42	66	Janssen, S. and Pectjare, O.	1957		299-1290		13.9	No details reported.
43	61	Wechsler, M.S. and Kernohan, R.H.	1958	A	4.0-322		6.97	Calculated composition (15 a/o Al); polycrystalline; grain size 200 grains $\text{mm}^{-2}$ ; 0.125 in. in diameter; supplied by Metallurgical Division at ORNL; prepared from 99.92 pure electrolytic Cu and 99.97 pure Al; vacuum-cast, homogenized and swaged, annealed at about 1023 K for several hr, then cooled at $15 \text{ K h}^{-1}$ to room temperature.
44	63	Linde, J.O.	1954	B	293.2		2.5	Calculated composition (5.6 a/o Al); prepared from Johnson, Matthey spectroscopically standardized metals; rolled, homogenized at 1173 K for about 10 hr, rolled to about 0.4 mm <sup>2</sup> in cross-section; annealed in vacuum at 723 to 773 K for 15 hr; density $8.51 \text{ g cm}^{-3}$ .
45	63	Linde, J.O.	1954	B	293.2		4.5	Calculated composition (10.0 a/o Al); similar to the above specimen except density $8.23 \text{ g cm}^{-3}$ .
46 *	63	Linde, J.O.	1954	B	293.2		6.3	Calculated composition (13.6 a/o Al); similar to the above specimen except density $7.98 \text{ g cm}^{-3}$ .
47 *	63	Linde, J.O.	1954	B	293.2		8.6	Calculated composition (18.1 a/o Al); similar to the above specimen except density $7.68 \text{ g cm}^{-3}$ .
48	60	Hibbard, W.R., Jr.	1959		20-299		1.81	Calculated composition (4.15 a/o Al); annealed at 470 C for 2 hr; grain size 0.0117 mm.
49	68	Galewick, J.J. and Wasik, C.A.	1971	A	4.2, 295		64 46	Polycrystalline; 49 cm wide and 0.048 cm thick.

\* Not shown in figure.

TABLE 4. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + ALUMINUM ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Al	Composition (continued), Specifications, and Remarks
50	69	Sinha, T.H.D. and Prasad, R.S.	1970	A	302-792	AC <sub>1</sub>	99.57 0.43	Prepared by National Metallurgical Laboratory, Jamshedpur; residual electrical resistivity 0.5 $\mu\Omega$ cm.
51	69	Sinha, T.H.D. and Prasad, R.S.	1970	A	302-773	AC <sub>2</sub>	98.44 1.56	Prepared by National Metallurgical Laboratory, Jamshedpur; residual electrical resistivity 3.5 $\mu\Omega$ cm.
52	69	Sinha, T.H.D. and Prasad, R.S.	1970	A	307-817	AC <sub>3</sub>	94.4 5.6	Prepared by National Metallurgical Laboratory, Jamshedpur; residual electrical resistivity 7.5 $\mu\Omega$ cm.
53	69	Sinha, T.H.D. and Prasad, R.S.	1970	A	303-779	AC <sub>4</sub>	91.4 8.6	Prepared by National Metallurgical Laboratory, Jamshedpur; residual electrical resistivity 9.7 $\mu\Omega$ cm.
54	69	Watanabe, K. and Katsui, A.	1968		288-807		54.3 45.7	3 mm diameter cylindrical specimen; prepared by melting together in an argon atmosphere 99.99 pure Al and Cu; annealed for 2 days at 500 C in vacuum and cooled at 50-60 C/h.
55	71, 72	Chu, T.K. and Lipschultz, F.P.	1972	A	1.1-273	B1	6.97	Same supplier and dimensions as the above specimen C1 for data set No. 28; annealed at 1273 K for 48 h.
56*	71, 72	Chu, T.K. and Lipschultz, F.P.	1972	A	1.1-273	B2		The above specimen fatigued for 500 cycles with maximum load 8.3 kg mm <sup>-2</sup> .
57*	71, 72	Chu, T.K. and Lipschultz, F.P.	1972	A	1.1-273	B3		The above specimen fatigued for 10 <sup>4</sup> cycles.
58*	71, 72	Chu, T.K. and Lipschultz, F.P.	1972	A	1.1-273	B4		The above specimen fatigued for 10 <sup>5</sup> cycles.
59*	71, 72	Chu, T.K. and Lipschultz, F.P.	1972	A	1.1-273	B5	6.97	Similar to the above specimen B1 except given a 5% plastic deform under uniaxial stress.
60*	71, 72	Chu, T.K. and Lipschultz, F.P.	1972	A	4.2-273	B6		The above specimen fatigued for 2 x 10 <sup>5</sup> cycles with maximum load 8.3 kg mm <sup>-2</sup> .
61*	90	Leaver, A.D.W. and Charsley, P.	1971		4.2	2 Al	0.83	Polycrystalline; 3 mm diameter x 10 cm long; obtained from International Research and Development Co.; annealed at 750 C for 15 h.
62*	90	Leaver, A.D.W. and Charsley, P.	1971		4.2	2 Al		The above specimen tensile strained 8.2% under a stress of 16.93 kg mm <sup>-2</sup> .
63*	90	Leaver, A.D.W. and Charsley, P.	1971		4.2	12 Al	5.56	Polycrystalline; obtained from International Research and Development Co.
64*	90	Leaver, A.D.W. and Charsley, P.	1971		4.2	12 Al		The above specimen tensile strained 1.6% under a stress of 16.38 kg mm <sup>-2</sup> .
65*	70	Weinberg, I.	1965		4.2		0.33	Calculated composition (0.77 a/o Al).
66*	70	Weinberg, I.	1965		4.2		1.02	Calculated composition (2.37 a/o Al).
67*	67	Köster, W. and Rothenbacher, P.	1967		298.2		87.5 12.5	Recrystallized.
68*	67	Köster, W. and Rothenbacher, P.	1967		298.2		91.7 8.3	Recrystallized.
69*	67	Köster, W. and Rothenbacher, P.	1967		298.2		87.5 12.5	Recrystallized.

\* Not shown in figure.



TABLE 4. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + ALUMINUM ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Al	Composition (continued), Specifications, and Remarks
70*	87	Köster, W. and Rothenbacher, P.	1967		298.2		84 16	Recrystallized.
71*	91	Eretnov, K.I. and Lyubimov, A.P.	1967		1373, 1473		1.7	Calculated composition (4 a/o Al); prepared from 99.95 pure electrolytic copper and 99.9 pure aluminum; in liquid state.
72*	91	Eretnov, K.I. and Lyubimov, A.P.	1967		1373, 1473		4.5	Calculated composition (10 a/o Al); similar to the above specimen.
73*	91	Eretnov, K.I. and Lyubimov, A.P.	1967		1373, 1473		6.0	Calculated composition (13 a/o Al); similar to the above specimen.
74*	91	Eretnov, K.I. and Lyubimov, A.P.	1967		1373, 1473		8.5	Calculated composition (18 a/o Al); similar to the above specimen.
75*	91	Eretnov, K.I. and Lyubimov, A.P.	1967		1373, 1473		9.6	Calculated composition (20 a/o Al); similar to the above specimen.
76*	91	Eretnov, K.I. and Lyubimov, A.P.	1967		1373, 1473		10.7	Calculated composition (22 a/o Al); similar to the above specimen.
77*	91	Eretnov, K.I. and Lyubimov, A.P.	1967		1373, 1473		12.4	Calculated composition (25 a/o Al); similar to the above specimen.
78*	91	Eretnov, K.I. and Lyubimov, A.P.	1967		1373, 1473		13.6	Calculated composition (27 a/o Al); similar to the above specimen.
79*	91	Eretnov, K.I. and Lyubimov, A.P.	1967		1373, 1473		14.8	Calculated composition (29 a/o Al); similar to the above specimen.
80*	91	Eretnov, K.I. and Lyubimov, A.P.	1967		1373, 1473		17.9	Calculated composition (34 a/o Al); similar to the above specimen.
81*	91	Eretnov, K.I. and Lyubimov, A.P.	1967		1373, 1473		19.3	Calculated composition (36 a/o Al); similar to the above specimen.
82*	91	Eretnov, K.I. and Lyubimov, A.P.	1967		1373, 1473		23.5	Calculated composition (42 a/o Al); similar to the above specimen.
83*	91	Eretnov, K.I. and Lyubimov, A.P.	1967		1373, 1473		25.8	Calculated composition (45 a/o Al); similar to the above specimen.
84*	91	Eretnov, K.I. and Lyubimov, A.P.	1967		1373, 1473		29.8	Calculated composition (50 a/o Al); similar to the above specimen.
85*	91	Eretnov, K.I. and Lyubimov, A.P.	1967		1373, 1473		49.8	Calculated composition (70 a/o Al); similar to the above specimen.
86	57	Gaudig, W. and Warlimont, H.	1969	A	297-766		6.97	Calculated composition (1.5 a/o Al); wire specimen 0.9 mm in diameter; prepared from 99.999 pure copper and 99.99 pure aluminum; annealed in argon at 773 K and cooled slowly to room temperature.
87	57	Gaudig, W. and Warlimont, H.	1969	A	300-772			The above specimen measured at decreasing temperatures.

\* Not shown in figure.

TABLE 4. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + ALUMINUM ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Al	Composition (continued), Specifications, and Remarks
88	58	Panin, V. Ye., Zenkova, E. K., and Fadin, V. P.	1962		299-474		6.62	Calculated composition (14.3 a/o Al); 1 mm in diameter and 100 to 120 mm long; prepared from 99.99 pure Al and Cu; cast, forged, and rolled; heated at a rate of 6 C min <sup>-1</sup> and annealed.
89*	58	Panin, V. Ye., et al.	1962		300-404			The above specimen quenched from 400 C.
90*	59	Panin, V. Ye., et al.	1962		298-369			Similar to the above specimen for dataset No. 88 except heated at a rate of 0.6 C min <sup>-1</sup> and annealed.
91*	58	Panin, V. Ye., et al.	1962		334-388			The above specimen quenched from 400 C.
92*	59	Panin, V. Ye., Fadin, V. P., and Solov'yev, L. A.	1962		298-473		9.76	Calculated composition (20.3 a/o Al); prepared from 99.99 pure Al and Cu; vacuum cast; heated at a rate of 6 C min <sup>-1</sup> and annealed.
93*	59	Panin, V. Ye., et al.	1962		299-413			The above specimen quenched from 500 C.
94	59	Panin, V. Ye., et al.	1962		299-472			Similar to the above specimen for data set No. 92 except heated at a rate of 0.6 C min <sup>-1</sup> and annealed.
95*	59	Panin, V. Ye., et al.	1962		296-359			The above specimen quenched from 500 C.
96*	92	Sidorova, T. S., Panin, V. Ye., and Bol'shatina, N. A.	1962	B	311-586		6.62	Similar to the specimen for data set No. 88; annealed at 750 C for 2 h and cooled at a rate of 50 C h <sup>-1</sup> ; deformed 4% at room temperature by drawing; heated at a rate of 6 C min <sup>-1</sup> .
97	92	Sidorova, T. S., et al.	1962	B	307-762			Similar to the above specimen except deformed 27%.
98*	92	Sidorova, T. S., et al.	1962	B	307-772			Similar to the above specimen except deformed 44%.
99*	92	Sidorova, T. S., et al.	1962	B	446-567			Similar to the above specimen for data set No. 96 except heated at a rate of 0.6 C min <sup>-1</sup> .
100	92	Sidorova, T. S., et al.	1962	B	475-563			Similar to the above specimen except deformed 44%.
101	93	Scala, E. and Robertson, W. D.	1953	A	1381-1557		0.29	Calculated composition (0.95 a/o Al); impurities: <0.001 each of Fe, Mg, Si, and Sn; in liquid state; contained in a refractory insulating tube about 0.6 cm I.D. and 13 cm long.
102	77	Kapoor, A., Rowlands, J. A., and Woods, S. B.	1974	A	4.2		4.5	Calculated composition (10 a/o Al); 3.6 mm in diameter; prepared by melting pure metals in vacuum, then swaging to size; cold-worked.
103*	77	Kapoor, A., et al.	1974	A	4.2			The above specimen annealed in vacuum at 873 K for 12 h.
104*	77	Kapoor, A., et al.	1974	A	4.2			The above specimen reannealed in vacuum at 948 K for 12 h.
105	77	Kapoor, A., et al.	1974	A	4.2			The above specimen reannealed in vacuum at 1273 K for 12 h.
106*	87	Korol'kov, A. M. and Shahtkov, D. P.	1962	R	673-1023		54.0 46.0	In solid and liquid states.

\* Not shown in figure.

TABLE 5. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF COPPER + ALUMINUM ALLOYS (Temperature Dependence)

[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-4} \Omega m$ ]

T	$\rho$	T	$\rho$	Ann. Temp. (K)	$\rho$	T	$\rho$	T	$\rho$			
<u>DATA SET 1</u>			<u>DATA SET 10*</u>			<u>DATA SET 27 (cont.)*</u> (T = 4.2 K)			<u>DATA SET 37 (cont.)</u>			
293	14.7	4.2	2.07	<u>DATA SET 22*</u>		(T = 4.2 K)		570		39.0		
349	15.6	<u>DATA SET 11*</u>		Ann. Temp. (K)		660	7.601	645	47.5	772	31.0	
373	16.0	4.2	2.12	<u>DATA SET 23*</u>		732	7.553	775	54.5	875	35.0	
423	16.7	(T = 4.2 K)				1308	7.576	870	60.5	973	39.8	
473	17.5	<u>DATA SET 12*</u>				T		970	67.5	1072	45.3	
523	18.3	4.2		2.10	300	7.468	<u>DATA SET 28</u>		<u>DATA SET 38</u>			
<u>DATA SET 2</u>			4.2	6.50	373	7.450	1.1		1.068	275	15.0	
293.2	2.386	<u>DATA SET 13*</u>		693	7.463	713	7.404	4.2	1.066	375	18.4	
473.2	3.625	4.2	6.63	<u>DATA SET 24*</u>		77	1.302	77	1.302	475	21.7	
<u>DATA SET 3</u>			4.2	7.21	4.2	7.350	<u>DATA SET 29*</u>		560	24.0		
293.2	3.115	<u>DATA SET 14*</u>		Ann. Temp. (K)		1.1	1.071	870	36.0	680	29.0	
473.2	4.365	4.2	7.41	<u>DATA SET 25*</u>		4.2	1.067	975	53.7	780	33.0	
<u>DATA SET 4</u>			4.2	7.57	(T = 4.2 K)		77	1.301	1075	47.5*		
293.2	4.274	<u>DATA SET 15*</u>				273	2.664	1075	64.0	1205	54.0	
473.2	5.571	4.2	7.21	<u>DATA SET 16*</u>		<u>DATA SET 30*</u>		<u>DATA SET 35</u>		1275	49.0	
<u>DATA SET 5</u>			4.2	7.41	300	7.586	1.1		1.069	1315	53.0	
293.2	6.285	4.2	7.41	422	7.475	552	7.498	4.2	1.069	<u>DATA SET 39</u>		
473.2	7.692	<u>DATA SET 17*</u>		673	7.542	797	7.456	77	1.304	275	13.0	
<u>DATA SET 6</u>			4.2	7.57	920	7.453	<u>DATA SET 31*</u>		880	38.3	375	16.5
293.2	9.747	4.2	7.57	1202	7.441	1.1		1.066	970	44.5	480	20.1
473.2	11.353	<u>DATA SET 18*</u>		<u>DATA SET 26*</u>		4.2		1.068	1080	52.6	575	22.8
<u>DATA SET 7</u>			4.2	2.10	(T = 4.2 K)		77	1.294	<u>DATA SET 36</u>		675	25.0
293.2	11.320	4.2	2.10	360	7.567	1.1		1.066	280	13.5	765	30.3
473.2	13.072	<u>DATA SET 19*</u>		564	7.536	4.2		1.068	360	15.6	875	30.8
<u>DATA SET 8</u>			4.2	7.49	565	7.536	77	1.294	490	20.2	970	34.7
293.2	12.136	4.2	7.49	567	7.498	273	2.660	273	2.660	1050	36.0	
473.2	14.172	<u>DATA SET 20*</u>		570	7.498	<u>DATA SET 32*</u>		255	13.3	1110	40.0	
<u>DATA SET 9*</u>			4.2	7.617	(T = 4.2 K)		4.2	1.064	390	16.5	1175	38.0
293.2	12.440	<u>DATA SET 21*</u>		344	7.644	77		1.306	560	22.2	1255	31.7
473.2	14.428	4.2	7.617	670	7.625	273		2.665	793	32.2	1300	35.3
473.2	17.428	4.2	7.995	661	7.612	<u>DATA SET 33</u>		1062	41.2	<u>DATA SET 40</u>		
			4.2	7.995	(T = 4.2 K)		270	18.7	375	47.0	390	12.3
			4.2	7.995	344	7.644	270	18.7	475	16.3	375	13.9
			4.2	7.995	670	7.625	470	19.8	575	18.7	475	16.3
			4.2	7.995	661	7.612	575	23.2	630	19.4	575	18.7
			4.2	7.995	661	7.612	470	26.5	675	19.9	675	19.9
			4.2	7.995	661	7.612	470	26.5	825	23.2	825	23.2
			4.2	7.995	661	7.612	470	26.5	865	26.5	865	26.5

\* Not shown in figure.



EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF COPPER + ALUMINUM ALLOYS (Temperature Dependence) (continued)

DATA SET 86 (cont.)			DATA SET 88 (cont.)			DATA SET 90*			DATA SET 92 (cont.)*			DATA SET 94 (cont.)			DATA SET 96*		
T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$	
460	12.00		332.9	10.33*		297.6	10.10		396.5	11.92		309.5	11.12		311	10.15	
464	12.23*		338.6	10.38*		314.0	10.23		403.2	11.98		314.8	11.17*		321	10.22	
504	12.44*		342.9	10.42*		322.9	10.32		407.7	12.02		323.5	11.28*		332	10.30	
520	12.61		347.7	10.47		332.9	10.41		412.6	12.07		327.3	11.34*		342	10.41	
540	12.89*		354.1	10.53*		336.9	10.45		419.0	12.15		332.1	11.37		352	10.47	
564	13.20*		358.4	10.58*		344.3	10.50		424.0	12.20		336.8	11.42*		361	10.57	
585	13.44		363.9	10.66*		348.4	10.54		428.6	12.25		341.1	11.47*		371	10.67	
595	13.67		368.8	10.71		353.5	10.59		434.1	12.32		347.9	11.51*		379	10.75	
616	14.18		373.0	10.79*		358.8	10.65		438.1	12.36		353.6	11.58		390	10.84	
669	14.46*		379.0	10.84*		364.1	10.69		443.9	12.42		358.6	11.63*		399	10.93	
681	14.66*		384.9	10.90*		368.4	10.75		448.8	12.47		362.7	11.68*		412	11.04	
702	14.90		388.7	10.95*		373.0	10.81		452.4	12.53		367.9	11.73*		418	11.11	
723	15.18*		393.7	11.02*		378.6	11.02*		458.3	12.59		372.8	11.76		427	11.20	
745	15.47		399.2	11.05		383.2	11.07		463.1	12.65		378.3	11.80*		437	11.29	
763	15.74*		403.2	11.10*		387.6	11.12		468.9	12.69		383.8	11.85*		448	11.35	
786	16.04		409.1	11.16*		393.7	11.17		473.0	12.73		387.1	11.90*		458	11.44	
			417.4	11.20*		397.9	11.22					397.7	12.02		469	11.55	
			423.4	11.25*		403.2	11.27					404.3	12.08*		479	11.63	
			434.4	11.30		409.1	11.32					408.5	12.13*		490	11.73	
			444.6	11.35*		415.5	11.37					412.0	12.18*		498	11.79	
			454.6	11.40*		421.5	11.42					416.7	12.23*		510	11.89	
			463.3	11.45*		427.4	11.47					422.9	12.27		520	12.02	
			474.4	11.50*		433.4	11.52					428.4	12.33*		529	12.13	
						439.4	11.57					433.9	12.38*		537	12.24	
						445.4	11.62					438.0	12.43*		549	12.35	
						451.4	11.67					442.6	12.48		557	12.46	
						457.4	11.72					446.6	12.52*		565	12.55	
						463.4	11.77					452.0	12.58*		578	12.68	
						469.4	11.82					457.7	12.63*		586	12.86	
						475.4	11.87					462.8	12.68*				

\* Not shown in figure.

TABLE 5. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF COPPER + ALUMINUM ALLOYS (Temperature Dependence) (continued)

T	$\rho$	T	$\rho$	T	$\rho$
<u>DATA SET 97 (cont.)</u>		<u>DATA SET 99 (cont.)*</u>		<u>DATA SET 106*</u>	
537	12.65*	477	11.94	673	32.6
573	12.81	491	12.03	1020	55.0
598	13.03	502	12.11	1023	56.2
627	13.35	512	12.19		
655	13.58	522	12.26		
672	13.74*	532	12.37		
687	13.88	542	12.47		
703	14.04*	552	12.57		
719	14.18	567	12.73		
762	14.71				
<u>DATA SET 98*</u>		<u>DATA SET 100</u>			
307	11.12	475	12.50		
326	11.29	482	12.56*		
347	11.49	493	12.61		
364	11.67	503	12.66*		
387	11.89	513	12.68*		
404	12.04	524	12.73*		
425	12.31	533	12.83*		
444	12.37	543	12.91		
463	12.51	563	13.06		
473	12.60				
484	12.68	<u>DATA SET 101</u>			
493	12.75	1381	25.55		
503	12.80	1398	26.11*		
513	12.85	1416	26.40*		
523	12.90	1430	26.40*		
533	12.94	1451	27.35		
543	12.98	1470	27.35		
553	13.02	1520	28.34		
569	13.12	1529	28.66		
584	13.19	1557	29.42		
603	13.41				
623	13.61	<u>DATA SET 102</u>			
643	13.81	4.2	7.54		
662	13.95	<u>DATA SET 103*</u>			
680	14.03	4.2	6.79		
701	14.12	<u>DATA SET 104*</u>			
722	14.28	4.2	6.88		
740	14.50	<u>DATA SET 105</u>			
772	14.82	4.2	6.69		
<u>DATA SET 99*</u>					
446	11.72				
457	11.91				
469	11.87				

\* Not shown in figure.

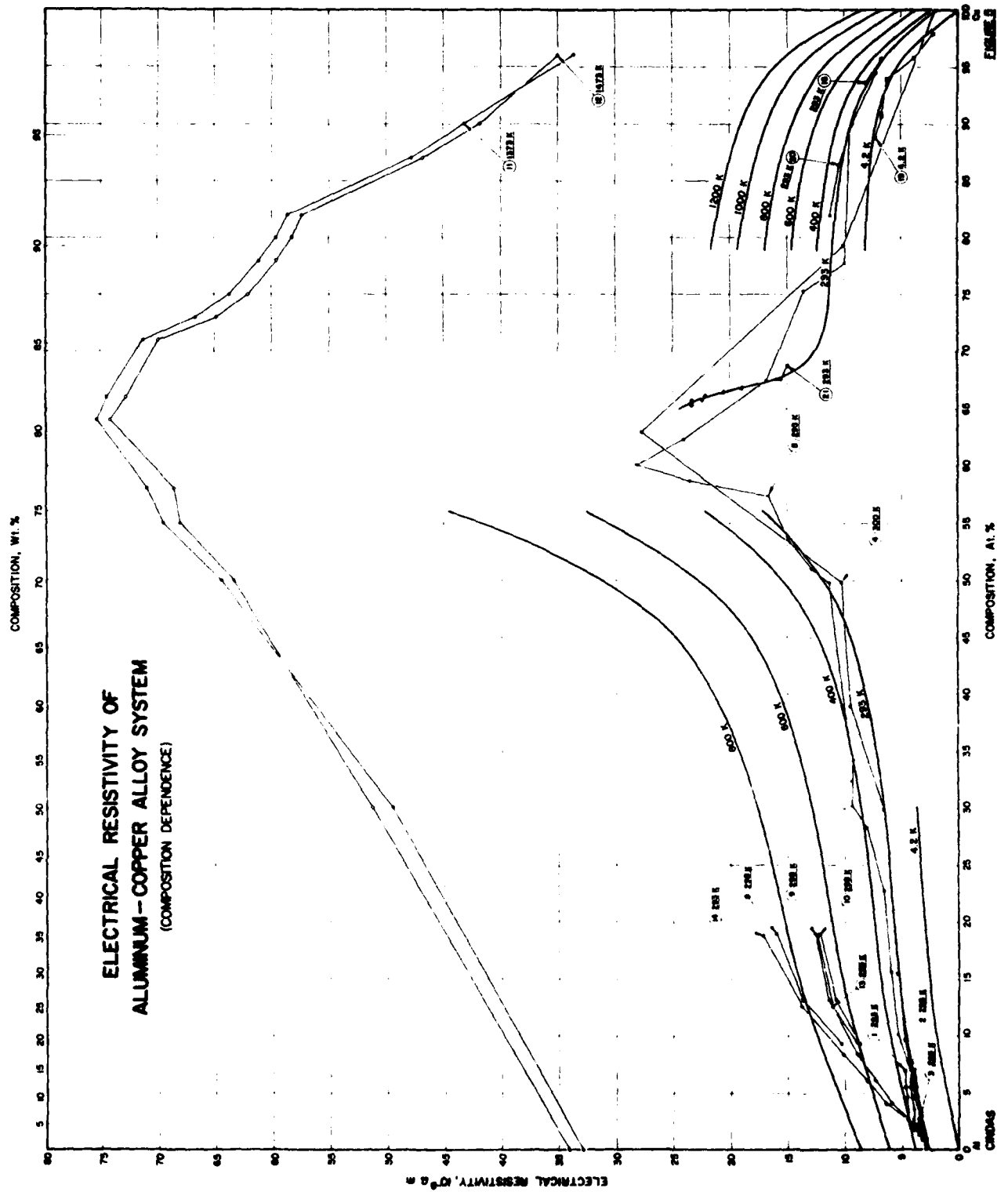


TABLE 6. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF ALUMINUM-COPPER ALLOY SYSTEM (Composition Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp., K	Composition Range (at. % of Cu)	Composition (weight percent), Specifications, and Remarks
1	94	Vozdvizhenskii, V. M.	1961		293.2	0.1-15	5.8 mm diameter x 60 mm long; cast; annealed at 819 K for 70 hr.
2	94	Vozdvizhenskii, V. M.	1961		293.2	1.7-7.0	Similar to the above specimens except annealed at 773 K for 200 hr.
3	94	Vozdvizhenskii, V. M.	1961		293.2	1.5-7.0	Similar to the above specimens except annealed at 723 K for 230 hr.
4	55	Smith, A. W.	1925	B	300	0-100	Specimens 1.9 cm in diameter and 10 cm long; prepared by melting 99.97% pure aluminum and copper supplied by Baker.
5	95	Aliev, N. A.	1953		295.2	4.6-89	No details reported.
6*	82	Tureball, D., Rosenbaum, H. S., and Treadle, H. N.	1960		273.2	0.4-2.2	Specimens 0.025 cm diameter x 25 cm long in the form of a coil 0.32 cm in diameter and 3.8 cm long; fabricated by cold-swaging, drawing, and rolling; homogenized at 823 K and quenched to 223 K.
7*	96	Bradley, J. M. and Stringer, J.	1974	A	293.2	0.2-1.5	40 x 5 x 0.5 mm; prepared from 99.999 pure metals by melting and homogenizing in vacuum, obtained ingots cold-rolled to 0.5 mm thick, cut to size; solution-treated at 773 K and water quenched immediately prior to tests.
8	97	Köster, W. and Rave, H.-P.	1961		298.2	9.3-19	Cold worked.
9	97	Köster, W. and Rave, H.-P.	1961		298.2	9.3-19	Recrystallized at 873 K for 20 min.
10	97	Köster, W. and Rave, H.-P.	1961		298.2	9.3-19	Ordered.
11	91	Eretnov, K. I. and Lyubimov, A. P.	1967		1373	0-96	Alloys prepared from 99.95 pure electrolytic copper and 99.9 pure aluminum; in liquid state.
12	91	Eretnov, K. I. and Lyubimov, A. P.	1967		1473	0-96	The above specimens.
13	98	Köster, W. and Rave, H.-P.	1964		293.2	1.0-19	Alloys recrystallized; measuring temperature not given, assigned here as 20 C.
14	98	Köster, W. and Rave, H.-P.	1964		293.2	1.0-19	The above specimens 90% cold-worked.
15*	62	Gulyaev, A. P. and Trusova, E. F.	1950		293.2	0.3-0.9	Measuring temperature assigned here as 20 C.
16	62	Gulyaev, A. P. and Trusova, E. F.	1950		293.2	94.96	Similar to the above specimens.
17*	87	Korol'kov, A. M. and Shashkov, D. P.	1962	R	673.2	0-33	Alloys in solid state.
18*	87	Korol'kov, A. M. and Shashkov, D. P.	1962	R	1023	0-33	The above specimens in liquid state.
19	74	Charley, P. and Salter, J. A. M.	1965	A	4.2	89-98	Polycrystalline specimens; grain size 0.1 to 0.3 mm; 3 mm diameter x 12 cm long; prepared by International Research and Development Co.; cast, machined, swaged, and drawn; annealed in vacuo at 1023 K for at least 14 hr and furnace cooled.
20	63	Linde, J. O.	1964	B	293.2	82-94	Specimens prepared from Johnson, Matthey spectroscopically standardized pure metals by melting together; rolled, homogenized at 1173 K for about 10 hr, and rolled to about 0.4 mm <sup>2</sup> in cross-section; annealed in vacuum at 723 to 773 K for 16 hr, cooled slowly.

\* Not shown in figure.



TABLE 6. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF ALUMINUM-COPPER ALLOY SYSTEM (Composition Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp., K	Composition Range (at. % of Cu)	Composition (weight percent), Specifications, and Remarks
21	66	Hishiyama, Y.	1967	A	293.3	66.25-66.65	Compositions calculated from electron to atom ratios of 1.698, 1.684, 1.685, 1.679, 1.672, 1.664, 1.648, and 1.627 for specimens in increasing order of Cu content; $\gamma$ -phase specimens 1.5 x 1.5 x 20 mm <sup>3</sup> prepared from 99.9 pure constituents by induction heating under flux of sodium and potassium chloride salt bath in high-purity graphite crucible, imbedding the 20g ingots in high-purity graphite flour in order to prevent volatilization, sealing in quartz or glass tubes in vacuo, annealing at 973 K for 7 days, furnace cooling, and checking homogeneity and concentration by chemical analysis; due to excessive brittleness specimens were ground on emery paper rather than cut by usual techniques; data extracted from table.

TABLE 7. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF ALUMINUM-COPPER ALLOY SYSTEM (Composition Dependence)

[Composition, C, at/o Cu; Electrical Resistivity, $\rho$ , $10^{-8}$ $\Omega$ m]			
C	$\rho$	C	$\rho$
DATA SET 1 (T = 293.2 K)			
0.04	2.61*		
0.09	2.69*		
0.34	2.80		
0.43	2.90*		
0.64	2.99*		
0.86	3.15		
1.08	3.36*		
1.80	3.50		
1.52	3.71*		
1.74	3.98*		
2.19	4.19*		
2.64	4.42		
3.56	4.44*		
4.51	4.59		
5.37	4.67		
6.97	4.75		
DATA SET 2 (T = 293.2 K)			
1.74	3.94		
1.96	3.95		
2.19	3.99		
2.64	3.96		
3.56	3.98		
4.51	4.12		
5.37	4.24		
6.97	4.28		
DATA SET 3 (T = 293.2 K)			
1.52	3.51		
1.74	3.52*		
1.96	3.52*		
2.19	3.53		
2.64	3.47		
3.56	3.57		
4.51	3.64		
5.37	3.78		
6.97	3.86		
DATA SET 4 (T = 300 K)			
0	2.96		
4.5	3.85		
9.6	4.78		
15.4	5.40		
29.8	6.54		
38.9	9.43		
49.8	10.25		
62.9	27.78		
79.3	10.02		
100	1.97		
DATA SET 5 (T = 296.2 K)			
4.62	4.72*		
10.02	5.32		
15.58	5.98		
22.64	6.55		
28.14	8.06		
30.18	9.36		
32.36	9.31		
34.15	9.24		
38.52	10.02		
49.73	11.30		
50.95	12.90		
53.42	14.90		
57.33	16.61		
58.68	23.53		
60.06	28.25		
67.44	16.81		
75.27	13.51		
77.86	9.96		
88.98	9.52		
DATA SET 6* (T = 273.2 K)			
0.4	2.82		
0.8	3.20		
1.3	3.69		
1.6	3.95		
1.7	3.76		
2.2	3.82		
2.2	3.97		
DATA SET 7* (T = 293.2 K)			
0.2	2.84		
0.4	3.02		
0.5	3.15		
0.6	3.17		
0.8	3.36		
1.0	3.55		
1.2	3.65		
1.5	3.74		
DATA SET 8 (T = 298.2 K)			
9.3	10.3		
13.1	13.7		
18.9	16.0		
DATA SET 9 (T = 298.2 K)			
9.3	8.95		
13.1	11.5		
18.9	12.6		
DATA SET 10 (T = 298.2 K)			
9.3	8.7		
13.1	10.8		
18.9	12.2		
DATA SET 11 (T = 1373 K)			
0	32.9		
30	49.5		
50	64.5		
55	69.6		
64	75.4		
66	74.5		
71	71.3		
73	68.7		
75	63.7		
78	61.1		
80	59.6		
DATA SET 12 (T = 1473 K)			
0	34.1		
30	51.3		
50	63.4		
55	68.1		
58	68.6		
64	74.2		
66	72.8		
71	70.0		
73	64.9		
75	62.1		
76	59.6		
80	58.2		
82	57.4		
87	46.8		
90	41.8		
96	35.0		
DATA SET 13 (T = 293.2 K)			
9.3	8.7		
13.1	10.8		
18.9	12.2		
DATA SET 14 (T = 293.2 K)			
1.0	2.89		
2.0	4.24*		
4.0	5.99		
6.0	7.32		
8.3	8.91		
12.5	11.08		
18.7	12.44		
DATA SET 15* (T = 293.2 K)			
0	54.7		
0.64	57.6		
1.16	65.5		
2.41	71.8		
5.23	51.8		
9.60	46.3		
13.46	44.6		
17.43	50.0		
20.58	47.3		
25.09	48.3		
33.27	56.2		
DATA SET 16 (T = 293.2 K)			
93.55	8.05		
95.7	6.64		
DATA SET 17* (T = 673.2 K)			
0	15.0		
0.64	18.1		
1.16	20.8		
2.41	28.4		
5.23	17.5		
9.60	14.4		
13.46	14.0		
17.43	16.2		
20.58	13.4		
25.09	12.7		
33.27	32.6		
DATA SET 18* (T = 1023 K)			
0	54.7		
0.64	57.6		
1.16	65.5		
2.41	71.8		
5.23	51.8		
9.60	46.3		
13.46	44.6		
17.43	50.0		
20.58	47.3		
25.09	48.3		
33.27	56.2		
DATA SET 19 (T = 4.2 K)			
0	54.7		
0.64	57.6		
1.16	65.5		
2.41	71.8		
5.23	51.8		
9.60	46.3		
13.46	44.6		
17.43	50.0		
20.58	47.3		
25.09	48.3		
33.27	56.2		
DATA SET 20 (T = 293.2 K)			
90.88	6.63		
93.90	5.20		
95.78	3.88		
97.91	2.12		
DATA SET 21 (T = 293.2 K)			
65.25	23.4		
65.65	23.4		
65.75	22.5		
66.05	22.2		
66.40	20.6		
66.80	19.0		
67.60	15.6		
68.65	15.0		

\* Not shown in figure.

### 3.2. Aluminum-Magnesium Alloy System

The aluminum-magnesium alloy system does not form a continuous series of solid solutions. The maximum solid solubility of magnesium in aluminum is 17.4% (18.9 at.%) at 723 K and the solubility decreases at higher and lower temperatures, being only 1.9% (2.1 at.%) at 373 K. The maximum solid solubility of aluminum in magnesium is 12.7% (11.6 at.%) at 710 K and likewise it decreases at higher and lower temperatures, being only about 1.5% (1.3 at.%) at 373 K. Thus the region of solid solution for this system is even more limited than that of the aluminum-copper alloy system.

There are 76 sets of experimental electrical resistivity data available for this system. Of the 43 data sets for Al + Mg alloys listed in table 9, tabulated in table 10, and shown in figure 8, 11 sets are merely single data points. Twelve of the 28 data sets for Mg + Al alloys listed in table 11, tabulated in table 12, and shown in figure 9 are single data points. Of the 5 resistivity-composition data sets listed in table 13, tabulated in table 14, and shown in figure 10, one set is for specimens in liquid state.

For the Al + Mg alloys, measurements were limited to alloys containing no more than 14% Mg. Recommended values were, therefore, generated only for 0.5 to 10% Mg alloys. For the electrical resistivity of these alloys, it appears that the deviation from the Matthiessen's rule due to alloying is small. Hence, a residual resistivity versus composition curve was constructed based on the slope of the data of Gulyaev and Trusova [62] (Al-Mg data set 4) measured at 293 K, and the total electrical resistivity of each alloy was then obtained by adding its residual electrical resistivity to the intrinsic electrical resistivity of pure Al [54]. The resulting recommended values agree well with the data of Hase et al. [99] (Al + Mg data sets 1-4) and of Cordier and Detert [100] (Al + Mg data sets 12-14) above room temperature, and agree with the data of Seth and Wood [101] (Al + Mg data sets 9-11), Clark et al. [102,103] (Al + Mg data sets 15, 26, 27), and Clark and Tryon [104] (Al + Mg data sets 28-36) between 100 K and 300 K.

For the Mg + Al alloys, no measurements were made for alloys containing more than 12.2% Al. Accordingly, the recommended values were generated for 0.5 to 10% Al alloys only. For the electrical resistivity of these alloys, the deviation from the Matthiessen's rule due to alloying appears also to be

small. The residual resistivity values of these alloys were derived based on the data of Smith [55] (Al-Mg data set 5) measured at 292 K, and the total electrical resistivity of each alloy was obtained by adding its residual resistivity to the intrinsic electrical resistivity of pure Mg [51]. The resulting recommended values agree with the data of Staebler [105] (Mg + Al data sets 19-21) and of Powell et al. [106] (Mg + Al data set 28) above room temperature, and with the data of Hedgecock and Muir [107] (Mg + Al data sets 1-3) and Seth and Wood [101] (Mg + Al data sets 4-6) below room temperature.

The resulting recommended electrical resistivity values for Al, Mg, and for 10 Al-Mg binary alloys are presented in table 8 and shown in figures 6, 7, and 10. No values were generated for alloys containing 15 to 85% Mg. The recommended values for Al and for Mg are for well-annealed high-purity specimens, but those values for temperatures below about 100 K are applicable only to Al and Mg having residual electrical resistivities as given at 1 K in table 8. The alloys for which the recommended values are generated are not ordered and have not been quenched or cold-worked severely. The recommended values cover a full range of temperature from 1 K to the solidus temperature of the alloy where melting starts. These values are not corrected for the thermal expansion of the material. The estimated uncertainties in the values for the various alloys and for different temperature ranges are explicitly stated in a footnote to table 8. Some of the values in table 8 are indicated as provisional because their uncertainties are greater than  $\pm 5\%$ .

TABLE 8. RECOMMENDED ELECTRICAL RESISTIVITY OF ALUMINUM-MAGNESIUM ALLOY SYSTEM†

[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Al: 100.00% (100.00 At.%) Mg: 0.00% (0.00 At.%)		Al: 99.50% (99.45 At.%) Mg: 0.50% (0.55 At.%)		Al: 99.00% (98.99 At.%) Mg: 1.00% (1.11 At.%)		Al: 97.00% (96.68 At.%) Mg: 3.00% (3.32 At.%)		Al: 95.00% (94.48 At.%) Mg: 5.00% (5.52 At.%)		Al: 90.00% (89.02 At.%) Mg: 10.00% (10.98 At.%)	
T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
1	0.00100*	1	0.254*	1	0.512*	1	1.53*	1	2.54*	1	4.93*
4	0.00100	4	0.254*	4	0.512*	4	1.53*	4	2.54*	4	4.93*
7	0.00100	7	0.254*	7	0.512*	7	1.53*	7	2.54*	7	4.93*
10	0.00103	10	0.254*	10	0.512*	10	1.53*	10	2.54*	10	4.93*
15	0.00117	15	0.254*	15	0.512*	15	1.53*	15	2.54*	15	4.93*
20	0.00168	20	0.255*	20	0.513*	20	1.53*	20	2.54*	20	4.93*
25	0.00296	25	0.258*	25	0.514*	25	1.53*	25	2.54*	25	4.93*
30	0.00564	30	0.259*	30	0.517*	30	1.54*	30	2.55*	30	4.94*
40	0.0138	40	0.273*	40	0.531*	40	1.55*	40	2.56*	40	4.95*
50	0.0480	50	0.302*	50	0.561*	50	1.56*	50	2.56*	50	4.96*
60	0.0960	60	0.351*	60	0.609*	60	1.63*	60	2.64*	60	5.04*
70	0.163	70	0.418*	70	0.673*	70	1.70*	70	2.71*	70	5.12*
80	0.246	80	0.500*	80	0.754*	80	1.78*	80	2.80*	80	5.21*
90	0.340	90	0.593*	90	0.851*	90	1.88*	90	2.90*	90	5.31*
100	0.443	100	0.697*	100	0.958*	100	1.96*	100	3.01*	100	5.42*
150	1.010	150	1.27*	150	1.53*	150	2.56*	150	3.58*	150	6.02*
200	1.593	200	1.85*	200	2.11*	200	3.15*	200	4.19*	200	6.63*
250	2.167	250	2.43*	250	2.69*	250	3.74*	250	4.78*	250	7.24*
273	2.429	273	2.69*	273	2.96*	273	4.01*	273	5.05*	273	7.52*
293	2.653	293	2.92*	293	3.18*	293	4.23*	293	5.28*	293	7.76*
300	2.731	300	2.99*	300	3.26*	300	4.31*	300	5.36*	300	7.85*
350	3.292	350	3.56*	350	3.83*	350	4.86*	350	5.93*	350	8.43*
400	3.854	400	4.12*	400	4.39*	400	5.45*	400	6.51*	400	9.02*
500	4.992	500	5.26*	500	5.53*	500	6.58*	500	7.67*	500	10.2*
600	6.155	600	6.42*	600	6.69*	600	7.75*	600	8.86*	600	11.4*
700	7.353	700	7.62*	700	7.90*	700	8.97*	700	10.1*	700	12.6*
800	8.622	800	8.90*	800	9.18*	800	10.3*	800	11.4*	800	13.7*
900	10.019	900	10.3	900	10.6	900	11.4*	900	12.1*	900	14.8*
933.52	10.518*	933	10.6	933	10.8			949			

† Uncertainties in the electrical resistivity values are as follows:

- 100.00 Al - 0.00 Mg:  $\pm 2\%$  below 80 K,  $\pm 1\%$  from 80 to 200 K,  $\pm 0.5\%$  above 200 K, and  $\pm 5\%$  above 500 K.  
 99.50 Al - 0.50 Mg:  $\pm 10\%$  up to 200 K and  $\pm 5\%$  above 200 K.  
 99.00 Al - 1.00 Mg:  $\pm 10\%$  up to 200 K and  $\pm 5\%$  above 200 K.  
 97.00 Al - 3.00 Mg:  $\pm 10\%$  up to 200 K and  $\pm 5\%$  above 200 K.  
 95.00 Al - 5.00 Mg:  $\pm 10\%$  up to 200 K and  $\pm 5\%$  above 200 K.  
 90.00 Al - 10.00 Mg:  $\pm 10\%$  up to 200 K and  $\pm 5\%$  above 200 K.

\* Provisional value.

\* In temperature range where no experimental data are available.

TABLE 8. RECOMMENDED ELECTRICAL RESISTIVITY OF ALUMINUM-MAGNESIUM ALLOY SYSTEM† (continued)

[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Al: 10.00% (9.10 At.%) Mg: 90.00% (90.90 At.%)		Al: 5.00% (4.53 At.%) Mg: 95.00% (95.47 At.%)		Al: 3.00% (2.71 At.%) Mg: 97.00% (97.29 At.%)		Al: 1.00% (0.90 At.%) Mg: 99.00% (99.10 At.%)		Al: 0.50% (0.45 At.%) Mg: 99.50% (99.55 At.%)		Al: 0.00% (0.00 At.%) Mg: 100.00% (100.00 At.%)	
T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
1	13.1*	1	9.02*	1	5.64*	1	1.87*	1	0.936*	1	0.0062
4	13.1	4	9.02	4	5.64	4	1.87	4	0.936	4	0.0062
7	13.1	7	9.02	7	5.64	7	1.87	7	0.937	7	0.0064
10	13.1	10	9.02	10	5.64	10	1.87	10	0.937	10	0.0069
15	13.1	15	9.02	15	5.64	15	1.87	15	0.939	15	0.0066†
20	13.1	20	9.03	20	5.65	20	1.88	20	0.942	20	0.0124†
25	13.1	25	9.04	25	5.66	25	1.89	25	0.949	25	0.0194†
30	13.1	30	9.05	30	5.67	30	1.90	30	0.960	30	0.0310
40	13.1	40	9.09	40	5.70	40	1.94	40	1.00	40	0.0746
50	13.2	50	9.17	50	5.78	50	2.02	50	1.08	50	0.152
60	13.3	60	9.28	60	5.90	60	2.14	60	1.19	60	0.262
70	13.4	70	9.42	70	6.04	70	2.27	70	1.34	70	0.400
80	13.6	80	9.58	80	6.20	80	2.43	80	1.50	80	0.559
90	13.6	90	9.75	90	6.37	90	2.60	90	1.67	90	0.731
100	14.0	100	9.93	100	6.55	100	2.78	100	1.85	100	0.912
150	14.9	150	10.9	150	7.50	150	3.73	150	2.80	150	1.84
200	15.8	200	11.8	200	8.42	200	4.65	200	3.72	200	2.75
250	16.7	250	12.7	250	9.30	250	5.53	250	4.59	250	3.61
273	17.1	273	13.1	273	9.69	273	5.92	273	4.98	273	4.05
293	17.4	293	13.4	293	10.0	293	6.25	293	5.31	293	4.39
300	17.6	300	13.5	300	10.1	300	6.37	300	5.43	300	4.51
350	18.4	350	14.3	350	11.0	350	7.20	350	6.26	350	5.36
400	19.2	400	15.2	400	11.8*	400	8.03	400	7.10*	400	6.18
500	20.9*	500	16.9*	500	13.5*	500	9.74	500	8.79*	500	7.52
600	22.7*	600	18.6*	600	15.2*	600	11.5	600	10.5*	600	9.44
700	24.4*	700	20.4*	700	17.0*	700	13.2	700	12.3*	700	11.0
756	25.2*	800	22.1*	800	18.7*	800	14.9*	800	14.0*	800	12.6
		839	22.7*	872	20.0*	900	16.6*	900	15.7*	900	14.2
						903	16.6*	913	15.9*	922	14.5(s)
										923	26.1†(L)
										1000	26.0†
										1100	25.8†
										1200	25.6†

† Uncertainties in the electrical resistivity values are as follows:

- 10.00 Al - 90.00 Mg:  $\pm 5\%$  below 200 K,  $\pm 3\%$  from 200 to 600 K, and  $\pm 5\%$  above 600 K.
- 5.00 Al - 95.00 Mg:  $\pm 5\%$  below 200 K,  $\pm 3\%$  from 200 to 600 K, and  $\pm 5\%$  above 600 K.
- 3.00 Al - 97.00 Mg:  $\pm 5\%$  below 200 K,  $\pm 3\%$  from 200 to 600 K, and  $\pm 5\%$  above 600 K.
- 1.00 Al - 99.00 Mg:  $\pm 5\%$  below 200 K,  $\pm 2\%$  from 200 to 600 K, and  $\pm 5\%$  above 600 K.
- 0.50 Al - 99.50 Mg:  $\pm 5\%$  below 200 K,  $\pm 2\%$  from 200 to 600 K, and  $\pm 5\%$  above 600 K.
- 0.00 Al - 100.00 Mg:  $\pm 5\%$  up to 10 K,  $\pm 6\%$  between 10 and 30 K,  $\pm 5\%$  from 30 to 100 K,  $\pm 3\%$  above 100 K to 600 K,  $\pm 5\%$  above 600 K to 922 K, and  $\pm 10\%$  above 922 K.

\* Provisional value.

† In temperature range where no experimental data are available.

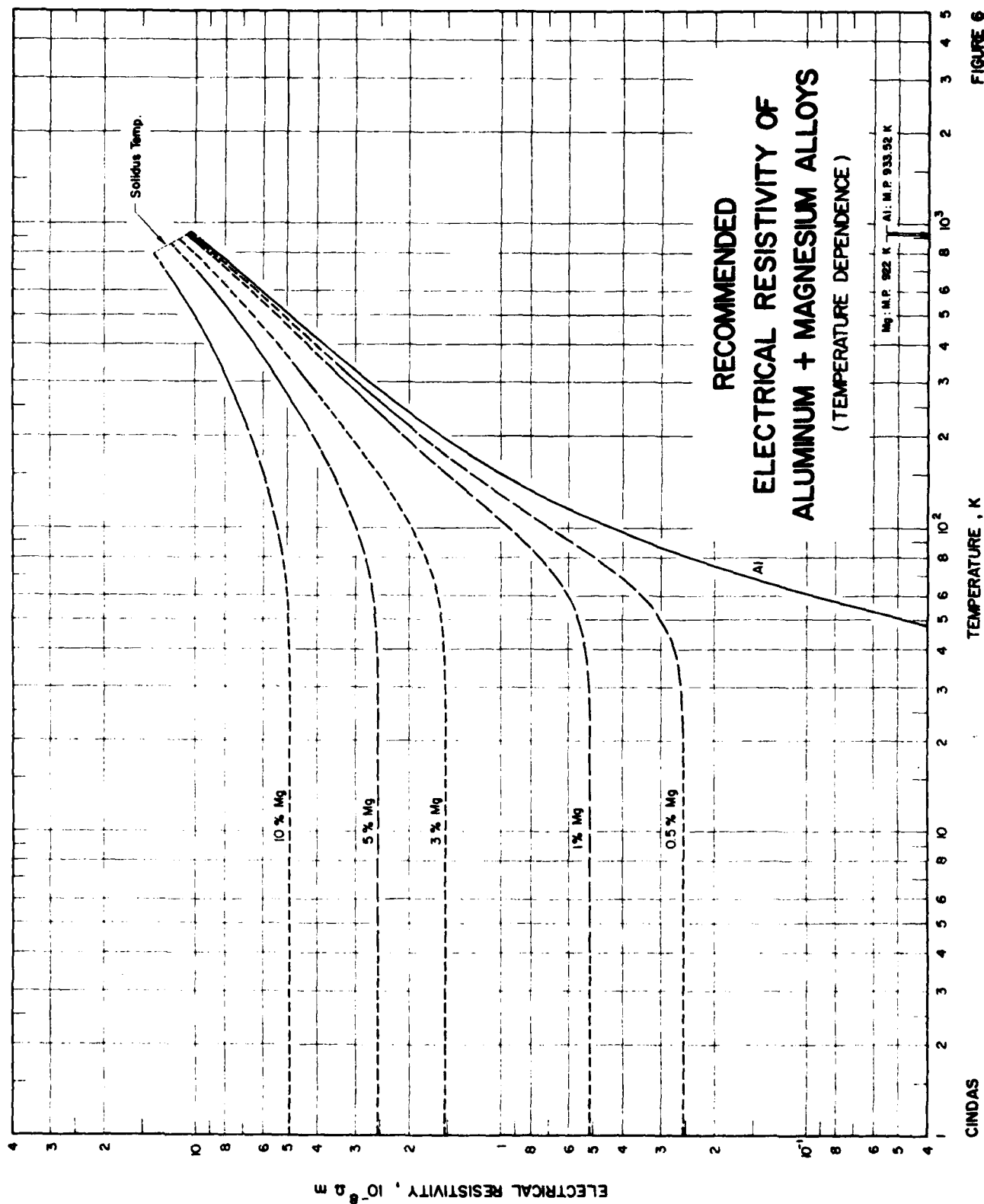
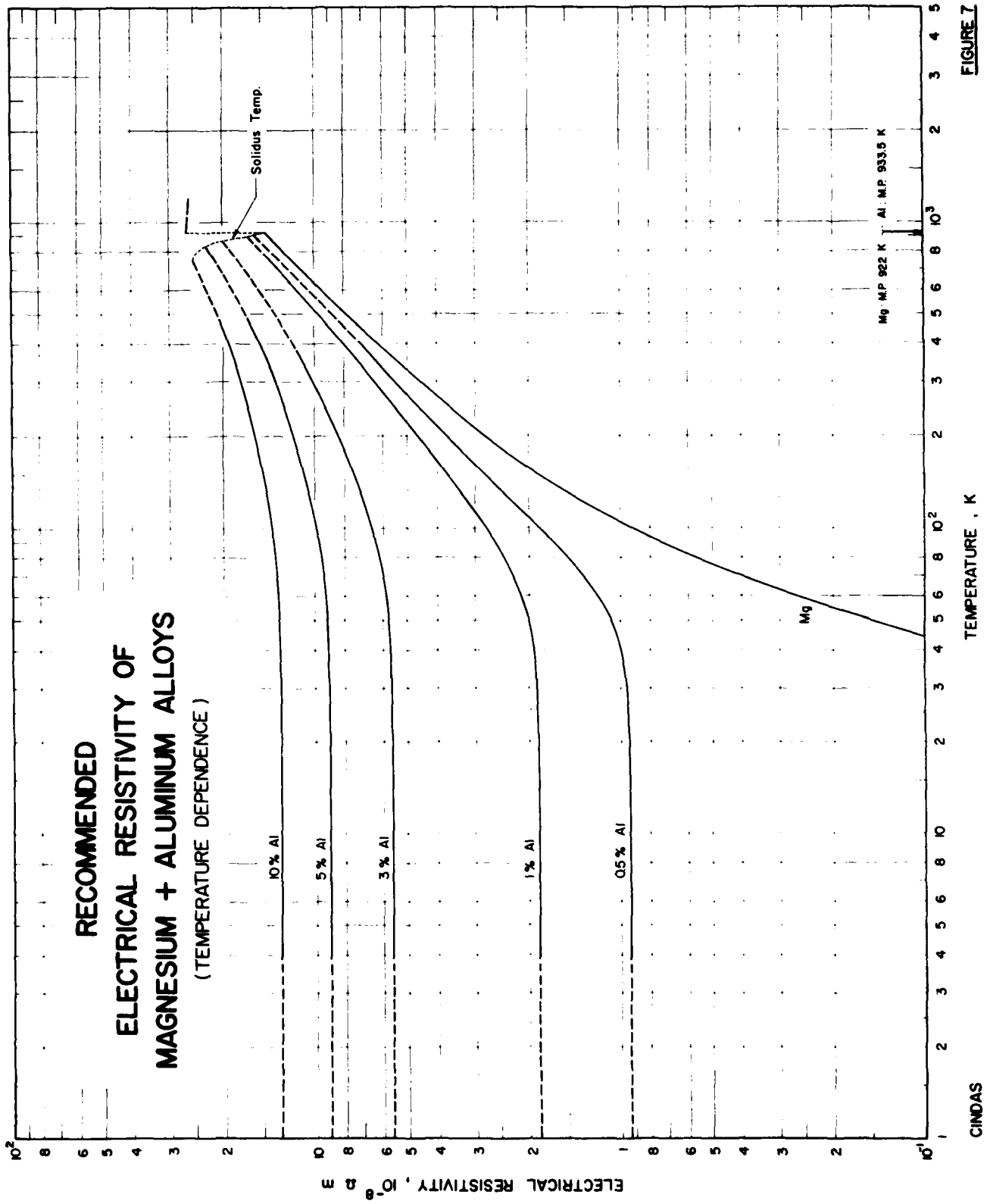
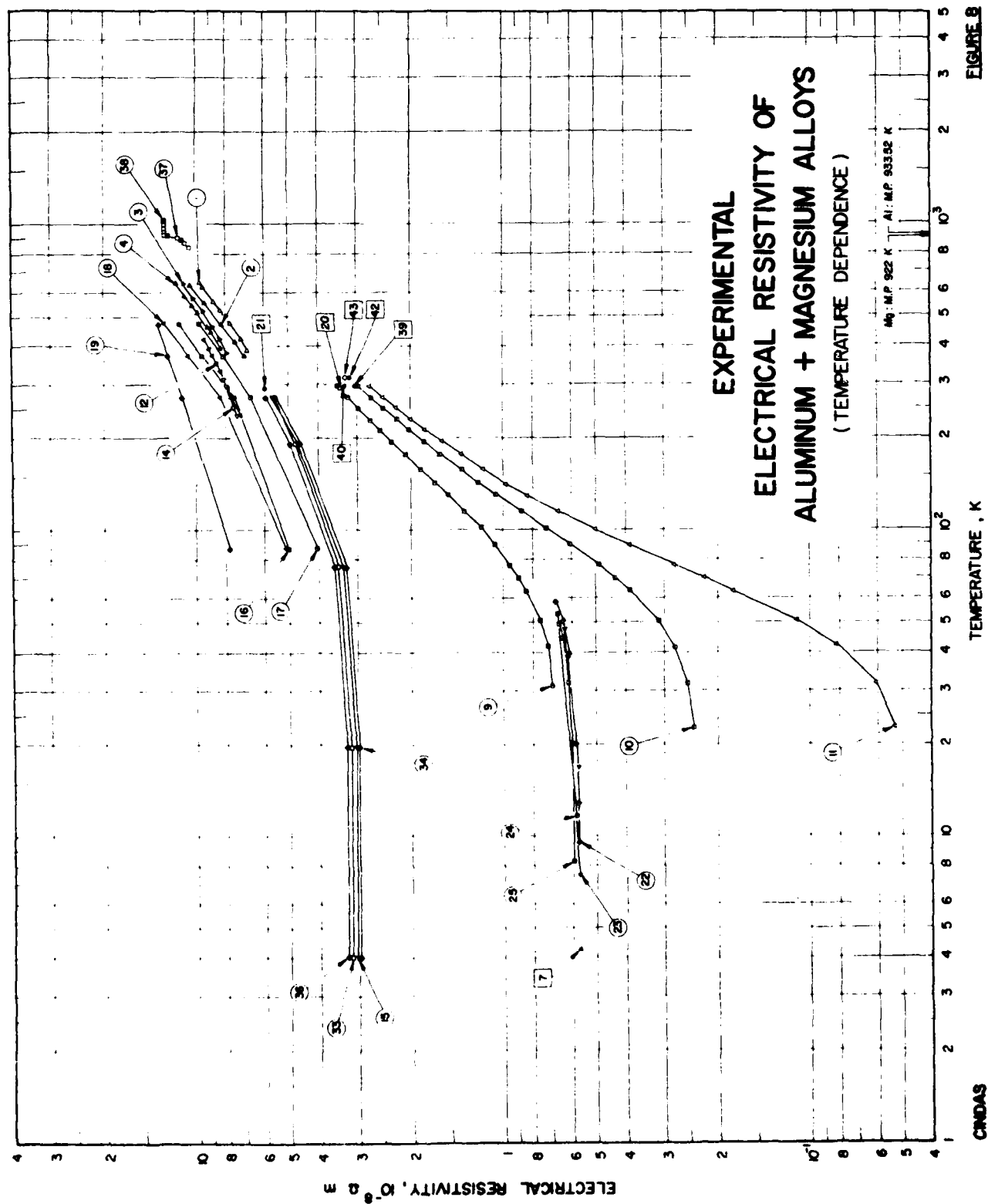


FIGURE 6







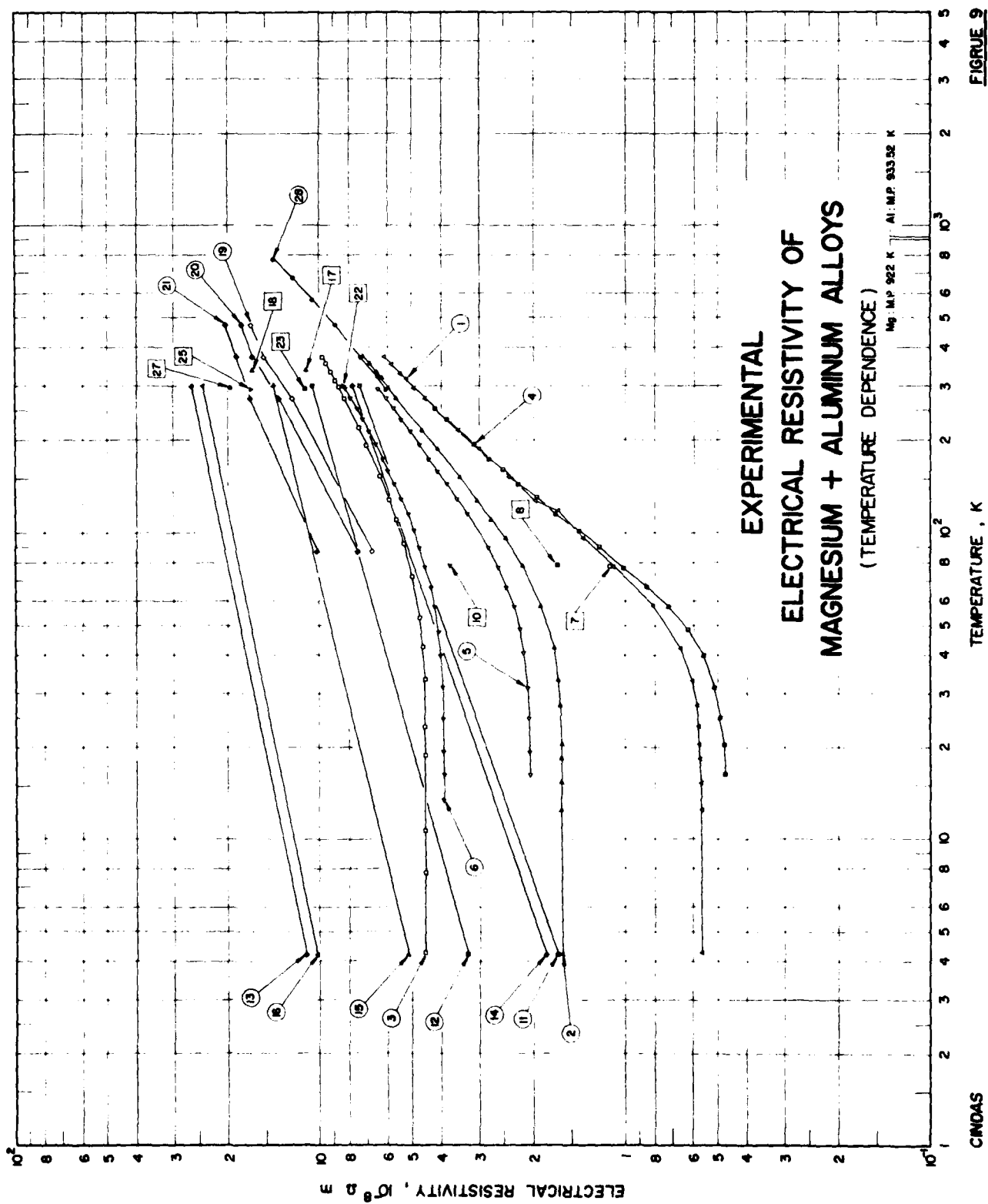


TABLE 9. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF ALUMINUM + MAGNESIUM ALLOYS (Temperature Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Al Mg	Composition (continued), Specifications, and Remarks
1	99	Hase, R., Heterberg, R., and Walkenhorst, W.	1940	V	387-650	Hydronalium 5	5	Nominal composition: 6 to 10 mm in diameter and 25 cm long; cast at 973 K into molds at 473 K; rolled and drawn, then turned into rods. Similar to the above specimen except containing some Si as impurity.
2	99	Hase, R., et al.	1940	V	374-638	Hy 51		Nominal composition: same dimensions and fabrication method as the specimen for data set No. 1.
3	99	Hase, R., et al.	1940	V	384-641	Hy 7	7	Similar to the above specimen except containing some Si as impurity.
4	99	Hase, R., et al.	1940	V	396-643	Hy 71		Wires 0.127 mm in diameter drawn from 1.016 mm diameter wire supplied by Cominco (nominally 99.9999 aluminum and dopants); etched with a hot mixture of 95% phosphoric acid and 5% nitric acid between drawings; annealed in air; suddenly heated to 723 K and furnace cooled to room temperature.
5*	108	Doyama, M., Kochler, J.S., Lwin, Y.N., Ryan, E.A., and Shaw, D.G.	1971	A	4.2	LD <sub>2</sub>	0.08	Similar to the above specimen.
6*	106	Doyama, M., et al.	1971	A	4.2	B1	0.1	Similar to the above specimen.
7	106	Doyama, M., et al.	1971	A	4.2	22	1.6	Similar to the above specimen.
8*	106	Doyama, M., et al.	1971	A	4.2	61	0.03	Similar to the above specimen.
9	101	Seth, R.S. and Woods, S.B.	1970	A	31-297		1.580	Calculated composition (1.75 at/o Mg); made by melting freshly cleaned pellets of silver and cadmium in sealed quartz tube of about 10 torr of hydrogen gas, homogenized by shaking periodically during a melt period lasting several h, then annealed just below the solidification temperature for several more h, about 6 mm diameter rod reduced to 1.5 mm by drawing through steel dies and reduced to 0.5 mm by drawing through diamond dies; annealed at 500 C for 12 h in a vacuum of 10 <sup>-4</sup> torr.
10	101	Seth, R.S. and Woods, S.B.	1970	A	23-298		0.580	Calculated composition (0.61 at/o Mg); similar to the above specimen.
11	101	Seth, R.S. and Woods, S.B.	1970	A	23-297		0.117	Calculated composition (0.13 at/o Mg); similar to the above specimen.
12	100	Cordier, H. and Detert, K.	1961	A	238-425		10.0	<0.01 each of Cu, Fe, and Si; polycrystalline; hot-pressed 12 mm thick rod cold-hammered to 5 mm thick, then cold-drawn to 2 mm or 0.5 mm thick, homogenized at 683 K and cooled slowly in air.
13*	100	Cordier, H. and Detert, K.	1961	A	238, 293			Similar to the above except aged at 293 K.
14	100	Cordier, H. and Detert, K.	1961	A	238-314			Similar to the above except quenched from 1273 K.
15	102	Clart, A.F., Chulda, G.E., and Wallace, G.H.	1969	A	4-273	Al 5083	94	0.63 Mn, 0.19 Fe, 0.13 Cr, the rest Cu, Ni, Si, Ti, V, and Zn; average grain size 6.077 mm.
16	79	Manchen, W.	1931	V	87-476		92.0	Cast.
17	79	Manchen, W.	1931	V	87-476		92.0	Annealed.
18	79	Manchen, W.	1931	V	87-476		98.0	Cast.
19	79	Manchen, W.	1931	V	87-476		96.0	Annealed.

\* Not shown in figure.

TABLE 9. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF ALUMINUM + MAGNESIUM ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Al Mg	Composition (continued), Specifications, and Remarks
20	109	Materials in Design Engineering	1959		293.2	5005	0.8	Nominal composition; annealed at 617 K.
21	109	Materials in Design Engineering	1959		293.2	5056	4.7~5.6	0.05-0.20 Cr and 0.05~0.20 Mn (nominal composition); annealed at 617 K.
22	110	Täubert, P., Thom, F., and Gammert, U.	1973		9.5-58	AlMg 1		Composition not reported; 1 mm diameter x 8 cm long; homogenized at 450 C for 4.5 h, then quenched in ice water.
23	110	Täubert, P., et al.	1973		7.4-47	AlMg 1		The above specimen; second run.
24	110	Täubert, P., et al.	1973		12.54	AlMg 1		The above specimen reannealed at 30 C for 20 h and at 50 C for 20 h.
25	110	Täubert, P., et al.	1973		8.2-49	AlMg 1		The above specimen reannealed at 70 C for 20 h and at 70 C for 40 h.
26*	103	Clark, A.F., Childs, G.E., and Wallace, G.H.	1970	A	4.0-273	Al 5083	~94 4.75	0.63 Mn, 0.19 Fe, 0.13 Cr, and <0.1 each of Cu, Ni, Si, Ti, V, and Zn; 6.35 mm diameter x 152 mm long; machined and heat-treated (H113); grain size 0.02 mm; measured across grain; reported error ~1%.
27*	103	Clark, A.F., et al.	1970	A	4.0-273	Al 5083		The above specimen measured with grain.
28*	104	Clark, A.F. and Tryon, P.V.	1972	A	4.0-273	5056	4.95	0.11 Fe, 0.09 Mn, 0.08 Si, 0.04 Zn, 0.02 Cu, 0.01 Ni, and 0.01 Ti; 0.635 cm diameter x 15.2 cm long; machined and ground; as received.
29*	104	Clark, A.F. and Tryon, P.V.	1972	A	4.0-273	5056		From the same heat as the above specimen.
30*	104	Clark, A.F. and Tryon, P.V.	1972	A	4.0-273	5056		From the same heat as the above specimen.
31*	104	Clark, A.F. and Tryon, P.V.	1972	A	4.0-273	5056		From the same heat as the above specimen.
32*	104	Clark, A.F. and Tryon, P.V.	1972	A	4.0-273	5056	4.93	0.12 Fe, 0.09 Mn, 0.08 Si, 0.03 Zn, 0.02 Cu, 0.01 Ni, and 0.01 Ti; 0.635 cm diameter x 15.2 cm long; from a different heat of the same manufacturer of the above specimen; as received.
33	104	Clark, A.F. and Tryon, P.V.	1972	A	4.0-273	5056	5.17	0.14 Fe, 0.10 Si, 0.09 Mn, 0.03 Cu, and 0.01 each of Ni, Ti, and Zn; similar to the above specimen except from another heat of the same manufacturer.
34	104	Clark, A.F. and Tryon, P.V.	1972	A	4.0-273	5056	4.94	0.12 Fe, 0.10 Mn, 0.09 Si, and 0.01 each of Cu, Ni, and Ti; similar to the above specimen except from another heat of the same manufacturer.
35*	104	Clark, A.F. and Tryon, P.V.	1972	A	4.0-273	5093	5	Nominal composition; 0.635 cm diameter x 15.2 cm long; from a different manufacturer of the above specimen; machined and ground; as received.
36	104	Clark, A.F. and Tryon, P.V.	1972	A	4.0-273	5083	5	Similar to the above specimen except from another manufacturer.
37	111	Romanov, O.V. and Persson, Z.V.	1973		843-1008	3	1.00	Calculated composition (1.11 a/o Mg); in solid and liquid states.
38	111	Romanov, O.V. and Persson, Z.V.	1973		893-1036	4	0.50	Calculated composition (0.56 a/o Mg); in solid and liquid states.
39	85	Robinson, A.T. and Dorn, J.E.	1951	B	297.2		0.499	Calculated composition (0.564 a/o Mg); impurities: 0.007 Cu, 0.003 Fe, and 0.003 Si; sheet specimen 0.070 in. thick and 0.250 in. wide; reported error $\pm 0.5\%$ .

\* Not shown in figure.

TABLE 2. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF ALUMINUM + MAGNESIUM ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Al Mg	Composition (continued), Specifications, and Remarks
40	85	Robinson, A. T. and Dorn, J. E.	1951	B	297.2		0.990	Calculated composition (1.097 at/o Mg); impurities: 0.007 Cu, 0.004 Fe, and 0.004 Si; sheet specimen 0.070 in. thick and 0.250 in. wide.
41*	85	Robinson, A. T. and Dorn, J. E.	1951	B	297.2		1.459	Calculated composition (1.617 at/o Mg); impurities: 0.006 Cu, 0.004 Fe, and 0.003 Si; sheet specimen 0.070 in. thick and 0.250 in. wide.
42	112	Rapp, O. and Fogelholm, R.	1974		318.2		0.24	Calculated composition (0.27 at/o Mg); $\rho/\delta T = 0.0114 \mu\Omega \text{ cm K}^{-1}$ at 318.2 K.
43	112	Rapp, O. and Fogelholm, R.	1974		318.2		0.43	Calculated composition (0.46 at/o Mg); $\rho/\delta T = 0.0114 \mu\Omega \text{ cm K}^{-1}$ at 318.2 K.

\* Not shown in figure.

TABLE 10.  
EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF ALUMINUM + MAGNESIUM ALLOYS (Temperature Dependence)

T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P	T	P
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**\* Not shown in figure.**

TABLE 10. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF ALUMINUM + MAGNESIUM ALLOYS (Temperature Dependence) (continued)

T	P	T	P
<u>DATA SET 33</u>		<u>DATA SET 38 (cont.)</u>	
4.0	2.152	966	26.60*
19.65	3.168	989	26.67*
75.75	3.455	1036	27.40
192.4	4.729	<u>DATA SET 39</u>	
273.2	5.690*	297.2	3.030
<u>DATA SET 34</u>		<u>DATA SET 40</u>	
4.0	2.929	297.2	3.296
19.65	3.002	<u>DATA SET 41*</u>	
75.75	3.265	297.2	3.530
192.4	4.642	<u>DATA SET 42</u>	
273.2	5.592	318.2	3.19
<u>DATA SET 35*</u>		<u>DATA SET 43</u>	
4.0	3.003	318.2	3.29
19.65	3.097	<u>DATA SET 36</u>	
75.75	3.288	4.0	2.261
192.4	4.640	19.65	3.254
273.2	5.591	75.75	3.536
<u>DATA SET 36</u>		192.4	4.946
4.0	2.261	273.2	5.918
19.65	3.254	<u>DATA SET 37</u>	
75.75	3.536	943	10.55
192.4	4.946	972	10.93
273.2	5.918	996	11.12*
<u>DATA SET 37</u>		996	11.47*
943	10.55	927	27.40
972	10.93	948	27.40
996	11.12*	969	27.63*
996	11.47*	977	27.63
927	27.40	1008	27.86
948	27.40	<u>DATA SET 38</u>	
969	27.63*	893	11.27
977	27.63	920	12.24
1008	27.86	931	26.60*
<u>DATA SET 38</u>		940	26.60*

\* Not shown in figure.

TABLE 11. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF MAGNESIUM + ALUMINUM ALLOYS (Temperature Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Mg Al	Composition (continued), Specifications, and Remarks
1	107	Hedgcock, F.T. and Muri, W.B.	1964	A	4.2-373	400	0.32	Calculated composition (0.29 a/o Al); 4 x 0.125 x 0.01 in.; specimen cut from the supplied ingots, rolled into 0.01 in. strips, etched, cut to size; annealed in a helium atmosphere at 7 cm of Hg at 450 C for 12 h; electrical resistivity data calculated from the resistance ratio $R(T)/R(273)$ reported by the author; electrical resistivity reported as 4.545 $\mu\Omega$ cm at 273 K; reported error $\pm 0.2\%$ .
2	107	Hedgcock, F.T. and Muri, W.B.	1964	A	4.2-373	402	0.89	Calculated composition (0.80 a/o Al); similar to the above specimen except electrical resistivity reported as 5.624 $\mu\Omega$ cm at 273 K.
3	107	Hedgcock, F.T. and Muri, W.B.	1964	A	4.3-373	405	2.67	Calculated composition (2.41 a/o Al); similar to the above specimen except electrical resistivity reported as 8.333 $\mu\Omega$ cm at 273 K.
4	101	Seth, R.S. and Woods, S.B.	1970	A	16-295		0.24	Calculated composition (0.22 a/o Al); 99.98% pure; prepared by Dow Chemical Co. from sublimed magnesium; obtained in the form of a non-uniform 0.035 in. diameter wire, drawn through the 0.032 in. diamond die to produce uniform, smooth wires; annealed at 350 C for 8 h in a hydrogen gas.
5	101	Seth, R.S. and Woods, S.B.	1970	A	16-294		1.10	Calculated composition (0.98 a/o Al); similar to the above specimen.
6	101	Seth, R.S. and Woods, S.B.	1970	A	13-294		2.10	Calculated composition (1.90 a/o Al); similar to the above specimen.
7	113	Blatt, F.J.	1968		78		0.32	No details reported.
8	113	Blatt, F.J.	1968		78		0.59	No details reported.
9*	113	Blatt, F.J.	1968		78		0.89	No details reported.
10	113	Blatt, F.J.	1968		78		1.82	No details reported.
11	114	Collings, E.W., Hedgcock, F.T., Muir, W.B., and Muto, Y.	1964		4.2, 300	402	0.90	No details reported.
12	114	Collings, E.W., et al.	1964		4.2, 300	403	1.84	No details reported.
13	114	Collings, E.W., et al.	1964		4.2, 300	407	6.10	No details reported.
14	114	Collings, E.W., et al.	1964		4.2, 300	91076	1.03	0.061 Mn.
15	114	Collings, E.W., et al.	1964		4.2, 300	91077	2.87	0.056 Mn.
16	114	Collings, E.W., et al.	1964		4.2, 300	91078	5.51	0.050 Mn.
17	55	Smith, A.W.	1925	B	336.2		95.82	0.028 Fe and 0.019 Si; supplied by Aluminum Co. of America.
18	55	Smith, A.W.	1925	B	336.2		99.82	0.023 Si and 0.028 Fe; supplied by Aluminum Co. of America.
19	105, 79	Saebler, J.; Mannchen, W.	1929	V	87-476		94.0	Cast.
20	105, 79	Saebler, J.; Mannchen, W.	1929	V	87-476		92.0	Cast.

\* Not shown in figure.



TABLE 11. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF MAGNESIUM + ALUMINUM ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent)		Composition (continued), Specifications, and Remarks
							Mg	Al	
21	106	Stachler, J. J.	1929	V	87-476		88	12	Cast.
22	115	Kibuchi, R.	1932		300.2		97.9	2.1	No details reported.
23	115	Kibuchi, R.	1932		295.5		95.8	4.2	No details reported.
24*	115	Kibuchi, R.	1932		295.1		93.8	6.2	No details reported.
25	115	Kibuchi, R.	1932		295.1		91.8	8.2	No details reported.
26*	115	Kibuchi, R.	1933		292.5		89.7	10.3	No details reported.
27	115	Kibuchi, R.	1933		293.8		87.8	12.2	No details reported.
28	106	Powell, R. W., Hickman, M. J., and Tye, R. P.	1964		293-773	Magnox B		1.0	0.002 to 0.003 Be; 2.5 cm diameter x 20" cm long.

\* Not shown in figure.



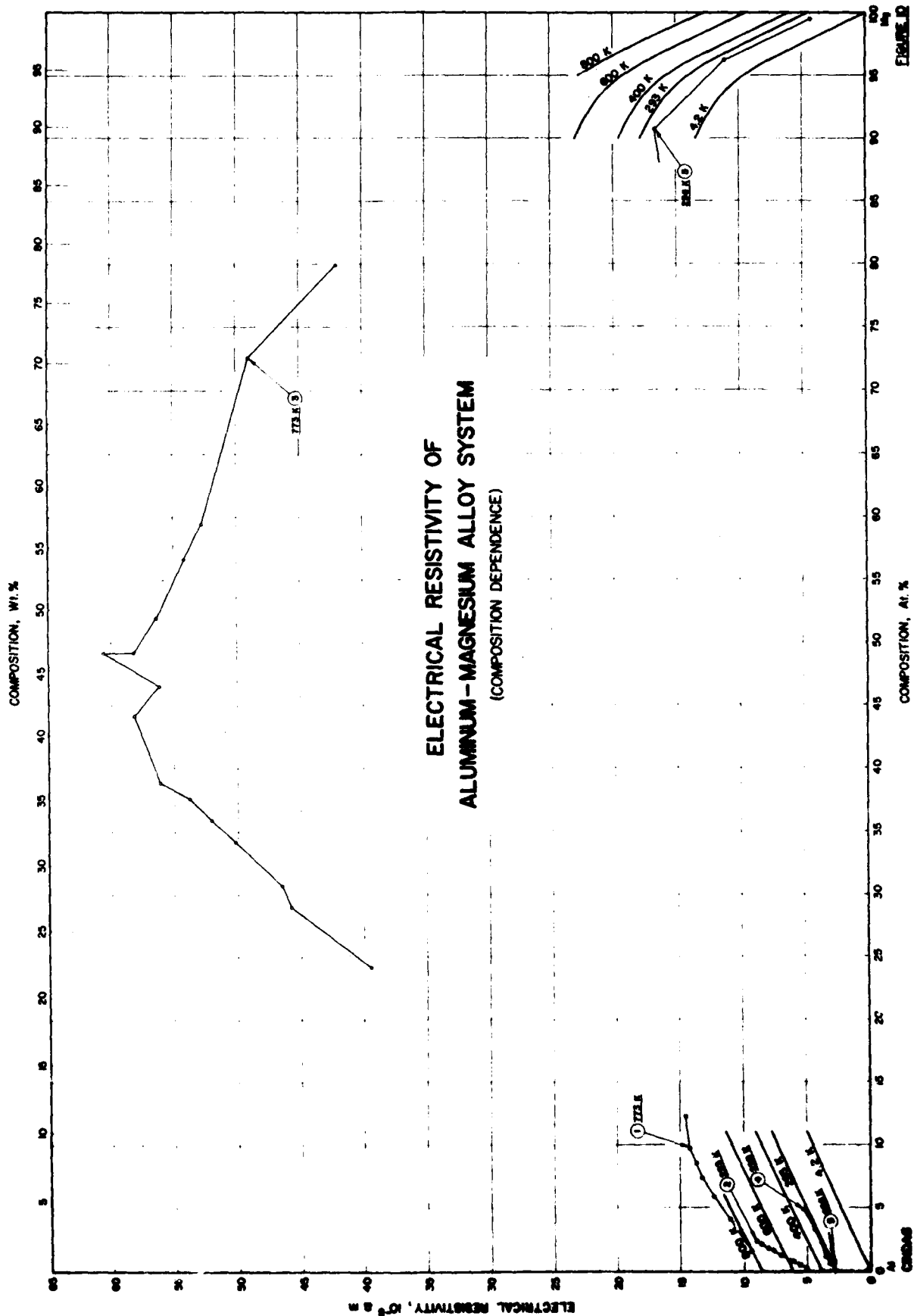


TABLE 13. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF ALUMINUM-MAGNESIUM ALLOY SYSTEM (Composition Dependence)

Data Ref. Set No.	Author(s)	Year	Method Used	Temp., K	Composition Range (At. % of Mg)	Composition (weight percent), Specifications, and Remarks
1 116	Matsuda, T. and Sato, T.	1966	A	773.2	1.5-12	50 x 3 x 2 mm; prepared from 99.999 Al and 99.99 Mg by melting in an induction furnace; homogenized in vacuum, cut and rolled.
2 117	Salkovitz, E. I., Schindler, A. I., and Kammer, E. W.	1967	B	293.2	0.3-2.4	Polycrystallines; specimens 9.54 to 29.30 cm long, 0.412 to 0.640 cm wide, and 0.212 to 0.240 cm thick; extruded and annealed; $\rho/\ell T$ ranging from 1.605 to $1.494 \times 10^{-8} \mu\Omega \text{ cm } ^\circ\text{C}^{-1}$ at room temperature.
3 118	Steeb, S. and Wörner, S.	1968		773.2	24-80	Alloys in liquid state.
4 62	Gulyaev, A. P. and Trusova, E. F.	1950		293.2	0.6-4.9	Measuring temperature assigned here as 20 C.
5 55	Smith, A. W.	1925	B	296.2	0-100	Specimens about 0.3 cm <sup>3</sup> in cross-section and 8 cm long; supplied by Aluminum Co. of America.

TABLE 14. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF ALUMINUM-MAGNESIUM ALLOY SYSTEM (Composition Dependence)

[Composition, C, at % Mg; Electrical Resistivity,  $\rho$ ,  $10^{-4}$   $\Omega$  m]

C	$\rho$	C	$\rho$
DATA SET 1 ( $T = 773.2$ K)		DATA SET 4 ( $T = 293.2$ K)	
1.54	9.40	0.57	3.19
4.02	11.05	1.14	3.505
5.91	12.35	2.2	3.86
7.41	13.22	3.27	4.315
8.51	13.72	4.89	5.24
9.70	14.23		
12.29	14.54		
DATA SET 2 ( $T = 293.2$ K)		DATA SET 5 ( $T = 296.2$ K)	
0.29	5.032	0	2.73
0.53	5.538	90.73	16.67
0.80	6.091	96.22	11.04
0.88	6.216	99.95	4.26
1.31	7.036		
1.64	7.688		
1.82	8.081		
2.18	8.646		
2.41	9.064		
DATA SET 3 ( $T = 773.2$ K)			
24.1	39.5		
28.9	45.8		
30.6	48.5		
34.1	50.2		
35.8	52.1		
37.5	53.8		
38.8	56.2		
44.1	58.2		
46.5	56.2		
49.1	60.6		
49.1	59.2		
51.9	56.4		
56.6	54.2		
59.4	52.8		
72.5	49.0		
79.9	42.0		

### 3.3. Copper-Gold Alloy System

The copper-gold alloy system forms a continuous series of solid solutions over the entire range of compositions. However, ordered structures are formed at temperatures below 663 K for compositions ranging from about 40 to 63% Au (17.7 to 35.5 at.% Au) and below 683 K for compositions ranging from about 63 to 94% Au (35.5 to 83.5 at.% Au). These ordered structures are due to the formation of intermetallic compounds  $\text{Cu}_3\text{Au}$  (50.85% Au),  $\text{CuAu}$  (75.63% Au), and  $\text{CuAu}_3$  (90.30% Au). For  $\text{Cu}_3\text{Au}$  and  $\text{CuAu}$ , the ordered state possesses a remarkably lower resistivity than that of the disordered state. However, for  $\text{CuAu}_3$ , it has been found that the electrical resistivity increases as ordering proceeds below the transition point, but the residual resistivity is slightly lower in the highly ordered state [119-121].

The experiments by Johansson and Linde [122] (Cu-Au data sets 1-3) showed the dependence of the electrical resistivity on the degree of ordering. Solid solutions of metals are usually of the substitutional type in which the different atoms are positioned randomly at sites of the crystal lattice. In a number of alloys with stoichiometric composition and at sufficiently low temperatures, the atoms are arranged in such a manner that each species of atom occupies only a certain type of site in the crystal lattice; an alloy in this state is called ordered. As the temperature increases, there is a transition of some of the atoms from their sites to foreign sites; such an alloy is called partially ordered. The concentration of atoms of a given type on foreign sites increases with temperature and at some temperature the concentration of atoms on sites of different types becomes identical; such an alloy is called disordered. The temperature at which such a transition occurs is called the order-disorder phase transition temperature or the critical temperature. The order-disorder phase transition occurs not only in alloys with stoichiometric composition and the critical temperature is a well defined function of composition. In this work experimental data on intermetallic compounds and ordered alloys are excluded, though data on partially ordered alloys are compiled and presented in the figures and tables. However, these data are not evaluated or analyzed since recommended values for disordered alloys only are generated.

There are 243 sets of experimental data available for this alloy system. These experimental data sets are listed in tables 16, 18, and 20, tabulated in

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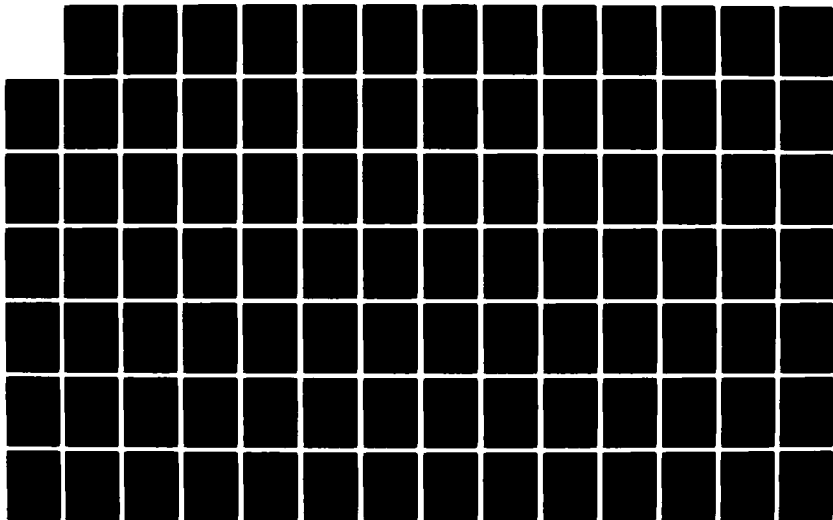
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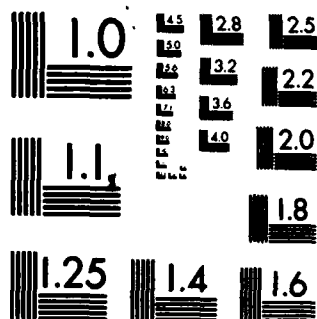
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tables 17, 19, and 21 , and shown partially in figures 13, 14, and 15. Unfortunately, most of the available data are for specimens in partially ordered state. For the disordered state, more important experimental data and information are from the following. Johansson and Linde [122] (Cu-Au data sets 1-3) reported a complete curve of resistivity versus composition at 293 K. Borelius, Johansson, and Linde [123] (Cu + Au data sets 6-8 and Au + Cu data sets 1-22) measured a series of alloys containing 20.2, 22.6, 25.0, 27.7, 30.1, 45.0, 50.0, and 55.0 at.% Au during both heating and cooling of the specimens. Passaglia and Love [124] (Au + Cu data sets 23-26) measured the resistivity of Au + Cu alloys from liquid helium temperature to about 90 K for both quenched and annealed specimens. Tainsh and White [125] (Cu + Au data sets 9, 10) reported the resistivity of alloys containing 20.1 and 38.0 wt.% Au at 4.2, 90, and 293 K. Linde [126] (Cu + Au data sets 1-5) reported the resistivity at 291.2 K of alloys containing 1.53, 3.00, 5.92, 7.05, and 8.75 wt.% Au.

The smoothing and synthesizing of the electrical resistivity data was based mainly on the results reported in the above five research papers. Since Cu, Ag, and Au are in the same column of the periodic table, the properties of Au + Cu alloys and Au + Ag alloys may be similar in some respects. Accordingly, in the composition range where ordering occurs, some of the resistivity versus temperature curves have been obtained by taking a point on the resistivity versus composition isotherm and drawing a curve parallel to the electrical resistivity curve of an Au + Ag alloy of the same atomic concentration; values obtained in this way for the electrical resistivity are considered only provisional. Using this method, the values obtained for the gold-rich alloys would appear to be more reliable than those for the copper-rich alloys.

The resulting recommended electrical resistivity values for Cu, Au, and for 25 Cu-Au binary alloys are presented in table 15 and shown in figures 11, 12, and 15. The recommended values for Cu and for Au are for well-annealed high-purity specimens, but those values for temperatures below about 100 K are applicable only to Cu and Au having residual electrical resistivities as given at 1 K in table 15. The alloys for which the recommended values are generated are not ordered and have not been quenched or cold-worked severely. The recommended values cover a full range of temperature from 1 K to the solidus temperature of the alloy where melting starts. These values are not corrected for the thermal expansion of the material. The estimated uncertainties in

the values for the various alloys and for different temperature ranges are explicitly stated in a footnote to table 15. Many of the values in table 15 are indicated as provisional because their uncertainties are greater than  $\pm 5\%$ .

TABLE 15. RECOMMENDED ELECTRICAL RESISTIVITY OF COPPER-GOLD ALLOY SYSTEM†

[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Cu: 100.00% (100.00 At.%) Au: 0.00% (0.00 At.%)		Cu: 99.50% (99.84 At.%) Au: 0.50% (0.16 At.%)		Cu: 99.00% (99.68 At.%) Au: 1.00% (0.32 At.%)		Cu: 97.00% (99.01 At.%) Au: 3.00% (0.99 At.%)		Cu: 95.00% (98.33 At.%) Au: 5.00% (1.67 At.%)		Cu: 90.00% (96.54 At.%) Au: 10.00% (3.46 At.%)	
T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
1	0.00200	1	0.100 <sup>±</sup>	1	0.200 <sup>±</sup>	1	0.520 <sup>±</sup>	1	0.870 <sup>±</sup>	1	1.72 <sup>±</sup>
4	0.00200	4	0.100 <sup>±</sup>	4	0.200 <sup>±</sup>	4	0.530 <sup>±</sup>	4	0.870 <sup>±</sup>	4	1.72 <sup>±</sup>
7	0.00200	7	0.100 <sup>±</sup>	7	0.200 <sup>±</sup>	7	0.530 <sup>±</sup>	7	0.870 <sup>±</sup>	7	1.72 <sup>±</sup>
10	0.00202	10	0.100 <sup>±</sup>	10	0.200 <sup>±</sup>	10	0.530 <sup>±</sup>	10	0.870 <sup>±</sup>	10	1.72 <sup>±</sup>
15	0.00218	15	0.100 <sup>±</sup>	15	0.200 <sup>±</sup>	15	0.530 <sup>±</sup>	15	0.870 <sup>±</sup>	15	1.72 <sup>±</sup>
20	0.00280	20	0.100 <sup>±</sup>	20	0.200 <sup>±</sup>	20	0.530 <sup>±</sup>	20	0.870 <sup>±</sup>	20	1.72 <sup>±</sup>
25	0.00450	25	0.102 <sup>±</sup>	25	0.200 <sup>±</sup>	25	0.530 <sup>±</sup>	25	0.870 <sup>±</sup>	25	1.72 <sup>±</sup>
30	0.00530	30	0.108 <sup>±</sup>	30	0.201 <sup>±</sup>	30	0.531 <sup>±</sup>	30	0.871 <sup>±</sup>	30	1.73 <sup>±</sup>
40	0.0240	40	0.120 <sup>±</sup>	40	0.210 <sup>±</sup>	40	0.540 <sup>±</sup>	40	0.879 <sup>±</sup>	40	1.75 <sup>±</sup>
50	0.0520	50	0.141 <sup>±</sup>	50	0.229 <sup>±</sup>	50	0.559 <sup>±</sup>	50	0.901 <sup>±</sup>	50	1.78 <sup>±</sup>
60	0.0974	60	0.180 <sup>±</sup>	60	0.261 <sup>±</sup>	60	0.591 <sup>±</sup>	60	0.941 <sup>±</sup>	60	1.84 <sup>±</sup>
70	0.154	70	0.237 <sup>±</sup>	70	0.311 <sup>±</sup>	70	0.641 <sup>±</sup>	70	0.997 <sup>±</sup>	70	1.90 <sup>±</sup>
80	0.216	80	0.302 <sup>±</sup>	80	0.378 <sup>±</sup>	80	0.709 <sup>±</sup>	80	1.06 <sup>±</sup>	80	1.96 <sup>±</sup>
90	0.282	90	0.370 <sup>±</sup>	90	0.450 <sup>±</sup>	90	0.780 <sup>±</sup>	90	1.14 <sup>±</sup>	90	2.04 <sup>±</sup>
100	0.349	100	0.439 <sup>±</sup>	100	0.520 <sup>±</sup>	100	0.850 <sup>±</sup>	100	1.21 <sup>±</sup>	100	2.11 <sup>±</sup>
150	0.701	150	0.790 <sup>±</sup>	150	0.871 <sup>±</sup>	150	1.20 <sup>±</sup>	150	1.56 <sup>±</sup>	150	2.46 <sup>±</sup>
200	1.048	200	1.14 <sup>±</sup>	200	1.22 <sup>±</sup>	200	1.56 <sup>±</sup>	200	1.91 <sup>±</sup>	200	2.50 <sup>±</sup>
250	1.388	250	1.49 <sup>±</sup>	250	1.57 <sup>±</sup>	250	1.91 <sup>±</sup>	250	2.25 <sup>±</sup>	250	3.14 <sup>±</sup>
273	1.544	273	1.64	273	1.73	273	2.06	273	2.41	273	3.29
293	1.678	293	1.77 <sup>±</sup>	293	1.86 <sup>±</sup>	293	2.20 <sup>±</sup>	293	2.54 <sup>±</sup>	293	3.42 <sup>±</sup>
300	1.725	300	1.82 <sup>±</sup>	300	1.91 <sup>±</sup>	300	2.24 <sup>±</sup>	300	2.59 <sup>±</sup>	300	3.46 <sup>±</sup>
350	2.051	350	2.14 <sup>±</sup>	350	2.24 <sup>±</sup>	350	2.58 <sup>±</sup>	350	2.92 <sup>±</sup>	350	3.79 <sup>±</sup>
400	2.398	400	2.48 <sup>±</sup>	400	2.58 <sup>±</sup>	400	2.91 <sup>±</sup>	400	3.26 <sup>±</sup>	400	4.12 <sup>±</sup>
500	3.079	500	3.17 <sup>±</sup>	500	3.27 <sup>±</sup>	500	3.60 <sup>±</sup>	500	3.95 <sup>±</sup>	500	4.81 <sup>±</sup>
600	3.771	600	3.88 <sup>±</sup>	600	3.98 <sup>±</sup>	600	4.31 <sup>±</sup>	600	4.66 <sup>±</sup>	600	5.52 <sup>±</sup>
700	4.481	700	4.58 <sup>±</sup>	700	4.68 <sup>±</sup>	700	5.01 <sup>±</sup>	700	5.37 <sup>±</sup>	700	6.22 <sup>±</sup>
800	5.213	800	5.31 <sup>±</sup>	800	5.42 <sup>±</sup>	800	5.74 <sup>±</sup>	800	6.10 <sup>±</sup>	800	6.94 <sup>±</sup>
900	5.973	900	6.06 <sup>±</sup>	900	6.16 <sup>±</sup>	900	6.50 <sup>±</sup>	900	6.86 <sup>±</sup>	900	7.69 <sup>±</sup>
1000	6.768	1000	6.86 <sup>±</sup>	1000	6.96 <sup>±</sup>	1000	7.29 <sup>±</sup>	1000	7.64 <sup>±</sup>	1000	8.48 <sup>±</sup>
1100	7.596	1100	7.69 <sup>±</sup>	1100	7.80 <sup>±</sup>	1100	8.11 <sup>±</sup>	1100	8.46 <sup>±</sup>	1100	9.30 <sup>±</sup>
1200	8.470	1200	8.55 <sup>±</sup>	1200	8.66 <sup>±</sup>	1200	8.98 <sup>±</sup>	1200	9.33 <sup>±</sup>	1200	10.16 <sup>±</sup>
1300	9.395	1300	9.41 <sup>±</sup>	1300	9.52 <sup>±</sup>	1300	9.85 <sup>±</sup>	1300	10.18 <sup>±</sup>	1300	11.03 <sup>±</sup>
1357.6	9.946 (s)	1357.6	9.95 <sup>±</sup>	1357.6	9.96 <sup>±</sup>	1357.6	10.25 <sup>±</sup>	1357.6	10.55 <sup>±</sup>	1357.6	11.19 <sup>±</sup>
1358	21.01 (z)										
1700	24.41										

† Uncertainties in the electrical resistivity values are as follows:

- 100.00 Cu - 0.00 Au:  $\pm 5\%$  up to 100 K,  $\pm 1\%$  above 100 K to 250 K,  $\pm 0.5\%$  above 250 K to 500 K,  $\pm 1\%$  above 500 K to 1357.6 K, and  $\pm 5\%$  above 1357.6 K.  
 99.50 Cu - 0.50 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 99.00 Cu - 1.00 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 97.00 Cu - 3.00 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 95.00 Cu - 5.00 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 90.00 Cu - 10.00 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.

‡ Provisional value.

§ In temperature range where no experimental data are available.

TABLE 18. RECOMMENDED ELECTRICAL RESISTIVITY OF COPPER-GOLD ALLOY SYSTEM† (continued)  
[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Cu: 85.00% (94.61 At.%) Au: 15.00% (5.39 At.%)		Cu: 80.00% (82.54 At.%) Au: 20.00% (7.46 At.%)		Cu: 75.00% (90.29 At.%) Au: 25.00% (9.71 At.%)		Cu: 70.00% (87.85 At.%) Au: 30.00% (12.15 At.%)		Cu: 65.00% (85.20 At.%) Au: 35.00% (14.80 At.%)		Cu: 60.00% (82.30 At.%) Au: 40.00% (17.70 At.%)	
T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
1	2.50 <sup>†</sup>	1	3.52 <sup>†</sup>	1	4.45 <sup>†</sup>	1	5.47 <sup>†</sup>	1	6.52 <sup>†</sup>	1	7.52 <sup>†</sup>
4	2.52 <sup>†</sup>	4	3.52 <sup>†</sup>	4	4.45 <sup>†</sup>	4	5.47 <sup>†</sup>	4	6.52 <sup>†</sup>	4	7.52 <sup>†</sup>
7	2.53 <sup>†</sup>	7	3.52 <sup>†</sup>	7	4.45 <sup>†</sup>	7	5.47 <sup>†</sup>	7	6.52 <sup>†</sup>	7	7.52 <sup>†</sup>
10	2.53 <sup>†</sup>	10	3.52 <sup>†</sup>	10	4.45 <sup>†</sup>	10	5.47 <sup>†</sup>	10	6.52 <sup>†</sup>	10	7.52 <sup>†</sup>
15	2.53 <sup>†</sup>	15	3.52 <sup>†</sup>	15	4.45 <sup>†</sup>	15	5.47 <sup>†</sup>	15	6.52 <sup>†</sup>	15	7.52 <sup>†</sup>
20	2.53 <sup>†</sup>	20	3.52 <sup>†</sup>	20	4.45 <sup>†</sup>	20	5.47 <sup>†</sup>	20	6.52 <sup>†</sup>	20	7.52 <sup>†</sup>
25	2.60 <sup>†</sup>	25	3.53 <sup>†</sup>	25	4.46 <sup>†</sup>	25	5.48 <sup>†</sup>	25	6.54 <sup>†</sup>	25	7.54 <sup>†</sup>
30	2.61 <sup>†</sup>	30	3.54 <sup>†</sup>	30	4.47 <sup>†</sup>	30	5.51 <sup>†</sup>	30	6.57 <sup>†</sup>	30	7.59 <sup>†</sup>
40	2.63 <sup>†</sup>	40	3.57 <sup>†</sup>	40	4.49 <sup>†</sup>	40	5.54 <sup>†</sup>	40	6.61 <sup>†</sup>	40	7.66 <sup>†</sup>
50	2.68 <sup>†</sup>	50	3.61 <sup>†</sup>	50	4.53 <sup>†</sup>	50	5.57 <sup>†</sup>	50	6.67 <sup>†</sup>	50	7.72 <sup>†</sup>
60	2.73 <sup>†</sup>	60	3.67 <sup>†</sup>	60	4.58 <sup>†</sup>	60	5.63 <sup>†</sup>	60	6.74 <sup>†</sup>	60	7.78 <sup>†</sup>
70	2.80 <sup>†</sup>	70	3.74 <sup>†</sup>	70	4.64 <sup>†</sup>	70	5.70 <sup>†</sup>	70	6.80 <sup>†</sup>	70	7.85 <sup>†</sup>
80	2.87 <sup>†</sup>	80	3.80 <sup>†</sup>	80	4.72 <sup>†</sup>	80	5.77 <sup>†</sup>	80	6.87 <sup>†</sup>	80	7.92 <sup>†</sup>
90	2.94 <sup>†</sup>	90	3.86 <sup>†</sup>	90	4.79 <sup>†</sup>	90	5.84 <sup>†</sup>	90	6.93 <sup>†</sup>	90	7.95 <sup>†</sup>
100	3.01 <sup>†</sup>	100	3.93 <sup>†</sup>	100	4.86 <sup>†</sup>	100	5.91 <sup>†</sup>	100	7.00 <sup>†</sup>	100	8.04 <sup>†</sup>
150	3.36 <sup>†</sup>	150	4.30	150	5.22	150	6.26	150	7.33 <sup>†</sup>	150	8.37
200	3.70 <sup>†</sup>	200	4.65	200	5.59	200	6.62	200	7.66 <sup>†</sup>	200	8.70
250	4.04 <sup>†</sup>	250	4.99	250	5.93	250	6.96	250	8.00 <sup>†</sup>	250	9.03
273	4.20	273	5.15	273	6.09	273	7.12	273	8.16	273	9.18
293	4.33	293	5.28	293	6.23	293	7.25	293	8.29	293	9.31
300	4.38 <sup>†</sup>	300	5.32	300	6.28	300	7.30	300	8.34 <sup>†</sup>	300	9.36
350	4.71 <sup>†</sup>	350	5.65	350	6.63	350	7.64	350	8.69 <sup>†</sup>	350	9.70
400	5.05 <sup>†</sup>	400	5.99	400	6.98	400	7.99	400	9.04 <sup>†</sup>	400	10.05
500	5.72 <sup>†</sup>	500	6.66 <sup>†</sup>	500	7.66 <sup>†</sup>	500	8.67 <sup>†</sup>	500	9.72 <sup>†</sup>	500	10.75 <sup>†</sup>
600	6.42 <sup>†</sup>	600	7.37 <sup>†</sup>	600	8.37 <sup>†</sup>	600	9.37 <sup>†</sup>	600	10.43 <sup>†</sup>	600	11.47 <sup>†</sup>
700	7.13 <sup>†</sup>	700	8.08 <sup>†</sup>	700	9.08 <sup>†</sup>	700	10.08 <sup>†</sup>	700	11.14 <sup>†</sup>	700	12.19 <sup>†</sup>
800	7.87 <sup>†</sup>	800	8.82 <sup>†</sup>	800	9.81 <sup>†</sup>	800	10.81 <sup>†</sup>	800	11.87 <sup>†</sup>	800	12.93 <sup>†</sup>
900	8.62 <sup>†</sup>	900	9.55 <sup>†</sup>	900	10.56 <sup>†</sup>	900	11.55 <sup>†</sup>	900	12.61 <sup>†</sup>	900	13.69 <sup>†</sup>
1000	9.39 <sup>†</sup>	1000	10.30 <sup>†</sup>	1000	11.32 <sup>†</sup>	1000	12.30 <sup>†</sup>	1000	13.37 <sup>†</sup>	1000	14.49 <sup>†</sup>
1100	10.18 <sup>†</sup>	1100	11.07 <sup>†</sup>	1100	12.09 <sup>†</sup>	1100	13.08 <sup>†</sup>	1100	14.15 <sup>†</sup>	1100	15.31 <sup>†</sup>
1200	11.00 <sup>†</sup>	1200	11.86 <sup>†</sup>	1200	12.88 <sup>†</sup>	1200	13.87 <sup>†</sup>	1200	14.96 <sup>†</sup>	1200	16.15 <sup>†</sup>
1303	11.84 <sup>†</sup>	1289	12.56 <sup>†</sup>	1277	13.49 <sup>†</sup>	1265	14.38 <sup>†</sup>	1255	15.41 <sup>†</sup>	1245	16.53 <sup>†</sup>

† Uncertainties in the electrical resistivity values are as follows:

85.00 Cu - 15.00 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 80.00 Cu - 20.00 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 75.00 Cu - 25.00 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 70.00 Cu - 30.00 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 65.00 Cu - 35.00 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 60.00 Cu - 40.00 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.

† Provisional value.

\* In temperature range where no experimental data are available.

TABLE 16. RECOMMENDED ELECTRICAL RESISTIVITY OF COPPER-GOLD ALLOY SYSTEM<sup>†</sup> (continued)  
[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Cu: 55.00% (79.12 At. %) Au: 45.00% (20.88 At. %)			Cu: 50.00% (75.61 At. %) Au: 50.00% (24.39 At. %)			Cu: 45.00% (71.72 At. %) Au: 55.00% (28.28 At. %)			Cu: 40.00% (67.39 At. %) Au: 60.00% (32.61 At. %)			Cu: 35.00% (63.54 At. %) Au: 65.00% (37.46 At. %)			Cu: 30.00% (57.05 At. %) Au: 70.00% (42.95 At. %)		
T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$	
1	8.48*		1	9.34*		1	10.14*		1	10.88*		1	11.43*		1	11.82*	
4	8.48*		4	9.34*		4	10.14*		4	10.88*		4	11.43*		4	11.82*	
7	8.48*		7	9.34*		7	10.14*		7	10.88*		7	11.43*		7	11.82*	
10	8.48*		10	9.34*		10	10.14*		10	10.88*		10	11.43*		10	11.82*	
15	8.48*		15	9.35*		15	10.15*		15	10.88*		15	11.43*		15	11.83*	
20	8.49*		20	9.37*		20	10.17*		20	10.89*		20	11.44*		20	11.84*	
25	8.51*		25	9.39*		25	10.19*		25	10.90*		25	11.46*		25	11.86*	
30	8.54*		30	9.42*		30	10.22*		30	10.93*		30	11.50*		30	11.90*	
40	8.61*		40	9.48*		40	10.28*		40	11.00*		40	11.57*		40	11.97*	
50	8.67*		50	9.55*		50	10.36*		50	11.08*		50	11.65*		50	12.04*	
60	8.74*		60	9.62*		60	10.42*		60	11.15*		60	11.73*		60	12.12*	
70	8.80*		70	9.68*		70	10.49*		70	11.22*		70	11.80*		70	12.20*	
80	8.87*		80	9.74*		80	10.56*		80	11.30*		80	11.87*		80	12.27*	
90	8.94*		90	9.81*		90	10.63*		90	11.37*		90	11.95*		90	12.35*	
100	9.00*		100	9.88*		100	10.70*		100	11.44*		100	12.02*		100	12.43*	
150	9.33		150	10.22		150	11.05*		150	11.90*		150	12.40*		150	12.81*	
200	9.66		200	10.57		200	11.41*		200	12.18*		200	12.78*		200	13.20*	
250	10.00		250	10.91		250	11.76*		250	12.54*		250	13.16*		250	13.59*	
273	10.16		273	11.07		273	11.92		273	12.70		273	13.33		273	13.77	
293	10.30		293	11.20		293	12.06		293	12.85		293	13.48		293	13.93	
300	10.35		300	11.25		300	12.11*		300	12.90*		300	13.54*		300	13.99*	
350	10.68		350	11.60		350	12.46*		350	13.27*		350	13.89*		350	14.36*	
400	11.03		400	11.94		400	12.81		400	13.65		400	14.31*		400	14.78*	
500	11.72*		500	12.65*		500	13.54*		500	14.40*		500	15.09*		500	15.58*	
600	12.43*		600	13.38*		600	14.30*		600	15.17*		600	15.86*		600	16.40*	
700	13.16*		700	14.13*		700	15.07*		700	15.96*		700	16.69*		700	17.24*	
800	13.88*		800	14.91*		800	15.87*		800	16.76*		800	17.53*		800	18.11*	
900	14.71*		900	15.71*		900	16.71*		900	17.61*		900	18.43*		900	19.02*	
1000	15.59*		1000	16.59*		1000	17.60*		1000	18.52*		1000	19.39*		1000	20.00*	
1100	16.38*		1100	17.37*		1100	18.53*		1100	19.47*		1100	20.33*		1100	20.96*	
1200	17.19*		1200	18.23*		1200	19.47*		1200	20.48*		1200	21.30*		1200	21.89*	
1236	17.49*		1236	18.45*		1236	19.63*		1236	20.61*							

<sup>†</sup> Uncertainties in the electrical resistivity values are as follows:

- 55.00 Cu - 45.00 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.
- 50.00 Cu - 50.00 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.
- 45.00 Cu - 55.00 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.
- 40.00 Cu - 60.00 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.
- 35.00 Cu - 65.00 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.
- 30.00 Cu - 70.00 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.

\* Provisional value.

† In temperature range where no experimental data are available.

TABLE 15. RECOMMENDED ELECTRICAL RESISTIVITY OF COPPER-GOLD ALLOY SYSTEM† (continued)

[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8}$   $\Omega$  m]

Cu: 25.00% (50.82 At. %) Au: 75.00% (49.18 At. %)		Cu: 20.00% (42.66 At. %) Au: 80.00% (57.34 At. %)		Cu: 15.00% (35.36 At. %) Au: 85.00% (64.64 At. %)		Cu: 10.00% (25.62 At. %) Au: 90.00% (74.38 At. %)		Cu: 5.00% (14.03 At. %) Au: 95.00% (85.97 At. %)		Cu: 3.00% (8.75 At. %) Au: 97.00% (91.25 At. %)	
T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
1	11.95*	1	11.72*	1	10.75*	1	8.72*	1	5.27*	1	3.44*
4	11.85*	4	11.72*	4	10.75*	4	8.72*	4	5.27*	4	3.44*
7	11.85*	7	11.72*	7	10.75*	7	8.72*	7	5.27*	7	3.44*
10	11.96*	10	11.72*	10	10.75*	10	8.72*	10	5.28*	10	3.44*
15	11.97*	15	11.73*	15	10.76*	15	8.72*	15	5.29*	15	3.45*
20	11.96*	20	11.75*	20	10.77*	20	8.73*	20	5.31*	20	3.45*
25	12.00*	25	11.77*	25	10.78*	25	8.75*	25	5.33*	25	3.45*
30	12.04*	30	11.80*	30	10.83*	30	8.78*	30	5.36*	30	3.52*
40	12.12*	40	11.87*	40	10.91*	40	8.86*	40	5.44*	40	3.60*
50	12.20*	50	11.95*	50	10.98*	50	8.94*	50	5.52*	50	3.68*
60	12.27*	60	12.03*	60	11.07*	60	9.02*	60	5.60*	60	3.76*
70	12.35*	70	12.10*	70	11.15*	70	9.10*	70	5.68*	70	3.84*
80	12.43*	80	12.18*	80	11.23*	80	9.17*	80	5.75*	80	3.92*
90	12.51*	90	12.26*	90	11.30*	90	9.25*	90	5.83*	90	4.00*
100	12.59*	100	12.35*	100	11.38*	100	9.33*	100	5.91*	100	4.08*
150	12.98	150	12.74*	150	11.78*	150	9.73	150	6.30*	150	4.48*
200	13.36	200	13.12*	200	12.18*	200	10.12	200	6.68*	200	4.86*
250	13.75	250	13.52*	250	12.57*	250	10.52	250	7.07*	250	5.23*
273	13.83	273	13.70*	273	12.75	273	10.70	273	7.25	273	5.47
293	14.00	293	13.86*	293	12.91	293	10.86	293	7.41*	293	5.64
300	14.14	300	13.92*	300	12.98*	300	10.91	300	7.46*	300	5.69*
350	14.54	350	14.32*	350	13.36*	350	11.31	350	7.87	350	6.10
400	14.94	400	14.73*	400	13.77	400	11.72	400	8.28	400	6.49
500	15.78	500	15.57*	500	14.60*	500	12.58*	500	9.14*	500	7.33*
600	16.63	600	16.41*	600	15.44*	600	13.47*	600	10.02*	600	8.13*
700	17.48*	700	17.26*	700	16.29*	700	14.38*	700	10.92*	700	9.10*
800	18.38*	800	18.14*	800	17.16*	800	15.32*	800	11.86*	800	10.05*
900	19.31*	900	19.09*	900	18.15*	900	16.30*	900	12.84*	900	11.03*
1000	20.32*	1000	20.13*	1000	19.23*	1000	17.36*	1000	13.90*	1000	12.13*
1100	21.34*	1100	21.14*	1100	20.30*	1100	18.47*	1100	15.01*	1100	13.30*
1184	22.28*	1182	22.09*	1185	21.30*	1199	19.61*	1200	16.14*	1200	14.56*
								1241	16.60*	1270	15.44*

† Uncertainties in the electrical resistivity values are as follows:

- 25.00 Cu - 75.00 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 20.00 Cu - 80.00 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 15.00 Cu - 85.00 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 10.00 Cu - 90.00 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 5.00 Cu - 95.00 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 3.00 Cu - 97.00 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.

\* Provisional values.

\* In temperature range where no experimental data are available.

TABLE 15. RECOMMENDED ELECTRICAL RESISTIVITY OF COPPER-GOLD ALLOY SYSTEM† (continued)  
[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Cu: 1.00% ( 3.04 At. %) Au: 99.00% (96.96 At. %)		Cu: 0.50% ( 1.53 At. %) Au: 99.50% (98.47 At. %)		Cu: 0.00% ( 0.00 At. %) Au: 100.00% (100.00 At. %)	
T	$\rho$	T	$\rho$	T	$\rho$
1	1.40 <sup>‡</sup>	1	0.770 <sup>‡</sup>	1	0.0220
4	1.40 <sup>‡</sup>	4	0.770 <sup>‡</sup>	4	0.0220
7	1.40 <sup>‡</sup>	7	0.770 <sup>‡</sup>	7	0.0221
10	1.40 <sup>‡</sup>	10	0.770 <sup>‡</sup>	10	0.0226
15	1.40 <sup>‡</sup>	15	0.770 <sup>‡</sup>	15	0.0258
20	1.41 <sup>‡</sup>	20	0.780 <sup>‡</sup>	20	0.0346 <sup>‡</sup>
25	1.43 <sup>‡</sup>	25	0.801 <sup>‡</sup>	25	0.0503 <sup>‡</sup>
30	1.45 <sup>‡</sup>	30	0.826 <sup>‡</sup>	30	0.0727 <sup>‡</sup>
40	1.51 <sup>‡</sup>	40	0.908 <sup>‡</sup>	40	0.141 <sup>‡</sup>
50	1.60 <sup>‡</sup>	50	0.968 <sup>‡</sup>	50	0.232
60	1.68 <sup>‡</sup>	60	1.04 <sup>‡</sup>	60	0.309
70	1.76 <sup>‡</sup>	70	1.13 <sup>‡</sup>	70	0.396
80	1.84 <sup>‡</sup>	80	1.21 <sup>‡</sup>	80	0.482
90	1.92 <sup>‡</sup>	90	1.29 <sup>‡</sup>	90	0.567
100	2.00 <sup>‡</sup>	100	1.37 <sup>‡</sup>	100	0.652
150	2.41	150	1.76 <sup>*</sup>	150	1.063
200	2.81	200	2.18 <sup>*</sup>	200	1.464
250	3.22	250	2.59 <sup>*</sup>	250	1.865
273	3.40	273	2.78	273	2.052
293	3.57	293	2.94 <sup>*</sup>	293	2.214
300	3.63	300	3.00 <sup>*</sup>	300	2.271
350	4.03	350	3.41 <sup>*</sup>	350	2.683
400	4.45	400	3.82 <sup>*</sup>	400	3.102
500	5.36 <sup>‡</sup>	500	4.68 <sup>‡</sup>	500	3.962
600	6.16 <sup>‡</sup>	600	5.63 <sup>‡</sup>	600	4.853
700	7.10 <sup>‡</sup>	700	6.81 <sup>‡</sup>	700	5.780
800	8.07 <sup>‡</sup>	800	7.45 <sup>‡</sup>	800	6.755
900	9.10 <sup>‡</sup>	900	8.47 <sup>‡</sup>	900	7.787
1000	10.23 <sup>‡</sup>	1000	9.57 <sup>‡</sup>	1000	8.884
1100	11.46 <sup>‡</sup>	1100	10.77 <sup>‡</sup>	1100	10.057
1200	12.77 <sup>‡</sup>	1200	12.10 <sup>‡</sup>	1200	11.312
1296	14.07 <sup>‡</sup>	1300	13.47 <sup>‡</sup>	1300	12.632
		1323	13.73 <sup>‡</sup>	1323.56	13.146(a)
				1336	31.06(c)
				1700	36.26

† Uncertainties in the electrical resistivity values are as follows:

1.00 Cu - 99.00 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.

0.50 Cu - 99.50 Au:  $\pm 7\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.

0.00 Cu - 100.00 Au:  $\pm 1\%$  up to 10 K,  $\pm 2.5\%$  above 10 K to 15 K,  $\pm 6\%$  above 15 K to 40 K,  $\pm 7\%$  above 40 K to 60 K,  $\pm 1\%$  above 60 K to 800 K, and  $\pm 2.5\%$  above 800 K.

<sup>‡</sup> Provisional value.

<sup>\*</sup> In temperature range where no experimental data are available.

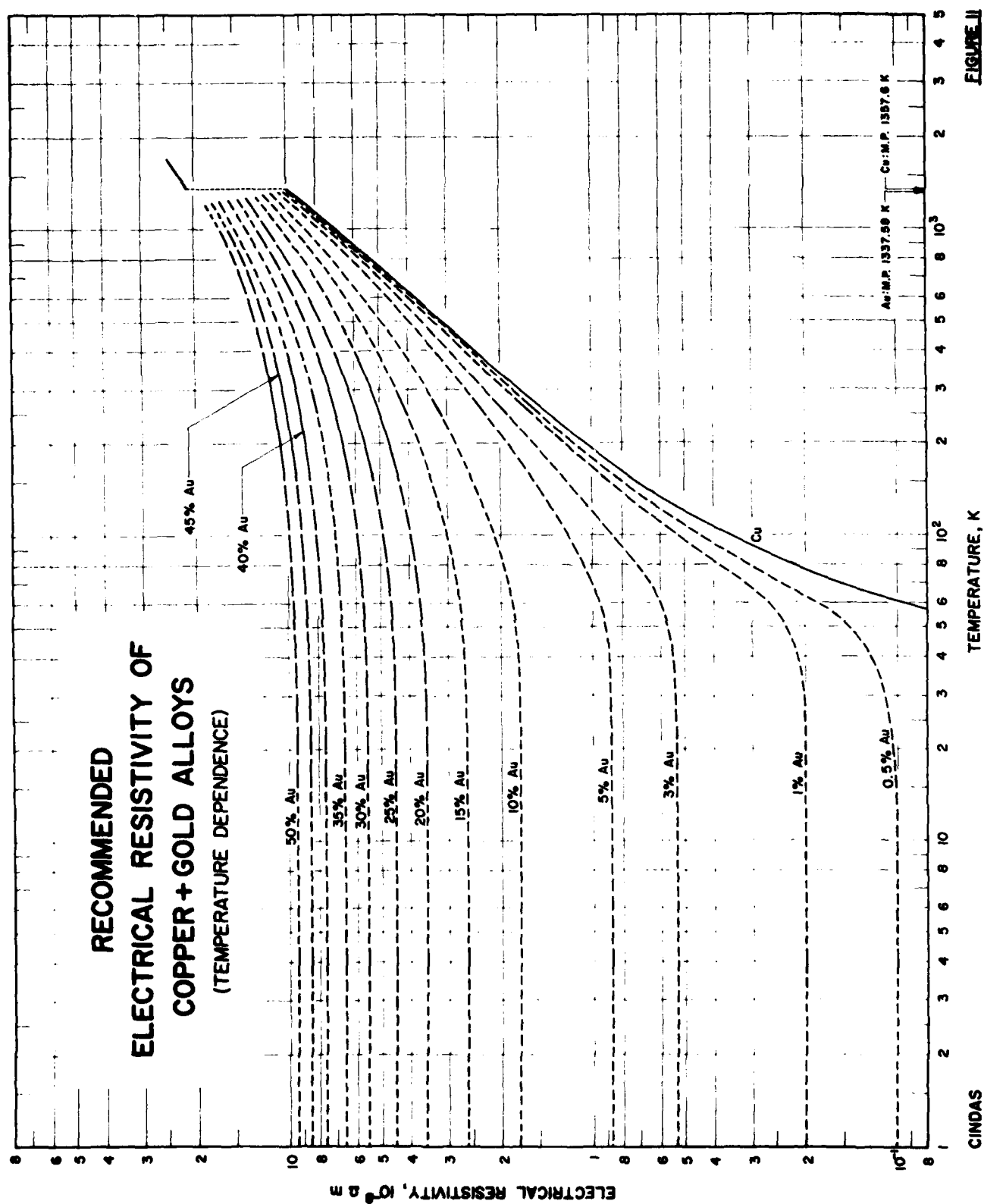
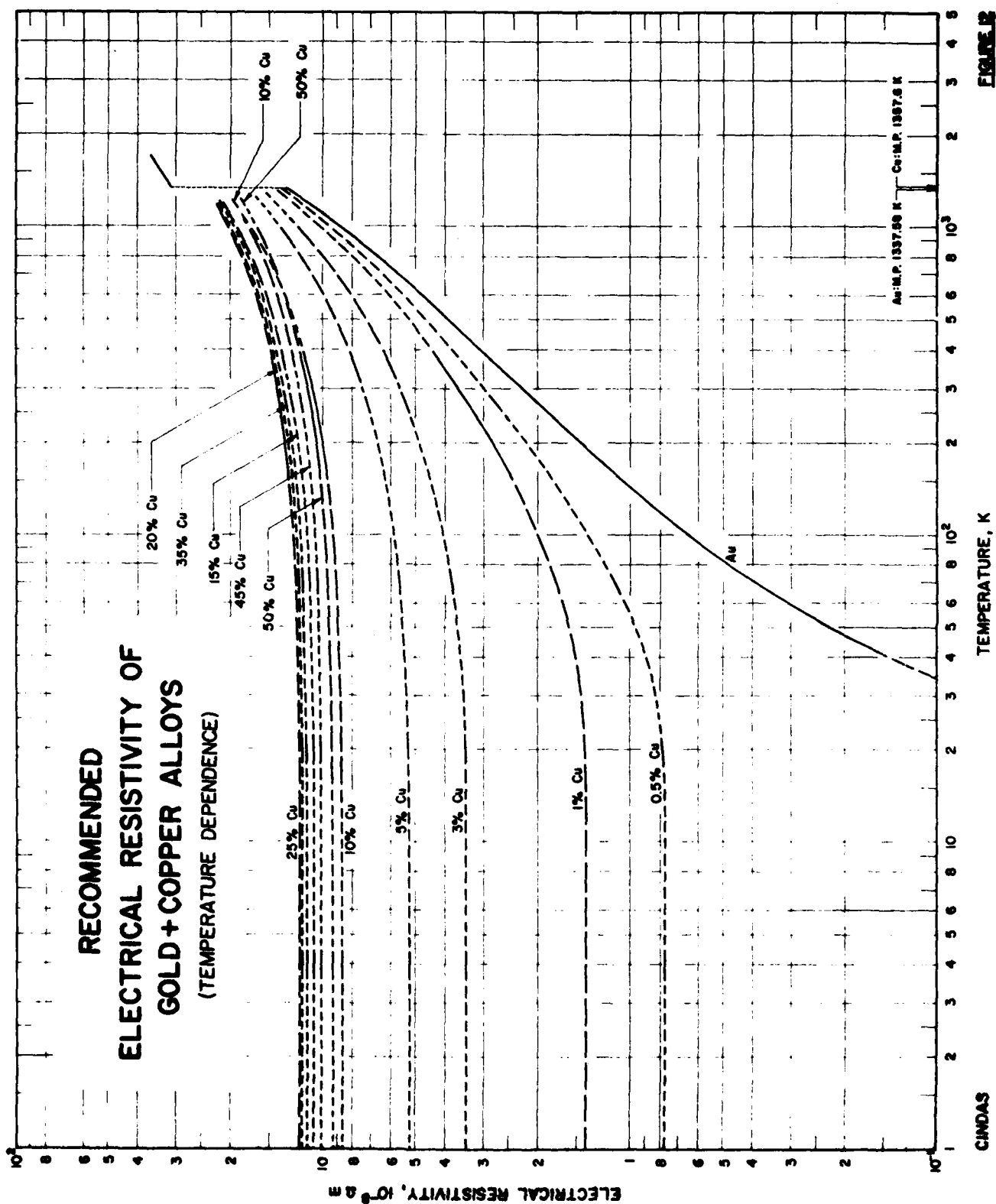
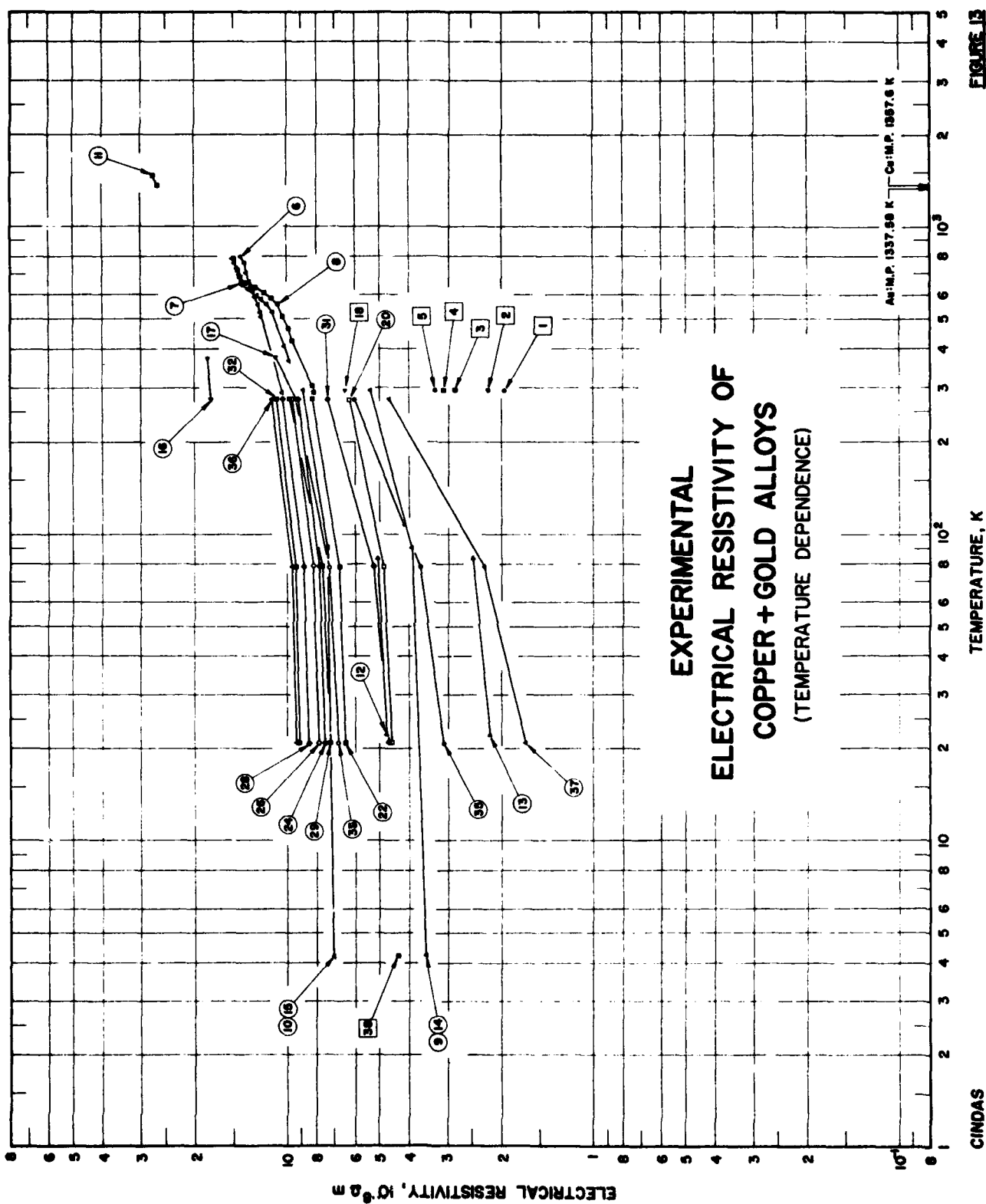


FIGURE II







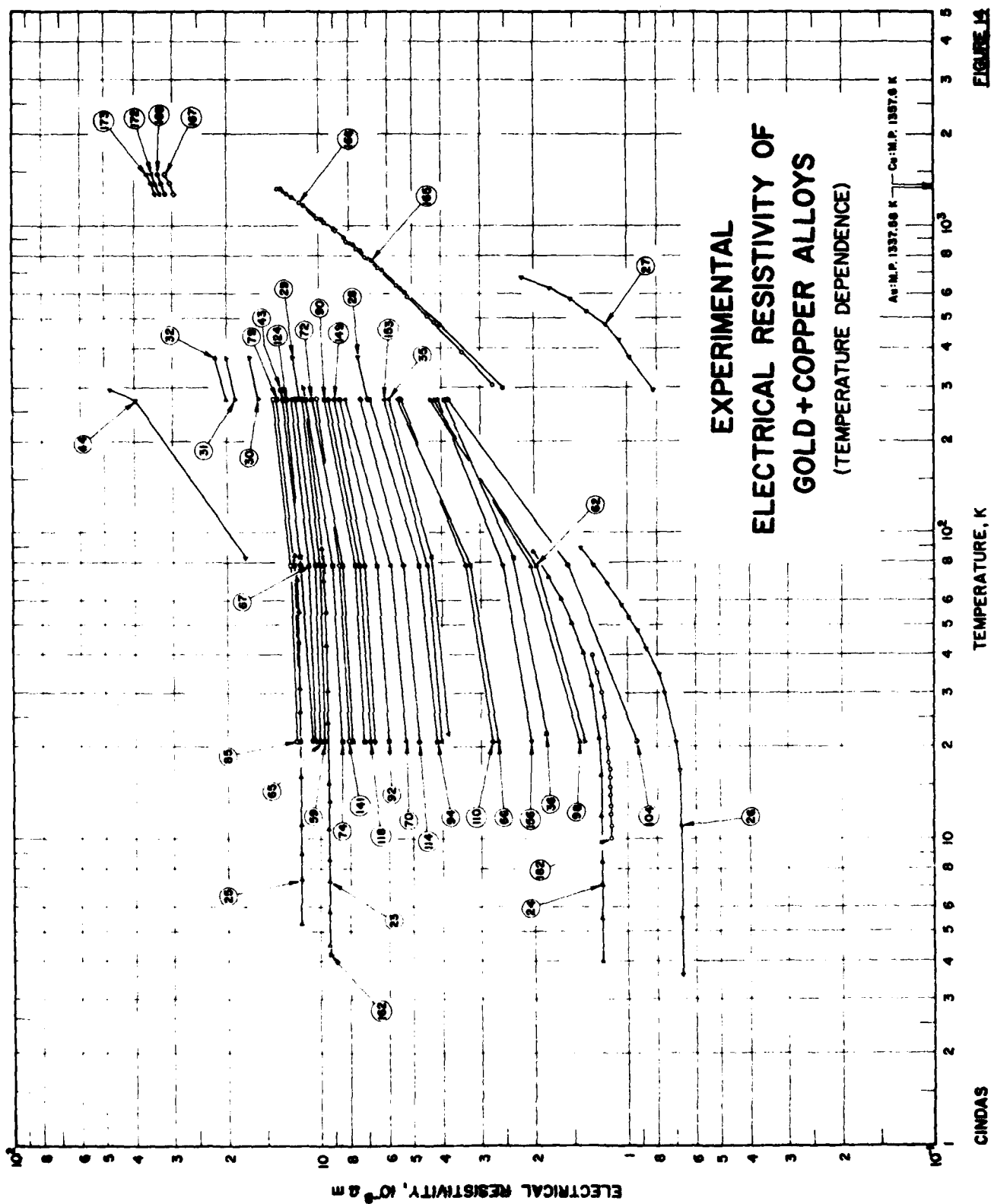


TABLE 16. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + GOLD ALLOYS (Temperature Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu	Composition (continued), Specifications, and Remarks
1	124	Linde, J.O.	1932		291.2		1.53	Calculated composition (0.500 a/o Au).
2	126	Linde, J.O.	1932		291.2		3.00	Calculated composition (0.988 a/o Au).
3	126	Linde, J.O.	1932		291.2		5.92	Calculated composition (1.99 a/o Au).
4	126	Linde, J.O.	1932		291.2		7.05	Calculated composition (2.39 a/o Au).
5	126	Linde, J.O.	1932		291.2		8.75	Calculated composition (3.00 a/o Au).
6	123	Borelius, G., Johansson, C.H., and Linde, J.O.	1928	V	286-795		43.97	Calculated composition (20.2 a/o Au); 1 x 1 x 20 mm; initial materials, melted in quartz tube, heat-treated near melting temperature and cooled in a way to keep rolling condition, rolled to 1 x 1 mm square wire, heat-treated again near melting temperature; measurements made during both heating and cooling.
7	123	Borelius, G., et al.	1928	V	363-787		47.51	Calculated composition (22.6 a/o Au); similar to the above specimen; measurements made during heating.
8	123	Borelius, G., et al.	1928	V	286-787		47.51	The above specimen measured during cooling.
9	125	Tainsh, R.J. and White, G.K.	1962	A	4.2-293		20.1	Calculated composition (7.5 a/o Au); rod specimen 3 to 5 mm in diameter and 6 cm long; annealed at 750 C.
10	125	Tainsh, R.J. and White, G.K.	1962	A	4.2-293		38.0	Calculated composition (16.5 a/o Au); similar to the above specimen.
11	127	Roll, A. and Motz, H.	1957	R	1373, 1473		30.7	Calculated composition (12.5 a/o Au); prepared from electrolytic copper and 99.95 pure gold; $d\rho/dT = 0.0106 \mu\Omega \text{ cm K}^{-1}$ .
12	128	Gruncisen, E. and Reddemann, H.	1934		22, 83		75.2	Calculated composition; polycrystalline; form factor $1.53 \times 10^3$ .
13	128	Gruncisen, E. and Reddemann, H.	1934		22, 83		87.4	Calculated composition; polycrystalline; form factor $2.61 \times 10^3$ .
14	129	Kemp, W.R.G., Klemens, P.G., and Tainsh, R.J.	1957	A	4.2-293		20.1	Calculated composition (7.5 a/o Au); specimen 8 cm long and 0.5 cm in diameter; annealed at 750 C for 1 hr.
15	129	Kemp, W.R.G., et al.	1957	A	4.2-293		38.0	Calculated composition (16.5 a/o Au); similar to the above specimen.
16	130	Sedstrom, E.	1919		273, 373		55.3	Calculated composition (79.3 a/o Cu); rolled and drawn to wire 1 mm diameter; heated to near melting point for 1/2 hr.
17	130	Sedstrom, E.	1919		273, 373		73.5	Calculated composition (89.6 a/o Cu); similar to the above specimen.
18	131	Röhl, H.	1933		291.2	I	25.05	5 mm diameter x 10 cm long; heat-treated at 1073 K for 4 hr, then quenched.
19*	131	Röhl, H.	1933		291.2	I		Similar to the above specimen but heat-treated at 573 to 673 K for 20 hr and slowly cooled.
20	132	Peppisli, V.	1933	V	21-273	1	25.6	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density $10.60 \text{ g cm}^{-3}$ .
21*	132	Peppisli, V.	1933	V	21-273	1		The above specimen annealed at 300 to 400 C for one week.
22	132	Peppisli, V.	1933	V	21-273	2	35.4	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density $10.84 \text{ g cm}^{-3}$ .

\* Not shown in figure.

TABLE 16. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + GOLD ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Au	Composition (continued), Specifications, and Remarks
23*	132	Pospisil, V.	1933	V	21-273	2	35.4	The above specimen annealed at 300 to 400 C for one week.
24	132	Pospisil, V.	1933	V	21-273	3	39.7	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 11.18 g cm <sup>-3</sup> .
25*	132	Pospisil, V.	1933	V	21-273	3	39.7	The above specimen annealed at 300 to 400 C for one week.
26	132	Pospisil, V.	1933	V	21-273	4	42.0	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 11.41 g cm <sup>-3</sup> .
27*	132	Pospisil, V.	1933	V	21-273	4	42.0	The above specimen annealed at 300 to 400 C for one week.
28	132	Pospisil, V.	1933	V	21-273	5	43.7	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 11.50 g cm <sup>-3</sup> .
29	132	Pospisil, V.	1933	V	21-273	5	43.7	The above specimen annealed at 300 to 400 C for one week.
30*	132	Pospisil, V.	1933	V	21-273	5a	45.3	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 11.69 g cm <sup>-3</sup> .
31	132	Pospisil, V.	1933	V	21-273	5a	45.3	The above specimen annealed at 300 to 400 C for one week.
32	132	Pospisil, V.	1933	V	21-273	6	46.0	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 11.71 g cm <sup>-3</sup> .
33	132	Pospisil, V.	1933	V	21-273	6	46.0	The above specimen annealed at 300 to 400 C for one week.
34*	132	Pospisil, V.	1933	V	21-273	7	48.0	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 11.81 g cm <sup>-3</sup> .
35	132	Pospisil, V.	1933	V	21-273	7	48.0	The above specimen annealed at 300 to 400 C for one week.
36	132	Pospisil, V.	1933	V	21-273	7	49.6	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 12.05 g cm <sup>-3</sup> .
37	132	Pospisil, V.	1933	V	21-273	7	49.6	The above specimen annealed at 300 to 400 C for one week.
38	90	Leaver, A.D.W. and Charley, P.	1971	A	4.2	10 Au	25.4	Polycrystalline; obtained from International Research and Development Co.; annealed.
39*	90	Leaver, A.D.W. and Charley, P.	1971	A	4.2	10 Au	25.4	The above specimen tensile strained 13.4% under a stress of 36.66 kg mm <sup>-2</sup> .

\* Not shown in figure.

TABLE 17. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF COPPER + GOLD ALLOYS (Temperature Dependence)

[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $\cdot 10^{-9} \Omega \cdot m$ ]

[illegible]

\* Not shown in figure.

TABLE 16. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD + COPPER ALLOYS (Temperature Dependence)

Data Set No.	Ref.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Au Cu	Composition (continued), Specifications, and Remarks
1*	123	Borelius, G., et al.	1928	V	289-723		49.18	Calculated composition (75 a/o Cu); 1 x 1 x 20 mm; initial materials melted in quartz tube, heat-treated near melting temperature, cooled in a way to keep rolling condition, rolled to 1 x 1 mm square wire; heat-treated again near melting temperature; measured during heating.
2*	123	Borelius, G., et al.	1928	V	283-723		49.18	The above specimen measured during cooling.
3*	123	Borelius, G., et al.	1928	V	277-656		49.18	The above specimen quenched; measured during heating.
4*	123	Borelius, G., et al.	1928	V	381-732		45.71	Calculated composition (72.3 a/o Cu); same dimensions, fabrication method, and heat-treatment as the above specimen for curve no. 1; measured during heating.
5*	123	Borelius, G., et al.	1928	V	282-732		45.71	The above specimen measured during cooling.
6*	123	Borelius, G., et al.	1928	V	287-700		42.83	Calculated composition (69.9 a/o Cu); similar to the above specimen; measured during heating.
7*	123	Borelius, G., et al.	1928	V	283-700		42.83	The above specimen measured during cooling.
8*	123	Borelius, G., et al.	1928	V	283-707		28.28	Calculated composition (55 a/o Cu); similar to the above specimen; measured during heating.
9*	123	Borelius, G., et al.	1928	V	286-707		28.28	The above specimen measured during cooling.
10*	123	Borelius, G., et al.	1928	V	18-327		24.39	Calculated composition (50 a/o Cu); same dimensions, fabrication method, and heat-treatment as the above specimen; quenched; measured during heating.
11*	123	Borelius, G., et al.	1928	V	15-327		24.39	The above specimen measured during cooling.
12*	123	Borelius, G., et al.	1928	V	15-419		24.39	The above specimen remeasured during heating.
13*	123	Borelius, G., et al.	1928	V	17-419		24.39	The above specimen again measured during cooling.
14*	123	Borelius, G., et al.	1928	V	17-427		24.39	The above specimen reheated for measurement.
15*	123	Borelius, G., et al.	1928	V	404-427		24.39	The above specimen remeasured.
16*	123	Borelius, G., et al.	1928	V	448-467		24.39	The above specimen remeasured during heating.
17*	123	Borelius, G., et al.	1928	V	10-448		24.39	The above specimen measured during cooling.
18*	123	Borelius, G., et al.	1928	V	10-379		24.39	The above specimen reheated.
19*	123	Borelius, G., et al.	1928	V	100-379		24.39	The above specimen measured during cooling.
20*	123	Borelius, G., et al.	1928	V	293		24.39	The above specimen repeatedly annealed.
21*	123	Borelius, G., et al.	1928	V	290-723		20.96	Calculated composition (45 at. % Cu); similar to the specimen for Data Set No. 1; measured during heating.
22*	123	Borelius, G., et al.	1928	V	290-723		20.96	The above specimen measured during cooling.

\* Not shown in figure.

TABLE 18. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD + COPPER ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Au	Composition Cu	Composition (continued), Specifications, and Remarks
23	124	Passaglia, E. and Love, W.F.	1955	A	4.5-89			49.18	Calculated composition (75 a/o Cu); prepared from 99.999 pure copper obtained from Johnson-Matthey and 99.99 pure gold obtained from J. Blahop and Co. by melting in sealed Y-poor tubes under high vacuum, shaken vigorously to provide a homogeneous melt; quenched; electrical resistivity values converted from the reported $\gamma$ , the ratio of resistance and ice point resistance, by the equation, $\rho(T) = \gamma/\rho_i$ , using the ice point resistivity $\rho_i$ value extrapolated from resistivity value at 20 C of Johansson and Linde, ( $\rho_i = 11.201 \mu\Omega \text{ cm}$ ).
24	124	Passaglia, E. and Love, W.F.	1955	A	4.0-86			49.18	The above specimen annealed at 850 C for 48 hr ( $\rho_i = 4.49 \mu\Omega \text{ cm}$ ).
25	124	Passaglia, E. and Love, W.F.	1955	A	5.3-84			24.39	Calculated composition (50 a/o Cu); similar to the above specimen for Data Set No. 23 ( $\rho_i = 13.915 \mu\Omega \text{ cm}$ ).
26	124	Passaglia, E. and Love, W.F.	1955	A	3.6-89			24.39	The above specimen annealed at 850 C for 48 hr ( $\rho_i = 3.63 \mu\Omega \text{ cm}$ ).
27	133	Jones, F.W. and Sykes, C.	1938		293-673			49.2	Calculated composition (75 a/o Cu); 0.4 cm diameter x 12.5 cm long; obtained from Johnson-Matthey; cold-rolled to 0.1 cm thick, annealed at 1223 K for 4 hr, cold-drawn to 0.04 cm in diameter, and reannealed at 823 K for 30 min in vacuum; smoothed values reported.
28	130	Sedström, E.	1919		273, 373		96.73	3.27	Calculated composition (9.5 a/o Cu); rolled and drawn to 1 mm diameter wire; annealed close to melting point for 1/2 hr.
29	130	Sedström, E.	1919		273, 373		92.55	7.45	Calculated composition (20.0 a/o Cu); similar to the above specimen.
30	130	Sedström, E.	1919		273, 373		87.77	12.23	Calculated composition (30.2 a/o Cu); similar to the above specimen.
31	130	Sedström, E.	1919		273, 373			19.2	Calculated composition (42.4 a/o Cu); similar to the above specimen.
32	130	Sedström, E.	1919		273, 373		59.25	40.75	Calculated composition (58.1 a/o Cu); similar to the above specimen.
33*	128	Gruneisen, E. and Reddemann, H.	1934		83		89.6	10.4	Calculated composition; polycrystalline; cast.
34*	128	Gruneisen, E. and Reddemann, H.	1934		273		89.6	10.4	The above specimen annealed at 365 C for 40 hr.
35	128	Gruneisen, E. and Reddemann, H.	1934		22-273		96.9	3.10	Calculated composition; polycrystalline; cast.
36	128	Gruneisen, E. and Reddemann, H.	1934		22-273		98.43	1.57	Similar to the above specimen.
37*	128	Gruneisen, E. and Reddemann, H.	1934		83		50.1	49.9	Calculated composition; polycrystalline; cast; quenched from 800 C.
38*	128	Gruneisen, E. and Reddemann, H.	1934		83, 273		50.1	49.9	The above specimen annealed at ~400 C for 20 hr.
39*	128	Gruneisen, E. and Reddemann, H.	1934		83, 273		50.1	49.9	The above specimen annealed at ~360 C for 33 hr.
40*	128	Gruneisen, E. and Reddemann, H.	1934		273		50.1	49.9	The above specimen annealed at ~850 C for 3 hr and then quenched.

\* Not shown in figure.



TABLE 18. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD + COPPER ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Au Cu	Composition (continued), Specifications, and Remarks
41*	128	Gruneisen, E. and Reddemann, H.	1934		83, 273		50.1 49.9	The above specimen measured after 5 months.
42*	128	Gruneisen, E. and Reddemann, H.	1934		22, 83		50.1 49.9	The above specimen annealed at ~325 C for 30 hr.
43	128	Gruneisen, E. and Reddemann, H.	1934		83-292		75.6 24.4	Calculated composition; polycrystalline; cast; quenched from 800 C.
44	128	Gruneisen, E. and Reddemann, H.	1934		83-293		75.6 24.4	The above specimen annealed at 360 C for 22 hr.
45*	128	Gruneisen, E. and Reddemann, H.	1934		83, 273		75.6 24.4	The above specimen annealed at 345 C for 30 hr.
46*	128	Gruneisen, E. and Reddemann, H.	1934		83, 273		75.6 24.4	The above specimen annealed at 325 C for 30 hr.
47*	128	Gruneisen, E. and Reddemann, H.	1934		83		75.6 24.4	The above specimen annealed at 800 C for 2 hr, then quenched.
48*	128	Gruneisen, E. and Reddemann, H.	1934		83		75.6 24.4	The above specimen measured after 4 months.
49*	128	Gruneisen, E. and Reddemann, H.	1934		83, 273		75.6 24.4	The above specimen annealed at ~325 C for 30 hr.
50*	131	Röhl, H.	1933		291.2	II	50.75 49.25	5 mm diameter x 10 cm long; heat-treated at 800 C for 4 hr and quenched.
51*	131	Röhl, H.	1933		291.2	II	50.75 49.25	The above specimen annealed at 300 to 400 C for 20 hr and cooled slowly.
52*	131	Röhl, H.	1933		291.2	II	50.75 49.25	The above specimen reheated to 800 C for 4 hr and quenched.
53*	131	Röhl, H.	1933		291.2	II	50.75 49.25	The above specimen reheated at 300 to 400 C for 20 hr and cooled slowly.
54*	131	Röhl, H.	1933		291.2	IIIa	75.70 24.30	5 mm diameter x 10 cm long; heat-treated at 800 C for 4 hr and quenched.
55*	131	Röhl, H.	1933		291.2	IIIa	75.70 24.30	The above specimen annealed at 300 to 400 C for 20 hr and cooled slowly.
56*	131	Röhl, H.	1933		291.2	IV	89.63 10.37	5 mm diameter x 10 cm long; annealed at 300 to 400 C for 20 hr and cooled slowly.
57*	131	Röhl, H.	1933		291.2	IV	89.63 10.37	The above specimen reheated to 800 C for 4 hr and quenched.
58*	131	Röhl, H.	1933		291.2	IV	89.63 10.37	The above specimen reheated at 300 to 400 C for 20 hr and cooled slowly.
59	132	Pospisil, V.	1933	V	21-273	9	50.8 49.2	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 12.00 g cm <sup>-3</sup> .
60*	132	Pospisil, V.	1933	V	21-273	9	50.8 49.2	The above specimen annealed at 300 to 400 C for one week.
61*	132	Pospisil, V.	1933	V	21-273	9a	50.8 49.2	Similar to the specimen for Data Set No. 59 but density 12.10 g cm <sup>-3</sup> .
62	132	Pospisil, V.	1933	V	21-273	9a	50.8 49.2	The above specimen annealed at 300 to 400 C for one week.
63*	132	Pospisil, V.	1933	V	21-273	10	52.2 47.8	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 11.97 g cm <sup>-3</sup> .
64*	132	Pospisil, V.	1933	V	21-273	10	52.2 47.8	The above specimen annealed at 300 to 400 C for one week.

\* Not shown in figure.

TABLE 18. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD + COPPER ALLOYS (Temperature Dependence) (continued)

Data Ref. Set No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Au Cu	Composition (continued), Specifications, and Remarks
65 132	Pospisil, V.	1933	V	21-273	11	54.2 45.8	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 12.34 g cm <sup>-3</sup> .
66 132	Pospisil, V.	1933	V	21-273	11	54.2 45.8	The above specimen annealed at 300 to 400 C for one week.
67 132	Pospisil, V.	1933	V	21-273	12	55.9 44.1	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 12.45 g cm <sup>-3</sup> .
68* 132	Pospisil, V.	1933	V	21-273	12	55.9 44.1	The above specimen annealed at 300 to 400 C for one week.
69* 132	Pospisil, V.	1933	V	21-273	13	57.0 43	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 12.43 g cm <sup>-3</sup> .
70 132	Pospisil, V.	1933	V	21-273	13	57.0 43	The above specimen annealed at 300 to 400 C for one week.
71* 132	Pospisil, V.	1933	V	21-273	14	59.3 40.7	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 12.78 g cm <sup>-3</sup> .
72 132	Pospisil, V.	1933	V	21-273	14	59.3 40.7	The above specimen annealed at 300 to 400 C for one week.
73* 132	Pospisil, V.	1933	V	21-273	15	60.5 39.5	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 13.07 g cm <sup>-3</sup> .
74 132	Pospisil, V.	1933	V	21-273	15	60.5 39.5	The above specimen annealed at 300 to 400 C for one week.
75* 132	Pospisil, V.	1933	V	21-273	16	61.4 38.6	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 13.00 g cm <sup>-3</sup> .
76* 132	Pospisil, V.	1933	V	21-273	16	61.4 38.6	The above specimen annealed at 300 to 400 C for one week.
77* 132	Pospisil, V.	1933	V	21-273	17	62.6 37.4	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 13.16 g cm <sup>-3</sup> .
78* 132	Pospisil, V.	1933	V	21-273	17	62.6 37.4	The above specimen annealed at 300 to 400 C for one week.
79 132	Pospisil, V.	1933	V	21-273	18	63.6 36.4	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 13.00 g cm <sup>-3</sup> .
80* 132	Pospisil, V.	1933	V	21-273	18	63.6 36.4	The above specimen annealed at 300 to 400 C for one week.
81* 132	Pospisil, V.	1933	V	21-273	19	64.7 35.3	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 12.93 g cm <sup>-3</sup> .
82* 132	Pospisil, V.	1933	V	21-273	19	64.7 35.3	The above specimen annealed at 300 to 400 C for one week.
83* 132	Pospisil, V.	1933	V	21-273	20	65.6 34.4	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 13.42 g cm <sup>-3</sup> .
84* 132	Pospisil, V.	1933	V	21-273	20	65.6 34.4	The above specimen annealed at 300 to 400 C for one week.
85* 132	Pospisil, V.	1933	V	21-273	21	66.5 33.5	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 13.39 g cm <sup>-3</sup> .
86* 132	Pospisil, V.	1933	V	21-273	21	66.5 33.5	The above specimen annealed at 300 to 400 C for one week.
87* 132	Pospisil, V.	1933	V	21-273	22	67.4 32.6	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 13.60 g cm <sup>-3</sup> .
88* 132	Pospisil, V.	1933	V	21-273	22	67.4 32.6	The above specimen annealed at 300 to 400 C for one week.
89* 132	Pospisil, V.	1933	V	21-273	23	68.4 31.6	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 13.76 g cm <sup>-3</sup> .

\* Not shown in figure.

TABLE 18. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD + COPPER ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Au Cu	Composition (continued), Specifications, and Remarks
90	132	Pospisil, V.	1933	V	21-273	23	68.4 31.6	The above specimen annealed at 300 to 400 C for one week.
91*	132	Pospisil, V.	1933	V	21-273	24	69.7 30.3	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 14.01 g cm <sup>-3</sup> .
92	132	Pospisil, V.	1933	V	21-273	24	69.7 30.3	The above specimen annealed at 300 to 400 C for one week.
93*	132	Pospisil, V.	1933	V	21-273	25	71.0 29	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 14.18 g cm <sup>-3</sup> .
94	132	Pospisil, V.	1933	V	21-273	25	71.0 29	The above specimen annealed at 300 to 400 C for one week.
95*	132	Pospisil, V.	1933	V	21-273	26	72.6 27.4	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 14.37 g cm <sup>-3</sup> .
96*	132	Pospisil, V.	1933	V	21-273	26	72.3 27.4	The above specimen annealed at 300 to 400 C for one week.
97*	132	Pospisil, V.	1933	V	21-273	27	73.8 26.2	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 14.62 g cm <sup>-3</sup> .
98	132	Pospisil, V.	1933	V	21-273	27	73.8 26.2	The above specimen annealed at 300 to 400 C for one week.
99*	132	Pospisil, V.	1933	V	21-273	28	74.9 25.1	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 14.71 g cm <sup>-3</sup> .
100*	132	Pospisil, V.	1933	V	21-273	28	74.9 25.1	The above specimen annealed at 300 to 400 C for one week.
101*	132	Pospisil, V.	1933	V	21-273	29	75.8 24.4	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 14.77 g cm <sup>-3</sup> .
102*	132	Pospisil, V.	1933	V	21-273	29	75.6 24.4	The above specimen annealed at 300 to 400 C for one week.
103*	132	Pospisil, V.	1933	V	21-273	29a	75.6 24.4	Similar to the specimen for Data Set No. 21 but density 14.64 g cm <sup>-3</sup> .
104	132	Pospisil, V.	1933	V	21-273	29a	75.6 24.4	The above specimen annealed at 300 to 400 C for one week.
105*	132	Pospisil, V.	1933	V	21-273	30	76.4 23.6	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 14.63 g cm <sup>-3</sup> .
106*	132	Pospisil, V.	1933	V	21-273	30	76.4 23.6	The above specimen annealed at 300 to 400 C for one week.
107*	132	Pospisil, V.	1933	V	21-273	31	77.5 22.5	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 14.88 g cm <sup>-3</sup> .
108*	132	Pospisil, V.	1933	V	21-273	31	77.5 22.5	The above specimen annealed at 300 to 400 C for one week.
109*	132	Pospisil, V.	1933	V	21-273	32	78.5 21.5	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 15.15 g cm <sup>-3</sup> .
110	132	Pospisil, V.	1933	V	21-273	32	78.5 21.5	The above specimen annealed at 300 to 400 C for one week.
111*	132	Pospisil, V.	1933	V	21-273	33	79.2 20.8	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 15.32 g cm <sup>-3</sup> .
112*	132	Pospisil, V.	1933	V	21-273	33	79.2 20.8	The above specimen annealed at 300 to 400 C for one week.
113*	132	Pospisil, V.	1933	V	21-273	34	80.5 19.5	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 15.46 g cm <sup>-3</sup> .
114	132	Pospisil, V.	1933	V	21-273	34	80.5 19.5	The above specimen annealed at 300 to 400 C for one week.

\* Not shown in figure.

TABLE 18. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD + COPPER ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Au Cu	Composition (continued), Specifications, and Remarks
115*	132	Pospisil, V.	1933	V	21-273	35	81.7 18.3	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 15.73 g cm <sup>-3</sup> .
116*	132	Pospisil, V.	1933	V	21-273	35	81.7 18.3	The above specimen annealed at 300 to 400 C for one week.
117*	132	Pospisil, V.	1933	V	21-273	36	82.4 17.6	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 15.85 g cm <sup>-3</sup> .
118	132	Pospisil, V.	1933	V	21-273	36	82.4 17.6	The above specimen annealed at 300 to 400 C for one week.
119*	132	Pospisil, V.	1933	V	21-273	37	83.3 16.7	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 15.94 g cm <sup>-3</sup> .
120*	132	Pospisil, V.	1933	V	21-273	37	83.3 16.7	The above specimen annealed at 300 to 400 C for one week.
121*	132	Pospisil, V.	1933	V	21-273	38	84.2 15.8	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 16.17 g cm <sup>-3</sup> .
122*	132	Pospisil, V.	1933	V	21-273	38	84.2 15.8	The above specimen annealed at 300 to 400 C for one week.
123*	132	Pospisil, V.	1933	V	21-273	39	84.7 18.3	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 16.17 g cm <sup>-3</sup> .
124	132	Pospisil, V.	1933	V	21-273	39	84.7 18.3	The above specimen annealed at 300 to 400 C for one week.
125*	132	Pospisil, V.	1933	V	21-273	40	85.3 14.7	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 16.31 g cm <sup>-3</sup> .
126*	132	Pospisil, V.	1933	V	21-273	40	85.3 14.7	The above specimen annealed at 300 to 400 C for one week.
127*	132	Pospisil, V.	1933	V	21-273	41	86.4 13.6	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 16.45 g cm <sup>-3</sup> .
128*	132	Pospisil, V.	1933	V	21-273	41	86.4 13.6	The above specimen annealed at 300 to 400 C for one week.
129*	132	Pospisil, V.	1933	V	21-273	42	87.4 12.6	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 16.67 g cm <sup>-3</sup> .
130*	132	Pospisil, V.	1933	V	21-273	42	87.4 12.6	The above specimen annealed at 300 to 400 C for one week.
131*	132	Pospisil, V.	1933	V	21-273	43	88.4 11.6	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 16.80 g cm <sup>-3</sup> .
132*	132	Pospisil, V.	1933	V	21-273	43	88.4 11.6	The above specimen annealed at 300 to 400 C for one week.
133*	132	Pospisil, V.	1933	V	21-273	44	89.2 10.8	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 16.87 g cm <sup>-3</sup> .
134*	132	Pospisil, V.	1933	V	21-273	44	89.2 10.8	The above specimen annealed at 300 to 400 C for one week.
135*	132	Pospisil, V.	1933	V	21-273	45	89.9 10.1	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 16.96 g cm <sup>-3</sup> .
136*	132	Pospisil, V.	1933	V	21-273	45	89.9 10.1	The above specimen annealed at 300 to 400 C for one week.
137*	132	Pospisil, V.	1933	V	21-273	46	90.4 9.6	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 17.03 g cm <sup>-3</sup> .
138*	132	Pospisil, V.	1933	V	21-273	46	90.4 9.6	The above specimen annealed at 300 to 400 C for one week.

\* Not shown in figure.

TABLE 18. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD + COPPER ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Au Cu	Composition (continued), Specifications, and Remarks
139*	132	Pospisil, V.	1933	V	21-273	47	90.8 9.2	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 17.20 g cm <sup>-3</sup> .
140*	132	Pospisil, V.	1933	V	21-273	47	90.8 9.2	The above specimen annealed at 300 to 400 C for one week.
141	132	Pospisil, V.	1933	V	21-273	48	91.3 8.7	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 17.26 g cm <sup>-3</sup> .
142*	132	Pospisil, V.	1933	V	21-273	48	91.3 8.7	The above specimen annealed at 300 to 400 C for one week.
143*	132	Pospisil, V.	1933	V	21-273	49	92.0 8.0	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 17.49 g cm <sup>-3</sup> .
144*	132	Pospisil, V.	1933	V	21-273	49	92.0 8.0	The above specimen annealed at 300 to 400 C for one week.
145*	132	Pospisil, V.	1933	V	21-273	50	92.6 7.4	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 17.64 g cm <sup>-3</sup> .
146*	132	Pospisil, V.	1933	V	21-273	50	92.6 7.4	The above specimen annealed at 300 to 400 C for one week.
147*	132	Pospisil, V.	1933	V	21-273	51	93.1 6.9	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 17.64 g cm <sup>-3</sup> .
148*	132	Pospisil, V.	1933	V	21-273	51	93.1 6.9	The above specimen annealed at 300 to 400 C for one week.
149	132	Pospisil, V.	1933	V	21-273	52	93.7 6.3	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 17.71 g cm <sup>-3</sup> .
150*	132	Pospisil, V.	1933	V	21-273	52	93.7 6.3	The above specimen annealed at 300 to 400 C for one week.
151*	132	Pospisil, V.	1933	V	21-273	53	94.6 5.4	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 17.89 g cm <sup>-3</sup> .
152*	132	Pospisil, V.	1933	V	21-273	53	94.6 5.4	The above specimen annealed at 300 to 400 C for one week.
153	132	Pospisil, V.	1933	V	21-273	54	95.6 3.4	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 18.35 g cm <sup>-3</sup> .
154*	132	Pospisil, V.	1933	V	21-273	54	96.6 3.4	The above specimen annealed at 300 to 400 C for one week.
155*	132	Pospisil, V.	1933	V	21-273	55	98.3 1.7	4 mm diameter x 6 cm long; quenched in ice-water from 500 C; density 18.86 g cm <sup>-3</sup> .
156	132	Pospisil, V.	1933	V	21-273	55	98.3 1.7	The above specimen annealed at 300 to 400 C for one week.
157*	134	Davis, T.H. and Rayne, J.A.	1972	4.2-395			82.3 17.7	Calculated composition (90.0 a/o Au); 1.25 x 0.100 x 0.100 in.; prepared from 99.999% pure gold and 99.9999% pure copper by induction melting; heat-treated at 425 C.
158*	135	Haray, P.G., Roberts, L.D., and Thomson, J.D.	1971	4.2			50.82 49.18	Calculated composition (25 a/o Au); foil specimen 0.001 in. thick; prepared from 99.99% pure Cu and Au by arc-melting in argon and drop-casting in a cylindrical mold; alloy cylinder annealed in an inert atmosphere for 5 days within 50 C of the melting point, cut lengthwise and rolled with intermittent annealing; heat-treated at 368 C for 100 hr, furnace-cooled to 250 C, maintained for 150 hr and quenched in water at room temperature; ordered.
159*	135	Ferry, P.G., et al.	1971	4.2-395			50.82 49.18	The above specimen annealed in vacuum at 410 C for 30 min and quenched in vacuum-pump oil at room temperature.

\* Not shown in figure.

TABLE 18. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD + COPPER ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Au Cu	Composition (continued), Specifications, and Remarks
160*	135	Huray, P.G., Roberts, L.D., and Thomson, J.D.	1971		4.2-296		50.82 49.18	The above specimen quenched from 450 C.
161*	135	Huray, P.G., et al.	1971		4.2-296		50.82 49.18	The above specimen quenched from 500 C.
162	135	Huray, P.G., et al.	1971		4.2-296		50.82 49.18	The above specimen quenched from 600 C; disordered.
163*	135	Huray, P.G., et al.	1971		4.2-296		50.82 49.18	The above specimen quenched from 700 C; disordered.
164*	135	Huray, P.G., et al.	1971		4.2-296		50.82 49.18	The above specimen quenched from 800 C; disordered.
165	136	Ascoli, A., Bergamini, P., and Queirolo, G.T.	1972		297-1311		0.32	Calculated composition (1.0 at/o Cu); 0.25 cm diameter x 10 cm long; prepared from 99.9985 pure Au and 99.999 pure Cu.
166	136	Ascoli, A., et al.	1972		301-1317		0.16	Calculated composition (0.5 at/o Cu); similar to the above specimen.
167	127	Roll, A. and Motz, H.	1957	R	1273-1473		50.8 49.2	Calculated composition (25 at/o Au); in liquid state; prepared from electrolytic copper and 99.95 pure gold; $d\rho/dT = 0.0091 \mu\Omega \text{ cm K}^{-1}$ .
168*	127	Roll, A. and Motz, H.	1957	R	1273-1473		54.7 45.3	Calculated composition (28 at/o Au); similar to the above specimen but $d\rho/dT = 0.0093 \mu\Omega \text{ cm K}^{-1}$ .
169	127	Roll, A. and Motz, H.	1957	R	1273-1473		57.1 42.9	Calculated composition (30 at/o Au); similar to the above specimen but $d\rho/dT = 0.0089 \mu\Omega \text{ cm K}^{-1}$ .
170*	127	Roll, A. and Motz, H.	1957	R	1273-1473		59.3 40.7	Calculated composition (32 at/o Au); similar to the above specimen but $d\rho/dT = 0.0109 \mu\Omega \text{ cm K}^{-1}$ .
171*	127	Roll, A. and Motz, H.	1957	R	1273-1473		61.5 38.5	Calculated composition (34 at/o Au); similar to the above specimen but $d\rho/dT = 0.0104 \mu\Omega \text{ cm K}^{-1}$ .
172	127	Roll, A. and Motz, H.	1957	R	1273-1473		67.4 32.6	Calculated composition (40 at/o Au); similar to the above specimen but $d\rho/dT = 0.0096 \mu\Omega \text{ cm K}^{-1}$ .
173	127	Roll, A. and Motz, H.	1957	R	1273-1473		75.6 24.4	Calculated composition (50 at/o Au); similar to the above specimen but $d\rho/dT = 0.0118 \mu\Omega \text{ cm K}^{-1}$ .
174*	127	Roll, A. and Motz, H.	1957	R	1273-1473		82.3 17.7	Calculated composition (60 at/o Au); similar to the above specimen but $d\rho/dT = 0.0107 \mu\Omega \text{ cm K}^{-1}$ .
175*	127	Roll, A. and Motz, H.	1957	R	1273-1473		86.1 13.9	Calculated composition (66.7 at/o Au); similar to the above specimen but $d\rho/dT = 0.0120 \mu\Omega \text{ cm K}^{-1}$ .
176*	127	Roll, A. and Motz, H.	1957	R	1273-1473		87.2 12.8	Calculated composition (68.7 at/o Au); similar to the above specimen but $d\rho/dT = 0.0136 \mu\Omega \text{ cm K}^{-1}$ .
177*	127	Roll, A. and Motz, H.	1957	R	1273-1473		87.9 12.1	Calculated composition (70 at/o Au); similar to the above specimen but $d\rho/dT = 0.0124 \mu\Omega \text{ cm K}^{-1}$ .
178*	127	Roll, A. and Motz, H.	1957	R	1273-1473		88.9 11.1	Calculated composition (72 at/o Au); similar to the above specimen but $d\rho/dT = 0.0115 \mu\Omega \text{ cm K}^{-1}$ .
179*	127	Roll, A. and Motz, H.	1957	R	1273-1473		90.3 9.7	Calculated composition (75 at/o Au); similar to the above specimen but $d\rho/dT = 0.0134 \mu\Omega \text{ cm K}^{-1}$ .

\* Not shown in figure.

TABLE 18. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD + COPPER ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Au Cu	Composition (continued), Specifications, and Remarks
180*	127	Roll, A. and Metz, H.	1957	R	1273-1473		91.2 8.8	Calculated composition (77 s/o Au); similar to the above specimen.
181*	127	Roll, A. and Metz, H.	1957	R	1273-1473		96.6 4.4	Calculated composition (87.5 s/o Au); similar to the above specimen but $d\rho/dT = 0.0153 \mu\Omega \text{ cm K}^{-1}$ .
182	137	Damon, D.H., Mathur, M.P., and Klemens, P.G.	1968	A	10.1-40.0		0.0085	Wire specimen diameter 0.7 to 0.025 cm and about 20 cm long, annealed at 750 K from 8 to 24 h; loosely wound on 1.5 cm diameter lava sample holder; residual resistivity $1.136 \mu\Omega \text{ cm}$ .

\* Not shown in figure.





[illegible]

**\* Not shown in figure.**



T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
<u>DATA SET 94*</u>		<u>DATA SET 92</u>		<u>DATA SET 100*</u>		<u>DATA SET 108*</u>		<u>DATA SET 116*</u>		<u>DATA SET 124</u>			
20.8	9.24	20.8	5.97	20.8	1.48	20.8	1.96	20.8	6.15	20.8	10.50		
78.2	9.74	78.2	6.50	78.2	2.09	78.2	2.58	78.2	6.70	78.2	11.06		
273.2	11.61	273.2	8.55	273.2	4.28	273.2	4.79	273.2	8.69	273.2	12.90		
<u>DATA SET 95*</u>		<u>DATA SET 93*</u>		<u>DATA SET 101*</u>		<u>DATA SET 109*</u>		<u>DATA SET 117*</u>		<u>DATA SET 125*</u>			
20.8	11.96	20.8	12.10	20.8	12.29	20.8	11.66	20.8	11.02	20.8	10.42		
78.2	12.45	78.2	12.54	78.2	12.75	78.2	12.12	78.2	11.53	78.2	10.92		
273.2	14.08	273.2	14.13	273.2	14.37	273.2	13.74	273.2	13.23	273.2	12.61		
<u>DATA SET 96*</u>		<u>DATA SET 94</u>		<u>DATA SET 102*</u>		<u>DATA SET 110</u>		<u>DATA SET 118</u>		<u>DATA SET 126*</u>			
20.8	8.41	20.8	4.17	20.8	1.65	20.8	2.73	20.8	6.86	20.8	10.45		
78.2	8.94	78.2	4.75	78.2	2.28	78.2	3.33	78.2	7.41	78.2	10.99		
273.2	10.95	273.2	8.91	273.2	4.58	273.2	5.49	273.2	9.38	273.2	12.76		
<u>DATA SET 97*</u>		<u>DATA SET 95*</u>		<u>DATA SET 103*</u>		<u>DATA SET 111*</u>		<u>DATA SET 119*</u>		<u>DATA SET 127*</u>			
20.8	11.39	20.8	11.92	20.8	11.99	20.8	11.97	20.8	10.97	20.8	10.18		
78.2	11.84	78.2	12.34	78.2	12.44	78.2	12.44	78.2	11.47	78.2	10.69		
273.2	13.40	273.2	13.93	273.2	14.03	273.2	14.10	273.2	13.09	273.2	12.40		
<u>DATA SET 98*</u>		<u>DATA SET 96*</u>		<u>DATA SET 104</u>		<u>DATA SET 112*</u>		<u>DATA SET 120*</u>		<u>DATA SET 128*</u>			
20.8	8.11	20.8	3.89	20.8	0.93	20.8	3.96	20.8	7.16	20.8	10.25		
78.2	8.61	78.2	4.41	78.2	1.56	78.2	4.57	78.2	7.70	78.2	10.78		
273.2	10.53	273.2	6.42	273.2	3.83	273.2	6.82	273.2	9.65	273.2	12.53		
<u>DATA SET 99*</u>		<u>DATA SET 97*</u>		<u>DATA SET 105*</u>		<u>DATA SET 113*</u>		<u>DATA SET 121*</u>		<u>DATA SET 129*</u>			
20.8	11.38	20.8	12.05	20.8	12.09	20.8	12.14	20.8	10.60	20.8	9.73		
78.2	11.82	78.2	12.49	78.2	12.52	78.2	12.67	78.2	11.12	78.2	10.24		
273.2	13.37	273.2	14.08	273.2	14.11	273.2	14.49	273.2	12.80	273.2	11.97		
<u>DATA SET 90</u>		<u>DATA SET 98</u>		<u>DATA SET 106*</u>		<u>DATA SET 114</u>		<u>DATA SET 122*</u>		<u>DATA SET 130*</u>			
20.8	7.17	20.8	1.43	20.8	1.28	20.8	4.78	20.8	10.52	20.8	9.85		
78.2	7.67	78.2	2.05	78.2	1.89	78.2	5.34	78.2	11.08	78.2	10.37		
273.2	9.60	273.2	4.25	273.2	4.12	273.2	7.96	273.2	12.91	273.2	12.10		
<u>DATA SET 91*</u>		<u>DATA SET 99*</u>		<u>DATA SET 107*</u>		<u>DATA SET 115*</u>		<u>DATA SET 123*</u>		<u>DATA SET 131*</u>			
20.8	11.37	20.8	11.96	20.8	11.72	20.8	11.28	20.8	10.63	20.8	9.30		
78.2	11.84	78.2	12.39	78.2	12.18	78.2	11.76	78.2	11.14	78.2	9.61		
273.2	13.32	273.2	13.96	273.2	13.78	273.2	13.97	273.2	12.85	273.2	11.58		

**Not shown in figure.**



TABLE 19. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF GOLD + COPPER ALLOYS (Temperature Dependence) (continued)

T	$\rho$
<u>DATA SET 177*</u>	
1273	34.1
1373	35.3
1473	36.5
<u>DATA SET 178*</u>	
1273	34.4
1373	35.6
1473	36.8
<u>DATA SET 179*</u>	
1273	33.9
1373	35.2
1473	36.5
<u>DATA SET 180*</u>	
1273	33.5
1373	34.9
1473	36.2
<u>DATA SET 181*</u>	
1273	32.0
1373	33.5
1473	35.0
<u>DATA SET 182</u>	
10.1	1.1376
11.1	1.1381
12.1	1.1391
12.9	1.1404
14.0	1.1421
14.9	1.1441
16.0	1.1464
17.0	1.1490
17.9	1.1521
19.9	1.1594
25.0	1.1845
30.0	1.2162
35.0	1.2540
40.0	1.2945

\* Not shown in figure.

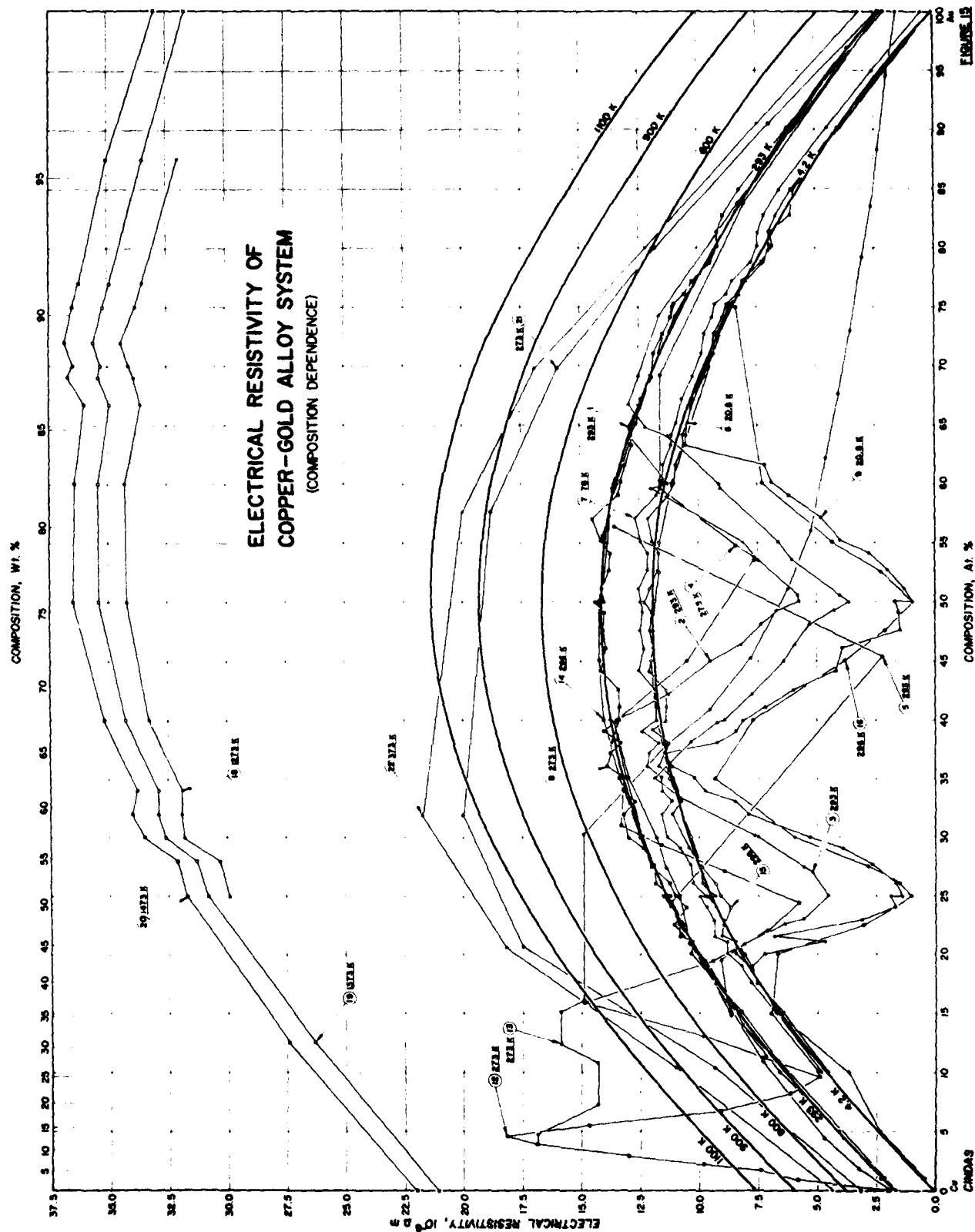


FIGURE 15

TABLE 20. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER-GOLD ALLOY SYSTEM (Composition Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp., K	Composition Range (At. % of Au)	Composition (weight percent), Specifications, and Remarks
1	122	Johansson, C.H. and Linde, J.D.	1936		293.2	4.4-97	Alloys quenched from 923 K.
2	122	Johansson, C.H. and Linde, J.D.	1936		293.2	38-64	Alloys furnace-cooled.
3	122	Johansson, C.H. and Linde, J.D.	1936		293.2	0-100	Alloys annealed at 473 K.
4	138	Sedström, E.	1924		273.2	0-100	Specimens rolled and drawn to 1 mm <sup>2</sup> in cross-section and 3 cm long; annealed close to melting point for 30 min.
5	139	Eitel-McCullough, Inc.	1963		293.2		0.010 in. thick wire specimens; measured at room temperature, assigned as 20 C.
6	132	Pospisil, V.	1933		20.8	10-100	4 mm diameter x 6 cm long; quenched in ice water from 773 K.
7	132	Pospisil, V.	1933		78.2	10-100	The above specimens.
8	132	Pospisil, V.	1933		273.2	10-100	The above specimens.
9	132	Pospisil, V.	1933		20.8	10-100	The above specimens annealed at 573 to 673 K for one week.
10*	132	Pospisil, V.	1933		78.2	10-100	The above specimens.
11*	132	Pospisil, V.	1933		273.2	10-100	The above specimens.
12	140	Broniewski, W. and Wesolowski, K.	1934		273.2	0-94	Prepared in graphite crucible in high frequency furnace, homogenized for 50 h at 1073 K, then drawn to 5 mm diameter; heated for 1 h at 923 K and quenched in water; data extracted from smoothed curve.
13	140	Broniewski, W. and Wesolowski, K.	1934		273.2	0-100	Similar to the above specimens, but cooled slowly to 573 K over a period of 50 h rather than quenched.
14	135	Huray, P.G., Roberts, L.D., and Thomson, J.D.	1971		296.2	1.0-75	Foil specimens 0.001 in. thick; prepared from 99.99 pure metals by arc-melting in argon atmosphere, drop-casting into a cylindrical mold, obtained cylindrical alloys annealed in an inert atmosphere within 50 K of the melting points for 5 days, cut lengthwise and rolled with intermittent annealing; heat-treated in vacuum at 873 K for 30 min. and quenched in vacuum pump oil at room temperature.
15	135	Huray, P.G., et al.	1971		4.2	1.0-75	The above specimens.
16	135	Huray, P.G., et al.	1971		4.2	10-75	The above specimens annealed in argon at 941 K for 100 h, furnace-cooled to 523 K, maintained for 120 h, then quenched in water at room temperature.
17*	141	Lengeler, B., Schilling, W., and Wenzl, H.	1970	V	4.2	0.08-1.8	Wire specimens 150 to 270 $\mu$ m thick and about 30 cm long; supplied by DEGUSSA.
18	127	Roll, A. and Motz, H.	1957	R	1273	25-88	Specimens prepared from electrolytic copper and 99.95 pure gold; in liquid state; reported error $\pm 1\%$ .
19	127	Roll, A. and Motz, H.	1957	R	1373	0-100	The above specimens.
20	127	Roll, A. and Motz, H.	1957	R	1473	0-100	The above specimens.
21	130	Sedström, E.	1919		273.2	0-100	Wire specimens 1 mm in diameter; rolled and drawn; annealed at near melting point for 30 min.
22	130	Sedström, E.	1919		373.2	0-100	The above specimens.

\* Not shown in figure.





TABLE 21. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF COPPER-GOLD ALLOY SYSTEM (Composition Dependence) (continued)

DATA SET 8 (cont.) (T = 273.2 K)			DATA SET 9 (cont.) (T = 20.8 K)			DATA SET 10 (cont.) <sup>a</sup> (T = 18.2 K)			DATA SET 11 <sup>a</sup> (T = 273.2 K)			DATA SET 12 (cont.) (T = 273.2 K)		
C	P		C	P		C	P		C	P		C	P	
48.07	13.93		27.61	2.43		10.00	4.81		75.22	9.34		52.61	4.79	
47.59	14.08		29.01	3.89		15.01	6.82		76.09	8.76		54.06	5.49	
49.03	13.95		29.94	5.29		17.51	7.89		77.19	8.58		55.11	6.82	
49.97	14.37		31.97	7.89		18.93	8.16		78.76	7.78		57.09	7.36	
49.97	14.03		33.05	8.48		20.01	7.78		80.14	7.33		59.00	8.69	
51.07	14.11		33.90	9.75		21.07	5.21		81.31	7.44		60.15	9.38	
52.62	13.78		35.05	10.19		21.55	7.30		82.74	7.08		61.66	9.65	
54.06	13.74		36.03	11.12		22.93	3.68		84.96	6.48		63.21	12.91	
55.11	14.10		37.14	11.42		24.08	2.27		90.16	4.48		64.09	12.90	
57.09	14.49		38.07	9.24		24.98	2.46		94.91	2.54		65.16	12.78	
59.00	13.37		39.02	8.41		24.98	1.97		100.00	0.477		67.19	12.53	
60.15	13.23		39.99	8.11		26.04	2.25					69.10	12.10	
61.66	13.09		41.10	7.17		27.61	3.23					71.07	11.66	
63.21	12.80		42.58	5.97		29.01	4.48					72.70	11.50	
64.09	12.85		44.11	4.17		29.94	5.89					74.16	11.15	
65.16	12.61		46.07	3.89		31.97	8.48					75.22	10.95	
67.19	12.40		47.59	1.43		33.05	9.05					76.09	10.46	
69.10	11.97		49.03	1.48		33.90	10.30					77.19	10.28	
71.07	11.85		49.97	1.65		35.05	10.72					78.76	9.45	
72.70	11.47		51.07	1.28		36.03	11.93					80.14	9.01	
74.16	11.14		51.07	1.28		37.14	11.93					81.31	9.13	
75.22	10.99		52.61	1.96		38.07	9.74					82.74	8.78	
76.09	10.52		54.06	2.73		39.02	8.94					84.96	8.20	
77.19	10.22		55.11	3.96		39.99	8.61					90.16	6.14	
78.76	9.47		57.09	4.78		41.10	7.67					94.91	4.17	
80.14	9.18		59.00	6.15		42.58	6.50					100.00	2.08	
81.31	9.15		60.15	6.86		44.11	4.75							
82.74	8.90		61.66	7.16		46.07	4.41							
84.96	8.24		63.21	10.52		47.59	2.05							
90.16	6.16		64.09	10.50		49.03	2.09							
94.91	4.20		65.16	10.45		49.97	2.28							
100.00	2.07		67.19	10.25		49.97	1.56							
			69.10	9.85		51.07	1.89							
			71.07	9.43		52.61	2.58							
			72.70	9.29		54.06	3.33							
			74.16	8.94		55.11	4.57							
			75.22	8.73		57.09	5.34							
			76.09	8.28		59.00	6.70							
			77.19	8.08		60.15	7.41							
			78.76	7.25		61.66	7.70							
			80.14	6.94		63.21	11.08							
			81.31	6.94		64.09	11.06							
			82.74	6.01		65.16	10.99							
			84.96	5.98		67.19	10.79							
			90.16	3.99		69.10	10.37							
			94.91	2.05		71.07	9.94							
			100.00	0.0373		72.70	9.79							
						74.16	9.45							

DATA SET 9 (T = 20.8 K)			DATA SET 12 (T = 273.2 K)			DATA SET 13 (T = 273.2 K)			DATA SET 14 (T = 286.2 K)		
C	P		C	P		C	P		C	P	
10.00	4.51		0.0	2.00		0.0	2.00		1	2.15	
15.01	6.48		0.9	5.81		0.9	5.81		1	2.28	
17.51	7.54		1.7	7.41		1.7	7.41		2	2.70	
18.93	7.75		2.2	8.00		2.2	8.00		5	4.10	
20.01	7.22		2.9	13.00		2.9	13.00		10	6.00	
21.07	4.68		3.9	16.90		3.9	16.90		15	8.68	
21.55	6.80		4.6	18.20		4.6	18.20		20	10.05	
22.93	3.09		5.5	14.70		5.5	14.70		25	11.98	
24.08	1.66		6.7	9.09		6.7	9.09		34	16.55	
24.98	1.37		8.2	6.13		8.2	6.13		51.1	6.21	
26.04	1.63		9.6	4.90		9.6	4.90		55.9	5.21	
27.61	1.58		11.2	7.25		11.2	7.25		62.2	4.57	
29.01	6.78		13.1	9.90		13.1	9.90		67.6	4.03	
29.94	8.21		19.4	4.28		19.4	4.28		72.9	3.53	
31.97	10.66		22.4	5.75		22.4	5.75		79.2	3.00	
33.05	11.07		27.1	4.13		27.1	4.13		83.5	2.83	
33.90	12.27		31.97	10.66		31.97	10.66		88.3	2.33	
35.05	12.58		33.90	12.27		33.90	12.27		94.5	1.93	
36.03	13.48		35.05	12.58		35.05	12.58		100.0	1.55	
37.14	13.80		37.14	13.80		37.14	13.80				
38.07	11.61		39.02	10.95		39.02	10.95				
39.99	10.53		39.99	10.53		39.99	10.53				
41.10	9.60		42.58	8.55		42.58	8.55				
44.11	6.81		44.11	6.81		44.11	6.81				
46.07	6.42		46.07	6.42		46.07	6.42				
47.59	4.25		47.59	4.25		47.59	4.25				
49.03	4.28		49.03	4.28		49.03	4.28				
49.97	9.94		49.97	9.94		49.97	9.94				
51.07	9.79		51.07	9.79		51.07	9.79				
52.61	9.45		52.61	9.45		52.61	9.45				

<sup>a</sup> Not shown in figure.



### 3.4. Copper-Nickel Alloy System

Sixty-five data-source references are available in the literature on the electrical resistivity of the Cu-Ni alloy system. From these references, a total of 244 data sets, 18 of which are merely single data points, have been extracted along with sample characterization and measurement information. One hundred and fifty-five of the data sets are for the electrical resistivity of Cu + Ni alloys as a function of temperature covering the temperature range from 1 to 1273 K, which are listed in table 23, tabulated in table 24, and shown in figure 19; 71 of the data sets are for the electrical resistivity of Ni + Cu alloys as a function of temperature covering the range from 1.8 to 1273 K, which are listed in table 25, tabulated in table 26, and shown in figure 20; and 18 of the data sets are isotherms of the electrical resistivity as a function of composition, which are listed in table 27, tabulated in table 28, and shown in figure 22. In order to show both functional dependencies of the electrical resistivity, a few of the data sets have been presented in the tables and figures both for temperature dependence and for composition dependence.

Considerable evidence now exists to support the assertion that the Cu-Ni alloy system does not form a continuous series of solid solutions with truly random atomic arrangements. Rather it appears that, for certain alloy compositions and thermal treatments, Ni atoms tend to segregate from the random mixture to form short-range clusters. Some of the earliest evidence for this effect came from electrical resistivity and Hall constant measurements by Köster and Schüle [142] who concluded that short-range ordering occurs in alloys with from 15 to 45 wt.% Ni at temperatures below 923 K, with a maximum degree of order at about 723 K. The effect of the short-range ordering was to decrease the electrical resistivity. In a similar series of experiments, Schüle and Kehrner [14] (Cu + Ni data sets 56-63 and Ni + Cu data sets 40-41) concluded that clustering begins to form below 873 K and increases down to 623 K under conditions of slow cooling. In specimens quenched from homogenization temperatures of greater than 1273 K at a rate of at least  $10,000 \text{ K s}^{-1}$ , Schüle and Kehrner found a greater degree of clustering at lower temperatures, which they attributed to the presence of excess vacancies frozen-in by the quench from high temperatures. Hedman and Mattuck [144] (Cu + Ni data sets 16-18) quenched alloys with 41 to 47 wt.% Ni from 1273 K and found that subsequent annealing at

temperatures from about 473 to 723 K resulted in a decrease in the room temperature resistivity, with a minimum resistivity for annealing temperatures between 553 K and 608 K. They explained these results by suggesting that quenched-in vacancies, by their migration, allow the formation of clusters which produce the observed decrease in resistivity. Robbins et al. [145] came to a very similar interpretation from an analysis of magnetic susceptibility and cluster specific heat data. Mozer et al. [146] found direct evidence for the existence of Ni clusters in a diffuse neutron scattering experiment on a 47.5 at.% Ni specimen prepared with  $\text{Ni}^{62}$  to enhance the difference in scattering powers of Cu and Ni. The specimen, which was furnace-cooled from a temperature of 1294 K, showed a type of clustering in which "the probability of finding a Ni atom in the first-neighbor shell around a Ni atom increases from the random probability of 0.475 to 0.539. The clustering was found to be most pronounced for the first-neighbor shell, and in fact the solid solution is essentially random beyond this shell." By a consideration of diffusion rates in alloys near the composition of their specimen, Mozer et al. concluded that no cooling rate would successfully quench-in the local atomic arrangements characteristic of the alloy at 1273 K because the diffusion rates are too great at this elevated temperature. On the other hand, according to the authors, at around 773 to 873 K the diffusion rates become slow enough that any reasonable cooling rate, even furnace cooling, will freeze-in the atomic structure and prevent changes below 773 K. Hicks et al. [147] have confirmed this short-range clustering effect in neutron experiments of their own on specimens with compositions near the ferromagnetic critical composition.

The magnetic critical composition for Cu-Ni alloys and the Curie temperature for specific compositions are dependent on the degree of clustering present. Hedman and Mattuck [144] and Kussmann and Wollenberger [148] prepared specimens with different degrees of clustering by using appropriate heat and mechanical treatments. Their results are shown in figure 21. In both cases, specimens annealed in the 553 to 673 K range, where clustering is greatest, showed higher Curie temperatures than alloys quenched from 1073 or 1273 K. Cold-work, which presumably reduces the degree of clustering, produced specimens whose Curie temperatures were much lower than those of specimens quenched from higher temperatures. The effect of clustering on the Curie temperature is evident in alloys with as much as 70 at.% Ni and as little as 43 at.% Ni.

In addition to the data mentioned above, figure 21 shows a rather eclectic collection of measurements of Curie temperatures produced by a wide diversity of methods for alloy compositions from the critical composition to pure Ni. Mott [44] predicted a linear dependence of the Curie temperature on atomic composition and predicted a critical composition of 40 at.% Ni. As shown in the figure, this linear dependence is supported by the data for compositions with more than about 75 at.% Ni, but for lower concentrations of Ni, a discrepancy arises. In this region, the data of Hedman and Mattuck [144] and Kussmann and Wollenberger [148] for specimens quenched from high temperature fall very near a straight line which is determined by a critical composition of 40 at.% Ni (at which the Curie temperature is zero) and a Curie temperature for pure Ni of 631 K (represented by the dashed line in figure 21). However, measurements by Hicks et al. [147], Rode et al. [150], Ahmad and Greig [151], and Ahern et al. [152] show Curie temperatures which are not only considerably lower, but also show a non-linear dependence on composition near the critical composition. Two considerations might contribute to an explanation of this discrepancy, which is particularly evident for the composition range studied by Hedman and Mattuck. First, the degree of clustering may be lower for the non-linear group of measurements, resulting in lowered Curie temperatures. Indeed, the Curie temperatures determined by Hedman and Mattuck and by Kussman and Wollenberger for deformed specimens with a low degree of clustering show remarkable agreement with the Curie temperatures determined by Hicks et al., Rode et al., and Ahmad and Greig. However, with the information available, it is difficult to determine what specific differences in thermal and mechanical histories of the specimens gave rise to this difference in degree of clustering. For example, both Hicks et al. and Hedman and Mattuck quenched specimens from 1273 K after homogenization, yet apparently produced specimens of greatly differing degrees of clustering. Second, the experimental arrangement used by Hedman and Mattuck did not allow measurements of the magnetic susceptibility at sufficiently low temperatures, and in some cases required extrapolation of the inverse susceptibility from temperatures as much as 165 K above the Curie temperature.

In this work the lower set of non-linear Curie temperatures, which are believed to be representative of the Curie temperatures of alloy specimens with a low degree of clustering, has been selected. The selected Curie temperatures are shown in figure 21 by a solid line and follow the data of Hicks et al.,

Rode et al., and Ahmad and Grieg closely. The data of Ahern et al. are greater in magnitude than the selected values, though they follow the trend of the selected values as a function of composition. Above about 77 at.% Ni, the solid line and the dashed line, representing the linear relationship suggested by Mott, are congruent.

A number of features of the temperature dependence of the electrical resistivity have received considerable interest in the research literature and are worth noting. First, at low temperatures (up to about 80 K), alloys with from about 30 to 45 wt.% Ni show shallow minima in the electrical resistivity, as shown in figure 19. This feature has been noted and discussed by Skoskiewicz and Baranowski [170], Crangle and Butcher [171], Houghton et al. [172,173], Kondorskii et al. [174], Ahmad and Greig [175], Eagen [176], Eagen and Legvold [177], and Legvold et al. [178], amongst others. Second, alloys with from about 35 to 55 wt.% Ni also show a high temperature (near 700 K) minimum in the electrical resistivity, which has been noted and discussed by Ahmad and Greig [151,175], Houghton et al. [173], and Schüle and Kehrler [143], amongst others. Third, according to Ahmad and Greig [151] and Schüle and Kehrler [143], the electrical resistivity of specimens with from about 30 to 60 wt.% Ni measured during heating is appreciably greater than the electrical resistivity measured during cooling, for temperatures below about 700 to 770 K. Above 700 to 770 K, the heating and cooling curves are the same. Ahmad and Greig found differences of about 1% and Schüle and Kehrler found differences of as much as  $3.5 \times 10^{-6} \Omega \text{m}$ . Ahmad and Greig suggested differences in the degree of clustering as an explanation of this phenomenon.

One feature of the composition-dependent resistivity worthy of note is the shape of the  $\rho_{4.2\text{K}}$  isotherm shown in figure 22. It does not have the rounded, parabolic shape characteristic of the Cu-Au alloy system, but instead is similar to the Matterhorn silhouette of the Cu-Pd alloy system. As the maximum (about 48 at.% Ni) of the residual electrical resistivity is approached from the Ni-rich side, a rapid increase in slope is observed at around 60 to 65 at.% Ni. Another feature is the slight change of slope in the isotherms at the transition from ferromagnetic to paramagnetic behavior. This can be seen, for example, at around 70 at.% Ni in the 293 K isotherm of figure 22. Finally, it should be noted that, as the temperature increases, the maxima in the isotherms of the electrical resistivity as a function of composition are shifted to increasingly greater Ni concentrations.

The recommended values for the electrical resistivity of Cu-Ni alloys were generated by a careful examination and critical analysis of the entire body of experimental data shown in figures 19, 20, and 22. In order to assist in the generation of recommended values for the selected compositions, selected more reliable data were plotted as isotherms of the electrical resistivity as a function of composition at 4.2, 30, 80, 150, 230, 300, 400, 500, 600, 700, 900, and 1100 K. The data were graphically interpolated and smoothed as a function of composition and temperature successively in order to produce the final recommendations, which are shown in figures 16, 17, and 18 as a function of temperature, in figure 22 as a function of composition at selected temperatures, and in table 22. The additional figure 18 in linear scale is given for showing more clearly the recommended curves at high temperatures.

In the generation of recommended values for the resistivity of the Cu-Ni system, we have relied heavily upon the data of Ahmad and Greig [151,175] (Cu + Ni data sets 145-147, 152, Ni + Cu data sets 56-59, and Cu-Ni data sets 14-15), which constitute one of the most comprehensive sets of data in their coverage of a wide range of both temperature and composition. From 30 to 50 wt.% Ni at low temperatures, the data of Houghton et al. [172] (Cu + Ni data sets 38-44) are about  $1.0$  to  $1.5 \times 10^{-8} \Omega\text{m}$  higher than the data of Ahmad and Greig, while the data of Crangle and Butcher [171] (Cu + Ni data sets 51-55) are about the same amount lower. The data of Legvold et al. [178] (Cu + Ni data sets 103-108) are in good agreement. In this range, the recommended values are consistent with all these data. However, near 50 wt.% Ni at low temperatures, only the specimens of Ahmad and Greig [151] (Ni + Cu data sets 58-59) and Crangle and Butcher [171] (Ni + Cu data set 39) are available, yet their resistivities differ by as much as  $6 \times 10^{-8} \Omega\text{m}$ . (Near room temperature the resistivities for these specimens are in good agreement.) In this region of composition, the data of Ahmad and Greig have been given preference and a large uncertainty is assigned to the recommended values. Again at high temperatures, there is a conflict between the data of Ahmad and Greig [151] (Ni + Cu data sets 56-59) for alloys with greater than 50 wt.% Ni and the data of Yao [158] (Ni + Cu data sets 60-63) and Svensson [164] (Ni + Cu data sets 19-22). Both Yao and Svensson report resistivities which are as much as  $3$  to  $4 \times 10^{-8} \Omega\text{m}$  higher than the data of Ahmad and Greig. The data of Yao are believed to show too great an increase with temperature above the Curie

temperature, as can be seen even with his pure Ni specimen. The data of Svensson were disregarded because they were often too high over the whole composition range and because the specimens were not well characterized.

The resulting recommended electrical resistivity values for Cu, Ni, and for 25 Cu-Ni binary alloys are presented in table 22 and shown in figures 16, 17, 18, and 22. The recommended values for Cu and for Ni are for well-annealed high-purity specimens, but those values for temperatures below about 100 K are applicable only to Cu and Ni having residual electrical resistivities as given at 1 K in table 22. The alloys for which the recommended values are generated are well-homogenized alloys with a low degree of clustering and have not been quenched or cold-worked severely. The recommended values cover a full range of temperature from 1 K to 1100 K. These values are not corrected for the thermal expansion of the material. The estimated uncertainties in the values for the various alloys and for different temperature ranges are explicitly stated in a footnote to table 22. A few of the low-temperature values for Cu + 50 % Ni alloy and the two values for molten Ni are indicated as provisional because their uncertainties are greater than  $\pm 5\%$ .



TABLE 22. RECOMMENDED ELECTRICAL RESISTIVITY OF COPPER-NICKEL ALLOY SYSTEM†

(Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ )

Cu: 100.00% (100.00 At.%) Ni: 0.00% (0.00 At.%)		Cu: 99.50% (99.48 At.%) Ni: 0.50% (0.52 At.%)		Cu: 99.00% (98.92 At.%) Ni: 1.00% (1.08 At.%)		Cu: 97.00% (96.26 At.%) Ni: 3.00% (3.24 At.%)		Cu: 95.00% (94.61 At.%) Ni: 5.00% (5.39 At.%)		Cu: 90.00% (89.27 At.%) Ni: 10.00% (10.73 At.%)	
T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
1	0.00200	1	0.58*	1	1.16*	1	3.53*	1	5.89	1	11.79
4	0.00200	4	0.58	4	1.16	4	3.53	4	5.89	4	11.79
7	0.00200	7	0.58*	7	1.16*	7	3.53*	7	5.89	7	11.79
10	0.00202	10	0.58*	10	1.16*	10	3.53*	10	5.89	10	11.79
15	0.00218	15	0.58*	15	1.16*	15	3.53*	15	5.89	15	11.79
20	0.00250	20	0.59*	20	1.17	20	3.53	20	5.89	20	11.79
25	0.00450	25	0.59*	25	1.17	25	3.53	25	5.90	25	11.79
30	0.00830	30	0.59*	30	1.17	30	3.54	30	5.90	30	11.80
40	0.0210	40	0.61*	40	1.19	40	3.56	40	5.93	40	11.80
50	0.0520	50	0.63*	50	1.21	50	3.59	50	5.97	50	11.82
60	0.0974	60	0.67*	60	1.25	60	3.62	60	6.00	60	11.88
70	0.154	70	0.71	70	1.28	70	3.67	70	6.02	70	11.91
80	0.216	80	0.77	80	1.33	80	3.71	80	6.09	80	11.97
90	0.232	90	0.82*	90	1.39	90	3.76	90	6.13	90	12.01
100	0.349	100	0.86*	100	1.45	100	3.82	100	6.19	100	12.08
150	0.701	150	1.21*	150	1.81	150	4.15	150	6.52	150	12.46
200	1.048	200	1.59	200	2.20	200	4.55	200	6.94	200	12.93
250	1.388	250	1.95	250	2.56	250	4.97	250	7.38	250	13.44
273	1.544	273	2.10	273	2.71	273	5.17	273	7.60	273	13.69
293	1.678	293	2.23	293	2.85	293	5.32	293	7.77	293	13.89
300	1.725	300	2.30	300	2.91	300	5.39	300	7.82	300	13.96
350	2.061	350	2.66	350	3.27	350	5.77	350	8.22	350	14.40
400	2.398	400	3.00	400	3.62	400	6.18	400	8.62	400	14.81
500	3.079	500	3.69	500	4.34	500	6.96	500	9.47	500	15.36
600	3.771	600	4.37	600	5.03	600	7.65	600	10.18	600	16.28
700	4.481	700	5.09	700	5.78	700	8.37	700	10.90	700	16.96
800	5.213	800	5.87	800	6.54	800	9.13	800	11.69	800	17.62
900	5.973	900	6.61	900	7.31	900	9.88	900	12.54	900	18.36
1000	6.766	1000	7.39*	1000	8.10*	1000	10.78*	1000	13.41*	1000	19.10*
1100	7.596	1100	8.25*	1100	8.99*	1100	11.69*	1100	14.31*	1100	19.99*
1200	8.470										
1300	9.395										
1357.6	9.946(a)										
1358	21.01(c)										
1700	24.41										

† Uncertainties in the electrical resistivity values are as follows:

100.00 Cu - 0.00 Ni:  $\pm 3\%$  up to 100 K,  $\pm 1\%$  above 100 K to 250 K,  $\pm 0.5\%$  above 250 K to 500 K,  $\pm 1\%$  above 500 K to 1357.6 K, and  $\pm 5\%$  above 1357.6 K.99.50 Cu - 0.50 Ni:  $\pm 5\%$ .99.00 Cu - 1.00 Ni:  $\pm 5\%$ .97.00 Cu - 3.00 Ni:  $\pm 5\%$ .95.00 Cu - 5.00 Ni:  $\pm 5\%$  below 400 K and  $\pm 4\%$  from 400 to 1100 K.90.00 Cu - 10.00 Ni:  $\pm 5\%$  below 400 K and  $\pm 3\%$  from 400 to 1100 K.

\* In temperature range where no experimental data are available.

TABLE 22. RECOMMENDED ELECTRICAL RESISTIVITY OF COPPER-NICKEL ALLOY SYSTEM† (continued)  
[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Cu: 85.00% (83.96 At. %) Ni: 15.00% (16.04 At. %)		Cu: 80.00% (78.71 At. %) Ni: 20.00% (21.29 At. %)		Cu: 75.00% (73.49 At. %) Ni: 25.00% (26.51 At. %)		Cu: 70.00% (68.31 At. %) Ni: 30.00% (31.69 At. %)		Cu: 65.00% (63.18 At. %) Ni: 35.00% (36.82 At. %)		Cu: 60.00% (58.09 At. %) Ni: 40.00% (41.91 At. %)	
T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
1	17.69*	1	23.61	1	29.55*	1	35.48*	1	41.43*	1	46.19*
4	17.69	4	23.61	4	29.55	4	35.47	4	41.39	4	46.17
7	17.69*	7	23.61	7	29.55*	7	35.45*	7	41.35	7	46.13*
10	17.69*	10	23.61	10	29.55*	10	35.42*	10	41.32	10	46.11*
15	17.69*	15	23.61	15	29.55*	15	35.41*	15	41.29	15	46.09*
20	17.69	20	23.61	20	29.55	20	35.40	20	41.27	20	46.04
25	17.69	25	23.61	25	29.55	25	35.40	25	41.25	25	46.02
30	17.69	30	23.61	30	29.55	30	35.41	30	41.24	30	45.99
40	17.71	40	23.63	40	29.58	40	35.42	40	41.23	40	45.93
50	17.75	50	23.68	50	29.60	50	35.48	50	41.27	50	45.90
60	17.79	60	23.71	60	29.62	60	35.52	60	41.30	60	45.86
70	17.83	70	23.73	70	29.68	70	35.58	70	41.32	70	45.82
80	17.90	80	23.79	80	29.71	80	35.63	80	41.37	80	45.80
90	17.97	90	23.82	90	29.78	90	35.69	90	41.39	90	45.78
100	18.01	100	23.89	100	29.82	100	35.73	100	41.40	100	45.76
150	18.39	150	24.22*	150	30.16	150	36.05	150	41.48	150	45.70
200	18.87	200	24.69*	200	30.56	200	36.32	200	41.50	200	45.62
250	19.39	250	25.21*	250	30.99	250	36.58	250	41.50	250	45.51
273	19.63	273	25.46	273	31.19	273	36.67	273	41.50	273	45.43
293	19.83	293	25.66	293	31.35	293	36.72	293	41.49	293	45.38
300	19.90	300	25.72	300	31.40	300	36.76	300	41.48	300	45.35
350	20.32	350	26.12*	350	31.72	350	36.85	350	41.40	350	45.20
400	20.70	400	26.44*	400	31.92	400	36.89	400	41.30	400	45.01
500	21.42	500	26.96*	500	32.18	500	36.82	500	41.10	500	44.70
600	22.04	600	27.41*	600	32.39	600	36.82	600	40.90	600	44.40
700	22.59	700	27.98*	700	32.60	700	36.92	700	40.90	700	44.29
800	23.31	800	28.50*	800	32.88	800	37.10	800	41.09	800	44.52
900	23.95	900	28.98*	900	33.19	900	37.46	900	41.46	900	44.98
1000	24.45*	1000	29.38*	1000	33.60*	1000	37.82	1000	41.86	1000	45.42*
1100	25.06*	1100	29.79*	1100	34.18*	1100	38.34	1100	42.32	1100	45.92*

† Uncertainties in the electrical resistivity values are as follows:

85.00 Cu - 15.00 Ni:  $\pm 5\%$  below 400 K and  $\pm 3\%$  from 400 to 1100 K.

80.00 Cu - 20.00 Ni:  $\pm 5\%$  below 400 K and  $\pm 3\%$  from 400 to 1100 K.

75.00 Cu - 25.00 Ni:  $\pm 5\%$  below 400 K and  $\pm 3\%$  from 400 to 1100 K.

70.00 Cu - 30.00 Ni:  $\pm 4\%$ .

65.00 Cu - 35.00 Ni:  $\pm 4\%$ .

60.00 Cu - 40.00 Ni:  $\pm 4\%$ .

\* In temperature range where no experimental data are available.

TABLE 22. RECOMMENDED ELECTRICAL RESISTIVITY OF COPPER-NICKEL ALLOY SYSTEM† (continued)

[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-4} \Omega \text{ m}$ ]

Cu: 55.00% (53.04 At.%) Ni: 45.00% (46.96 At.%)		Cu: 50.00% (48.02 At.%) Ni: 50.00% (51.98 At.%)		Cu: 45.00% (43.05 At.%) Ni: 55.00% (56.95 At.%)		Cu: 40.00% (38.12 At.%) Ni: 60.00% (61.88 At.%)		Cu: 35.00% (33.22 At.%) Ni: 65.00% (66.78 At.%)		Cu: 30.00% (28.37 At.%) Ni: 70.00% (71.63 At.%)	
T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
1	49.00*	1	46.60†*	1	39.34*	1	32.16*	1	27.30*	1	23.40*
4	49.00	4	46.73†	4	39.41	4	32.19	4	27.37	4	23.43
7	49.00*	7	46.87†*	7	39.48*	7	32.21*	7	27.43*	7	23.50*
10	49.00*	10	47.01†*	10	39.57*	10	32.24*	10	27.50*	10	23.56*
15	49.00*	15	47.23†*	15	39.70*	15	32.32*	15	27.62*	15	23.67*
20	49.00	20	47.49†	20	39.85	20	32.42	20	27.76*	20	23.78
25	49.00	25	47.73†	25	40.01	25	32.55	25	27.90*	25	23.89
30	49.00	30	47.98†	30	40.20	30	32.69	30	28.06*	30	24.01
40	49.00	40	48.40†	40	40.63	40	33.03	40	28.39*	40	24.28
50	49.00	50	48.82†	50	41.09	50	33.47	50	28.75*	50	24.56
60	49.00	60	49.21†	60	41.59	60	33.97	60	29.17*	60	24.89
70	49.00	70	49.52†	70	42.12	70	34.60	70	29.59	70	25.28
80	49.00	80	49.81†	80	42.70	80	35.31	80	30.08	80	25.71
90	49.00	90	50.02†	90	43.30	90	36.02	90	30.62*	90	26.20
100	49.00	100	50.22†	100	43.92	100	36.77	100	31.27*	100	26.73
150	49.00*	150	50.68	150	47.18	150	40.56	150	35.05*	150	29.98
200	48.89*	200	50.56	200	48.44	200	44.27	200	39.41*	200	33.91
250	48.72*	250	50.31	250	49.07	250	46.92	250	43.28*	250	38.29
273	48.65	273	50.19	273	49.23	273	47.42	273	44.62	273	40.19
293	48.58	293	50.06	293	49.34	293	47.73	293	45.47	293	41.79
300	48.55	300	50.01	300	49.38	300	47.82	300	45.67	300	42.34
350	48.33*	350	49.73	350	49.47	350	48.28	350	46.61*	350	44.51
400	48.09*	400	49.50	400	49.40	400	48.49	400	47.18*	400	45.40
500	47.60	500	49.03	500	49.19	500	48.68	500	47.76*	500	46.39
600	47.17	600	48.74	600	48.02	600	48.81	600	48.15*	600	47.08
700	46.98	700	48.63	700	49.18	700	49.18	700	48.68*	700	47.83
800	47.19	800	48.86	800	49.62	800	49.72	800	49.53*	800	48.82
900	47.62	900	49.39	900	50.29	900	50.53	900	50.39*	900	49.96
1000	48.07	1000	49.92	1000	51.10*	1000	51.49	1000	51.61*	1000	51.27*
1100	48.58	1100	50.49*	1100	52.00*	1100	52.83	1100	53.10*	1100	52.76*

† Uncertainties in the electrical resistivity values are as follows:

- 55.00 Cu - 45.00 Ni:  $\pm 4\%$   
 50.00 Cu - 50.00 Ni:  $\pm 10\%$  below 40 K,  $\pm 6\%$  from 40 up to 150 K, and  $\pm 4\%$  from 150 to 1100 K.  
 45.00 Cu - 55.00 Ni:  $\pm 5\%$   
 40.00 Cu - 60.00 Ni:  $\pm 5\%$   
 35.00 Cu - 65.00 Ni:  $\pm 5\%$   
 30.00 Cu - 70.00 Ni:  $\pm 5\%$

\* Provisional values.

\* In temperature range where no experimental data are available.

TABLE 22. RECOMMENDED ELECTRICAL RESISTIVITY OF COPPER-NICKEL ALLOY SYSTEM† (continued)

[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Cu: 25.00% (23.55 At.%) Ni: 75.00% (76.45 At.%)		Cu: 20.00% (18.76 At.%) Ni: 80.00% (81.24 At.%)		Cu: 15.00% (14.02 At.%) Ni: 85.00% (85.98 At.%)		Cu: 10.00% (9.31 At.%) Ni: 90.00% (90.69 At.%)		Cu: 5.00% (4.64 At.%) Ni: 95.00% (95.36 At.%)		Cu: 3.00% (2.78 At.%) Ni: 97.00% (97.22 At.%)	
T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
1	19.50*	1	15.60*	1	11.70*	1	7.81*	1	3.90*	1	2.34*
4	19.53	4	15.64	4	11.72	4	7.82	4	3.90	4	2.34
7	19.58*	7	15.68*	7	11.76*	7	7.83*	7	3.90*	7	2.34*
10	19.61*	10	15.72*	10	11.79*	10	7.85*	10	3.91	10	2.35*
15	19.70*	15	15.78*	15	11.82*	15	7.89*	15	3.92*	15	2.37*
20	19.80	20	15.85	20	11.89	20	7.92	20	3.97	20	2.38*
25	19.91	25	15.92	25	11.95	25	7.97	25	4.00	25	2.40*
30	20.03	30	16.01	30	12.01	30	8.02	30	4.03	30	2.42*
40	20.25	40	16.18	40	12.13	40	8.12	40	4.14	40	2.50*
50	20.50	50	16.36	50	12.30	50	8.27	50	4.28	50	2.61*
60	20.79	60	16.56	60	12.49	60	8.41	60	4.41	60	2.73*
70	21.11	70	16.79	70	12.69	70	8.58	70	4.59	70	2.89
80	21.48	80	17.02	80	12.92	80	8.76	80	4.78	80	3.09
90	21.83	90	17.35	90	13.19	90	9.00	90	4.98	90	3.28*
100	22.22	100	17.66	100	13.49	100	9.28	100	5.20	100	3.50*
150	24.68*	150	19.68	150	15.30	150	10.90	150	6.57	150	4.79*
200	27.82*	200	22.37	200	17.62	200	12.93	200	8.26	200	6.41*
250	31.56*	250	25.64	250	20.50	250	15.38	250	10.37	250	8.34*
273	33.48	273	27.38	273	22.00	273	16.65	273	11.49	273	9.33
35.11		293	28.98	293	23.35	293	17.82	293	12.50	293	10.26
300	35.69	300	29.53	300	23.85	300	18.26	300	12.90	300	10.58
350	39.67*	350	33.63	350	27.60	350	21.51	350	15.69	350	13.14*
400	42.81*	400	37.57	400	31.38	400	25.19	400	18.78	400	15.96*
500	44.60*	500	42.12	500	38.61	500	32.70	500	25.77	500	22.69*
600	45.68*	600	43.86	600	41.28	600	37.78	600	33.17	600	30.60*
700	46.80*	700	45.23	700	43.21	700	40.42	700	36.67	700	34.99
800	47.77*	800	46.56	800	44.72*	800	42.37	800	39.11*	800	37.68
900	49.13*	900	47.92	900	46.19*	900	44.02	900	41.43*	900	40.31
1000	50.61*	1000	49.39*	1000	47.88*	1000	46.02*	1000	43.89*	1000	42.94*
1100	52.10*	1100	51.11*	1100	49.80*	1100	48.47*	1100	46.43*	1100	45.57*

† Uncertainties in the electrical resistivity values are as follows:

25.00 Cu - 75.00 Ni:  $\pm 5\%$   
 20.00 Cu - 80.00 Ni:  $\pm 5\%$   
 15.00 Cu - 85.00 Ni:  $\pm 5\%$   
 10.00 Cu - 90.00 Ni:  $\pm 5\%$   
 5.00 Cu - 95.00 Ni:  $\pm 5\%$   
 3.00 Cu - 97.00 Ni:  $\pm 5\%$

\* In temperature range where no experimental data are available.

TABLE 22. RECOMMENDED ELECTRICAL RESISTIVITY OF COPPER-NICKEL ALLOY SYSTEM<sup>†</sup> (continued)  
[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Cu: 1.00% (0.92 At.%) Ni: 99.00% (99.08 At.%)		Cu: 0.50% (0.46 At.%) Ni: 99.50% (99.54 At.%)		Cu: 0.00% (0.00 At.%) Ni: 100.00% (100.00 At.%)		
T	$\rho$	T	$\rho$	T	$\rho$	
1	0.78*	1	0.39*	1	0.00320	
4	0.78	4	0.39	4	0.00360	
7	0.78*	7	0.39*	7	0.00443	
10	0.78	10	0.39	10	0.00573	
15	0.78*	15	0.39*	15	0.00901	
20	0.79	20	0.39	20	0.0140	
25	0.80*	25	0.40*	25	0.0212	
30	0.82	30	0.41	30	0.0317	
40	0.88	40	0.44	40	0.0678	
50	0.97*	50	0.52*	50	0.135	
60	1.08*	60	0.62*	60	0.242	
70	1.21	70	0.76	70	0.377	
80	1.37	80	0.93	80	0.545	
90	1.57*	90	1.13*	90	0.741	
100	1.61*	100	1.36*	100	0.959	
150	3.09*	150	2.62*	150	2.21	
200	4.57*	200	4.10*	200	3.67	
250	6.32*	250	5.82*	250	5.32	
273	7.23	273	6.71	273	6.16	
293	8.08	293	7.52	293	6.93	
300	8.37	300	7.82	300	7.20	
350	10.63*	350	10.01*	350	9.34	
400	13.18*	400	12.49*	400	11.78	
500	19.33*	500	18.48*	500	17.67	
600	27.28*	600	26.39*	600	25.54	
700	33.11*	700	32.64*	700	32.14	
800	36.39*	800	35.94*	800	35.52	
900	39.19*	900	38.58*	900	38.58	
1000	41.93*	1000	41.60*	1000	41.41	
1100	44.61*	1100	44.27*	1200	46.62	
				1400	51.73	
				1600	56.94	
				1728	60.23(a)	
				1729	82.24*(c)	
				2000	85.23*	

<sup>†</sup> Uncertainties in the electrical resistivity values are as follows:

1.00 Cu - 99.00 Ni:  $\pm 5\%$  below 400 K and  $\pm 3\%$  from 400 to 1100 K.

0.50 Cu - 99.50 Ni:  $\pm 5\%$  below 400 K and  $\pm 3\%$  from 400 to 1100 K.

0.00 Cu - 100.00 Ni:  $\pm 3\%$  below 150 K,  $\pm 3\%$  from 150 to 1300 K,  $\pm 5\%$  above 1300 K to 1728 K, and  $\pm 10\%$  above 1728 K.

\* Provisional value.

c In temperature range where no experimental data are available.

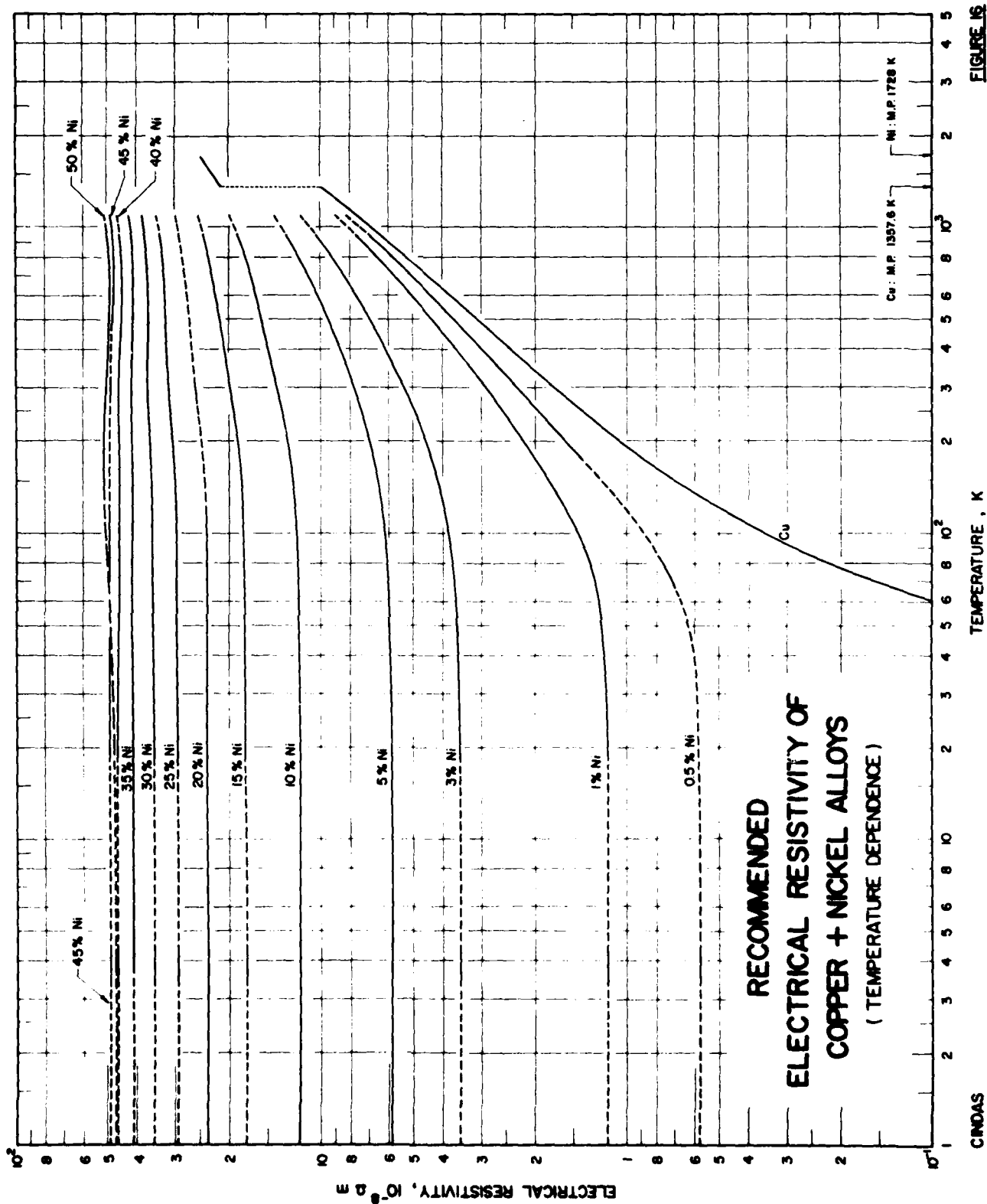


FIGURE 16

TEMPERATURE, K

CINDAS

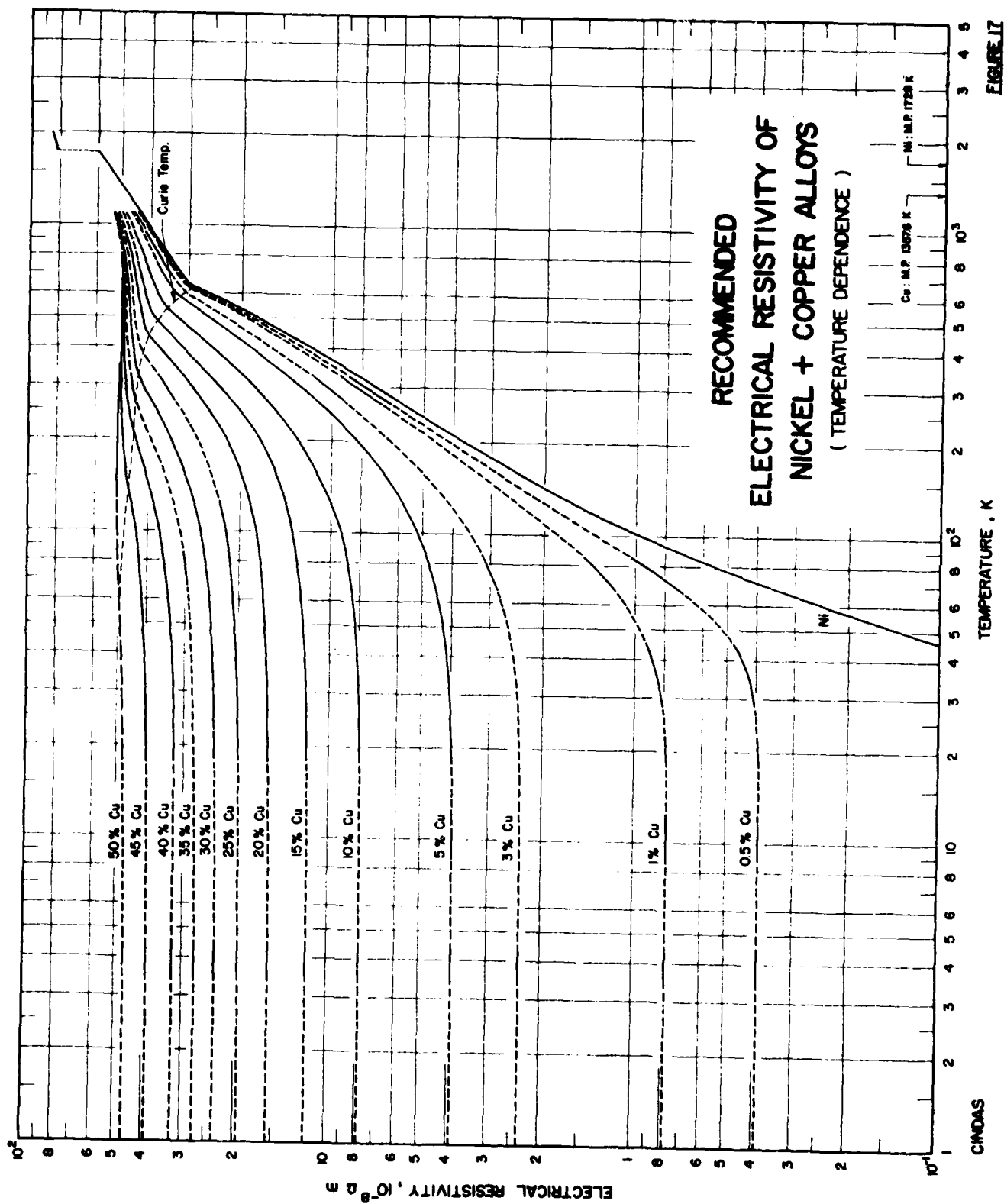


FIGURE 17

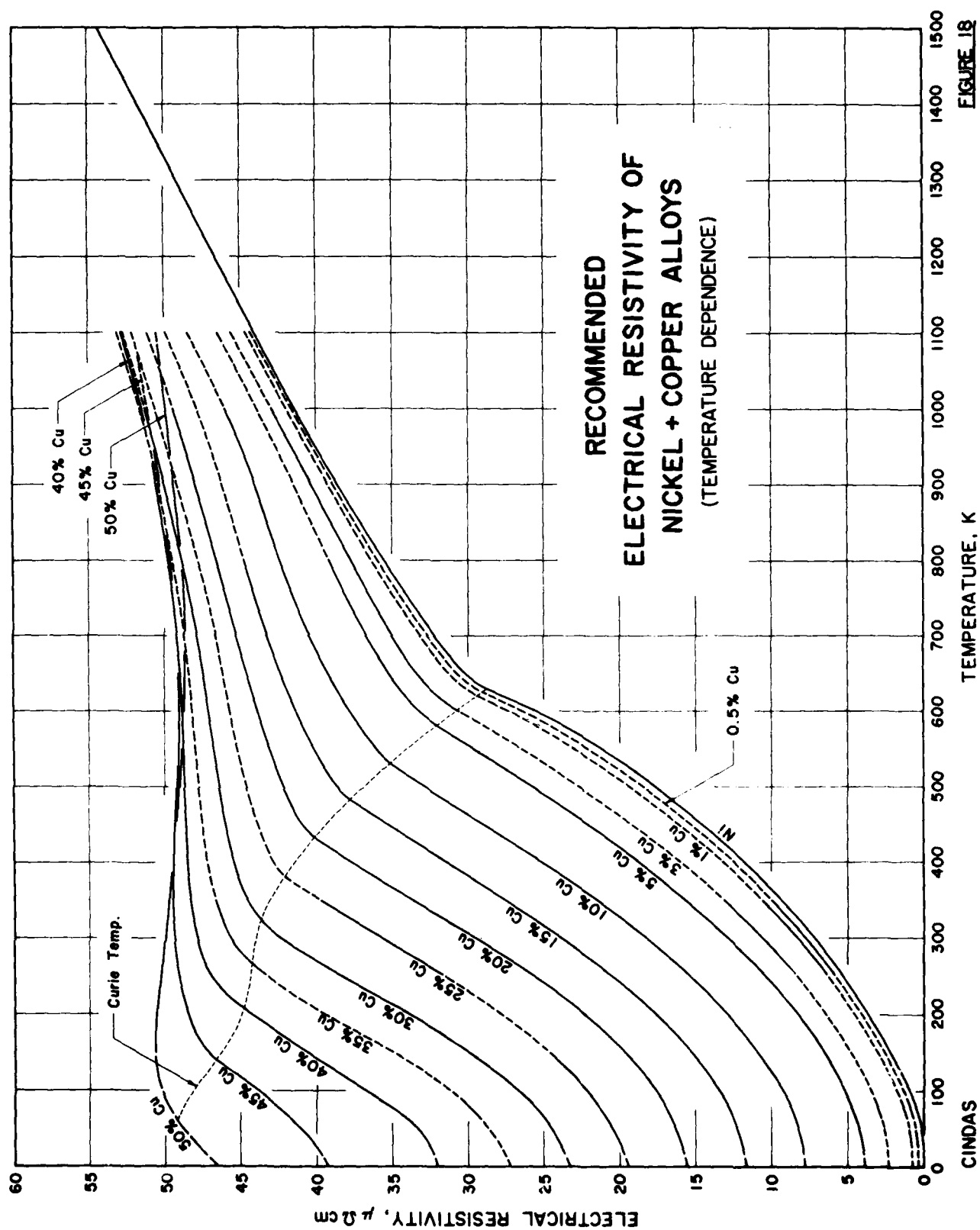
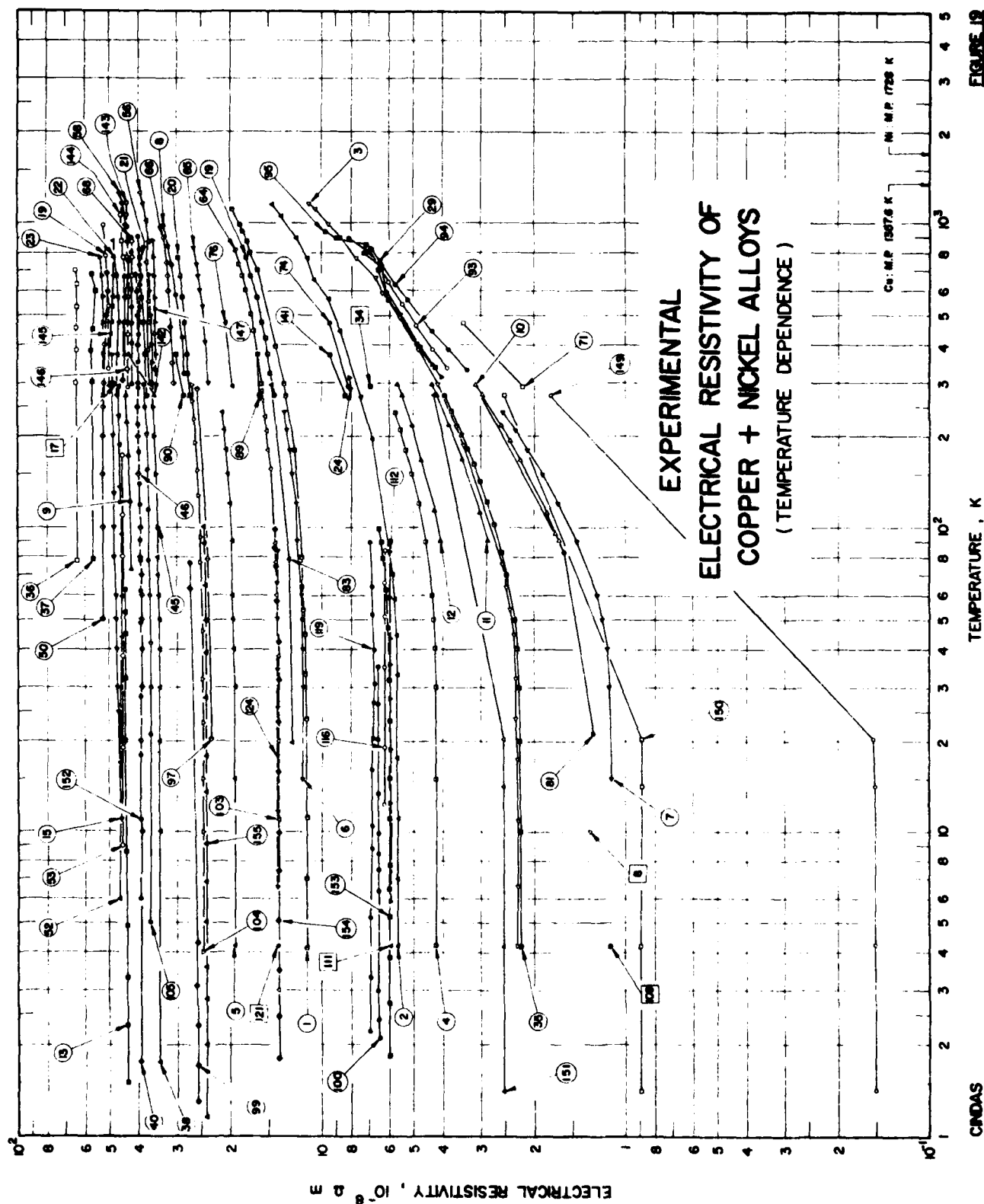


FIGURE 18





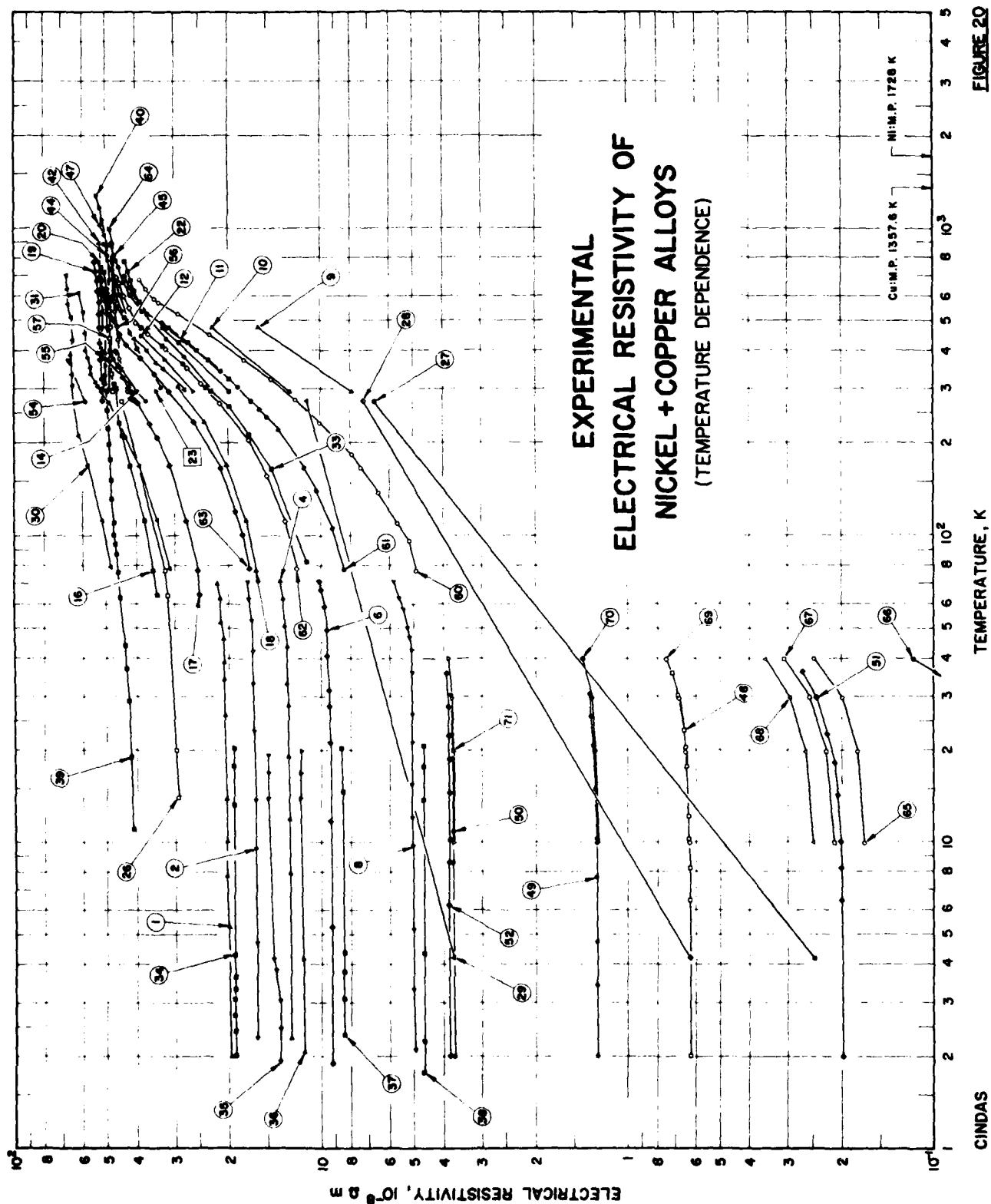


FIGURE 20

TABLE 23. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + NICKEL ALLOYS (Temperature Dependence)

Date Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Ni	Composition (continued), Specifications, and Remarks
1	179	Domenicali, C.A. and Christensen, E.L.	1961	A	4.1-1132		9.33	Calculated composition (10.01 a/o Ni); prepared by melting copper from American Smelting and Refining Co. and nickel from Johnson-Matthey in sealed quartz; annealed at about 500 C for several hours in vacuum.
2	179	Domenicali, C.A. and Christensen, E.L.	1961	A	4.2-1164		4.66	Calculated composition (5.02 a/o Ni); similar to the above specimen.
3	179	Domenicali, C.A. and Christensen, E.L.	1961	A	4.2-1172		1.85	Calculated composition (2.0 a/o Ni); similar to the above specimen.
4	180	Schroeder, P.A., Wolf, R., and Woolam, J.A.	1965	A	4.2-240		3.20	Calculated composition (3.45 a/o Ni); prepared from copper purchased from the American Smelting and Refining Company (R <sub>TP</sub> /R <sub>4</sub> = 1090) and nickel in the form of sponge obtained from Johnson, Matthey, and Company; by melting the components in an alumina crucible by induction heating, and chill casting, ingot drawn down to wire of 0.2 mm diameter, and annealed in vacuum at 680 C.
5	180	Schroeder, P.A., et al.	1965	A	4.2-240		16.01	Calculated composition (17.10 a/o Ni); similar to the above specimen.
6	180	Schroeder, P.A., et al.	1965	A	4.2-240		9.92	Calculated composition (10.65 a/o Ni); similar to the above specimen.
7	180	Schroeder, P.A., et al.	1965	A	4.2-240		0.79	Calculated composition (0.85 a/o Ni); similar to the above specimen.
8	181	Bulow, C.L.	1964		10	E	0.92	Calculated composition (1.0 a/o Ni).
9	182	Javitz, A.E.	1961		73-673	Constantan	60	No details reported.
10	126	Linde, J.O.	1932		90-296		1.04	Calculated composition (1.12 a/o Ni).
11	126	Linde, J.O.	1932		90-296		2.00	Calculated composition (2.16 a/o Ni).
12	126	Linde, J.O.	1932		90-296		3.06	Calculated composition (3.32 a/o Ni).
13	174	Kondorski, E.I., Galkina, O.S., and Chernikova, L.A.	1960	A	1.5-63	1	38	No details reported.
14*	174	Kondorski, E.I., et al.	1960	A	2.0-68	2a	59.35	No details reported.
15	149	Furukawa, G.T., Rellly, M.L., and Saba, W.G.	1964		11-372	Constantan	67	Wire specimen; supplied by Driver-Harris Co.; electrical resistivity values calculated from RT/R <sub>0</sub> °C ratios reported by the author (composition comparable to "Advance", probably an error).
16*	144	Hedman, L.E. and Matlack, R.D.	1962		298.2		46.81	Calculated composition (48.78 a/o Ni); quenching from 1000 C.
17	144	Hedman, L.E. and Matlack, R.D.	1962		298.2		45.04	Calculated composition (47.00 a/o Ni); similar to the above specimen.
18*	144	Hedman, L.E. and Matlack, R.D.	1962		298.2		40.87	Calculated composition (42.79 a/o Ni); similar to the above specimen.
19	164	Svensson, B.	1936		273-773		10.78	Calculated composition (11.56 a/o Ni).
20	164	Svensson, B.	1936		273-773		21.42	Calculated composition (22.78 a/o Ni).
21	164	Svensson, B.	1936		273-773		30.20	Calculated composition (31.89 a/o Ni).
22	164	Svensson, B.	1936		273-773		39.71	Calculated composition (41.62 a/o Ni).
23	164	Svensson, B.	1936		273-773		49.80	Calculated composition (51.78 a/o Ni).

\* Not shown in figure.

TABLE 23. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + NICKEL ALLOYS (Temperature Dependence) (continued)

Data Ref. Set No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Ni	Composition (continued), Specifications, and Remarks
24 184	Pollock, D.D. and Fuch, D.I.	1956		288, 313		5.286	< 0.008 Mg, < 0.004 Si, and < 0.001 each of others.
25* 184	Pollock, D.D. and Fuch, D.I.	1956		288, 313		4.327	Similar to the above specimen.
26* 184	Pollock, D.D. and Fuch, D.I.	1956		288, 313		3.101	Similar to the above specimen.
27* 184	Pollock, D.D. and Fuch, D.I.	1956		288, 313		2.012	Similar to the above specimen.
28* 184	Pollock, D.D. and Fuch, D.I.	1956		288, 313		1.029	Similar to the above specimen.
29 185	Mikryukov, V.E.	1956		312-830		0.90	0.18 Be; normalized.
30* 185	Mikryukov, V.E.	1956		343-830		0.62	0.14 Be; normalized.
31* 126	Linde, J.O.	1932		291		1.04	Calculated composition (1.12 a/o Ni); obtained from Kahlbaum; annealed at 800 C.
32* 126	Linde, J.O.	1932		291		2.00	Calculated composition (2.16 a/o Ni); obtained from Kahlbaum; annealed at 800 C.
33* 126	Linde, J.O.	1932		291		3.08	Calculated composition (3.32 a/o Ni); obtained from Kahlbaum; annealed at 800 C.
34 126	Linde, J.O.	1932		291		4.24	Calculated composition (4.57 a/o Ni); obtained from Hilger.
35 186	Kierspe, W.	1967	V	4, 2-273		1.85	Calculated composition (2.0 a/o Ni); 3 mm diameter specimen.
36 187	Dutta-Roy, S.K. and Subrahmanyam, A.V.	1969	V	78-706		49.98	Calculated composition (51.96 a/o Ni); spectrographically pure; procured from Johnson Matthey and Co.; 30 x 6 x 0.3 mm <sup>2</sup> ; annealed for 24 h at 800 C in a vacuum furnace.
37 187	Dutta-Roy, S.K. and Subrahmanyam, A.V.	1969	V	79-688		39.99	Calculated composition (41.80 a/o Ni); similar to the above specimen.
38 172	Houghton, R.W., Sarachuk, M.P., and Kouvel, J.S.	1970	A	1.75-80		28.37	Paramagnetic alloy; thin bar specimen; cut from cold-worked ingot, annealed and water-quenched.
39* 172	Houghton, R.W., et al.	1970	A	1.75-80		30.30	Similar to the above specimen.
40 172	Houghton, R.W., et al.	1970	A	1.75-80		32.25	Similar to the above specimen.
41* 172	Houghton, R.W., et al.	1970	A	1.75-80		34.20	Similar to the above specimen.
42* 172	Houghton, R.W., et al.	1970	A	1.75-80		36.16	Similar to the above specimen.
43* 172	Houghton, R.W., et al.	1970	A	1.75-80		38.12	Similar to the above specimen.
44* 172	Houghton, R.W., et al.	1970	A	1.75-80		40.09	Similar to the above specimen.
45 172	Houghton, R.W., et al.	1970	A	60-680		28.37	Calculated composition (30 a/o Ni).
46 172	Houghton, R.W., et al.	1970	A	50-680		32.25	Calculated composition (34 a/o Ni).
47* 172	Houghton, R.W., et al.	1970	A	50-680		36.16	Calculated composition (38 a/o Ni).

\* Not shown in figure.

TABLE 23. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + NICKEL ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Ni	Composition (continued), Specifications, and Remarks
48*	172	Houghton, R.W., Sarachik, M.P., and Kovel, J.S.	1970	A	50-600		40.09	Calculated composition (42 a/o Ni).
49*	172	Houghton, R.W., et al.	1970	A	50-690		44.04	Calculated composition (46 a/o Ni).
50	172	Houghton, R.W., et al.	1970	A	50-690		48.02	Calculated composition (50 a/o Ni).
51*	171	Crangle, J. and Butcher, P.J.L.	1970	A	5-294		45.43	Calculated composition (47.4 a/o Ni), prepared by melting together spectrographically pure Cu and Ni in an argon arc furnace and annealing the melts in vacuo for three days at above 1300 K, long, thin cylinder specimen accurately machined.
52	171	Crangle, J. and Butcher, P.J.L.	1970	A	6-294		42.75	Calculated composition (44.7 a/o Ni); similar to the above specimen.
53	171	Crangle, J. and Butcher, P.J.L.	1970	A	9-179		41.07	Calculated composition (43.0 a/o Ni); similar to the above specimen.
54*	171	Crangle, J. and Butcher, P.J.L.	1970	A	7-294		39.40	Calculated composition (41.3 a/o Ni); similar to the above specimen.
55*	171	Crangle, J. and Butcher, P.J.L.	1970	A	6-294		34.59	Calculated composition (36.4 a/o Ni); similar to the above specimen.
56	143	Schüle, W. and Kehr, H.P.	1961	V	295-1273		28.37	Calculated composition (30 a/o Ni); 0.4 mm diameter wire specimen; prepared by melting the components in high frequency furnace under argon atmosphere; 90% deformed; measured during heating.
57*	143	Schüle, W. and Kehr, H.P.	1961	V	319-1273		28.37	The above specimen measured during cooling.
58	143	Schüle, W. and Kehr, H.P.	1961	V	308-1273		32.25	Calculated composition (34 a/o Ni); similar to the above specimen, measured during heating.
59*	143	Schüle, W. and Kehr, H.P.	1961	V	311-1273		32.25	The above specimen measured during cooling.
60*	143	Schüle, W. and Kehr, H.P.	1961	V	320-1273		35.87	Calculated composition (37.5 a/o Ni); similar to the above specimen; measured during heating.
61*	143	Schüle, W. and Kehr, H.P.	1961	V	311-1273		35.87	The above specimen measured during cooling.
62*	143	Schüle, W. and Kehr, H.P.	1961	V	295-1193		43.05	Calculated composition (45 a/o Ni); similar to the above specimen; measured during heating.
63*	143	Schüle, W. and Kehr, H.P.	1961	V	295-1193		43.05	The above specimen measured during cooling.
64	188	Lecordier, J.C. and Colombani, A.	1968		298-820		9.5	Calculated composition (10.2 a/o Ni); film specimen prepared by evaporation and deposition at 1500 C.
65	188	Lecordier, J.C. and Colombani, A.	1968		300-839		16.1	Calculated composition (17.2 a/o Ni); similar to the above specimen.
66	188	Lecordier, J.C. and Colombani, A.	1968		300-9-0		23.7	Calculated composition (25.2 a/o Ni); similar to the above specimen.

\* Not shown in figure.

TABLE 23. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + NICKEL ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Ni	Composition (continued), Specifications, and Remarks
67*	188	Lecordier, J.C. and Colombani, A.	1968		303-897		32.1	Calculated composition (33.8 a/o Ni); similar to the above specimen.
68	188	Lecordier, J.C. and Colombani, A.	1968		297-900		32.9	Calculated composition (34.7 a/o Ni); similar to the above specimen.
69*	188	Lecordier, J.C. and Colombani, A.	1968		298-893		40.3	Calculated composition (42.2 a/o Ni); similar to the above specimen.
70*	188	Lecordier, J.C. and Colombani, A.	1968		299-896		47.8	Calculated composition (49.8 a/o Ni); similar to the above specimen.
71	64	Smith, C.S. and Palmer, E.W.	1935	B	293 173	Bar 107	99.73 0.28	Specimen 0.75 in. in diameter and 8 in. long; supplied by American Brass Co.; annealed at 1073 K for 2 h; $\rho_p/dT = 0.00315 \mu\Omega \text{ cm K}^{-1}$ at 293 K.
72*	64	Smith, C.S. and Palmer, E.W.	1935	B	293, 473	Bar 108	99.47 0.54	0.04 Mg and 0.02 Fe; similar to the above specimen; $\rho_p/dT = 0.00270 \mu\Omega \text{ cm K}^{-1}$ at 293 K.
73*	64	Smith, C.S. and Palmer, E.W.	1935	B	293, 473	Bar 109	97.94 1.97	0.04 Mg and 0.02 Fe; similar to the above specimen except annealed at 800 C for 4 h; $\rho_p/dT = 0.001613 \mu\Omega \text{ cm K}^{-1}$ at 293 K.
74	64	Smith, C.S. and Palmer, E.W.	1935	B	293, 473	Bar 110	94.92 5.09	0.03 Mg, and 0.01 Fe; similar to the above specimen; $\rho_p/dT = 0.000912 \mu\Omega \text{ cm K}^{-1}$ at 293 K.
75*	64	Smith, C.S. and Palmer, E.W.	1935	B	293, 473	Bar 111	89.90 10.07	0.03 Mg, 0.024 C, and 0.02 Fe; similar to the above specimen; $\rho_p/dT = 0.000524 \mu\Omega \text{ cm K}^{-1}$ at 293 K.
76	64	Smith, C.S. and Palmer, E.W.	1935	B	293, 473	Bar 125	84.85 15.07	0.05 Fe, 0.03, and 0.01 Mg; similar to the above specimen; $\rho_p/dT = 0.000344 \mu\Omega \text{ cm K}^{-1}$ at 293 K.
77*	64	Smith, C.S. and Palmer, E.W.	1935	B	293, 473	Bar 124	69.54 30.23	0.013 Mn, 0.05 Fe, and 0.05 Mg; similar to the above specimen; $\rho_p/dT = 0.000048 \mu\Omega \text{ cm K}^{-1}$ at 293 K.
78*	189	Sager, G.F.	1930	B	321-984		79.8 20.0	0.2 Mn, and trace Mg; 0.2 cm wire; chill cast, hot rolled and cold drawn; annealed at 700 C for 12 h.
79*	189	Sager, G.F.	1930	B	335-990		59.8 40.0	Similar to the above specimen.
80*	190	Darratt, T.	1914	B	279, 373	Eureka	60.0 40.0	0.0995 cm diameter x 28.14 cm long.
81	191	Gruncisen, E. and Goetz, E.	1927		21-273	Cu11	99.0 1.0	Rod specimen.
82*	192	Jaeger, W. and Dieselhorst, H.	1900	B	291-373	Constantan	60. 40.	Density 8.92 g cm <sup>-3</sup> at 18 C.
83	193, 194	Zimmerman, J.E. and Estermann, I.	1951		20-296	CN1	90. 10.	Machined from a forged bar supplied by Westinghouse.
84*	193, 194	Zimmerman, J.E. and Estermann, I.	1951		76-296	CN1	90. 10.	Cold worked by rolling from 0.25 in. thick to 0.14 in. before machined to size.
85*	193, 194	Zimmerman, J.E. and Estermann, E.	1951		79-298	CN3	90 10	Severely cold-worked.
86*	193, 194	Zimmerman, J.E. and Estermann, E.	1951		21-298	CN4	90 10	Single crystal.
87*	196	Berman, R.	1961		20-290	Constantan	60 40	Specimen consisted of 317 gauge 36 wires soldered together at ends.

\* Not shown in figure.

TABLE 23. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + NICKEL ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Ni	Composition (continued), Specifications, and Remarks	
88*	196	Ellis, W.C., Morgan, F. L., and Sager, F. G.	1928	A	305	Advance	55.0	45.0	0.25 cm diameter x 2.64 cm long; density 8.73 g cm <sup>-3</sup> .
89	130	Sedström, E.	1928		273, 373			10.1	Calculated composition (10.8 a/o Ni); prepared from Kahlbaum Cu and Ni; rolled and drawn to 1 mm thick; heated for 30 min. close to melting point.
90	130	Sedström, E.	1928		273, 373			20.1	Calculated composition (21.4 a/o Ni); similar to the above specimen.
91*	130	Sedström, E.	1928		273, 373			40.0	Calculated composition (41.9 a/o Ni); similar to the above specimen.
92*	197	Grüneisen, E.	1900		291		54	46	Density 8.89 g cm <sup>-3</sup> .
93	198	Mikryukov, V. E.	1957		336-825			0.70	0.15 Co and 0.1 Be.
94	198	Mikryukov, V. E.	1957		333-900			0.90	0.10 Be and 0.10 Zr.
95	198	Mikryukov, V. E.	1957		336-947			0.80	0.20 Ti.
96*	198	Mikryukov, V. E.	1957		345-923			0.55	0.17 Zr.
97	199	Hulm, J. K.	1951		20-289		80	20	Average grain size 0.011 mm.
98*	200	Tekes, M.	1954		333-493	Constantan	60	40	Nominal composition.
99	201	Lerner, E. and Daunt, J. G.	1964	A	1.3-77	Advance	57	43	Nominal composition; 0.0015 in. wire specimen obtained from Driver-Harris Co.
100	202	Harvey, A. R., Legvold, S., and Peterson, D. T.	1971	A	2.1-35	Cu <sub>55</sub> Ni <sub>45</sub>		5.86	0.0143 Mn; calculated composition (6 a/o Ni); 0.040 in. diameter x 1 in. long; prepared from 99.99 pure Cu supplied by American Smelting and Refining Co. and 99.999 pure Ni supplied by Atomergic Chemicals Co. by arc-melting, swaging, and drawing; electropolished, annealed in vacuum at 1000 C for 3 days, then quenched in ice water.
101*	202	Harvey, A. R., et al.	1971	A	2.1-35	Cu <sub>55</sub> Ni <sub>45</sub>		5.86	0.0296 Mn; similar to the above specimen.
102*	202	Harvey, A. R., et al.	1971	A	2.2-35	Cu <sub>55</sub> Ni <sub>45</sub>		5.86	0.0530 Mn; similar to the above specimen.
103	178	Legvold, S., Peterson, D. T., Burgardt, P., Hoffer, R. J., Landell, B., and Vyrostek, T. A.	1974	A	3-283			11.3	Calculated composition (12.1 a/o Ni); 1 mm diameter x 5 in. long; 99.999 pure copper and 99.999 pure nickel from American Smelting and Refining Co. melted together, ingot swaged and drawn; annealed at 800 C for 3 days and quenched in ice-water.
104	178	Legvold, S., et al.	1974	A	4-297			20.96	Calculated composition (22.3 a/o Ni); similar to the above specimen.
105	178	Legvold, S., et al.	1974	A	5-296			31.18	Calculated composition (32.9 a/o Ni); similar to the above specimen.
106*	178	Legvold, S., et al.	1974	A	1-290			42.06	Calculated composition (44.0 a/o Ni); similar to the above specimen but quenched from 1000 C.
107*	178	Legvold, S., et al.	1974	A	5-291			42.06	The above specimen aged at room temperature for 3 years.
108*	178	Legvold, S., et al.	1974	A	5-297			42.06	The above specimen reannealed at 800 C for 3 days and quenched in ice-water.
109	141	Langeler, B., Schilling, W., and Wenzl, H.	1970	V	4.2			0.925	Calculated composition (1.000 a/o Ni); wire specimen 150 to 270 $\mu$ thick and about 30 cm long supplied by DEGUSSA.

\* Not shown in figure.

TABLE 23. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + NICKEL ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Ni	Composition (continued), Specifications, and Remarks
110*	183	Chang, H.	1970	C	308-1065	Cupron special	55 45	Nominal composition; 0.1265 in. diameter $\pm$ 0.002 in. long; reported error $\pm$ 2%.
111	177	Eagen, C.F. and Legvold, S.	1972	A	4.2		5.10	<0.00007 Fe; calculated composition (5.5 a/o Ni); 99.999% pure Cu from American Smelting and Refining Co. and 99.999% pure Ni from Atomergic Chemicals Co. electron-beam melted separately under high vacuum, then arc-melted together in a graphite crucible, resulting fingers swaged and drawn into 0.014 in. wire, cut to 1 in. long and electro-polished; annealed in vacuum at 1000 C for 3 days, then quenched in ice water.
112	177	Eagen, C.F. and Legvold, S.	1972	A	4.2		5.10	The above specimen measured in longitudinal magnetic fields ranging from 0.2 to 84 kOe.
113*	177	Eagen, C.F. and Legvold, S.	1972	A	2.9-40		5.10	0.0113 Cr and <0.00007 Fe; calculated composition (5.5 a/o Ni and 0.0138 a/o Cr); prepared from 99.999% pure Cu from American Smelting and Refining Co., 99.999% pure Ni from Atomergic Chemicals Co., and 99.99% pure Cr from Chromalloy Corp., same fabrication method and heat-treatment as the above specimen.
114*	177	Eagen, C.F. and Legvold, S.	1972	A	4.2		5.10	The above specimen measured in longitudinal magnetic fields ranging from 5.7 to 84 kOe.
115*	177	Eagen, C.F. and Legvold, S.	1972	A	2.4-55		5.10	0.0247 Cr and <0.00007 Fe; calculated composition (5.5 a/o Ni and 0.0301 a/o Cr); similar to the above specimen.
116	177	Eagen, C.F. and Legvold, S.	1972	A	4.2			The above specimen measured in longitudinal magnetic fields ranging from 12 to 85 kOe.
117*	177	Eagen, C.F. and Legvold, S.	1972	A	2.3-64		5.01	0.0530 Cr and <0.00007 Fe; calculated composition (5.4 a/o Ni and 0.0645 a/o Cr); similar to the above specimen.
118*	177	Eagen, C.F. and Legvold, S.	1972	A	4.2		5.01	The above specimen measured in longitudinal magnetic fields ranging from 8.6 to 84 kOe.
119	177	Eagen, C.F. and Legvold, S.	1972	A	2.2-90		5.10	0.0974 Cr and <0.00007 Fe; calculated composition (5.5 a/o Ni and 0.1185 a/o Cr); similar to the above specimen.
120*	177	Eagen, C.F. and Legvold, S.	1972	A	4.2		5.10	The above specimen measured in longitudinal magnetic fields ranging from 8.8 to 85 kOe.
121	177	Eagen, C.F. and Legvold, S.	1972	A	4.2		11.94	<0.000065 Fe; calculated composition (12.8 a/o Ni); prepared from American Smelting and Refining Co. 99.999% pure Cu and Atomergic Chemicals Co. 99.999% pure Ni by the same fabrication method and heat-treatment as the above specimen.
122*	177	Eagen, C.F. and Legvold, S.	1972	A	4.2		11.94	The above specimen measured in longitudinal magnetic fields ranging from 0.5 to 85 kOe.
123*	177	Eagen, C.F. and Legvold, S.	1972	A	3.4-38		11.76	0.0132 Cr and <0.000065 Fe; calculated composition (12.6 a/o Ni and 0.0160 a/o Cr); prepared from the same original Cu and Ni and Chromalloy Co. 99.99% pure Cr by the same fabrication method and heat-treatment as the above specimen.
124	177	Eagen, C.F. and Legvold, S.	1972	A	4.2		11.76	The above specimen measured in longitudinal magnetic fields ranging from 6.6 to 84 kOe.

\* Not shown in figure.



TABLE 23. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + NICKEL ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Ni	Composition (continued), Specifications, and Remarks
125*	177	Eagen, C.F. and Legvold, S.	1972	A	2.8-43		11.85	0.0230 Cr and <0.000065 Fe; calculated composition (12.7 a/o Ni and 0.0278 a/o Cr); similar to the above specimen.
126*	177	Eagen, C.F. and Legvold, S.	1972	A	4.2		11.85	The above specimen measured in longitudinal magnetic fields ranging from 9.0 to 85 kOe.
127*	177	Eagen, C.F. and Legvold, S.	1972	A	3.5-51		11.76	0.0439 Cr and <0.000065 Fe; calculated composition (12.6 a/o Ni and 0.0531 a/o Cr); similar to the above specimen.
128*	177	Eagen, C.F. and Legvold, S.	1972	A	4.2		11.76	The above specimen measured in longitudinal magnetic fields ranging from 9.1 to 85 kOe.
129*	177	Eagen, C.F. and Legvold, S.	1972	A	4.2-73		11.57	0.0996 Cr and <0.000065 Fe; calculated composition (12.4 a/o Ni and 0.1205 a/o Cr); similar to the above specimen.
130*	177	Eagen, C.F. and Legvold, S.	1972	A	4.2		11.57	The above specimen measured in longitudinal magnetic fields ranging from 9.4 to 85 kOe.
131*	177	Eagen, C.F. and Legvold, S.	1972	A	4.2		21.53	<0.000054 Fe; calculated composition (22.9 a/o Ni); prepared from American Smelting and Refining Co. 99.999+ pure Cu and Atomergic Chemicals Co. 99.999+ pure Ni by the same fabrication method and heat-treatment as the above specimen.
132*	177	Eagen, C.F. and Legvold, S.	1972	A	4.2		21.53	The above specimen measured in longitudinal magnetic fields ranging from 9.4 to 85 kOe.
133*	177	Eagen, C.F. and Legvold, S.	1972	A	4.2-31		21.53	0.0117 Cr and <0.000054 Fe; calculated composition (22.9 a/o Ni and 0.0140 a/o Cr); prepared from the same original Cu and Ni and Chromalloy Co. 99.99+ pure Cr by the same fabrication method and heat-treatment as the above specimen.
134*	177	Eagen, C.F. and Legvold, S.	1972	A	4.2		21.53	The above specimen measured in longitudinal magnetic fields ranging from 7.1 to 85 kOe.
135*	177	Eagen, C.F. and Legvold, S.	1972	A	3.0-36		21.54	0.0272 Cr and <0.000054 Fe; calculated composition (22.9 a/o Ni and 0.0327 Cr); similar to the above specimen.
136*	177	Eagen, C.F. and Legvold, S.	1972	A	4.2		21.54	The above specimen measured in longitudinal magnetic fields ranging from 13 to 85 kOe.
137*	177	Eagen, C.F. and Legvold, S.	1972	A	3.2-44		21.63	0.0515 Cr and <0.000054 Fe; calculated composition (23.0 a/o Ni and 0.0618 a/o Cr); similar to the above specimen.
138*	177	Eagen, C.F. and Legvold, S.	1972	A	4.2		21.63	The above specimen measured in longitudinal magnetic fields ranging from 7.2 to 85 kOe.
139*	177	Eagen, C.F. and Legvold, S.	1972	A	3.6-55		21.25	0.1007 Cr and <0.000054 Fe; calculated composition (22.6 a/o Ni and 0.1209 a/o Cr); similar to the above specimen.
140*	177	Eagen, C.F. and Legvold, S.	1972	A	4.2		21.25	The above specimen measured in longitudinal magnetic fields ranging from 6.1 to 85 kOe.
141	130	Sedström, E.	1919		273, 373		5.0	Calculated composition (5.4 a/o Ni); rolled and drawn to 1 mm in diameter; annealed at near melting point for 30 min.
142	130	Sedström, E.	1919		273, 373		30.4	Calculated composition (32.1 a/o Ni); similar to the above specimen.

\* Not shown in figure.

TABLE 23. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + NICKEL ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Ni	Composition (continued), Specifications, and Remarks
143	142	Köster, W. and Schüle, W.	1957		290-1173		30.0	Specimen deformed 70% into ribbon 0.1 mm thick; heated to 900 C for 30 min and furnace cooled.
144	142	Köster, W. and Schüle, W.	1957		287-1173		15.0	Specimen deformed 20% into ribbon 0.1 mm thick; heated to 900 C for 30 min and furnace cooled.
145	151	Ahmad, W.M. and Greig, D.	1974		293-873	Ni + 50% Cu	51.98 48.02	Composition given by authors as 50% Cu; composition not specified as at, or wt. % but review of other evidence, including comparison of resistivities with those of other investigators is most consistent with conclusion that compositions are at %; prepared by M.J. Walker from Johnson-Matthey Spec-Pure Ni and Cu; pre-melted in argon arc followed by homogenization melt in quartz tubes; ingot rolled and drawn to wire of 0.5 mm diameter and about 120 cm length, cut to 10 cm length, and annealed in vacuum at 1223 K for about 24 hrs; resistivities accurate to better than 0.1% and temperatures to better than 1%; data extracted from tables.
146	151	Ahmad, W.M. and Greig, D.	1974		293-873	Ni + 60% Cu	61.89 38.11	Composition given by authors as 60% Cu; similar to the above specimen.
147	151	Ahmad, W.M. and Greig, D.	1974		293-873	Ni + 70% Cu	71.64 28.36	Composition given by authors as 70% Cu; similar to the above specimen.
148	*203	Los, G.J. and Gerritsen, A.N.	1957		1.42-273		0.023	Alloy prepared by Leiden Laboratory from pure components obtained from Johnson-Matthey and Co. Ltd. melted in vacuum radiation furnace. Ingot rolled and drawn into wire of desired diameter.
149	203	Los, G.J. and Gerritsen, A.N.	1957		1.42-273		0.14	Similar to the above specimen.
150	203	Los, G.J. and Gerritsen, A.N.	1957		1.42-273		0.72	Similar to the above specimen.
151	203	Los, G.J. and Gerritsen, A.N.	1957		1.42-273		2.06	Similar to the above specimen.
152	175	Ahmad, H.M. and Greig, D.	1974		6-908		33.22	Calculated composition (35 at. % Ni); no measurement information or specimen characterization; data extracted from figure.
153	176	Eagen, C.F.	1972		1.83-99.46		5.10	Composition given by author as 5.5 at. % Ni; prepared from 99.999% pure Cu [ $<2.0$ wt. ppm As, $<2.0$ Fe, $<1.0$ Ni, $<1.0$ Pb, $<1.0$ Sb, $<1.0$ Se, $<1.0$ Sn, $<0.7$ Fe, $<0.5$ Cr, $<0.3$ Ag, $<0.1$ Bi, and $<0.1$ wt. ppm Si] from American Smelting and Refining Co. and 99.999% pure Ni [ $<5.0$ wt. ppm Si, $3.0$ Fe, $<1.0$ Al, $<1.0$ Ni, $0.5$ Ca, $0.5$ Cd, $0.5$ Cr, $0.5$ Cu, $0.5$ Pb, $0.1$ Co, $<0.1$ C, and $<0.1$ wt. ppm S] from Atomergic Chemicals Co.; constituents electron-beam melted under high vacuum to remove volatile impurities, arc-melted together in graphite crucible, swaged and drawn through tungsten carbide die into wire of about 0.04 in. diameter, cut to about an inch in length, electropolished in 3 parts methanol to one part nitric acid, sealed in evacuated quartz ampoules, annealed for 3 days at 1273 K, and quenched as rapidly as possible in ice water; author believed annealing time and temperature resulted in no short-range ordering; data extracted from table.

\* Not shown in figure.

TABLE 23. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + NICKEL ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Ni	Composition (continued), Specifications, and Remarks
154	176	Eagen, C. F.	1972		1.80-89		11.94	Composition given by author as 12.8 at. % Ni; similar to the above specimen.
155	176	Eagen, C. F.	1972		1.17-101		21.53	Composition given by author as 22.9 at. % Ni; similar to the above specimen.



[illegible]

**\* Not shown in figure.**













**TABLE 24. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF COPPER + NICKEL ALLOYS (Temperature Dependence) (continued)**

T	P	DATA SET 148*	T	P	DATA SET 152 (cont.)	T	P	DATA SET 153	T	P	DATA SET 154 (cont.)
1.42	0.037		139	39.759		1.83	5.98026		46.57	13.8512	
4.22	0.037		149	39.808		2.13	5.97999		50.90	13.9763	
14.25	0.036		159	39.845		2.72	5.97997		87.96	14.0237	
20.35	0.037		177	39.884		3.34	5.97984		66.23	14.0857	
273.15	1.601		200	39.929		4.23	5.97980		74.98	14.1658	
			218	39.955		5.25	5.98017		85.41	14.2444	
DATA SET 149			237	39.972		6.50	5.98020		98.57	14.3650	
1.42	0.152		256	39.982		7.76	5.98040				
4.22	0.152		288	39.977		11.16	5.98071				
14.25	0.151		308	39.974		15.22	5.98151				
20.35	0.153		330	39.965		19.36	5.98326		1.17	23.9469	
273.15	1.757		351	39.939		23.45	5.98674		1.68	23.9470	
			370	39.909		27.05	5.99166		2.00	23.9469	
DATA SET 150			390	39.879		31.92	6.00244		2.43	23.9470	
1.42	0.897		409	39.851		39.10	6.02621		2.81	23.9471	
4.22	0.896		434	39.821		47.38	6.06457		3.22	23.9471	
14.25	0.898		452	39.798		54.62	6.10788		3.59	23.9473	
20.45	0.890		471	39.768		63.00	6.16381		3.91	23.9475	
273.15	2.522		490	39.736		70.06	6.21496		4.20	23.9475	
			510	39.704		79.55	6.28636		5.04	23.9471	
DATA SET 151			532	39.676		90.06	6.37418		6.89	23.9465	
1.42	2.549		553	39.655		99.46	6.44993		9.13	23.9465	
4.22	2.542		570	39.631					9.92	23.9462	
14.25	2.535		589	39.612					11.12	23.9457	
20.45	2.534		609	39.594					13.76	23.9466	
273.15	4.291		631	39.594		1.80	13.8605		17.19	23.9462	
			653	39.575		1.90	13.8604		21.23	23.9481	
			669	39.571		2.47	13.8607		24.50	23.9500	
			691	39.580		3.01	13.8609		27.99	23.9539	
			712	39.617		3.51	13.8614		31.63	23.9639	
			732	39.660		3.99	13.8616		35.65	23.9766	
DATA SET 152			750	39.702		4.32	13.8625		39.23	23.9920	
6	39.513		770	39.742		5.09	13.8634		42.60	24.0071	
11	39.443		790	39.784		5.94	13.8639		44.33	24.0232	
18	39.386		810	39.817		7.40	13.8641		50.23	24.0584	
28	39.348		827	39.857		9.95	13.8643		54.95	24.0888	
41	39.330		848	39.906		12.57	13.8645		59.68	24.1206	
49	39.349		869	39.957		15.97	13.8650		65.19	24.1701	
62	39.375		889	40.017		19.11	13.8654		71.91	24.2239	
71	39.416		906	40.075		20.60	13.8659		78.82	24.3007	
78	39.470					24.28	13.8702		89.88	24.3962	
91	39.513					27.65	13.8764		100.65	24.5002	
99	39.577					31.99	13.8863				
110	39.635					38.00	13.9066				
119	39.674					42.50	13.9285				
128	39.715					45.26	13.9437				

**\*\* Not shown in figure.**

TABLE 25. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF NICKEL + COPPER ALLOYS (Temperature Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Ni Cu	Composition (continued), Specifications, and Remarks
1	204	Kondorskii, E.I., Galkina, O.S., and Chernukova, L.A.	1958	A	2.0-70		25.1	Wire specimens 0.1 to 0.2 mm in diameter; heated in a vacuum at 900 C for an hour and then cooled slowly in furnace; $\rho_0 = 20.25 \mu\Omega \text{ cm}$ .
2	204	Kondorskii, E.I., et al.	1958	A	2.3-71		20	Similar to the above specimen; $\rho_0 = 16.22 \mu\Omega \text{ cm}$ .
3*	204	Kondorskii, E.I., et al.	1958	A	2.2-72		20	The above specimen reheated to 900 C and quenched in air; $\rho_0 = 17.80 \mu\Omega \text{ cm}$ .
4*	204	Kondorskii, E.I., et al.	1958	A	1.8-72		15.1	Wire specimen 0.1 to 0.2 mm in diameter; heated in a vacuum at 900 C for an hour and then cooled slowly in furnace; $\rho_0 = 12.65 \mu\Omega \text{ cm}$ .
5	204	Kondorskii, E.I., et al.	1958	A	2.3-71		15.1	The above specimen reheated to 900 C and quenched in air; $\rho_0 = 14.26 \mu\Omega \text{ cm}$ .
6*	204	Kondorskii, E.I., et al.	1958	A	1.8-72		9.9	Wire specimen 0.1 to 0.2 mm in diameter; heated in a vacuum at 900 C for an hour and then cooled slowly in furnace; $\rho_0 = 9.27 \mu\Omega \text{ cm}$ .
7	204	Kondorskii, E.I., et al.	1958	A	1.9-71		9.9	The above specimen reheated to 900 C and quenched in air; $\rho_0 = 10.0 \mu\Omega \text{ cm}$ .
8	204	Kondorskii, E.I., et al.	1958	A	2.1-71		4.6	Wire specimen 0.1 to 0.2 mm in diameter; heated in a vacuum at 900 C for an hour and then cooled slowly in furnace; $\rho_0 = 4.95 \mu\Omega \text{ cm}$ .
9	160	Standley, K.J. and Reich, K.H.	1955		293, 473		0.87	Calculated composition (0.8 a/o Cu); supplied by Mond Nickel Co.; Curie temperature 347 C.
10	160	Standley, K.J. and Reich, K.H.	1955		293, 473		4.96	Calculated composition (4.6 a/o Cu); supplied by Mond Nickel Co.; Curie temperature 308 C.
11	160	Standley, K.J. and Reich, K.H.	1955		293-473		10.47	Calculated composition (9.75 a/o Cu); supplied by Mond Nickel Co.; Curie temperature 256 C.
12	160	Standley, K.J. and Reich, K.H.	1955		293-448		15.82	Calculated composition (14.8 a/o Cu); supplied by Mond Nickel Co.; Curie temperature 203 C.
13*	160	Standley, K.J. and Reich, K.H.	1955		293-358		26.2	Calculated composition (24.7 a/o Cu); supplied by Mond Nickel Co.; Curie temperature 105 C.
14	160	Standley, K.J. and Reich, K.H.	1955		293, 348		20	Calculated composition (26.4 a/o Cu); supplied by Mond Nickel Co.; Curie temperature reported as 89 C.
15*	160	Standley, K.J. and Reich, K.H.	1955		293.2		30.1	Calculated composition (28.5 a/o Cu); supplied by Mond Nickel Co.; Curie temperature 69 C.
16	205	Allison, F.E. and Pugh, E.M.	1956	A	64-310		60	2 cm wide and 1 mm thick with 4.5 cm in length exposed between copper lugs.
17	205	Allison, F.E. and Pugh, E.M.	1956	A	64-399		70	Similar to the above specimen.
18	205	Allison, F.E. and Pugh, E.M.	1956	A	77-302		80	Similar to the above specimen.
19	164	Svensson, B.	1936		273-773		41.37	Calculated composition (39.80 a/o Cu).
20	164	Svensson, B.	1936		273-773		29.72	Calculated composition (28.10 a/o Cu).
21*	164	Svensson, B.	1936		273-773		20.81	Calculated composition (19.64 a/o Cu).

\* Not shown in figure.

TABLE 25. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF NICKEL + COPPER ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Ni Cu	Composition (continued), Specifications, and Remarks
22	164	Svensson, B.	1936		273-773		10.32	Calculated composition (9.81 a/o Cu).
23	206	Galkina, O.S. and Chernikova, L.A.	1960		294		39.6	Annealed in vacuum at 900 C for six h and then slowly cooled at the rate of 50 C h <sup>-1</sup> .
24*	206	Galkina, O.S. and Chernikova, L.A.	1960		294		44.55	Similar to the above specimen.
25*	206	Galkina, O.S. and Chernikova, L.A.	1960		294		49.6	Similar to the above specimen.
26	220	Van Elst, H.C. and Gorter, C.J.	1954	B	14-273		60	Polycrystalline; wire specimen 1 mm <sup>2</sup> in cross-section and 20 mm long; supplied by Laboratories of the Philips Lampworks, Eindhoven.
27	207	Farrell, T. and Greig, D.	1968	A	4.2, 273		0.34	Calculated composition (0.31 a/o Cu); 3 mm diameter x 9 cm long; prepared by Metals Research Ltd.; machined; annealed at 850 C for 15 h; reported error $\pm 7\%$ .
28	207	Farrell, T. and Greig, D.	1968	A	4.2, 273		0.87	Nominal composition (calculated from 0.8 a/o Cu); similar to the above specimen.
29	207	Farrell, T. and Greig, D.	1968	A	4.2, 273		5.28	Nominal composition (calculated from 4.9 a/o Cu); similar to the above specimen.
30	187	Dutta-Roy, S.K. and Subrahmanyam, A.V.	1969	V	79-706		59.99	Calculated composition (81.87 a/o Ni); spectrographically pure; procured from Johnson Matthey and Co.; 30 x 8 x 0.3 mm <sup>2</sup> ; annealed for 24 h at 800 C in a vacuum furnace.
31	187	Dutta-Roy, S.K. and Subrahmanyam, A.V.	1969	V	78-718		70.05	Calculated composition (71.68 a/o Ni); similar to the above specimen.
32*	187	Dutta-Roy, S.K. and Subrahmanyam, A.V.	1969	V	79-726		79.99	Calculated composition (81.23 a/o Ni); similar to the above specimen.
33	187	Dutta-Roy, S.K. and Subrahmanyam, A.V.	1969	V	82-720		89.83	Calculated composition (90.62 a/o Ni); similar to the above specimen.
34	208	Kondorskiĭ, E.I., Galkina, O.S., and Chernikova, L.A.	1958	A	2.0-20		74.9	Wire specimen 0.1 to 0.2 mm in diameter; supplied by Central Scientific Research Institute of Ferrous Metallurgy; cold drawn; annealed in neutral gas at 900 C for 1 to 12 h and cooled slowly.
35	208	Kondorskiĭ, E.I., et al.	1958	A	0-19		80.9	Similar to the above specimen.
36	208	Kondorskiĭ, E.I., et al.	1958	A	0-20		84.9	Similar to the above specimen.
37	208	Kondorskiĭ, E.I., et al.	1958	A	0-20		90.1	Similar to the above specimen.
38	208	Kondorskiĭ, E.I., et al.	1958	A	0-20		95.4	Similar to the above specimen.
39	171	Crangle, J. and Butcher, P.J.L.	1970	A	11-295		50.02	Calculated composition (48 a/o Cu); prepared by melting together spectrographically pure Cu and Ni in an argon arc furnace and then annealing the melts in vacuo for three days at above 1300 K; long, thin cylinder specimen accurately machined.

\* Not shown in figure.

TABLE 25. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF NICKEL + COPPER ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Ni Cu	Composition (continued), Specifications, and Remarks
40	143	Schille, W. and Kehr, H. P.	1961	V	309-1273		58.09 41.91	Calculated composition (40 a/o Cu); 0.4 mm diameter wire specimen; prepared by melting the components in high frequency furnace under Argon atmosphere; 90% deformed; electrical resistivity measured during heating. The above specimen measured during cooling.
41*	143	Schille, W. and Kehr, H. P.	1961	V	308-1273		58.09 41.91	
42*	188	Lecordier, J. C. and Colombani, A.	1968		300-896		53.5 46.5	Calculated composition (53.5 a/o Ni); film specimen prepared by evaporation and deposition at 1500 C.
43*	188	Lecordier, J. C. and Colombani, A.	1968		301-896		59.6 40.4	Calculated composition (61.5 a/o Ni); similar to the above specimen.
44	188	Lecordier, J. C. and Colombani, A.	1968		301-886		73.1 26.9	Calculated composition (74.6 a/o Ni); similar to the above specimen.
45	188	Lecordier, J. C. and Colombani, A.	1968		294-889		84.5 15.5	Calculated composition (85.5 a/o Ni); similar to the above specimen.
46*	189	Sager, G. F.	1930	B	299-1029		60 40	0.2 Mn, and 0.17 Mg; 0.2 cm wire prepared from Mond Nickel by fusing, chill-casting, hot-rolling, and cold-drawing; annealed at 700 C for 12 h.
47	189	Sager, G. F.	1930	B	299-1017		80 20	Similar to the above specimen.
48	209	Greig, D. and Harrison, J. P.	1965	A	2.0-36	C	0.65	Calculated composition (0.6 a/o Cu); cylindrical specimen 4 mm in diameter; supplied by Johnson Matthey and Co.; chill cast from J. M. 890 Ni and J. M. 30 Cu; annealed at 850 C for 12 h; small grains, very fine grain boundaries.
49	209	Greig, D. and Harrison, J. P.	1965	A	2.0-31	D	1.62	Calculated composition (1.5 a/o Cu); similar to the above specimen except long grains running in one direction and very thick (~0.05 mm) grain boundaries.
50	209	Greig, D. and Harrison, J. P.	1965	A	2.0-30	E	4.53	Calculated composition (4.2 a/o Cu); similar to the above specimen except various sizes of grain and various thickness of grain boundaries.
51	209	Greig, D. and Harrison, J. P.	1965	A	2.0-36	F	0.35	Calculated composition (0.32 a/o Cu); similar to the above specimen except mostly small grains, but few long grains running from center.
52	209	Greig, D. and Harrison, J. P.	1965	A	2.0-36	G	5.39	Composition estimated from residual electrical resistivity measurement (5.0 a/o Cu); similar to the above specimen but many grains running from center.
53*	210	Farrell, T. and Greig, D.	1969	A	4.2, 273		0.34	Calculated composition (0.31 a/o Cu); about 3 mm in diameter and 9 cm long; supplied by Metals Research Ltd.; annealed at 1123 K for 15 h.
54	130	Sedström, E.	1919		273, 373		39.1	Calculated composition (62.8 a/o Ni); rolled and drawn to 1 mm in diameter; annealed at near melting point for 30 min.
55	130	Sedström, E.	1919		273, 373		18.4	Calculated composition (82.8 a/o Ni); similar to the above specimen.

\* Not shown in figure.

TABLE 25. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF NICKEL + COPPER ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Ni Cu	Composition (continued), Specifications, and Remarks
56	151	Ahmad, W.M. and Greig, D.	1974		293-873		70.37 29.63	Composition given by authors as 28% Cu; compositions not specified as at. or wt. % but review of other evidence, including comparison of resistivities with those of other investigators is most consistent with conclusion that compositions are at. %; prepared by M.J. Walker from Johnson-Matthey Spec-Pure Ni and Cu; melted in argon arc, ingot rolled and drawn to wire of 0.5 mm diameter and about 120 cm length, cut to 10 cm length, and annealed in vacuum at 1223 K for about 24 h; resistivities accurate to better than 0.1% and temperatures to better than 1%; data extracted from tables.
57	151	Ahmad, W.M. and Greig, D.	1974		293-873		60.11 39.89	Composition given by authors as 38% Cu; similar to the above specimen.
58*	151	Ahmad, W.M. and Greig, D.	1974		293-873		53.03 46.97	Composition given by authors as 45% Cu; similar to the above specimen.
59*	151	Ahmad, W.M. and Greig, D.	1974		293-873		48.02 51.98	Composition given by authors as 50% Cu; similar to the above specimen.
60	158	Yao, Y.D.	1978		77.8-681		4.96	Composition given by author as 4.6 at. % Cu; measured with conventional four-probe technique using a cryostat for 78 to 300 K and a Marshall furnace for 300 to 700 K; data extracted from figure.
61	158	Yao, Y.D.	1978		76.0-700		9.99	Composition given by author as 9.3 at. % Cu; similar to the above specimen.
62	158	Yao, Y.D.	1978		76.2-697		14.98	Composition given by author as 14.0 at. % Cu; similar to the above specimen.
63	158	Yao, Y.D.	1978		76.4-693		20.04	Composition given by author as 18.8 at. % Cu; similar to the above specimen.
64	211	Yarbrough, D.W., Williams, R.K., and Graves, R.S.	1979		300-1000		50 50	Cylindrical specimen approximately 1.0 by 7.6 cm; prepared by arc-casting equal weights of constituent elements, machining, and annealing prior to installation in apparatus; spectrographic analysis showed <0.3 wt. % As, 0.03 Cr, 0.03 wt. % Si, and small amounts of several other metals; analytical results showed C, H, N, and O in the low ppm range; total determinate error $\pm 0.4\%$ ; 13 data points smoothed by least squares fit to the equation $\rho = 0.511993 - 0.861508 \times 10^{-4} T + 0.600814 \times 10^{-16} T^3 + 0.214884 \times 10^8 T^{-2}$ (where T is the absolute temperature in Kelvin degrees) with an average deviation of 0.20% and variance of 0.0173; smoothed values extracted from table.
65	212	Greig, D. and Rowlands, J.A.	1974		0-40		0.011	Nominal composition given as 0.01 at. % Cu, but authors state "sample clearly contains an additional impurity"; measured by T. Farrell, but no measurement information reported; data extracted from table.
66	212	Greig, D. and Rowlands, J.A.	1974		0-40		0.043	Nominal composition given as 0.04 at. % Cu; measured by T. Farrell, but no measurement information reported; data extracted from table.
67	212	Greig, D. and Rowlands, J.A.	1974		0-40		0.173	Nominal composition given as 0.16 at. % Cu; measured by J.P. Harrison, but no measurement information reported; data extracted from table.

\* Not shown in figure.

TABLE 2' MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF NICKEL + COPPER ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Ni Cu	Composition (continued), Specifications, and Remarks
68	213	Greig, D. and Rowlands, J.A.	1974		0-40		0.336	Nominal composition given as 0.3 at. % Cu; measured by T. Farrell, but no measurement information reported; appears to be the same specimen as Ni + Cu data set 27; data extracted from table.
69	213	Greig, D. and Rowlands, J.A.	1974		0-40		0.87	Nominal composition given as 0.8 at. % Cu; measured by J. P. Harrison, but no measurement information reported; appears to be the same specimen as Ni + Cu data set 28; data extracted from table.
70	213	Greig, D. and Rowlands, J.A.	1974		0-40		1.73	Nominal composition given as 1.6 at. % Cu; measured by J. P. Harrison, but no measurement information reported; data extracted from table.
71	213	Greig, D. and Rowlands, J.A.	1974		0-40		5.28	Nominal composition given as 4.9 at. % Cu; measured by J. P. Harrison, but no measurement information reported; appears to be the same specimen as Ni + Cu data set 29; data extracted from table.











TABLE 26. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF NICKEL + COPPER ALLOYS (Temperature Dependence) (continued)

T	$\rho$	T	$\rho$
DATA SET 64		DATA SET 70	
300	49.0	0	1.261*
400	48.3	10.0	1.270
500	47.7*	20.0	1.295
600	47.4	30.0	1.339
700	47.3*	40.0	1.411
800	47.4	DATA SET 71	
900	47.9	0	3.701*
1000	48.6	10.0	3.713
DATA SET 65		20.0	3.739
0	0.169*	30.0	3.783
10.0	0.1701	40.0	3.888
20.0	0.1782	DATA SET 66	
30.0	0.2003	0	0.044*
40.0	0.2455	10.0	0.0453*
DATA SET 66		20.0	0.0559*
0	0.044*	30.0	0.0760*
10.0	0.0453*	40.0	0.1178
20.0	0.0559*	DATA SET 67	
30.0	0.0760*	0	0.209*
40.0	0.1178	10.0	0.213
DATA SET 67		20.0	0.226
0	0.209*	30.0	0.257
10.0	0.213	40.0	0.310
20.0	0.226	DATA SET 68	
30.0	0.257	0	0.247*
40.0	0.310	10.0	0.2508
DATA SET 68		20.0	0.2632
0	0.247*	30.0	0.2974
10.0	0.2508	40.0	0.358
20.0	0.2632	DATA SET 69	
30.0	0.2974	0	0.627*
40.0	0.358	10.0	0.632
DATA SET 69		20.0	0.651
0	0.627*	30.0	0.689
10.0	0.632	40.0	0.753
20.0	0.651		
30.0	0.689		
40.0	0.753		

\* Not shown in figure.

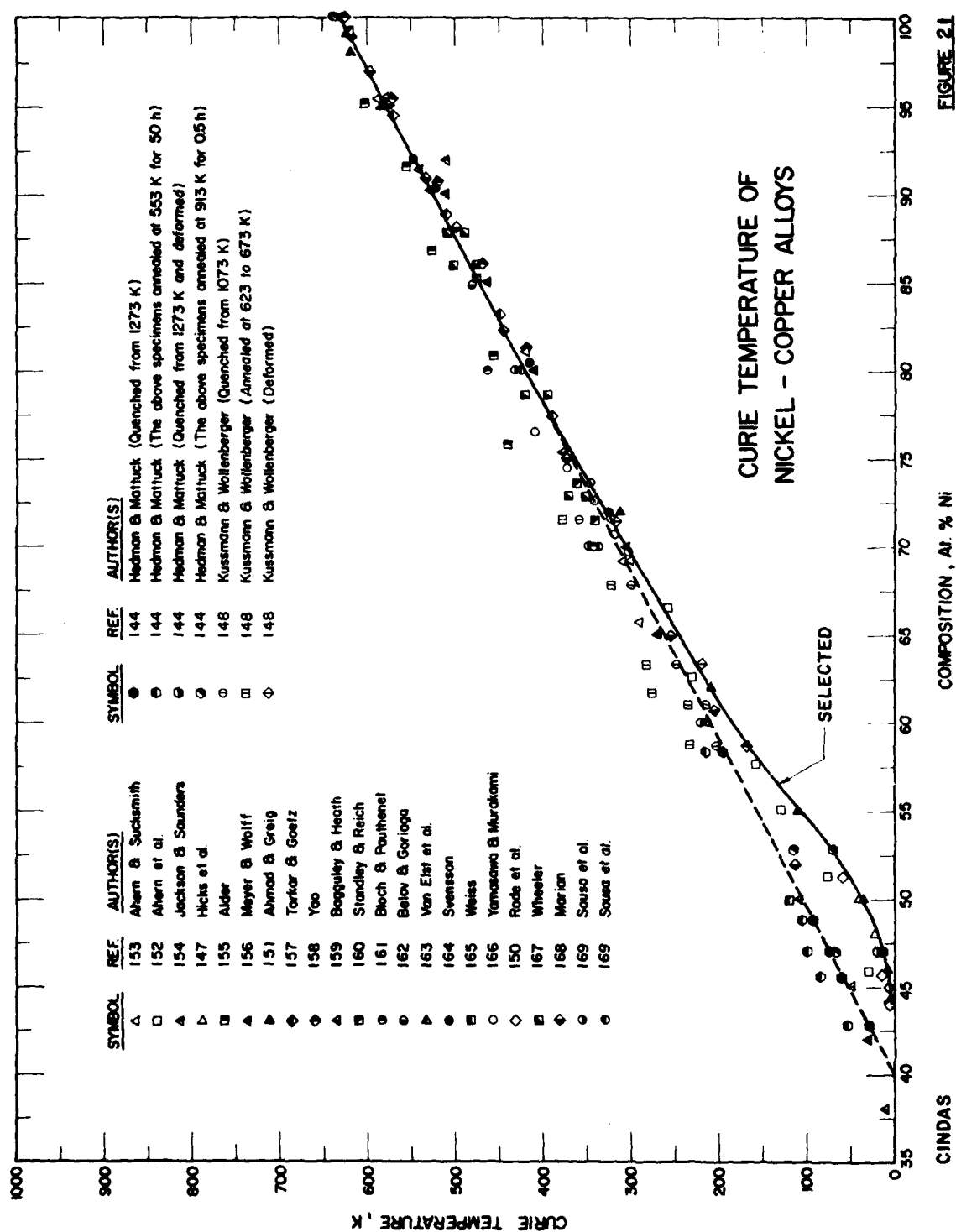


FIGURE 21

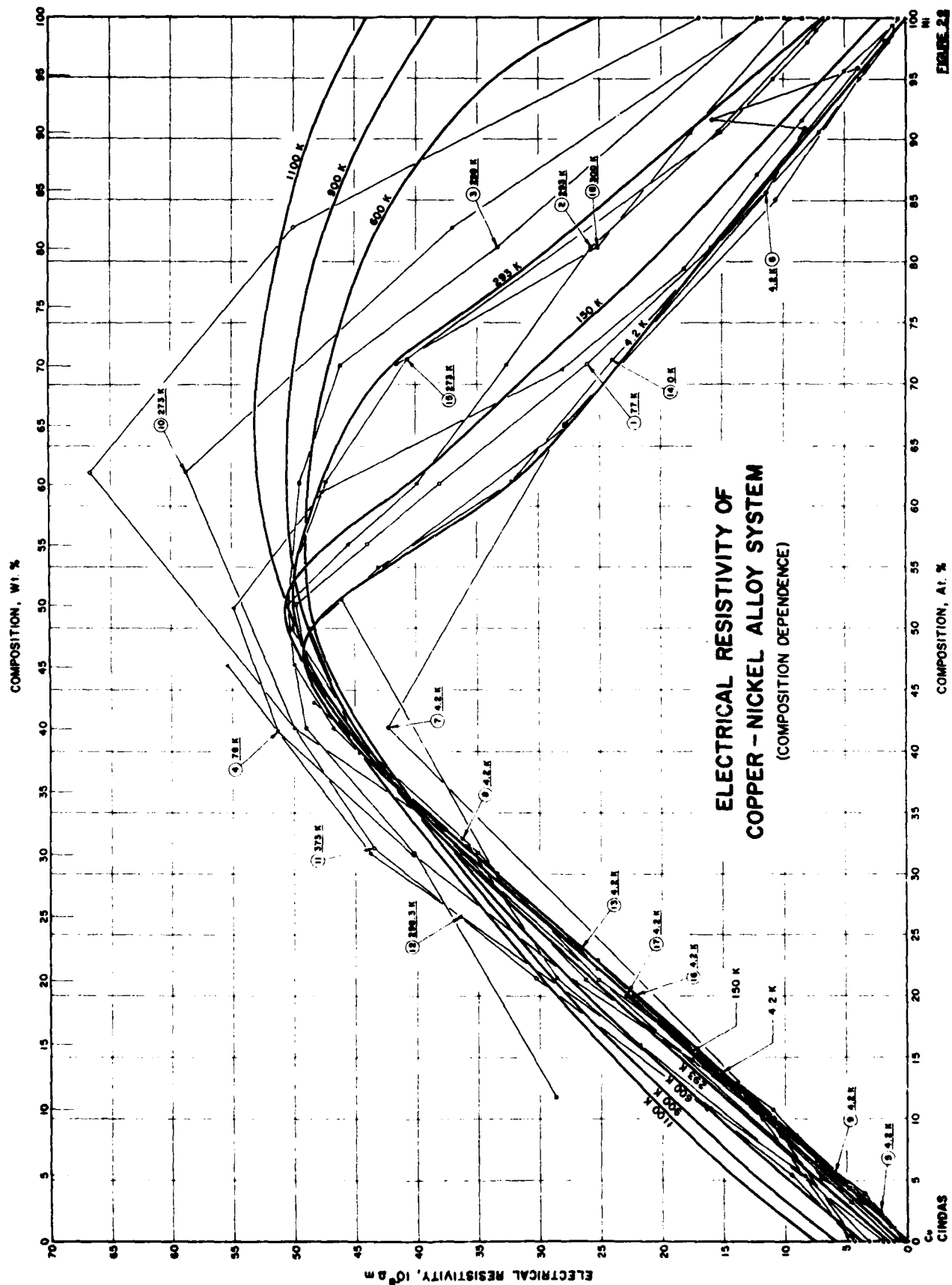


FIGURE 2.8

TABLE 27. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER-NICKEL ALLOY SYSTEM (Composition Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp., K	Composition Range (At. % of Ni)	Composition (weight percent), Specifications, and Remarks
1	213	Lihl, F. and Wildhack, H.	1971		77	0-100	Specimens prepared from 99.99 pure metals; annealed at 593 to 853 K for 1 h.
2	213	Lihl, F. and Wildhack, H.	1971		293	0-100	The above specimens.
3	55	Smith, A. W.	1925	B	298.2	11-100	Specimens 0.3 cm <sup>2</sup> in cross-section and 5 to 6 cm long; prepared from Baker's analyzed 99.97 pure copper and International Nickel Co. of America 99.75 to 99.85 pure nickel.
4	214	Aoyama, S. and Ito, T.	1940	V	78	0-99	Specimens prepared from 99.4 pure electrolytic nickel and 99.9 <sup>+</sup> pure electrolytic copper by fusing.
5	215	Erdmann, J. C. and Jahoda, J. A.	1968	A	4.2	2.5-99	Three single crystals 6.0 to 7.5 mm in diameter and 12 cm long; supplied by Materials Research Corp.; prepared by electron beam float zoning technique.
6	215	Erdmann, J. C. and Jahoda, J. A.	1968	A	4.2	10-96	Polycrystalline specimens 5.0 mm in diameter and 10 cm long; supplied by International Nickel Co., Inc.; vacuum-cast ingots hammer forged, hot-rolled to 18 mm in diameter and rough turned, swaged to 10 mm thick and machined to size; annealed in argon at 1203 K for 24 h, cooled slowly.
7	216	Erdmann, J. C. and Jahoda, J. A.	1964	A	4.2	2.1-98	Polycrystalline specimens 1.35 to 1.45 mm in diameter and 125 mm long; grain size 50 to 250 $\mu$ supplied by International Nickel Co.; vacuum-cast ingots hammer forged, hot-rolled to 18.5 mm in diameter, rough turned, cold-rolled to 6 mm thick and drawn to ~1.5 mm in diameter; annealed at 1273 K for 24 h.
8	178	Legvold, S., Peterson, D. T., Bargar, P., Hofer, R. J., Lutzell, B., and Vyrostek, T. A.	1974	A	4.2	0.3-44	Specimens 1 mm in diameter and 5 in. long; prepared from American Smelting and Refining Co.; 99.999 pure copper and nickel by melting; swaged and drawn; annealed at 1073 K for 3 days and quenched in ice water.
9	177	Eagen, C. F. and Legvold, S.	1972	A	4.2	0-23	Specimens prepared from American Smelting and Refining Co. 99.999 <sup>+</sup> pure copper and 99.999 pure nickel; metals electron-beam melted separately under high vacuum, then arc-melted together in a graphite crucible, swaged and drawn to 0.014 in. thick, cut to 1 in. long and electropolished; annealed in vacuum at 1273 K for 3 days and quenched in ice water.
10	130	Sedström, E.	1919		273.2	0-100	Wire specimens 1 mm in diameter; rolled and drawn; annealed at near melting point for 30 min.
11	130	Sedström, E.	1919		373.2	0-100	The above specimens.
12	142	Köster, W. and Schüle, W.	1957		298.2	0-45	Specimens containing 0, 15, 20, 25, 30, and 45% Ni deformed 70% into ribbons 1 mm thick; heated to 900 °C for 30 minutes and furnace cooled.
13	217	Gartner, H., Zrudsky, D. R., and Legvold, S.	1970		4.2	6.23-23.90	Prepared from high purity Cu and 99.999 pure Ni from American Smelting and Refining Co. by individually electron-beam melting the constituents in an ultra-high vacuum to remove occluded gases, arc melting the alloys several times on a spectrographically pure graphite hearth for homogeneity, sawing with a spark cutter, etching in HNO <sub>3</sub> , swaging inside Al jackets after which alloys were removed, carefully cleaned annealed for 72 h at 1273 K, quenched rapidly in ice water, spark cut and lightly polished to a nominal 0.8 x 0.8 mm <sup>2</sup> cross-section, and electropolished to remove damaged surface impurities; chemical analysis showed control of objectionable transition metal impurities and metallographs indicated absence of second phase formations; uncertainties in the resistivities of $\pm 0.01$ , $\pm 0.13$ and $\pm 0.26$ $\mu\Omega\text{cm}$ for the 6.23, 11.65, and 23.90 at. % Ni specimens respectively; data extracted from table.



TABLE 27. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER-NICKEL ALLOY SYSTEM (Composition Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Composition Range (At. % of Ni)	Composition (weight percent), Specifications, and Remarks
14	151	Ahmad, H. M. and Greig, D.	1974		0	0-100	Compositions were not specified as at. or wt. % but comparison of resistivities with those of other investigators is most consistent with conclusion that compositions are at. %; prepared by M. J. Walker from Johnson-Matthey Spec-Pure Ni and Cu, copper-rich specimens pre-melted in argon arc followed by homogenization melt in quartz tubes, nickel-rich alloys melted in argon arc only, ingots rolled and drawn to wires of 0.5 mm diameter and about 120 cm length, cut to 10 cm length, and annealed in vacuum at 1223 K for about 24 h; resistivities accurate to better than 0.1% and temperatures known to better than 1%; data extracted from tables.
15	151	Ahmad, H. M. and Greig, D.	1974		273	0-100	The above specimens.
16	218	Linz, R. J., Bowley, A. C., Klaffky, R. W., and Damon, D. H.	1975		4	4-20	Prepared from 99.999% pure Cu and 99.99% pure Ni, rods of 4 at. % Ni alloy swaged from 0.5 to 0.125 in. diameter and rods of other alloys swaged from 0.25 to 0.125 in. diameter at room temperature; measured in heavily deformed, cold-worked state; 1% absolute error in resistivity with precision typically within 0.1%; data obtained from 1 to 4 K was constant within measurement precision; data extracted from table.
17	218	Linz, R. J., Bowley, A. C., Klaffky, R. W., and Damon, D. H.	1975		4	4-20	The above specimens measured after annealing for 72 h within 20 K of the melting temperature i.e. 1348 K, 1353 K, and 1363 K for the 4, 10, and 20 at. % Ni specimens respectively; cooled slowly after anneal; data extracted from table.
18	219	Kaul, S. N.	1975		300	71.68-100	Spectrographically pure alloys from Johnson Matthey in the form of rectangular textured plates (25 x 5 x 0.3 mm <sup>2</sup> ) with their length along the rolling direction were the same alloys used in earlier work by Dutta-Roy and Subrahmanyam [Ni + Cu data sets 31-33]; pure Ni samples in order of decreasing resistivity were from E. Merck (nominal purity 99.9%), International Nickel Ltd. (99.99% pure), and Johnson Matthey and were of the same form as the alloys; annealed for 24 h at 1123 K in vacuum of $10^{-4}$ torr; data extracted from data table.

TABLE 26. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF COPPER-NICKEL ALLOY SYSTEM (Composition Dependence)  
[Composition, C, at/o Ni; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

C	$\rho$	DATA SET 1 (T = 77 K)	C	$\rho$	DATA SET 4 (T = 78 K)	C	$\rho$	DATA SET 8 (T = 4.2 K)	C	$\rho$	DATA SET 12 (T = 298.3 K)	C	$\rho$	DATA SET 16 (T = 4 K)
0.0	0.22	0.00	0.231	0.0	0.3	0.5	0.0	2.0	4	4.540				
3.2	3.1	1.11	1.039	1.2	1.3	16.0	10	11.263	10	11.263				
8.6	9.1	3.96	2.43	5.3	6.0	26.5	20	22.350	20	22.350				
10.7	11.0	10.2	11.9	5.3	6.7	31.7	40	43.8						
21.3	25.2	14.8	17.6	6.6	7.3	47.0	60	55.4						
31.7	35.0	21.1	27.1	11.6	13.1				DATA SET 17 (T = 4 K)					
41.9	46.1	31.6	40.3	12.2	13.6									
47.0	49.8	41.7	51.4	22.2	25.3									
52.0	49.8	51.7	54.9	24.1	27.0	6.23	6.947		4	4.920				
56.9	44.0	61.2	47.7	32.9	36.3	11.65	13.112		10	11.882				
61.9	38.1	71.2	28.0	44.0	48.4	23.90	26.592		20	22.974				
71.6	26.0	79.5	18.1					DATA SET 18 (T = 300 K)						
81.2	15.9	87.1	12.2	DATA SET 9 (T = 4.2 K)	0.00	0.10			71.68	41.60				
90.7	8.1	91.6	8.50	5.5	5.98				81.23	25.20				
100	0.65	95.7	5.00	12.8	14.0	0	0.0062		90.63	17.66				
		99.4	0.920	22.9	25.3	30	33.41		100	9.94 (Sample No. 2)				
						40	44.60		100	9.50 (Sample No. 3)				
						50	48.66		100	8.46 (Sample No. 1)				
						55	43.10							
						62	32.12							
						72	23.91							
						90.6	7.055							
						95	3.750							
						98	1.428							
						99	0.8627							
						100	0.0419							
								DATA SET 15 (T = 273 K)						
								0	1.593					
								30	34.89					
								40	43.87					
								50	50.08					
								55	49.92					
								62	47.39					
								72	40.72					
								90.6	15.123					
								95	10.84					
								98	8.064					
								99	7.297					
								100	6.313					

\* Not shown in figure.

### 3.5. Copper-Palladium Alloy System

The copper-palladium alloy system forms a continuous series of solid solutions over the entire range of compositions. However, ordered structures are formed at temperatures below about 775 K for compositions ranging from slightly below 10 to somewhat above 25 at.% (16 to 36 wt.%) palladium and at temperatures below about 975 K for compositions ranging from slightly below 30 to somewhat above 50 at.% (42 to 63 wt.%) palladium. The maxima of the temperatures of transformation suggest that these ordered structures are due to the formation of  $\text{PdCu}_5$  and  $\text{Pd}_3\text{Cu}_5$ , respectively. Structure ordering decreases the resistivity of these alloys.

There are 124 sets of experimental data available for this system, which are listed in tables 30, 32, and 34, tabulated in tables 31, 33, and 35, and shown in figures 25, 26, and 27. However, since many of the resistivity measurements were made to determine the maximum degree of order and the minimum value of electrical resistivity of individual alloys in this system with various compositions, a large portion of the data were not useful for the purpose of this study.

For this alloy system most of the data for disordered alloys are room-temperature values. The data of Svensson [221] (Cu-Pd data set 7) agree quite well with the pure element data and were chosen as the best resistivity versus composition curve at 291 K. Johansson and Linde [222] (Cu-Pd data set 3) also reported resistivity values for the entire spectrum of compositions; these data support the shape of Svensson's curve but are generally higher. At low Pd concentrations the agreement between the data of these authors is fairly good, but at concentrations above 60 wt.% Pd there is almost a 20% discrepancy. Pott's data [223] (Cu + Pd data sets 58-61 and Pd + Cu data sets 34-39) at 291 K agree quite well with the values from Svensson's isotherm. Köster and Lang [224] (Cu-Pd data sets 16-19) and Jaumot and Sawatzky [225] (Cu-Pd data sets 11-15) reported values only slightly higher than Svensson's.

At temperatures above and below 291 K there were not much useful data to work with. Otter [226] (Cu + Pd data sets 9-12) reported resistivity versus temperature for low Pd concentrations, but his data at higher Pd concentrations were limited and conflicting. As a result, a procedure similar to that used for the Cu-Au alloy system was followed. Since Cu, Ag, and Au lie in the same

column of the periodic table, it was assumed that their alloying properties with Pd should be similar. Accordingly, for a point on the 291 K resistivity versus composition isotherm a curve was drawn parallel to the recommended resistivity versus temperature curve for the same atomic percent of Ag in Pd. As in the case of the Cu-Au alloy system, values obtained in this way are only provisional. In the Ag-Pd alloy system at 65 to 70 wt.% Pd, the smoothed resistivity values as a function of temperature showed a change in slope (i.e., + to - to +). When these curves were followed for the Cu-Pd alloy system, a peculiar shape in the resistivity versus composition curves resulted. Since these values are only provisional, the bends were taken out so that the slope was always positive, and a family of resistivity versus composition isotherms similar to that of the Au-Pd alloy system was obtained.

Supporting data for the provisional values obtained by the above procedure are provided by Otter [226] (Cu + Pd data sets 9-12) at all temperatures for low Pd concentrations. At low temperatures Bäcklund [227] (Cu + Pd data sets 1-3) provides supporting data. The greatest uncertainty occurs between 60 and 80 wt.% Pd where few data exist and pure element resistivity provides no guidance.

The resulting recommended or provisional electrical resistivity values for Cu, Pd, and for 25 Cu-Pd binary alloys are presented in table 29 and shown in figures 23, 24, and 27. The recommended values for Cu and for Pd are for well-annealed high-purity specimens, but those values for temperatures below about 100 K are applicable only to Cu and Pd having residual electrical resistivities as given at 1 K in table 29. The alloys for which the recommended or provisional values are generated are not ordered and have not been quenched or cold-worked severely. The values cover a full range of temperature from 1 K to 1200 K, and are not corrected for the thermal expansion of the material. The estimated uncertainties in the values for the various alloys and for different temperature ranges are explicitly stated in a footnote to table 29. The values for the alloys at temperatures below 150 K and above 400 K are indicated as provisional because their uncertainties are greater than  $\pm 5\%$ .

TABLE 29. RECOMMENDED ELECTRICAL RESISTIVITY OF COPPER-PALLADIUM ALLOY SYSTEM†

[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-3} \Omega \text{ m}$ ]

Cu: 100.00% (100.00 At. %) Pd: 0.00% ( 0.00 At. %)		Cu: 99.50% (99.70 At. %) Pd: 0.50% ( 0.30 At. %)		Cu: 99.00% (99.40 At. %) Pd: 1.00% ( 0.60 At. %)		Cu: 97.00% (98.19 At. %) Pd: 3.00% ( 1.81 At. %)		Cu: 95.00% (96.95 At. %) Pd: 5.00% ( 3.05 At. %)		Cu: 90.00% (92.78 At. %) Pd: 10.00% ( 6.22 At. %)	
T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
1	0.00206	1	0.280**	1	0.580**	1	1.62**	1	2.70**	1	5.32**
4	0.00200	4	0.280*	4	0.580*	4	1.62*	4	2.70*	4	5.32*
7	0.00200	7	0.280*	7	0.580*	7	1.62*	7	2.70*	7	5.32*
10	0.00202	10	0.280**	10	0.580**	10	1.62**	10	2.70**	10	5.32**
15	0.00218	15	0.280**	15	0.580**	15	1.62**	15	2.70**	15	5.32**
20	0.00280	20	0.280**	20	0.580**	20	1.62**	20	2.70**	20	5.32**
25	0.00450	25	0.282**	25	0.580**	25	1.62**	25	2.70**	25	5.33*
30	0.00830	30	0.289**	30	0.580**	30	1.62**	30	2.70**	30	5.34*
40	0.0240	40	0.298**	40	0.592**	40	1.63**	40	2.70**	40	5.36*
50	0.0520	50	0.313**	50	0.617**	50	1.66**	50	2.71**	50	5.38*
60	0.0974	60	0.350**	60	0.652**	60	1.69**	60	2.74**	60	5.41*
70	0.154	70	0.407**	70	0.699**	70	1.75**	70	2.80**	70	5.46*
80	0.216	80	0.471**	80	0.763**	80	1.83**	80	2.86**	80	5.52*
90	0.282	90	0.540**	90	0.837**	90	1.90**	90	2.93**	90	5.60*
100	0.349	100	0.612**	100	0.909**	100	1.98**	100	2.99**	100	5.68*
150	0.701	150	0.962**	150	1.26*	150	2.35*	150	3.35*	150	6.07
200	1.048	200	1.33*	200	1.60*	200	2.68*	200	3.70*	200	6.40
250	1.388	250	1.67*	250	1.94*	250	3.01*	250	4.05*	250	6.74
273	1.544	273	1.83	273	2.10	273	3.17	273	4.21	273	6.88
293	1.678	293	1.96	293	2.23	293	3.31	293	4.35	293	7.03
300	1.725	300	2.01	300	2.27	300	3.36	300	4.40	300	7.08
350	2.061	350	2.34*	350	2.59	350	3.69	350	4.74	350	7.41
400	2.398	400	2.67*	400	2.92	400	4.02	400	5.08	400	7.74
500	3.079	500	3.32**	500	3.59*	500	4.67*	500	5.72*	500	9.40**
600	3.771	600	4.01**	600	4.29*	600	5.32*	600	6.39*	600	9.05**
700	4.481	700	4.72**	700	5.00*	700	6.03*	700	7.09*	700	9.20**
800	5.213	800	5.47**	800	5.74*	800	6.78*	800	7.82*	800	10.39**
900	5.973	900	6.25**	900	6.51*	900	7.53*	900	8.60*	900	11.12**
1000	6.766	1000	7.06**	1000	7.29**	1000	8.34**	1000	9.43**	1000	11.91**
1100	7.596	1100	7.88**	1100	8.11**	1100	9.20**	1100	10.30**	1100	12.74**
1200	8.470	1200	8.71**	1200	9.00**	1200	10.08**	1200	11.18**	1200	13.62**
1300	9.395										
1357.6	9.946(a)										
1358	21.01(c)										
1700	24.41										

† Uncertainties in the electrical resistivity values are as follows:

100.00 Cu - 0.00 Pd:  $\pm 5\%$  up to 100K,  $\pm 1\%$  above 100K to 250K,  $\pm 0.5\%$  above 250K to 350K,  $\pm 1\%$  above 350K to 500K,  $\pm 0.5\%$  above 500K to 1357.6K, and  $\pm 9\%$  above 1357.6K.  
 99.50 Cu - 0.50 Pd:  $\pm 5\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 99.00 Cu - 1.00 Pd:  $\pm 5\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 97.00 Cu - 3.00 Pd:  $\pm 5\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 95.00 Cu - 5.00 Pd:  $\pm 5\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 90.00 Cu - 10.00 Pd:  $\pm 5\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.

\* Provisional value.

\* In temperature range where no experimental data are available.

TABLE 29. RECOMMENDED ELECTRICAL RESISTIVITY OF COPPER-PALLADIUM ALLOY SYSTEM† (continued)

[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Cu: 85.00% (90.47 At.%) Pd: 15.00% (9.53 At.%)			Cu: 80.00% (87.01 At.%) Pd: 20.00% (12.99 At.%)			Cu: 75.00% (83.40 At.%) Pd: 25.00% (16.60 At.%)			Cu: 70.00% (79.62 At.%) Pd: 30.00% (20.38 At.%)			Cu: 65.00% (75.67 At.%) Pd: 35.00% (24.33 At.%)			Cu: 60.00% (71.52 At.%) Pd: 40.00% (28.48 At.%)		
T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$	
1	7.91*		1	10.43*		1	12.90*		1	15.30*		1	17.68*		1	20.01*	
4	7.91*		4	10.43*		4	12.90*		4	15.30*		4	17.68*		4	20.01*	
7	7.91*		7	10.43*		7	12.90*		7	15.30*		7	17.68*		7	20.01*	
10	7.91*		10	10.43*		10	12.90*		10	15.30*		10	17.68*		10	20.01*	
15	7.91*		15	10.43*		15	12.90*		15	15.30*		15	17.68*		15	20.01*	
20	7.91*		20	10.43*		20	12.90*		20	15.30*		20	17.68*		20	20.01*	
25	7.92*		25	10.43*		25	12.91*		25	15.31*		25	17.68*		25	20.02*	
30	7.92*		30	10.43*		30	12.91*		30	15.32*		30	17.69*		30	20.03*	
40	7.94*		40	10.44*		40	12.92*		40	15.34*		40	17.70*		40	20.05*	
50	7.98*		50	10.46*		50	12.96*		50	15.38*		50	17.71*		50	20.10*	
60	8.02*		60	10.50*		60	13.00*		60	15.42*		60	17.74*		60	20.16*	
70	8.09*		70	10.55*		70	13.05*		70	15.47*		70	17.79*		70	20.23*	
80	8.16*		80	10.61*		80	13.11*		80	15.53*		80	17.85*		80	20.30*	
90	8.23*		90	10.67*		90	13.19*		90	15.60*		90	17.92*		90	20.38*	
100	8.30*		100	10.74*		100	13.28*		100	15.67*		100	17.98*		100	20.43*	
150	8.66		150	11.11*		150	13.62*		150	16.01		150	18.33*		150	20.82	
200	9.00		200	11.47*		200	13.97*		200	16.37		200	18.71*		200	21.20	
250	9.32		250	11.81*		250	14.32*		250	16.73		250	19.09*		250	21.56	
273	9.48		273	11.99		273	14.48		273	16.87		273	19.26		273	21.73	
293	9.61		293	12.12		293	14.62		293	17.01		293	19.41		293	21.87	
300	9.66		300	12.16		300	14.67		300	17.06		300	19.46		300	21.92	
350	10.01		350	12.51*		350	15.03*		350	17.41		350	19.82		350	22.30	
400	10.36		400	12.87		400	15.39		400	17.78		400	20.17		400	22.69	
500	11.00*		500	13.51*		500	16.07*		500	18.47*		500	20.92*		500	23.39*	
600	11.64*		600	14.10*		600	16.70*		600	19.10*		600	21.44*		600	24.01*	
700	12.28*		700	14.78*		700	17.32*		700	19.72*		700	22.06*		700	24.61*	
800	12.98*		800	15.49*		800	17.96*		800	20.35*		800	22.68*		800	25.23*	
900	13.70*		900	16.18*		900	18.60*		900	21.00*		900	23.31*		900	25.88*	
1000	14.48*		1000	16.91*		1000	19.24*		1000	21.68*		1000	23.95*		1000	26.51*	
1100	15.31*		1100	17.70*		1100	19.91*		1100	22.39*		1100	24.62*		1100	27.20*	
1200	16.20*		1200	18.55*		1200	20.61*		1200	23.11*		1200	25.31*		1200	27.91*	

† Uncertainties in the electrical resistivity values are as follows:

- 85.00 Cu - 15.00 Pd:  $\pm 8\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 80.00 Cu - 20.00 Pd:  $\pm 8\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 75.00 Cu - 25.00 Pd:  $\pm 8\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 70.00 Cu - 30.00 Pd:  $\pm 8\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 65.00 Cu - 35.00 Pd:  $\pm 8\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 60.00 Cu - 40.00 Pd:  $\pm 8\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.

\* Provisional value.

\* In temperature range where no experimental data are available.

TABLE 29. RECOMMENDED ELECTRICAL RESISTIVITY OF COPPER-PALLADIUM ALLOY SYSTEM† (continued)  
[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-9} \Omega \text{ m}$ ]

Cu: 55.00% (57.18 At. %) Pd: 45.00% (32.82 At. %)			Cu: 50.00% (52.61 At. %) Pd: 50.00% (37.39 At. %)			Cu: 45.00% (57.81 At. %) Pd: 55.00% (42.19 At. %)			Cu: 40.00% (52.75 At. %) Pd: 60.00% (47.25 At. %)			Cu: 35.00% (47.41 At. %) Pd: 65.00% (52.59 At. %)			Cu: 30.00% (41.78 At. %) Pd: 70.00% (58.22 At. %)		
T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$	
1	22.60*		1	25.53*		1	29.00*		1	32.63*		1	40.00*		1	44.19*	
4	22.60*		4	25.53*		4	29.00*		4	32.63*		4	40.00*		4	44.19*	
7	22.60*		7	25.53*		7	29.00*		7	32.63*		7	40.00*		7	44.19*	
10	22.60*		10	25.53*		10	29.00*		10	32.63*		10	40.00*		10	44.19*	
15	22.60*		15	25.53*		15	29.00*		15	32.63*		15	40.00*		15	44.19*	
20	22.60*		20	25.53*		20	29.00*		20	32.63*		20	40.00*		20	44.19*	
25	22.60*		25	25.53*		25	29.00*		25	32.63*		25	40.00*		25	44.19*	
30	22.60*		30	25.53*		30	29.00*		30	32.63*		30	40.00*		30	44.19*	
40	22.60*		40	25.53*		40	29.00*		40	32.63*		40	40.00*		40	44.19*	
50	22.60*		50	25.53*		50	29.00*		50	32.63*		50	40.00*		50	44.19*	
60	22.60*		60	25.53*		60	29.00*		60	32.63*		60	40.00*		60	44.19*	
70	22.60*		70	25.53*		70	29.00*		70	32.63*		70	40.00*		70	44.19*	
80	22.60*		80	25.53*		80	29.00*		80	32.63*		80	40.00*		80	44.19*	
90	22.60*		90	25.53*		90	29.00*		90	32.63*		90	40.00*		90	44.19*	
100	22.60*		100	25.53*		100	29.00*		100	32.63*		100	40.00*		100	44.19*	
150	23.45*		150	26.52*		150	30.03*		150	34.05*		150	41.26*		150	45.47*	
200	23.87*		200	26.97*		200	30.48*		200	34.58*		200	41.69*		200	45.90*	
250	24.27*		250	27.42*		250	30.92*		250	35.08*		250	42.11*		250	46.31*	
273	24.47		273	27.62		273	31.12		273	35.31		273	42.31		273	46.50	
293	24.63		293	27.79		293	31.30		293	35.51		293	42.48		293	46.66	
300	24.68		300	27.86		300	31.36		300	35.57		300	42.53		300	46.71	
350	25.05*		350	28.25		350	31.29		350	36.03		350	42.97		350	47.11	
400	25.42*		400	28.64		400	32.22		400	36.47		400	43.38		400	47.47	
500	26.12*		500	29.40*		500	30.04*		500	37.30*		500	44.16*		500	48.13*	
600	26.78*		600	30.10*		600	33.90*		600	38.09*		600	44.90*		600	48.73*	
700	27.41*		700	30.75*		700	34.52*		700	38.82*		700	45.59*		700	49.27*	
800	28.05*		800	31.38*		800	35.20*		800	39.52*		800	46.21*		800	49.74*	
900	28.72*		900	32.01*		900	35.84*		900	40.20*		900	46.77*		900	50.23*	
1000	29.41*		1000	32.68*		1000	36.50*		1000	40.88*		1000	47.27*		1000	50.68*	
1100	30.13*		1100	33.37*		1100	37.19*		1100	41.60*		1100	47.71*		1100	51.07*	
1200	30.89*		1200	34.07*		1200	37.90*		1200	42.36*		1200	48.08*		1200	51.40*	

† Uncertainties in the electrical resistivity values are as follows:

- 55.00 Cu - 45.00 Pd:  $\pm 5\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.
- 50.00 Cu - 50.00 Pd:  $\pm 5\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.
- 45.00 Cu - 55.00 Pd:  $\pm 10\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.
- 40.00 Cu - 60.00 Pd:  $\pm 10\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.
- 35.00 Cu - 65.00 Pd:  $\pm 10\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.
- 30.00 Cu - 70.00 Pd:  $\pm 10\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.

\* Provisional value.

† In temperature range where no experimental data are available.

TABLE 29. RECOMMENDED ELECTRICAL RESISTIVITY OF COPPER-PALLADIUM ALLOY SYSTEM† (continued)

[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-4} \Omega \text{ m}$ ]

Cu: 25.00% (35.83 At. %) Pd: 75.00% (64.18 At. %)		Cu: 20.00% (29.51 At. %) Pd: 80.00% (70.49 At. %)		Cu: 15.00% (22.81 At. %) Pd: 85.00% (77.19 At. %)		Cu: 10.00% (15.69 At. %) Pd: 90.00% (84.31 At. %)		Cu: 5.00% (8.10 At. %) Pd: 95.00% (91.90 At. %)		Cu: 3.00% (4.92 At. %) Pd: 97.00% (95.08 At. %)	
T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
1	42.40*	1	36.26*	1	28.68*	1	20.10*	1	10.31*	1	6.20*
4	42.40*	4	36.26*	4	28.68*	4	20.10*	4	10.31*	4	6.20*
7	42.45*	7	36.33*	7	28.78*	7	20.17*	7	10.34*	7	6.20*
10	42.52*	10	36.43*	10	28.93*	10	20.28*	10	10.39*	10	6.20*
15	42.64*	15	36.59*	15	29.13*	15	20.44*	15	10.46*	15	6.21*
20	42.78*	20	36.75*	20	29.31*	20	20.58*	20	10.54*	20	6.25*
25	42.88*	25	36.90*	25	29.47*	25	20.72*	25	10.63*	25	6.30*
30	42.97*	30	37.03*	30	29.61*	30	20.85*	30	10.74*	30	6.35*
40	43.12*	40	37.29*	40	29.92*	40	21.11*	40	11.00*	40	6.60*
50	43.30*	50	37.54*	50	30.23*	50	21.40*	50	11.31*	50	6.89*
60	43.49*	60	37.80*	60	30.54*	60	21.70*	60	11.64*	60	7.24*
70	43.65*	70	38.06*	70	30.85*	70	22.00*	70	11.97*	70	7.66*
80	43.81*	80	38.32*	80	31.17*	80	22.31*	80	12.32*	80	8.09*
90	43.97*	90	38.58*	90	31.48*	90	22.64*	90	12.69*	90	8.52*
100	44.12*	100	38.83*	100	31.79*	100	23.00*	100	13.09*	100	8.97*
150	44.82*	150	40.06*	150	33.29	150	24.79*	150	15.12*	150	11.12*
200	45.45*	200	41.16*	200	34.66	200	26.52*	200	17.15*	200	13.20*
250	46.01*	250	42.13*	250	35.95	250	28.18*	250	19.11*	250	15.20*
300	46.25	300	42.54	300	36.52	300	28.90	300	20.00	300	16.09
350	46.45	350	42.87	350	36.99	350	29.51	350	20.75	350	16.85
400	46.52	400	43.74*	400	37.16	400	29.73	400	21.07	400	17.11
450	46.59*	450	44.16*	450	38.28	450	31.19*	450	22.84*	450	18.96*
500	46.73*	500	44.47*	500	39.35	500	32.56*	500	24.54*	500	20.76*
550	46.80*	550	44.74*	550	40.42*	550	33.09*	550	27.64*	550	24.20*
600	46.85*	600	44.99*	600	41.27*	600	34.67*	600	30.49*	600	27.36*
700	47.41*	700	47.68*	700	44.34*	700	39.65*	700	33.18*	700	30.23*
800	48.04*	800	48.48*	800	45.66*	800	41.60*	800	35.70*	800	32.90*
900	50.42*	900	49.22*	900	46.90*	900	43.35*	900	38.11*	900	35.38*
1000	50.85*	1000	49.91*	1000	48.10*	1000	45.02*	1000	40.45*	1000	37.78*
1100	51.21*	1100	50.54*	1100	49.23*	1100	46.62*	1100	42.88*	1100	40.11*
1200	51.49*	1200	51.07*	1200	50.26*	1200	48.11*	1200	44.76*	1200	42.33*

† Uncertainties in the electrical resistivity values are as follows:

- 25.00 Cu - 75.00 Pd:  $\pm 10\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 20.00 Cu - 80.00 Pd:  $\pm 10\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 15.00 Cu - 85.00 Pd:  $\pm 10\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 10.00 Cu - 90.00 Pd:  $\pm 10\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 5.00 Cu - 95.00 Pd:  $\pm 10\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.  
 3.00 Cu - 97.00 Pd:  $\pm 10\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.

\* Provisional value.

\* In temperature range where no experimental data are available.



TABLE 29. RECOMMENDED ELECTRICAL RESISTIVITY OF COPPER-PALLADIUM ALLOY SYSTEM† (continued)  
[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Cu: 1.00% (1.66 At. %) Pd: 99.00% (98.34 At. %)		Cu: 0.50% (0.83 At. %) Pd: 99.50% (98.17 At. %)		Cu: 0.00% (0.00 At. %) Pd: 100.00% (100.00 At. %)			
T	$\rho$	T	$\rho$	T	$\rho$		
1	2.10*	1	1.10*	1	0.0200		
4	2.10*	4	1.10*	4	0.0205		
7	2.10*	7	1.10*	7	0.0217		
10	2.10*	10	1.10*	10	0.0242		
15	2.10*	15	1.10*	15	0.0346		
20	2.11*	20	1.10*	20	0.0564		
25	2.14*	25	1.13*	25	0.0938		
30	2.21*	30	1.19*	30	0.151		
40	2.45*	40	1.40*	40	0.335		
50	2.78*	50	1.68*	50	0.607		
60	3.15*	60	2.00*	60	0.940		
70	3.32*	70	2.38*	70	1.32		
80	3.92*	80	2.79*	80	1.75		
90	4.34*	90	3.23*	90	2.19		
100	4.78*	100	3.68*	100	2.63		
150	6.91*	150	5.85*	150	4.81		
200	9.00*	200	7.92*	200	6.89		
250	11.00	250	9.91*	250	8.88		
273	11.90	273	10.29	273	9.78		
293	12.87	293	11.55	293	10.54		
300	12.83*	300	11.81	300	10.80		
350	14.82*	350	13.69*	350	12.66		
400	16.88*	400	15.54*	400	14.46		
500	20.19*	500	19.05*	500	17.89		
600	23.41*	600	22.28*	600	21.10		
700	26.42*	700	25.26*	700	24.10		
800	29.25*	800	28.07*	800	26.89		
900	31.7*	900	30.68*	900	29.50		
1000	34.27*	1000	33.09*	1000	31.92		
1100	36.46*	1100	35.32*	1200	36.21		
1200	38.50*	1200	37.40*	1400	39.80		
				1600	42.70		
				1827	45.14(a)		
				1830	82.0(4)		
				2000	83.0		

† Uncertainties in the electrical resistivity values are as follows:

1.00 Cu - 99.00 Pd:  $\pm 10\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.

0.50 Cu - 99.50 Pd:  $\pm 10\%$  below 150 K,  $\pm 5\%$  from 150 to 400 K, and  $\pm 10\%$  above 400 K.

0.00 Cu - 100.00 Pd:  $\pm 2\%$  up to 40 K,  $\pm 1\%$  above 40 K to 350 K,  $\pm 2\%$  above 350 K to 1600 K,  $\pm 2.5\%$  above 1600 K to 1837 K, and  $\pm 5\%$  above 1837 K.

\* Provisional value.

† In temperature range where no experimental data are available.

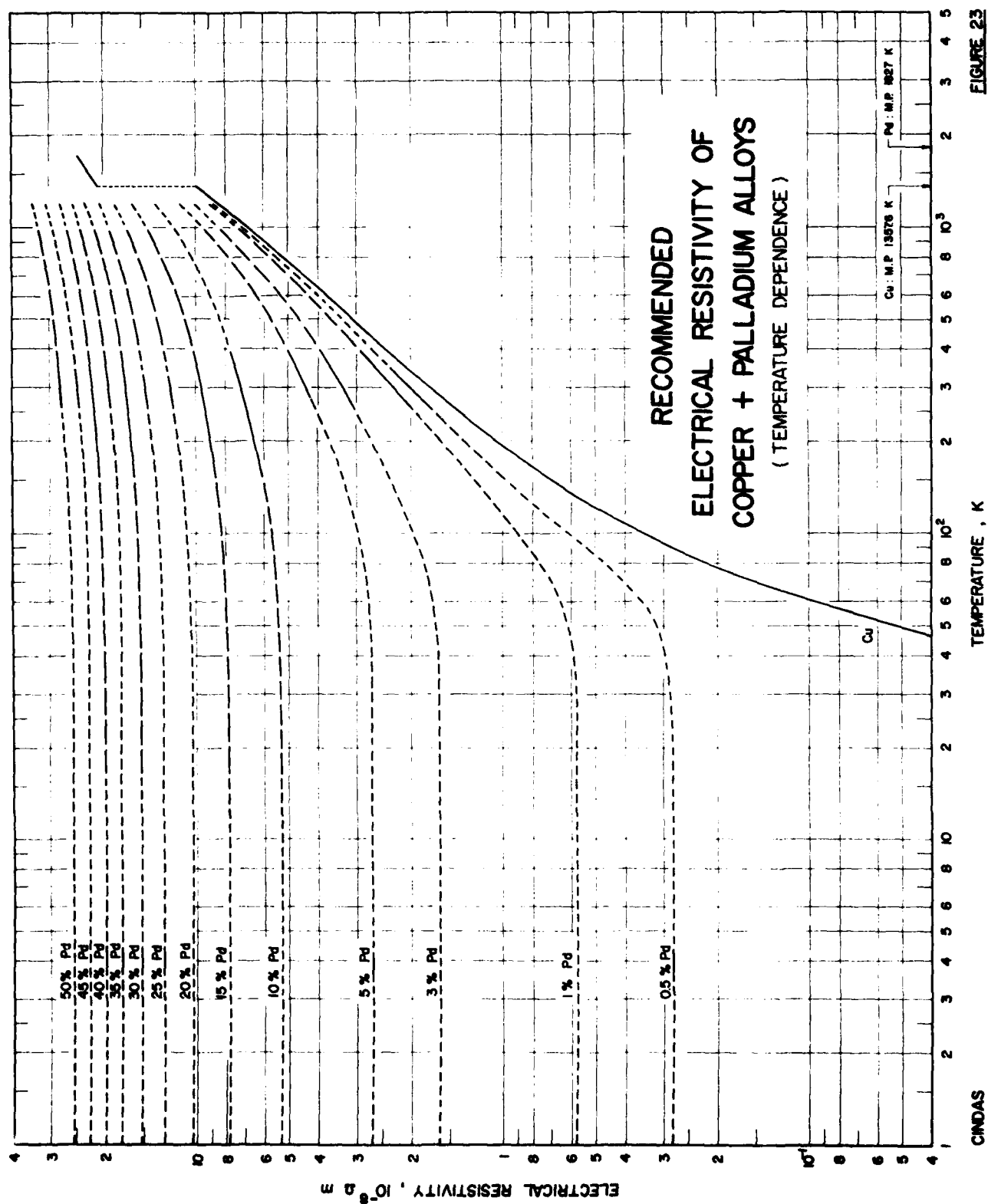
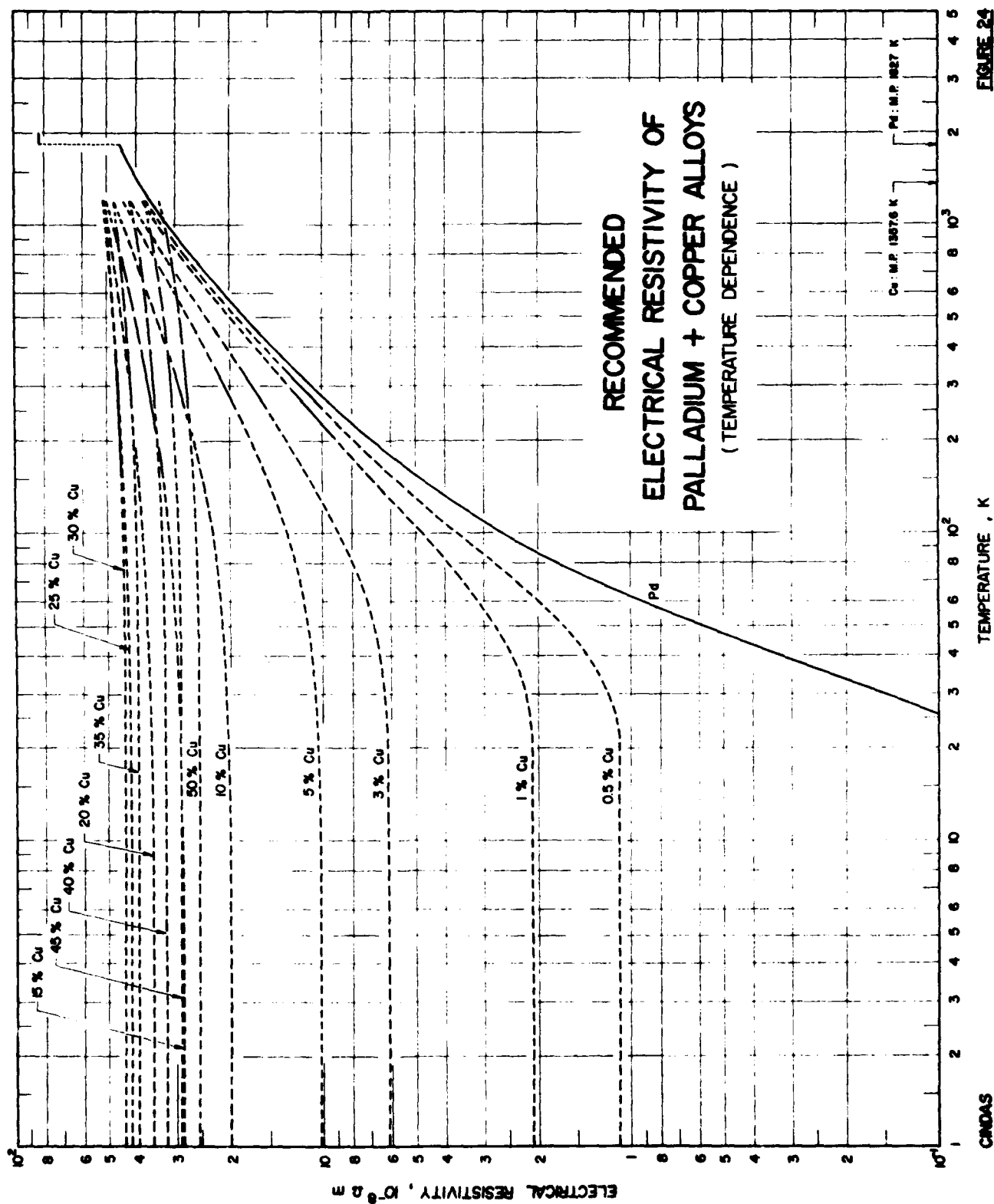


FIGURE 23



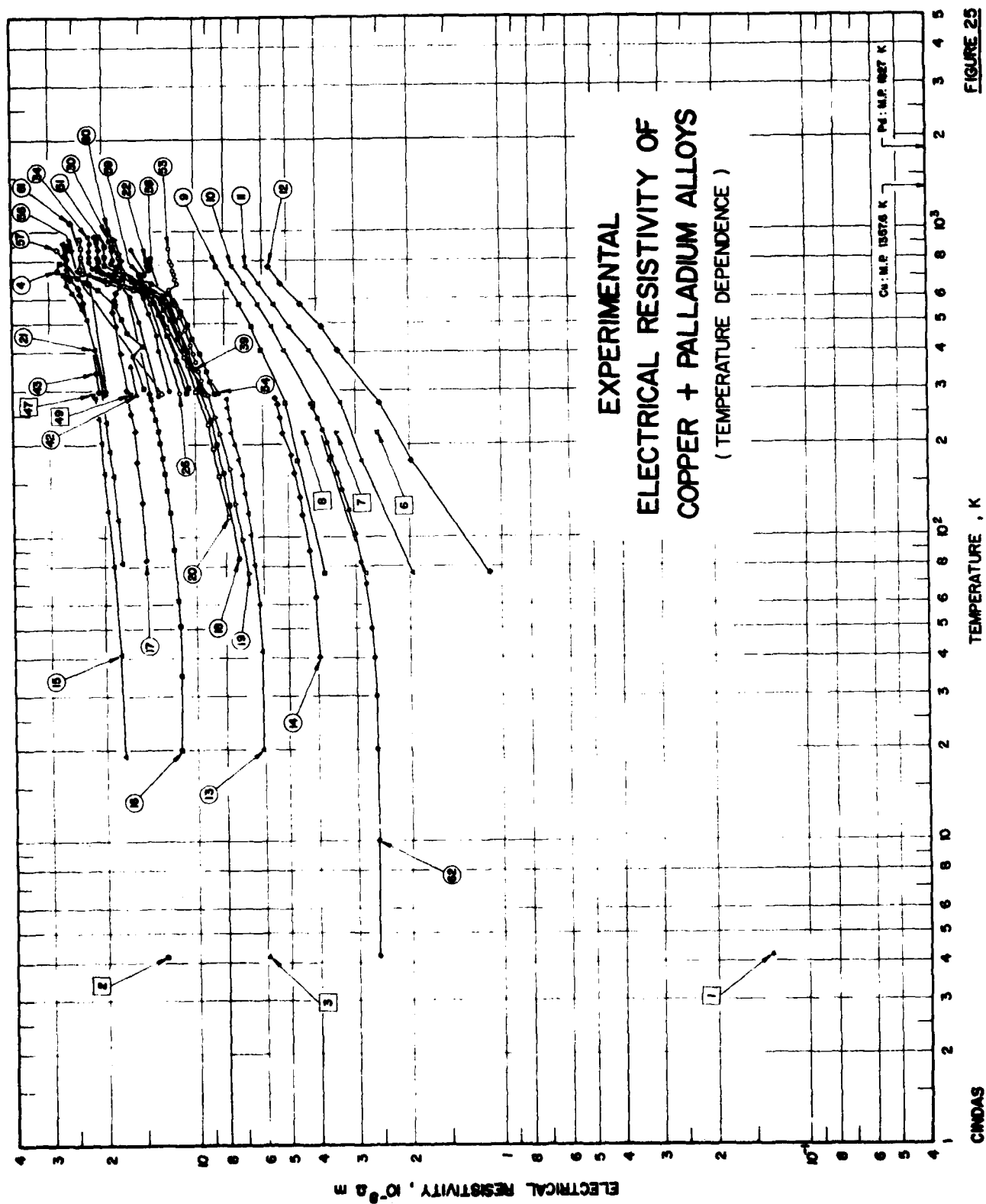


FIGURE 25

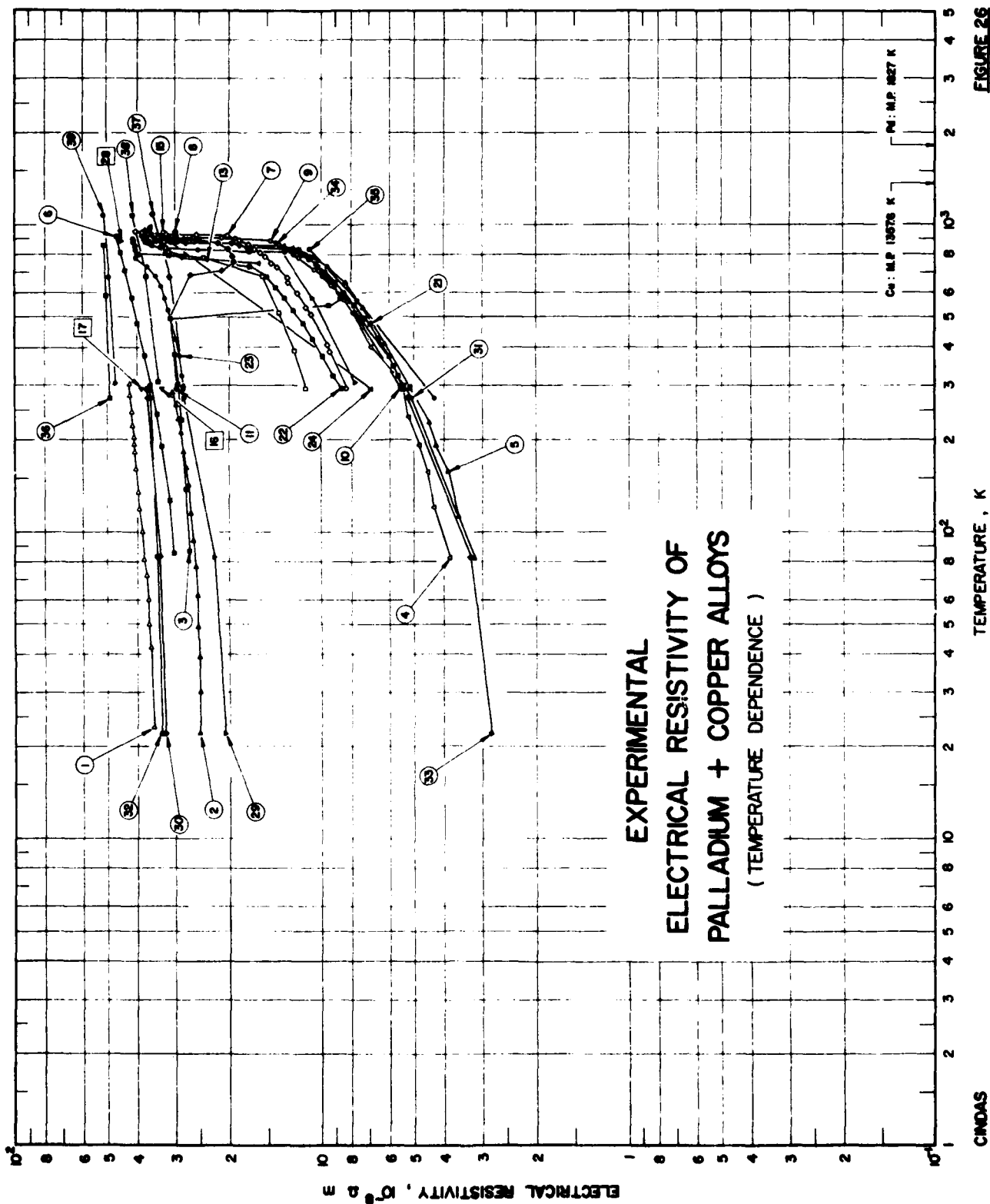


TABLE 30. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + PALLADIUM ALLOYS (Temperature Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Pd	Composition (continued), Specifications, and Remarks
1	227	Bücklund, N.	1958		4.2		0.318	Calculated composition (0.19 a/o Pd); prepared from spectroscopically pure materials supplied by Johnson, Matthey and Co. Ltd., London, by melting the components in evacuated silica tubes in a high-frequency furnace; after slight rolling, the ingot homogenized at 800-900 C for about 2 hr, drawn to wires of about 0.2 mm in diameter, then annealed at about 500 C prior to the measurements.
2	227	Bücklund, N.	1958		4.2		25.54	Calculated composition (17.0 a/o Pd); similar to the above specimen.
3	227	Bücklund, N.	1958		4.2		25.54	Similar to the above specimen; ordered.
4	228	Yonezumi, K. and Sato, T.	1958	A	300-792		35.83	Calculated composition (25 a/o Pd); 18.0 x 5.5 x 0.17 mm; measurements made while being cooled from temperature above Curie point with the cooling rate of 7.5, 8.5, 3, 20, and 5.0 K h <sup>-1</sup> at 733-713 K, 713-708 K, 708-678 K, 678-573 K, and 573-room temperature, respectively.
5*	228	Yonezumi, K. and Sato, T.	1958	A	300-611		35.82	The above specimen kept at 661 K for 27 hr, and at 539 K for 16 hr; same measuring condition as above.
6	126	Linde, J.O.	1932		291		1.66	Calculated composition (1.00 a/o Pd); prepared from Cu supplied by Iffigen, London, spectroscopically pure.
7	126	Linde, J.O.	1932		291		3.39	Calculated composition (2.05 a/o Pd); prepared from Cu supplied by Kahlbaum.
8	126	Linde, J.O.	1932		291		4.94	Calculated composition (3.01 a/o Pd); prepared from Cu supplied by Kahlbaum.
9	226	Otter, F.A., Jr.	1956		76-767		6.66	Calculated composition (4.09 a/o Pd); prepared from 99.99% copper supplied by Vacuum Metals Corp. and 99.99% palladium supplied by Bishop Platinum Company; components melted in vacuo in an aluminum crucible within an induction furnace, super heated approximately 100 C above the melting point, homogenized in vacuo for twenty four hours at 900 C, ingot swaged or drawn to approximately thirty mils in diameter; annealed for about one hour at 500 to 550 C before measuring.
10	226	Otter, F.A., Jr.	1956		76-767		5.00	Calculated composition (3.05 a/o Pd); similar to the above specimen.
11	226	Otter, F.A., Jr.	1956		76-767		3.32	Calculated composition (2.01 a/o Pd); similar to the above specimen.
12	226	Otter, F.A., Jr.	1956		76-767		1.66	Calculated composition (1.00 a/o Pd); similar to the above specimen.
13	229	Ugodnikova, L.A., Volkenshteyn, N.V., and Talovkin, Yu.N.	1971	A	20-280		84.31 15.69	Calculated composition (10 a/o Pd); cast and drawn; quenched in water from 600 K.
14	229	Ugodnikova, L.A., et al.	1971	A	40-284		91.9 8.1	Calculated composition (5 a/o Pd); similar to the above specimen.
15	229	Ugodnikova, L.A., et al.	1971	A	22-300		58.23 41.77	Calculated composition (30 a/o Pd); similar to the above specimen.
16	229	Ugodnikova, L.A., et al.	1971	A	19-282		70.49 29.51	Calculated composition (20 a/o Pd); similar to the above specimen.

\* Not shown in figure.

TABLE 30. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + PALLADIUM ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Pd	Composition (continued), Specifications, and Remarks
17	230	Remollet, G. and Seemann, H.J.	1969	A	84-960		70.5 29.5	Calculated composition (20 a/o Pd); cold-rolled; quenched from 700 C.
18	230	Remollet, G. and Seemann, H.J.	1969	A	85-931		70.5 29.5	The above specimen heated to 750 C, then cooled to room temperature in 21 days.
19	230	Remollet, G. and Seemann, H.J.	1969	A	76-944		77.19 22.81	Calculated composition (15 a/o Pd); cold-rolled.
20	230	Remollet, G. and Seemann, H.J.	1969	A	85-942		64.18 35.82	Calculated composition (25 a/o Pd); cold-rolled.
21	230	Remollet, G. and Seemann, H.J.	1969	A	82-917		58.23 41.77	Calculated composition (30 a/o Pd); cold-rolled.
22	123	Borelius, G., Johansson, C.H., and Linde, J.O.	1928	A	295-803		16.86	Calculated composition (10.8 a/o Pd); 20 x 1 x 1 mm; constituent metals melted in quartz tube, ingot heat-treated near melting point, cooled in a way to maintain rollable condition, rolled to 1 x 1 mm square wire, again heat-treated near melting point.
23*	123	Borelius, G., et al.	1928	A	524-803		16.86	The above specimen measured during cooling.
24*	123	Borelius, G., et al.	1928	A	524-764		16.86	The above specimen remeasured.
25*	123	Borelius, G., et al.	1928	A	287-771		16.86	The above specimen measured during cooling.
26	123	Borelius, G., et al.	1928	A	295-852		22.53	Calculated composition (14.8 a/o Pd); same dimensions, fabrication method, and heat-treatment as the above specimen.
27*	123	Borelius, G., et al.	1928	A	715-852		22.53	The above specimen measured during cooling.
28*	123	Borelius, G., et al.	1928	A	715-810		22.53	The above specimen remeasured.
29*	123	Borelius, G., et al.	1928	A	295-810		22.53	The above specimen measured during cooling.
30	123	Borelius, G., et al.	1928	A	300-980		25.54	Calculated composition (17.0 a/o Pd); same dimensions, fabrication method, and heat-treatment as the above specimen.
31*	123	Borelius, G., et al.	1928	A	295-980		25.54	The above specimen measured during cooling.
32*	123	Borelius, G., et al.	1928	A	298-982		28.20	Calculated composition (19.0 a/o Pd); same dimensions, fabrication method, and heat-treatment as the above specimen.
33*	123	Borelius, G., et al.	1928	A	293-993		30.80	Calculated composition (21.0 a/o Pd); similar to the above specimen.
34	123	Borelius, G., et al.	1928	A	289-962		30.80	The above specimen remeasured.
35*	123	Borelius, G., et al.	1928	A	294-962		30.80	The above specimen measured during cooling.
36*	123	Borelius, G., et al.	1928	A	299-959		33.34	Calculated composition (23.0 a/o Pd); same dimensions, fabrication method, and heat-treatment as the above specimen.
37*	123	Borelius, G., et al.	1928	A	293-758		33.34	The above specimen measured during cooling.
38*	123	Borelius, G., et al.	1928	A	294-742		35.82	Calculated composition (25.0 a/o Pd); same dimensions, fabrication method, and heat-treatment as the above specimen.
39	123	Borelius, G., et al.	1928	A	351-742		35.82	The above specimen measured during cooling.
40*	123	Borelius, G., et al.	1928	A	290-894		35.82	The above specimen remeasured.

\* Not shown in figure.

TABLE 30. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + PALLADIUM ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Pd	Composition (continued), Specifications, and Remarks
41*	123	Borelius, G., Johansson, C. H., and Linde, J. O.	1928	A	728-754		35.82	The above specimen measured during cooling.
42	123	Borelius, G., et al.	1928	A	293, 363		35.82	The above specimen.
43	123	Borelius, G., et al.	1928	A	296-899		39.44	Calculated composition (28.0 a/o Pd); same dimensions, fabrication method, and heat-treatment as the above specimen.
44*	123	Borelius, G., et al.	1928	A	299-692		39.44	The above specimen remeasured.
45*	123	Borelius, G., et al.	1928	A	670-750		39.44	The above specimen.
46*	123	Borelius, G., et al.	1928	A	285-750		39.44	The above specimen measured during cooling.
47	131	Röhl, H.	1933		291.2		35.82	Calculated composition (25 a/o Pd); 0.3 cm diameter x 11 cm long; obtained from Firma Siebert-Hansen; drawn.
48*	131	Röhl, H.	1933		291.2		35.82	The above specimen quenched.
49	131	Röhl, H.	1933		291.2		35.82	The above specimen tempered.
50*	131	Röhl, H.	1933		291.2		35.82	The above specimen reannealed.
51	231	Taylor, R.	1934	A	293-898		29.50	Calculated composition (19.95 a/o Pd); electrolytic copper (major impurity 0.002 %) and palladium sponge melted in high vacuum in a high frequency induction furnace, cooled and drawn into wires.
52*	231	Taylor, R.	1934	A	293-898		29.50	The above specimen measured at decreasing temperatures.
53	231	Taylor, R.	1934	A	293-873		15.65	Calculated composition (9.95 a/o Pd); similar to the above specimen.
54	231	Taylor, R.	1934	A	293, 873		23.10	Calculated composition (15 a/o Pd); similar to the above specimen.
55*	231	Taylor, R.	1934	A	293, 873		36.08	Calculated composition (25 a/o Pd); similar to the above specimen.
56	231	Taylor, R.	1934	A	293, 873		41.96	Calculated composition (30 a/o Pd); similar to the above specimen.
57	231	Taylor, R.	1934	A	293, 873		47.74	Calculated composition (35 a/o Pd); similar to the above specimen.
58	223	Pott, F. P.	1958		309-753		24.2	Calculated composition (16 a/o Pd); annealed at 600 to 700 C for 2 hr; ordered.
59	223	Pott, F. P.	1958		307-722		24.2	Similar to the above specimen but disordered.
60	223	Pott, F. P.	1958		292-1012		35.8	Calculated composition (25 a/o Pd); similar to the above specimen but ordered.
61	223	Pott, F. P.	1958		298-1073		35.8	Similar to the above specimen but disordered.
62*	186	Kierap, W.	1967	V	4.2-273		4.92	Calculated composition (3.0 a/o Pd); 3 mm diameter cylindrical specimen.

\* Not shown in figure.



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ELECTRICAL RESISTIVITY OF TEN SELECTED BINARY ALLOY  
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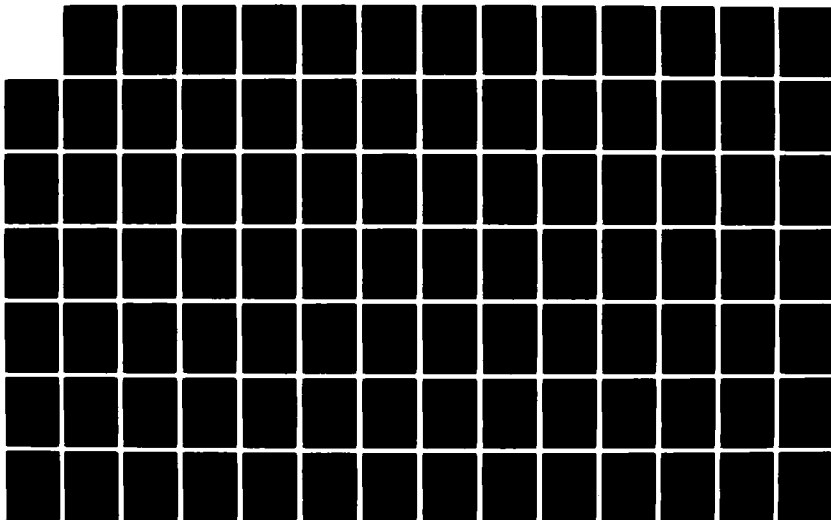
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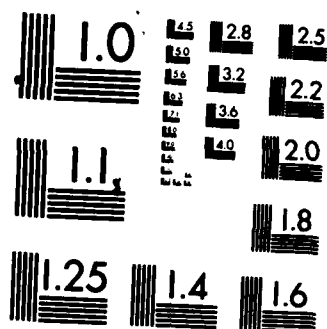
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**EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF COPPER + PALLADIUM ALLOYS (Temperature Dependence) (continued)**

[illegible]

\* Not shown in figure.



TABLE 32. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF PALLADIUM + COPPER ALLOYS (Temperature Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Pd Cu	Composition (continued), Specifications, and Remarks
1	229	Ugodnikova, L.A., Volkenshtein, N.V., and Tsvetkov, Yu. N.	1971	A	23-304		71.49 28.51	Calculated composition (40 a/o Cu); cast and drawn; quenched in water from 600 K.
2	229	Ugodnikova, L.A., et al.	1971	A	22-285		86.95 13.05	Calculated composition (20 a/o Cu); similar to the above specimen.
3	230	Remollet, G. and Seemann, H.J.	1969	A	87-959		54.8 45.20	Calculated composition (58 a/o Cu); cold-rolled; quenched from 700 C.
4	230	Remollet, G. and Seemann, H.J.	1969	A	82-890		54.8 45.20	The above specimen heated to 760 C, then cooled to room temperature in 21 days.
5	230	Remollet, G. and Seemann, H.J.	1969	A	82-851		57.76 42.23	Calculated composition (55 a/o Cu); cold-rolled.
6	230	Remollet, G. and Seemann, H.J.	1969	A	85-917		86.95 13.05	Calculated composition (20 a/o Cu); cold-rolled.
7	123	Borelius, G., Johansson, C. H., and Linde, J.O.	1928	A	293-947		62.42	Calculated composition (49.8 a/o Pd); 20 x 1 x 1 mm; metals melted in quartz tube, ingot heat-treated near melting point, cooled in a way to maintain rollable condition, rolled to 1 x 1 square wire, again heat-treated near melting point.
8	123	Borelius, G., et al.	1928	A	834-942		62.42	The above specimen measured during cooling.
9	123	Borelius, G., et al.	1928	A	297-926		56.72	Calculated composition (43.9 a/o Pd); same dimensions, fabrication method, and heat-treatment as the above specimen.
10	123	Borelius, G., et al.	1928	A	293-926		56.72	The above specimen measured during cooling.
11	123	Borelius, G., et al.	1928	A	292-897		50.65	Calculated composition (38.0 a/o Pd); same dimensions, fabrication method, and heat-treatment as the above specimen.
12*	123	Borelius, G., et al.	1928	A	859-906		50.65	The above specimen measured during cooling.
13	123	Borelius, G., et al.	1928	A	748-932		50.65	The above specimen measured during cooling.
14*	123	Borelius, G., et al.	1928	A	748-942		50.65	The above specimen remeasured.
15	123	Borelius, G., et al.	1928	A	291-942		50.65	The above specimen measured during cooling.
16	131	Röhl, H.	1933		291.2		62.58 37.42	Calculated composition (50 a/o Pd); 0.3 cm diameter x 11 cm long; obtained from Firma Siebert-Hannu; cold-drawn.
17	131	Röhl, H.	1933		291.2		62.58 37.42	The above specimen quenched.
18*	131	Röhl, H.	1933		291.2		62.58 37.42	The above specimen tempered.
19*	131	Röhl, H.	1933		291.2		57.81 42.19	Calculated composition (55 a/o Cu); same dimensions and supplier as the above specimen; quenched.
20*	131	Röhl, H.	1933		291.2		57.81 42.19	The above specimen tempered.
21	231	Taylor, R.	1934	A	293-896		54.69 45.31	Calculated composition (41.75 a/o Pd); electrolytic copper (major impurity 0.002 Fe) and palladium sponge melted in high vacuum, cooled, and drawn into wires.
22	231	Taylor, R.	1934	A	293-873		62.01 37.99	Calculated composition (49.24 a/o Pd); similar to the above specimen.

\* Not shown in figure.

TABLE 32. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF PALLADIUM + COPPER ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Pd Cu	Composition (continued), Specifications, and Remarks
23	231	Taylor, R.	1934	A	293-898		62.01 27.99	The above specimen measured during cooling.
24	231	Taylor, R.	1934	A	293, 873		52.57 47.43	Calculated composition (40 a/o Pd); similar to the above specimen.
25*	231	Taylor, R.	1934	A	293, 873		56.59 43.41	Calculated composition (42.5 a/o Pd); similar to the above specimen.
26*	231	Taylor, R.	1934	A	293, 873		58.21 41.79	Calculated composition (45 a/o Pd); similar to the above specimen.
27*	231	Taylor, R.	1934	A	293, 873		63.02 36.98	Calculated composition (50 a/o Pd); similar to the above specimen.
28	231	Taylor, R.	1934	A	873		67.49 32.51	Calculated composition (55 a/o Pd); similar to the above specimen.
29	126	Grüneisen, E. and Reddemann, H.	1934		22-273		9.2	Calculated composition (14.5 a/o Cu); polycrystalline; obtained from Heraeus.
30	126	Grüneisen, E. and Reddemann, H.	1934		22-292		37.3	Calculated composition (49.9 a/o Cu); annealed.
31	126	Grüneisen, E. and Reddemann, H.	1934		83-283		42.2	Calculated composition (55 a/o Cu); annealed.
32	126	Grüneisen, E. and Reddemann, H.	1934		22-292		42.2	The above specimen reannealed in vacuum at about 850 C for 2 h.
33	126	Grüneisen, E. and Reddemann, H.	1934		22-273		42.2	The above specimen reannealed at about 325 C for 30 h.
34	223	Pott, F. P.	1958		308-863		52.75 97.25	Calculated composition (40 a/o Pd); cut from a 0.2 mm thick sheet, cold-rolled; annealed at 650 C for 2 h; ordered.
35	223	Pott, F. P.	1958		273-833		57.81 42.19	Calculated composition (45 a/o Pd); similar to the above specimen.
36	223	Pott, F. P.	1958		273-853		70.67 29.33	Calculated composition (59 a/o Pd); similar to the above specimen.
37	223	Pott, F. P.	1958		298-1065		52.75 47.25	Calculated composition (40 a/o Pd); similar to the above specimen but disordered.
38	223	Pott, F. P.	1958		309-1073		57.81 42.19	Calculated composition (45 a/o Pd); similar to the above specimen.
39	223	Pott, F. P.	1958		305-1073		70.67 29.33	Calculated composition (59 a/o Pd); similar to the above specimen.

\* Not shown in figure.



[illegible]

Not shown in figure.



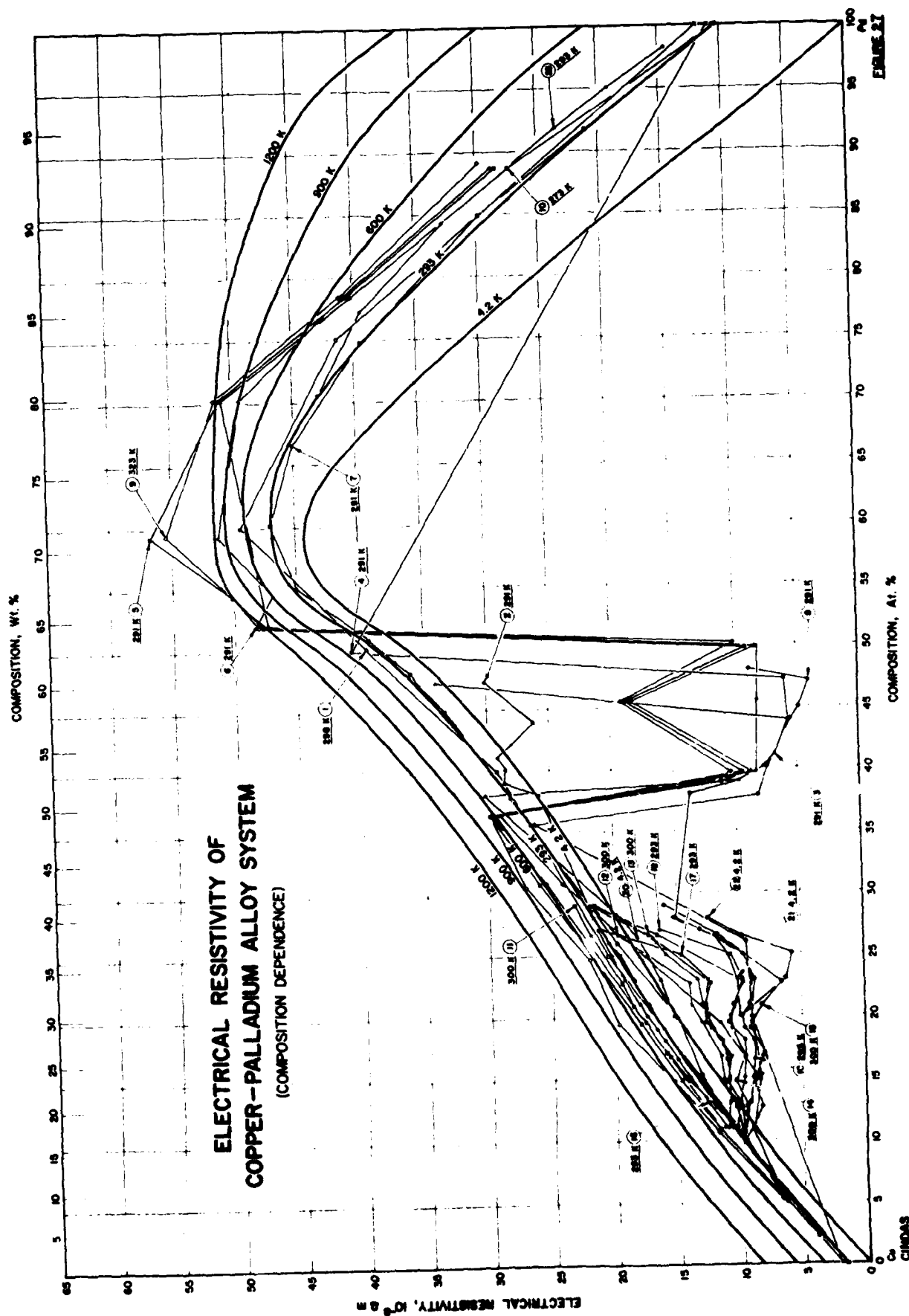


FIGURE 27

TABLE 34. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER-PALLADIUM ALLOY SYSTEM (Composition Dependence)

Date Ref. Set No.	Author(s)	Year	Method Used	Temp., K	Composition Range (At. % of Pd)	Composition (weight percent), Specifications, and Remarks
1 232	Kim, M.J. and Flanagan, W.F.	1967		298.2	0-100	Alloys prepared by induction melting mixtures of 99.97 pure palladium from Engelhard Industries, Inc. and 99.999 pure copper from American Smelting and Refining Co. in a vacuum of $10^{-4}$ torr; homogenized at 1173 K for one week in vacuum, ground; heat-treated at 973 K for 2 days.
2 222	Johansson, C.H. and Linde, J.O.	1927		291.2	17-47	Specimens quenched and hard-worked.
3 222	Johansson, C.H. and Linde, J.O.	1927		291.2	17-47	The above specimens annealed at below melting point for 2 to 3 h.
4 222	Johansson, C.H. and Linde, J.O.	1927		291.2	0-100	The above specimens tempered.
5 222	Johansson, C.H. and Linde, J.O.	1927		291.2	5.4-89	Data of Sedström (Dissertation, Stockholm, 1924) reported by the authors.
6 222	Johansson, C.H. and Linde, J.O.	1927		291.2	2.1-89	The above specimens tempered.
7 221	Svensson, B.	1932	B	291.2	0-100	Alloys prepared from physically pure palladium supplied by Heraeus and technically pure copper; $40 \times 1 \times 1$ mm; homogenized at 1273 to 1323 K for 10 to 12 h.
8 221	Svensson, B.	1932	B	291.2	11-48	The above specimens heated to 973 K and cooled slowly to 773 K; ordered.
9 234	Holgersson, S. and Sedström, E.	1924		323.2	5.2-89	No details reported.
10 235	Sedström, E.	1924		273.2	0-100	Data taken from Schulze, A. (Z. Anorg. Chem., 159, 325-42, 1927).
11 225	Jaumot, F.E., Jr. and Sawatzky, A.	1956	A	300	6.4-29	99.99 <sup>+</sup> pure copper from Vacuum Metals Co. and 99.98 <sup>+</sup> pure palladium from J. Bishop Co. melted together in vacuum, forged and swaged, annealed in vacuum at 1073 K for 2 h and quenched; reported error $\pm 2\%$ .
12 225	Jaumot, F.E., Jr. and Sawatzky, A.	1956	A	300	6.4-29	Similar to the above specimens but furnace-cooled from 1073 K.
13 225	Jaumot, F.E., Jr. and Sawatzky, A.	1956	A	300	6.4-29	Similar to the above specimens but reannealed at 673 K for 24 h and quenched.
14 225	Jaumot, F.E., Jr. and Sawatzky, A.	1956	A	300	6.4-29	Similar to the above specimens but cooled from 1073 K at a rate $2 \text{ K day}^{-1}$ to 673 K and quenched.
15 225	Jaumot, F.E., Jr. and Sawatzky, A.	1956	A	300	6.4-29	Similar to the above specimens but cooled from 1073 K at a rate $2 \text{ K day}^{-1}$ to 528 K and quenched.
16 224	Köster, W. and Lang, W.	1958		293.2	11-27	High purity Cu and Pd mixed and melted in argon atmosphere, cast, cold-hammered to 3 to 4 mm in diameter with annealing at 973 K and subsequent quenching between hammerings; quenched from 783 K; measuring temperature not given, assigned here as 20 C.
17 224	Köster, W. and Lang, W.	1958		293.2	11-27	The above specimens annealed at 693 K for 5 days.
18 224	Köster, W. and Lang, W.	1958		293.2	11-27	The above specimens reannealed at 633 K for 2 days.
19 224	Köster, W. and Lang, W.	1958		293.2	11-27	The above specimens reannealed at 573 K for 2 days.
20 223	Buyanova, L.N., Syutkina, V.I., Shashkov, O.D., and Yakovlev, E.S.	1972	A	4.2	20-28	99.99 pure copper and 99.98 pure palladium vacuum-melted, cast, and drawn into wires; quenched in water from 913 K.
21 233	Buyanova, L.N., et al.	1972	A	4.2	20-28	Similar to the above specimens but heat-treated by cooling from 733 to 573 K at a rate of $100 \text{ K day}^{-1}$ .
22 233	Buyanova, L.N., et al.	1972	A	4.2	20-28	Similar to the above specimens but cooled at $2 \text{ K day}^{-1}$ .
23 236	Mallory Metallurgical Co.	1964		293.2	70-90	Measuring temperature not given, assigned here as 20 C.

TABLE 35. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF COPPER-PALLADIUM ALLOY SYSTEM (Composition Dependence)

[Composition, C, at/o Pd; Electrical Resistivity, $\rho$ , $10^{-9} \Omega \text{ m}$ ]					
C	$\rho$	C	$\rho$	C	$\rho$
DATA SET 4 (T = 291.2 K)					
0	2.2	36.4	29.0	44.9	4.8
5	6.5	39.7	8.3	47.0	4.0
10	9.8	45.5	8.1	48.0	8.8
15	20.5	49.8	8.1	DATA SET 9 (T = 293.2 K)	
20	29.2	51.9	48.4	10.6	6.8
25	40	54.4	46.9	14.7	15.4
30	39.5	59.3	51.2	18.4	18.8
35	49.4	70.2	49.3	24.7	22.0
40	49.4	78.4	40.8	30.5	27.0
45	41.3	88.7	28.7	36.4	29.8
50	11.8	DATA SET 7 (T = 291.2 K)			
55	40.1	0	1.7	38.9	11.9
60	7.7	11.0	10.7	39.7	10.3
65	41.1	15.2	13.2	45.5	19.1
70	43.9	19.9	15.0	50.2	10.0
75	47.2	26.0	19.8	51.9	48.1
80	49.9	35.6	26.5	54.4	50.1
85	100	37.8	28.1	59.4	55.4
90	14.9	44.9	33.3	70.2	51.4
95	16.2	47.0	35.4	78.4	41.1
100	17.4	48.0	36.0	89.1	29.7
DATA SET 5 (T = 291.2 K)					
17.0	14.9	49.0	37.3	DATA SET 10 (T = 273.2 K)	
19.0	16.2	50.8	39.3	0	1.90
21.0	17.4	55.0	45.0	5.4	6.90
23.0	18.5	60.2	47.0	10.8	11.8
25.0	19.6	66.6	45.1	30.8	25.4
26.0	20.4	74.7	39.5	36.4	29.7
28.0	21.6	84.9	29.8	38.9	11.1
30.0	25.9	91.8	21.0	39.7	8.77
32.0	28.6	100	10.6	45.5	18.3
34.0	28.5	DATA SET 8 (T = 291.2 K)			
36.0	28.5	11.0	9.1	48.8	8.26
38.0	28.6	15.2	8.4	51.9	47.4
40.1	28.5	19.9	9.0	70.2	50.8
42.1	28.5	26.0	9.4	78.4	40.2
44.1	29.1	35.6	26.2	88.7	27.3
46.1	29.1	37.8	8.2	100	10.3
48.9	26.2	40.1	7.4	DATA SET 11 (T = 300 K)	
49.2	30.1	43.9	5.5	6.4	7.33
DATA SET 3 (T = 291.2 K)					
17.0	11.2	51.9	47.7	9.7	9.82
19.0	12.2	54.4	50.1	12.6	8.37
21.0	13.0	59.3	56.7	14.7	8.97
23.0	13.4	70.2	51.0	16.7	8.22
25.0	15.9	78.4	40.6	17.0	8.13
26.0	16.6	88.7	28.4	19.5	8.00
28.0	30.2	DATA SET 6 (T = 291.2 K)			
30.0	9.7	11.0	9.1	20.9	8.28
32.0	7.7	15.2	8.4	22.7	6.62
34.0	6.8	19.9	9.0	23.0	11.78
36.0	9.1	26.0	9.4	29.0	16.00
38.0	5.6	35.6	26.2	DATA SET 12 (T = 300 K)	
40.1	34.6	37.8	8.2	6.4	7.33
42.9	47.2	40.1	7.4	9.7	9.63
DATA SET 1 (T = 298.2 K)					
0	2.2	43.9	40.7	12.6	10.43
5	6.5	49.9	40.7	14.7	11.46
10	9.8	100	11.0	16.7	11.55
15	20.5	DATA SET 2 (T = 291.2 K)			
20	29.2	17.0	14.9	19.5	8.28
25	40	19.0	16.2	20.9	6.22
30	39.5	21.0	17.4	22.7	11.78
35	49.4	23.0	18.5	29.0	16.00
40	49.4	25.0	19.6	DATA SET 13 (T = 300 K)	
45	41.3	26.0	20.4	6.4	7.33
50	11.8	28.0	21.6	9.7	9.82
55	40.1	30.0	25.9	12.6	8.37
60	49.4	32.0	28.6	14.7	8.97
65	41.3	34.0	28.5	16.7	8.22
70	11.8	36.0	28.5	17.0	8.13
75	41.3	38.0	28.6	19.5	8.00
80	11.8	40.1	29.1	20.9	8.28
85	40.1	42.1	29.1	22.7	6.62
90	7.7	44.1	29.1	23.0	11.78
95	41.1	46.1	29.1	29.0	16.00
100	43.9	48.9	26.2	DATA SET 14 (cont.) (T = 300 K)	
DATA SET 15 (T = 300 K)					
12.6	9.53	DATA SET 16 (T = 293.2 K)			
14.7	10.48	11	11.34		
16.7	9.66	13	12.70		
19.5	10.92	15	14.11		
20.9	10.57	16.5	15.06		
22.7	9.81	18	16.09		
26.6	16.66	19	16.74		
29.0	21.65	20	17.44		
DATA SET 17 (T = 293.2 K)					
11	11.34	21	18.05		
13	12.70	23	19.27		
15	14.11	25	20.36		
16.5	15.06	27	21.27		
18	16.09	DATA SET 18 (T = 300 K)			
19	16.74	6.4	7.33		
20	17.44	9.7	9.82		
21	18.05	12.6	10.43		
23	19.27	14.7	11.46		
25	20.36	16.7	11.55		
27	21.27	19.5	8.28		
DATA SET 19 (T = 300 K)					
6.4	7.33	20.9	8.28		
9.7	9.82	22.7	6.62		
12.6	10.43	23.0	11.78		
14.7	11.46	29.0	16.00		
16.7	11.55	DATA SET 20 (T = 300 K)			
19.5	8.00	29.0	16.00		
20.9	8.28	DATA SET 21 (T = 300 K)			
22.7	6.62	6.4	7.33		
23.0	11.78	9.7	9.82		
29.0	16.00	12.6	10.43		
DATA SET 22 (T = 300 K)					
6.4	7.33	14.7	11.46		
9.7	9.82	16.7	11.55		
12.6	10.43	19.5	8.28		
14.7	11.46	20.9	8.28		
16.7	11.55	22.7	6.62		
19.5	8.00	23.0	11.78		
20.9	8.28	29.0	16.00		
22.7	6.62	DATA SET 23 (T = 300 K)			
23.0	11.78	29.0	16.00		
29.0	16.00	DATA SET 24 (T = 300 K)			
DATA SET 25 (T = 300 K)					
6.4	7.33	6.4	7.33		
9.7	9.82	9.7	9.82		
12.6	10.43	12.6	10.43		
14.7	11.46	14.7	11.46		
16.7	11.55	16.7	11.55		
19.5	8.00	19.5	8.00		
20.9	8.28	20.9	8.28		
22.7	6.62	22.7	6.62		
23.0	11.78	23.0	11.78		
29.0	16.00	29.0	16.00		
DATA SET 26 (T = 300 K)					
6.4	7.33	6.4	7.33		
9.7	9.82	9.7	9.82		
12.6	10.43	12.6	10.43		
14.7	11.46	14.7	11.46		
16.7	11.55	16.7	11.55		
19.5	8.00	19.5	8.00		
20.9	8.28	20.9	8.28		
22.7	6.62	22.7	6.62		
23.0	11.78	23.0	11.78		
29.0	16.00	29.0	16.00		
DATA SET 27 (T = 300 K)					
6.4	7.33	6.4	7.33		
9.7	9.82	9.7	9.82		
12.6	10.43	12.6	10.43		
14.7	11.46	14.7	11.46		
16.7	11.55	16.7	11.55		
19.5	8.00	19.5	8.00		
20.9	8.28	20.9	8.28		
22.7	6.62	22.7	6.62		
23.0	11.78	23.0	11.78		
29.0	16.00	29.0	16.00		
DATA SET 28 (T = 300 K)					
6.4	7.33	6.4	7.33		
9.7	9.82	9.7	9.82		
12.6	10.43	12.6	10.43		
14.7	11.46	14.7	11.46		
16.7	11.55	16.7	11.55		
19.5	8.00	19.5	8.00		
20.9	8.28	20.9	8.28		
22.7	6.62	22.7	6.62		
23.0	11.78	23.0	11.78		
29.0	16.00	29.0	16.00		
DATA SET 29 (T = 300 K)					
6.4	7.33	6.4	7.33		
9.7	9.82	9.7	9.82		
12.6	10.43	12.6	10.43		
14.7	11.46	14.7	11.46		
16.7	11.55	16.7	11.55		
19.5	8.00	19.5	8.00		
20.9	8.28	20.9	8.28		
22.7	6.62	22.7	6.62		
23.0	11.78	23.0	11.78		
29.0	16.00	29.0	16.00		
DATA SET 30 (T = 300 K)					
6.4	7.33	6.4	7.33		
9.7	9.82	9.7	9.82		
12.6	10.43	12.6	10.43		
14.7	11.46	14.7	11.46		
16.7	11.55	16.7	11.55		
19.5	8.00	19.5	8.00		
20.9	8.28	20.9	8.28		
22.7	6.62	22.7	6.62		
23.0	11.78	23.0	11.78		
29.0	16.00	29.0	16.00		
DATA SET 31 (T = 300 K)					
6.4	7.33	6.4	7.33		
9.7	9.82	9.7	9.82		
12.6	10.43	12.6	10.43		
14.7	11.46	14.7	11.46		
16.7	11.55	16.7	11.55		
19.5	8.00	19.5	8.00		
20.9	8.28	20.9	8.28		
22.7	6.62	22.7	6.62		
23.0	11.78	23.0	11.78		
29.0	16.00	29.0	16.00		
DATA SET 32 (T = 300 K)					
6.4	7.33	6.4	7.33		
9.7	9.82	9.7	9.82		
12.6	10.43	12.6	10.43		
14.7	11.46	14.7	11.46		
16.7	11.55	16.7	11.55		
19.5	8.00	19.5	8.00		
20.9	8.28	20.9	8.28		
22.7	6.62	22.7	6.62		
23.0	11.78	23.0	11.78		
29.0	16.00	29.0	16.00		
DATA SET 33 (T = 300 K)					
6.4	7.33	6.4	7.33		
9.7	9.82	9.7	9.82		
12.6	10.43	12.6	10.43		
14.7	11.46	14.7	11.46		
16.7	11.55	16.7	11.55		
19.5	8.00	19.5	8.00		
20.9	8.28	20.9	8.28		
22.7	6.62	22.7	6.62		
23.0	11.78	23.0	11.78		
29.0	16.00	29.0	16.00		
DATA SET 34 (T = 300 K)					
6.4	7.33	6.4	7.33		
9.7	9.82	9.7	9.82		
12.6	10.43	12.6	10.43		
14.7	11.46	14.7	11.46		
16.7	11.55	16.7	11.55		
19.5	8.00	19.5	8.00		
20.9	8.28	20.9	8.28		
22.7	6.62	22.7	6.62		
23.0	11.78	23.0	11.78		
29.0	16.00	29.0	16.00		
DATA SET 35 (T = 300 K)					
6.4	7.33	6.4	7.33		
9.7	9.82	9.7	9.82		
12.6	10.43	12.6	10.43		
14.7	11.46	14.7	11.46		
16.7	11.55	16.7	11.55		
19.5	8.00	19.5	8.00		
20.9	8.28	20.9	8.28		
22.7	6.62	22.7	6.62		
23.0	11.78	23.0	11.78		
29.0	16.00	29.0	16.00		
DATA SET 36 (T = 300 K)					
6.4	7.33	6.4	7.33		
9.7	9.82	9.7	9.82		
12.6	10.43	12.6	10.43		
14.7	11.46	14.7	11.46		
16.7	11.55	16.7	11.55		
19.5	8.00	19.5	8.00		
20.9	8.28	20.9	8.28		
22.7	6.62	22.7	6.62		
23.0	11.78	23.0	11.78		
29.0	16.00	29.0	16.00		
DATA SET 37 (T = 300 K)					
6.4	7.33	6.4			

TABLE 35. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF COPPER-PALLADIUM ALLOY SYSTEM (Composition Dependence) (continued)

C	$\rho$	C	$\rho$
DATA SET 17 (cont.) (T = 283.2 K)		DATA SET 21 (T = 4.2 K)	
16.5	11.11	20	10.0
18	11.47	23	10.2
19	12.30	25	10.9
20	12.57	26.5	12.0
21	12.79	28	15.3
23	12.59		
25	14.62	DATA SET 22 (T = 4.2 K)	
27	20.90	20	9.9
DATA SET 18 (T = 283.2 K)		23	6.3
11	10.44	25	5.7
13	10.14	26.5	9.7
15	9.72	28	12.5
16.5	9.63	DATA SET 23 (T = 283.2 K)	
18	9.63	20.5	42.9
19	10.64	77.2	39.4
20	10.71	84.3	32.7
21	10.35	91.9	23.4
23	9.80	95.1	19.1
25	10.67	98.3	14.5
27	16.34		
DATA SET 19 (T = 283.2 K)			
11	9.85		
13	9.39		
15	9.02		
16.5	8.90		
18	8.90		
19	9.79		
20	9.84		
21	9.26		
23	9.16		
25	9.68		
27	13.11		
DATA SET 20 (T = 4.2 K)			
20	15.3		
23	16.4		
25	17.4		
26.5	18.2		
28	19.5		

### 3.6. Copper-Zinc Alloy System

The copper-zinc alloy system does not constitute a continuous series of solid solutions. The maximum solid solubility of zinc in copper is 39.0 wt.% (38.3 at.%) at 727 K and the solubility decreases at higher and lower temperatures. At lower temperatures, the attainment of equilibrium becomes very slow and the solubility data are uncertain. Massalski and Kittl [237] analyzed the existing data and concluded that the boundary lies at about 34 at.% Zn at 473 K and it may lie at less than 30 at.% Zn at room temperature. Shinoda and Amano [238] reported a much greater reduction in solubility at room temperature. The maximum solid solubility of copper in zinc is 2.7 wt.% (2.77 at.%) at 697 K and it decreases to 0.3 wt.% (0.31 at.%) at 373 K.

There are 175 sets of experimental data available for this alloy system. These data sets are listed in tables 37, 39, and 41, tabulated in tables 38, 40, and 42, and shown partially in figures 29, 30, and 31. Most of the available data are for copper-rich alloys in the solid solution region. Due to the scarcity of data for zinc-rich alloys, the available data for Zn + Cu alloys are not evaluated and no recommendations are made in the present work.

For Cu + Zn alloys, Argent and Lee [239] (Cu + Zn data sets 31-50) reported the electrical resistivity for various compositions ranging from pure copper to 34.2 wt.% Zn at 77, 195, 273, and 373 K. Henry and Schroeder [240] (Cu + Zn data sets 57-65) reported the electrical resistivity of alloys with compositions ranging from 0.93 to 35.97 wt.% Zn at temperatures below room temperature. The values of Argent and Lee scatter more than those of Henry and Schroeder on a plot of resistivity versus composition.

Fairbank [241] (Cu + Zn data sets 12-30) has investigated the electrical resistivity of a full range of alloy compositions in the alpha-phase solid solution region in the temperature range 14.3 K to room temperature. Measurements were made on both annealed and hard-drawn specimens. His values for the annealed specimens are higher than those reported by other authors.

The data outside the region of solid solution are fragmentary and conflicting. In addition, beta-brass (46-48 wt.% Zn) forms ordered structures below about 735 K. For these reasons the data smoothing and synthesis was limited to the alpha-phase solid solution and was based mainly on the results of Henry and

Schroeder, Smith [242] (Cu + Zn data sets 67-79), Smith and Palmer [64] (data set 80), and Kemp, Klemens, Tainsh, and White [129, 246, 247] (Cu + Zn data sets 81-94).

The resulting recommended electrical resistivity values for Cu, Zn, and for 9 Cu + Zn binary alloys are presented in table 36 and shown in figures 28 and 31. No values were generated for alloys containing 35 to 99.5% Zn. The recommended values for Cu and for Zn are for well-annealed high-purity specimens, but those values for temperatures below about 100 K are applicable only to Cu and Zn having residual electrical resistivities as given at 1 K in table 36. The alloys for which the recommended values are generated are not ordered and have not been quenched or cold-worked severely. The recommended values cover the temperature range from 1 K to 700 K. These values are not corrected for the thermal expansion of the material. The estimated uncertainties in the values for the various alloys and for different temperature ranges are explicitly stated in a footnote to table 36.



TABLE 36. RECOMMENDED ELECTRICAL RESISTIVITY OF COPPER-ZINC ALLOY SYSTEM†  
[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Cu: 100.00% (100.00 At. %) Zn: 0.00% ( 0.00 At. %)			Cu: 99.50% (99.51 At. %) Zn: 0.50% ( 0.49 At. %)			Cu: 99.00% (99.03 At. %) Zn: 1.00% ( 0.97 At. %)			Cu: 97.00% (97.08 At. %) Zn: 3.00% ( 2.92 At. %)			Cu: 95.00% (95.13 At. %) Zn: 5.00% ( 4.87 At. %)			Cu: 90.00% (90.25 At. %) Zn: 10.00% ( 9.75 At. %)		
T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$	
1	0.00200		1	0.150*		1	0.270		1	0.710		1	1.09		1	1.84	
4	0.00200		4	0.150		4	0.270		4	0.710		4	1.09		4	1.84	
7	0.00200		7	0.150		7	0.270		7	0.710		7	1.09		7	1.84	
10	0.00202		10	0.150		10	0.270		10	0.710		10	1.09		10	1.84	
15	0.00218		15	0.150		15	0.271		15	0.710		15	1.09		15	1.85	
20	0.00280		20	0.151		20	0.279		20	0.710		20	1.10		20	1.86	
25	0.00450		25	0.159		25	0.289		25	0.711		25	1.11		25	1.87	
30	0.00830		30	0.172		30	0.302		30	0.719		30	1.13		30	1.88	
40	0.0240		40	0.207		40	0.338		40	0.748		40	1.16		40	1.91	
50	0.0320		50	0.243		50	0.381		50	0.794		50	1.20		50	1.96	
60	0.0974		60	0.288		60	0.431		60	0.856		60	1.26		60	2.02	
70	0.154		70	0.341		70	0.489		70	0.924		70	1.33		70	2.10	
80	0.216		80	0.397		80	0.550		80	0.996		80	1.40		80	2.18	
90	0.282		90	0.452		90	0.610		90	1.06		90	1.47		90	2.26	
100	0.349		100	0.511		100	0.671		100	1.13		100	1.54		100	2.33	
150	0.701		150	0.846		150	0.999		150	1.48		150	1.90		150	2.72	
200	1.048		200	1.19		200	1.34		200	1.83		200	2.26		200	3.10	
250	1.393		250	1.53		250	1.68		250	2.18		250	2.61		250	3.48	
273	1.544		273	1.69		273	1.84		273	2.34		273	2.78		273	3.66	
293	1.678		293	1.82		293	1.97		293	2.48		293	2.92		293	3.81	
300	1.725		300	1.87		300	2.02		300	2.53		300	2.97		300	3.86	
350	2.061		350	2.22		350	2.36		350	2.88		350	3.33		350	4.25	
400	2.398		400	2.56		400	2.71		400	3.23		400	3.69		400	4.63	
500	3.079		500	3.25*		500	3.40*		500	3.93*		500	4.41*		500	5.40*	
600	3.771		600	3.94*		600	4.08*		600	4.63*		600	5.13*		600	6.17*	
700	4.481		700	4.63*		700	4.78*		700	5.33*		700	5.85*		700	6.94*	
800	5.213																
900	5.973																
1000	6.766																
1100	7.596																
1200	8.470																
1300	9.395																
1357.6	9.946(z)																
1358	21.01(z)																
1700	24.41																

† Uncertainties in the electrical resistivity values are as follows:

- 100.00 Cu - 0.00 Zn:  $\pm 3\%$  up to 100K,  $\pm 1\%$  above 100K to 250K,  $\pm 0.5\%$  above 250K to 350K,  $\pm 1\%$  above 350K to 500K,  $\pm 0.5\%$  above 500K to 1357.6K, and  $\pm 5\%$  above 1357.6K.
- 99.50 Cu - 0.50 Zn:  $\pm 3\%$  below 500 K and  $\pm 5\%$  from 500 to 700 K.
- 99.00 Cu - 1.00 Zn:  $\pm 3\%$  below 500 K and  $\pm 5\%$  from 500 to 700 K.
- 97.00 Cu - 3.00 Zn:  $\pm 3\%$  below 500 K and  $\pm 5\%$  from 500 to 700 K.
- 95.00 Cu - 5.00 Zn:  $\pm 3\%$  below 500 K and  $\pm 5\%$  from 500 to 700 K.
- 90.00 Cu - 10.00 Zn:  $\pm 3\%$  below 500 K and  $\pm 5\%$  from 500 to 700 K.

\* In temperature range where no experimental data are available.

TABLE 36. RECOMMENDED ELECTRICAL RESISTIVITY OF COPPER-ZINC ALLOY SYSTEM† (continued)  
[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Cu: 85.00% (85.36 At. %) Zn: 15.00% (14.64 At. %)		Cu: 80.00% (80.45 At. %) Zn: 20.00% (19.55 At. %)		Cu: 75.00% (75.53 At. %) Zn: 25.00% (24.47 At. %)		Cu: 70.00% (70.59 At. %) Zn: 30.00% (29.41 At. %)		Cu: 0.00% (0.00 At. %) Zn: 100.00% (100.00 At. %)	
T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
1	2.33	1	2.84	1	3.20*	1	3.39	1	0.0100*
4	2.38	4	2.84	4	3.20	4	3.39	4	0.0100
7	2.38	7	2.84	7	3.20	7	3.39	7	0.0102
10	2.38	10	2.84	10	3.20	10	3.39	10	0.0112
15	2.39	15	2.85	15	3.21	15	3.40	15	0.0190
20	2.40	20	2.86	20	3.22	20	3.41	20	0.0387
25	2.42	25	2.87	25	3.23	25	3.43	25	0.0788
30	2.44	30	2.89	30	3.24	30	3.46	30	0.142
40	2.48	40	2.93	40	3.30	40	3.51	40	0.306
50	2.54	50	3.00	50	3.36	50	3.58	50	0.507
60	2.60	60	3.08	60	3.44	60	3.68	60	0.715
70	2.68	70	3.17	70	3.54	70	3.78	70	0.931
80	2.77	80	3.26	80	3.64	80	3.88	80	1.15
90	2.85	90	3.35	90	3.74	90	3.98	90	1.37
100	2.93	100	3.44	100	3.83	100	4.08	100	1.60
150	3.35	150	3.89	150	4.32	150	4.60	150	2.71
200	3.77	200	4.35	200	4.81	200	5.12	200	3.83
250	4.18	250	4.80	250	5.29	250	5.63	250	4.95
273	4.37	273	5.01	273	5.52	273	5.87	273	5.46
293	4.54	293	5.19	293	5.71	293	6.08	293	5.90
300	4.60	300	5.26	300	5.78	300	6.15	300	6.06
350	5.02	350	5.71	350	6.27	350	6.67	350	7.20
400	5.44	400	6.17	400	6.76	400	7.19	400	8.37
500	6.29*	500	7.08*	500	7.74*	500	8.23*	500	10.82
600	7.12*	600	7.99*	600	8.72*	600	9.27*	600	13.49
700	7.95*	700	8.90*	700	9.70*	700	10.31*	692.73	16.23*

† Uncertainties in the electrical resistivity values are as follows:

- 85.00 Cu - 15.00 Zn:  $\pm 3\%$  below 500 K and  $\pm 5\%$  from 500 to 700 K.
- 80.00 Cu - 20.00 Zn:  $\pm 3\%$  below 500 K and  $\pm 5\%$  from 500 to 700 K.
- 75.00 Cu - 25.00 Zn:  $\pm 3\%$  below 500 K and  $\pm 5\%$  from 500 to 700 K.
- 70.00 Cu - 30.00 Zn:  $\pm 3\%$  below 500 K and  $\pm 5\%$  from 500 to 700 K.
- 0.00 Cu - 100.00 Zn:  $\pm 5\%$  below 50 K,  $\pm 3\%$  from 50 to 400 K, and  $\pm 4\%$  above 400 K.

\* In temperature range where no experimental data are available.

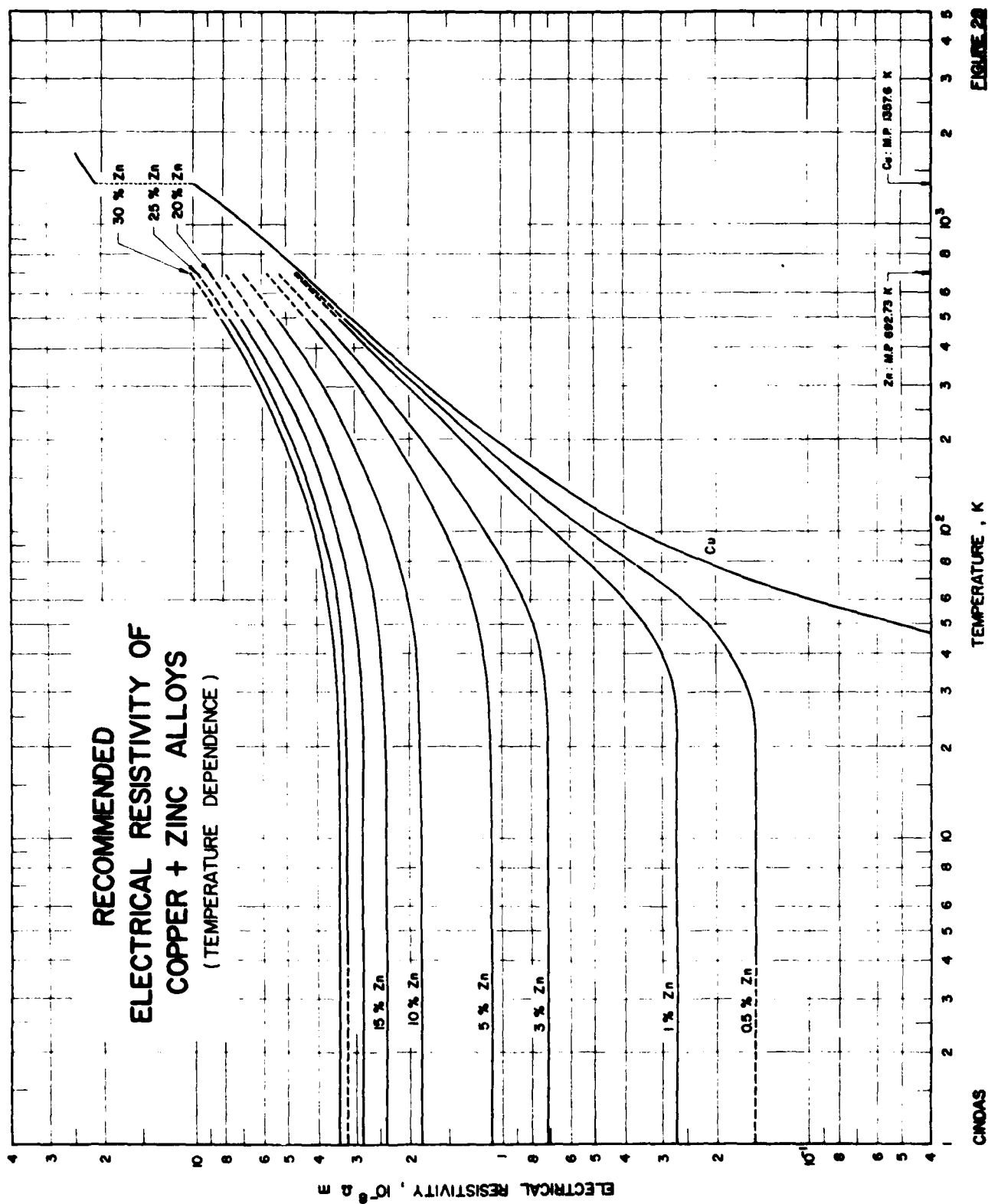


FIGURE 28

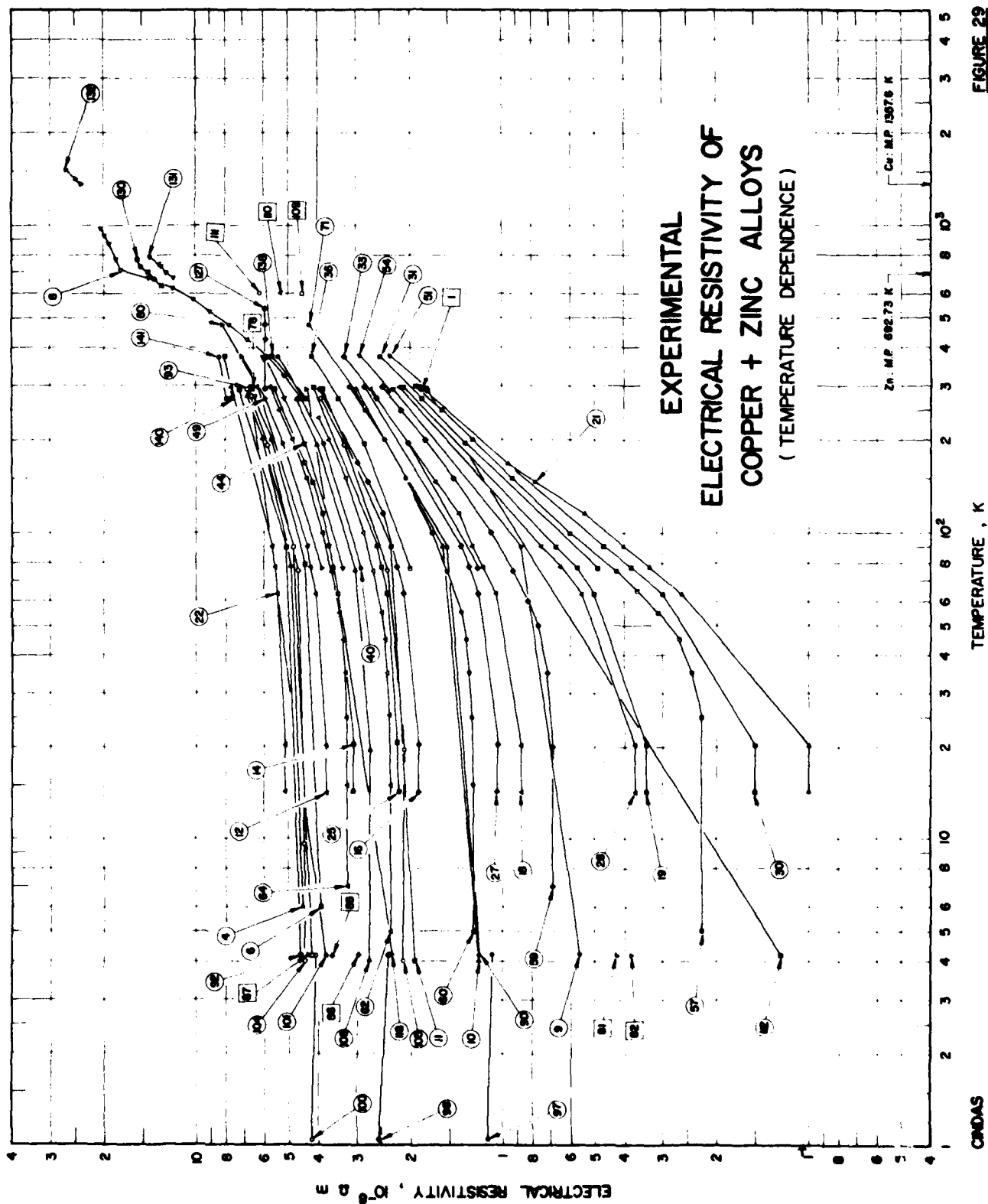


FIGURE 29

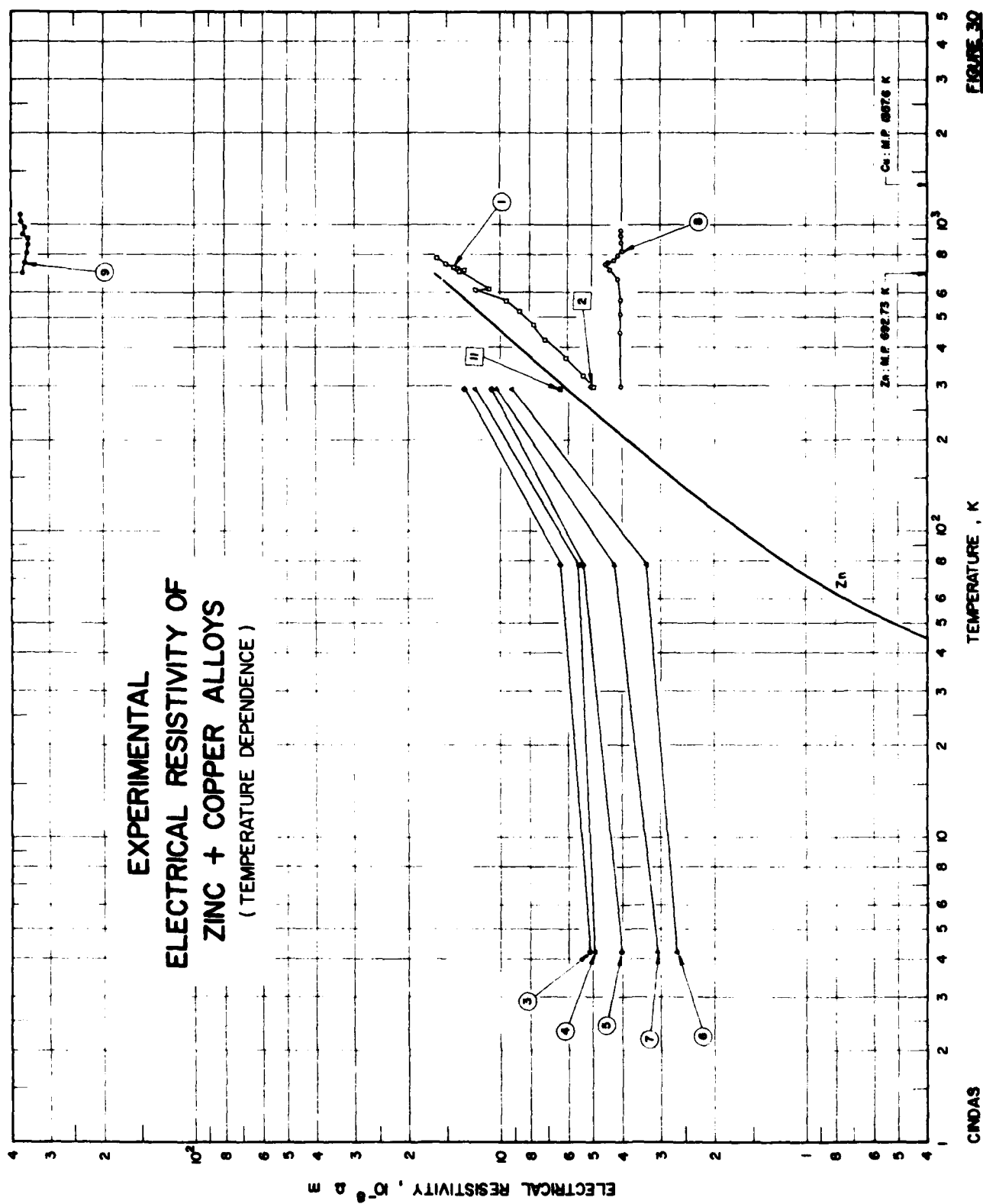


TABLE 37. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + ZINC ALLOYS (Temperature Dependence)

Data Set No.	Ref.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Zn	Composition (continued), Specifications, and Remarks
1	126	Linde, J.O.	1932		291		0.542	Calculated composition (0.527 s/o Zn).
2*	126	Linde, J.O.	1932		291		0.841	Calculated composition (0.818 s/o Zn).
3*	126	Linde, J.O.	1932		291		1.77	Calculated composition (1.724 s/o Zn).
4	41	Damask, A.C.	1956	A	6.0-279	$\alpha$ -brass	29.5	Wire specimen about 3 in. long and 0.015 in. in diameter; cold worked, annealed at 350 C, then drawn at room temperature to 60% reduction in area; annealed at 350 C for 1 hr, quenched in room temperature air stream to 50-60 C, then removed from air stream and allowed to cool to room temperature.
5*	41	Damask, A.C.	1956	A	6.0-277	$\alpha$ -brass	29.5	Similar to the above specimen except annealed for 30 min at 211 C after cold work.
6	41	Damask, A.C.	1956	A	6.0-278	$\alpha$ -brass	29.5	Similar to the above specimen except quenched from 320 C after completely annealed.
7*	41	Damask, A.C.	1956	A	5.0-278	$\alpha$ -brass	29.5	Similar to the above specimen except additionally annealed for 30 min at 211 C after completely annealed and quenched.
8	243	Takano, K.	1969	A	273-973		49.01	Calculated composition (48.3 s/o Zn); 20 x 5 x 0.1 mm; cut from ingots prepared by vacuum melting of high purity copper and zinc; heat-treated at appropriate temperature for two weeks.
9	244	Tainsh, R.J. and White, G.K.	1962		4.2-293		2.06	Calculated composition (2.0 s/o Zn); rod specimen 6 cm long and 3 to 5 mm in diameter; annealed at 850 C.
10	244	Tainsh, R.J. and White, G.K.	1962		4.2-293		5.14	Calculated composition (5.0 s/o Zn); similar to the above specimen.
11	244	Tainsh, R.J. and White, G.K.	1962		4.2-293		10.26	Calculated composition (10.0 s/o Zn); similar to the above specimen.
12	241	Fairbank, H.A.	1944	A	14-296		33.79	Alloy supplied by the American Brass Company of Waterbury, Connecticut; 0.04 in. diameter x 3 in. long; 0.75 in. rod hot-rolled flat to a strip 0.200 in. thick at 800 C, square rod 0.2 in. x 0.2 in. sawed out, swaged round, and drawn in round dies to 0.04 in. diameter, annealed at 700 C for one hour in a graphite mold and then rapidly quenched in cold water.
13*	241	Fairbank, H.A.	1944	A	14-293		29.39	Similar to the above specimen.
14	241	Fairbank, H.A.	1944	A	14-293		19.59	Similar to the above specimen.
15*	241	Fairbank, H.A.	1944	A	14-293		15.61	Similar to the above specimen.
16	241	Fairbank, H.A.	1944	A	14-293		9.66	Similar to the above specimen.
17*	241	Fairbank, H.A.	1944	A	14-293		4.65	Similar to the above specimen.
18	241	Fairbank, H.A.	1944	A	14-293		3.06	Similar to the above specimen.
19	241	Fairbank, H.A.	1944	A	14-296		0.93	Similar to the above specimen.
20*	241	Fairbank, H.A.	1944	A	14-293		0.50	Similar to the above specimen.
21	241	Fairbank, H.A.	1944	A	14-293		0.34	Similar to the above specimen.

\* Not shown in figure.

TABLE 37. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + ZINC ALLOYS (Temperature Dependence) (continued)

Date Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu      Zn	Composition (continued), Specifications, and Remarks
22	241	Fairbank, H.A.	1944	A	14-297		29.39	Alloy furnished by the American Brass Company of Waterbury, Connecticut; 0.04 in. diameter x 3 in. long; 0.75 in. rod hot-rolled flat to a strip 0.200 in. thick at 800 C, square rod 0.2 in. x 0.2 in. sawed out, swaged round, and drawn in round dies to 0.04 in. diameter.
23*	241	Fairbank, H.A.	1944	A	14-297		19.59	Similar to the above specimen.
24*	241	Fairbank, H.A.	1944	A	14-293		15.61	Similar to the above specimen.
25	241	Fairbank, H.A.	1944	A	14-297		9.66	Similar to the above specimen.
26*	241	Fairbank, H.A.	1944	A	14-297		4.65	Similar to the above specimen.
27	241	Fairbank, H.A.	1944	A	14-297		3.06	Similar to the above specimen.
28	241	Fairbank, H.A.	1944	A	14-297		0.93	Similar to the above specimen.
29*	241	Fairbank, H.A.	1944	A	14-296		0.50	Similar to the above specimen.
30	241	Fairbank, H.A.	1944	A	14-297		0.34	Similar to the above specimen.
31	239	Argent, B.B. and Lee, K.T.	1964	A	77-373		1.03	Calculated composition (1 a/o Zn); wire specimen 0.039 in. in diameter; prepared from 99.99% pure elements by melting, homogenizing at >1023 K for one week, then rolling and drawing; annealed at 400 C, then annealed at same temperature for one hour to give a grain size of 0.027 mm, and finally annealed at 400 C for 1 hr, cooled rapidly to room temperature.
32*	239	Argent, B.B. and Lee, K.T.	1964	A	77-373		2.06	Calculated composition (2 a/o Zn); similar to the above specimen.
33	239	Argent, B.B. and Lee, K.T.	1964	A	77-373		4.11	Calculated composition (4 a/o Zn); similar to the above specimen.
34*	239	Argent, B.B. and Lee, K.T.	1964	A	77-373		6.16	Calculated composition (6 a/o Zn); similar to the above specimen.
35*	239	Argent, B.B. and Lee, K.T.	1964	A	77-373		7.19	Calculated composition (7 a/o Zn); similar to the above specimen.
36	239	Argent, B.B. and Lee, K.T.	1964	A	77-373		8.21	Calculated composition (8 a/o Zn); similar to the above specimen.
37*	239	Argent, B.B. and Lee, K.T.	1964	A	77-373		10.26	Calculated composition (10 a/o Zn); similar to the above specimen.
38*	239	Argent, B.B. and Lee, K.T.	1964	A	77-373		12.30	Calculated composition (12 a/o Zn); similar to the above specimen.
39*	239	Argent, B.B. and Lee, K.T.	1964	A	77-373		14.36	Calculated composition (14 a/o Zn); similar to the above specimen.
40	239	Argent, B.B. and Lee, K.T.	1964	A	77-373		16.39	Calculated composition (16 a/o Zn); similar to the above specimen.
41*	239	Argent, B.B. and Lee, K.T.	1964	A	77-373		17.40	Calculated composition (17 a/o Zn); similar to the above specimen.

\* Not shown in figure.

TABLE 37. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + ZINC ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Zn	Composition (continued), Specifications, and Remarks
43*	239	Argent, B. B. and Lee, K. T.	1964	A	77-373		18.42	Calculated composition (18 a/o Zn); similar to the above specimen.
43*	239	Argent, B. B. and Lee, K. T.	1964	A	77-373		20.46	Calculated composition (20 a/o Zn); similar to the above specimen.
44	239	Argent, B. B. and Lee, K. T.	1964	A	77-373		22.49	Calculated composition (22 a/o Zn); similar to the above specimen.
45*	239	Argent, B. B. and Lee, K. T.	1964	A	77-373		24.52	Calculated composition (24 a/o Zn); similar to the above specimen.
46*	239	Argent, B. B. and Lee, K. T.	1964	A	77-373		26.55	Calculated composition (26 a/o Zn); similar to the above specimen.
47*	239	Argent, B. B. and Lee, K. T.	1964	A	77-373		28.58	Calculated composition (28 a/o Zn); similar to the above specimen.
48*	239	Argent, B. B. and Lee, K. T.	1964	A	77-373		30.60	Calculated composition (30 a/o Zn); similar to the above specimen.
49	239	Argent, B. B. and Lee, K. T.	1964	A	77-373		32.62	Calculated composition (32 a/o Zn); similar to the above specimen.
50*	239	Argent, B. B. and Lee, K. T.	1964	A	77-373		34.14	Calculated composition (33.5 a/o Zn); similar to the above specimen.
51	245	Crisp, R. S. and Henry, W. G.	1964	A	290, 375		0.48	Calculated composition (0.47 a/o Zn).
52*	245	Crisp, R. S. and Henry, W. G.	1964	A	290, 375		1.01	Calculated composition (0.98 a/o Zn).
53*	245	Crisp, R. S. and Henry, W. G.	1964	A	290, 375		1.39	Calculated composition (1.35 a/o Zn).
54	245	Crisp, R. S. and Henry, W. G.	1964	A	290, 375		2.43	Calculated composition (2.36 a/o Zn).
55*	245	Crisp, R. S. and Henry, W. G.	1964	A	290, 375		4.62	Calculated composition (4.50 a/o Zn).
56*	245	Crisp, R. S. and Henry, W. G.	1964	A	290, 375		6.92	Calculated composition (6.74 a/o Zn).
57	240	Henry, W. G. and Schroeder, P. A.	1963	A	5-298		0.93	Calculated composition (0.90 a/o Zn); American Smelting and Refining Co. copper ( $\rho(293\text{ K})/\rho(4.2\text{ K}) = 1320$ ) and consolidated Mining and Smelting Co.; 99.999 pure zinc; prepared by melting in alumina crucibles and chill-cast at least twice under purified argon into billets approximately 9 x 30 mm, annealed in Pyrex at 575 C in one-third of an atmosphere of purified argon for 7 days to remove local inhomogeneities, drawn to 0.5 mm wires, degassed and given a light etch in 10% nitric acid, then annealed in pyrex in one-third of an atmosphere of purified argon for 7 days at 575 C and allowed to cool in the air.
58*	240	Henry, W. G. and Schroeder, P. A.	1963	A	7-297		2.43	Calculated composition (2.36 a/o Zn); similar to the above specimen.

\* Not shown in figure.



TABLE 37. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + ZINC ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Zn	Composition (continued), Specifications, and Remarks
59	240	Henry, W. G. and Schroeder, P. A.	1963	A	7-298		3.01	Calculated composition (2.93 a/o Zn); similar to the above specimen.
60	240	Henry, W. G. and Schroeder, P. A.	1963	A	5-298		5.96	Calculated composition (5.83 a/o Zn); similar to the above specimen.
61*	240	Henry, W. G. and Schroeder, P. A.	1963	A	7-297		10.48	Calculated composition (10.22 a/o Zn); similar to the above specimen.
62	240	Henry, W. G. and Schroeder, P. A.	1963	A	5-237		14.22	Calculated composition (13.88 a/o Zn); similar to the above specimen.
63*	240	Henry, W. G. and Schroeder, P. A.	1963	A	7-299		18.89	Calculated composition (18.46 a/o Zn); similar to the above specimen.
64	240	Henry, W. G. and Schroeder, P. A.	1963	A	7-296		25.96	Calculated composition (25.44 a/o Zn); similar to the above specimen.
65*	240	Henry, W. G. and Schroeder, P. A.	1963	A	4-237		35.97	Calculated composition (35.32 a/o Zn); similar to the above specimen.
66*	60	Hibbard, W. R., Jr.	1959		20-299		8.61	Calculated composition (8.39 a/o Zn); annealed at 445 C for 2 hr; grain size 0.0125 mm.
67*	242	Smith, C. S.	1930		293	90	99.64	0.02 Fe and 0.01 Pb; polycrystalline; grain size 0.070 mm; cylindrical specimen 13.25 in. long and 0.750 in. in diameter; annealed at 680 C for 1 hr and cooled in air.
68*	242	Smith, C. S.	1930		293	89	99.45	0.01 Fe and 0.01 Pb; similar to the above specimen, but grain size 0.110 mm.
69*	242	Smith, C. S.	1930		293	73	88.93	0.02 Fe; similar to the above specimen, but annealing temperature 700 C and grain size 0.120 mm.
70*	242	Smith, C. S.	1930		293, 473	12	96.94	0.02 Fe; similar to the above specimen but annealing time 0.75 hr and grain size 0.100 mm.
71*	242	Smith, C. S.	1930		293, 473	13	95.21	Similar to the above specimen but grain size 0.085 mm.
72*	242	Smith, C. S.	1930		293	14	90.07	0.01 Fe and 0.01 Pb; similar to the above specimen but grain size 0.095 mm.
73*	242	Smith, C. S.	1930		293	15	83.20	0.03 Fe and 0.01 Pb; similar to the above specimen but grain size 0.125 mm.
74*	242	Smith, C. S.	1930		293	16	79.62	0.01 Fe and 0.02 Pb; similar to the above specimen but grain size 0.190 mm.
75*	242	Smith, C. S.	1930		293	22	59.20	0.02 Fe and 0.03 Pb; similar to the above specimen but grain size 0.070 mm.
76*	242	Smith, C. S.	1930		293	85	50.30	0.01 Fe and 0.04 Pb; similar to the above specimen but annealing time 2 hr and grain size 0.16 mm.
77*	242	Smith, C. S.	1930		293	18	69.14	0.03 Fe and 0.02 Pb; same structure and dimensions as the above specimen; grain size 0.075 mm; annealed at 650 C for 0.75 hr and cooled in air.
78	242	Smith, C. S.	1930		293	21	65.45	0.01 Fe and 0.03 Pb; similar to the above specimen but grain size 0.090 mm.

\* Not shown in figure.

TABLE 37. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + ZINC ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Zn	Composition (continued), Specifications, and Remarks
79* 243		Smith, C.S.	1930		293	88	54.96 45.02	0.01 Fe; similar to the above specimen but annealing time 2 hr and grain size 0.040 mm.
80 64		Smith, C.S. and Palmer, E.W.	1935	B	293, 473	19	66.24 33.72	0.03 Pb and 0.01 Fe; annealed at 650 C for 0.75 hr.
81 246		Kemp, W.R.G., Klemens, P.G., Tinsah, R.J., and White, G.K.	1957		4.2	25	1.63	~8 cm long and 0.5 cm in diameter; supplied by Johnson, Matthey and Co., Ltd.; as drawn.
82 246		Kemp, W.R.G., et al.	1957		4.2	2	1.63	The above specimen annealed at 500 C for 4 hr in He atmosphere.
83* 246		Kemp, W.R.G., et al.	1957		4.2	55	5.37	Same dimension and supplier as the above specimen; as drawn.
84* 246		Kemp, W.R.G., et al.	1957		4.2	5	5.37	The above specimen annealed at 500 C for 4 hr in a He atmosphere.
85* 246		Kemp, W.R.G., et al.	1957		4.2	10	9.98	Similar to the above specimen.
86 246		Kemp, W.R.G., et al.	1957		4.2	20	19.48	Similar to the above specimen.
87 246		Kemp, W.R.G., et al.	1957		4.2	308	31.87	Same dimension and supplier as the above specimen.
88* 246		Kemp, W.R.G., et al.	1957		4.2	30	31.87	The above specimen annealed in a He atmosphere at 500 C for 4 hr.
89* 139		Kemp, W.R.G., Klemens, P.G., Tinsah, R.J., and White, G.K.	1957		4.2-293		2	8 cm long and 0.5 cm in diameter; drawn and annealed at 850 C for 4 hr.
90 139		Kemp, W.R.G., et al.	1957		4.2-293		5	Similar to the above specimen.
91* 139		Kemp, W.R.G., et al.	1957		4.2-293		10	Similar to the above specimen.
92 247		Kemp, W.R.G., Klemens, P.G., and Tinsah, R.J.	1959		4.2-293		32	$\alpha$ -brass; machined from an annealed and torsionally deformed bar.
93 247		Kemp, W.R.G., et al.	1959		4.2-293		32	Similar to the above specimen but heated up to 250 C at a rate of 6 C min <sup>-1</sup> .
94* 247		Kemp, W.R.G., et al.	1959		4.2-293		32	Similar to the above specimen but heated up to 400 C at a rate of 6 C min <sup>-1</sup> .
95* 248		Eubank, A. and Neumann, O.	1924		90, 273	Red Brass	18	Polycrystalline; grain size 0.006 cm <sup>2</sup> .
96* 248		Eubank, A. and Neumann, O.	1924		90, 273	Red Brass	18	Polycrystalline; grain size 0.11 cm <sup>2</sup> .
97 249		Olsen, T.	1940		1.1, 4.2	24	95.4 4.60	0.01 Pb; cylindrical specimen 10 cm long; machined; annealed for 21 hr at 540 C.
98 249		Olsen, T.	1940		1.1, 4.2	215	94.53 5.47	0.03 Fe and 0.03 Pb; cylindrical specimen 10 cm long; machined; annealed for 21 hr at 540 C.
99* 249		Olsen, T.	1940		1.1, 4.2	230	86.56 13.43	0.01 Fe; cylindrical specimen 10 cm long; machined, cold worked, and annealed at 500 C for 17 hr.

\* Not shown in figure.

TABLE 37. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + ZINC ALLOYS (Temperature Dependence) (continued)

Data Ref. Set No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Zn	Composition (continued), Specifications, and Remarks
100 249	Olson, T.	1960		1.1, 4.2	Z30	69.95 30.02	0.02 Fe and 0.01 Pb; cylindrical specimen 10 cm long; machined and annealed at 540 C for 21 hr.
101 250	Gordon, J.E. and Amstutz, L.I.	1965		4.2, 298	CDA alloy No. 26	69.3 30.9 ±0.5 ±0.5	0.07 Si, 0.025 Pb, and <0.01 each of Fe, Co, and Ni; strip specimen 0.0150 cm in cross section and 3.46 cm long; supplied by Chase Brass and Copper Company; cold-rolled cartridge brass of nominal grain size 0.025-0.050 mm.
102* 251	Srivastava, B.N., Chatterjee, S., and Sen, S.K.	1969		4.2		1.96	Prepared from spectrographically pure rods of copper and zinc; supplied by Johnson, Matthey and Co. Ltd., by scaling the metals in appropriate portion in evacuated quartz tube, heating to 1100 C, shaking thoroughly, cooling to 900 C and maintaining for five days, rolled, then annealed at 500 C for 6 hr.
103* 251	Srivastava, B.N., et al.	1969		4.2		4.76	Same fabrication method as the above specimen.
104 103	Clark, A.F., Childs, G.E., and Wallace, G.H.	1970	V	4.0-273	Cartridge Brass	70.34 29.61	<0.1 each of Fe and Pb; 6.35 mm diameter x 152 mm long; machined; grain size 0.019 mm.
105 103	Clark, A.F., et al.	1970	V	4.0-273	Commercial Bronze	~89 10.01	<0.1 each of Fe and Pb; 6.35 mm diameter x 152 mm long; machined and annealed; grain size 0.051 mm.
106 103	Clark, A.F., et al.	1970	V	4.0-273	Red Brass	~84 15.33	<0.1 each of Fe, Mg, and Ag; 6.35 mm diameter x 152 mm long; cold-drawn; grain size 0.025 mm.
107* 253	Maxwell, E.	1946		293	Yellow Brass	80 20	Nominal composition; drawn; measuring temperature not given, here designated as 20 C.
108* 253	Maxwell, E.	1946		293	Yellow Brass	80 20	Similar to the above specimen but with machined surface.
109 253	Domenicali, C.A. and Otter, F.A.	1954		600		3.26	Calculated composition (3.17 a/o Zn).
110 253	Domenicali, C.A. and Otter, F.A.	1954		600		5.37	Calculated composition (5.23 a/o Zn).
111 253	Domenicali, C.A. and Otter, F.A.	1954		600		8.50	Calculated composition (8.28 a/o Zn).
112 254	Crisp, R.S., and Henry, W.G., and Schröder, P.A.	1964	A	4.2, 290		0.45	Calculated composition (0.44 a/o Zn); same original materials and fabrication method as the specimen for Data Set No. 57.
113* 254	Crisp, R.S., et al.	1964	A	4.2, 290		0.93	Calculated composition (0.90 a/o Zn); similar to the above specimen.
114* 254	Crisp, R.S., et al.	1964	A	4.2, 290		1.39	Calculated composition (1.35 a/o Zn); similar to the above specimen.
115* 254	Crisp, R.S., et al.	1964	A	4.2, 290		3.01	Calculated composition (2.93 a/o Zn); similar to the above specimen.
116* 254	Crisp, R.S., et al.	1964	A	4.2, 290		5.98	Calculated composition (5.83 a/o Zn); similar to the above specimen.
117* 254	Crisp, R.S., et al.	1964	A	4.2, 290		10.48	Calculated composition (10.23 a/o Zn); similar to the above specimen but quenched from 575 C after annealing.
118 254	Crisp, R.S., et al.	1964	A	4.2, 290		14.23	Calculated composition (13.98 a/o Zn); similar to the above specimen but without quenching.

\* Not shown in figure.

TABLE 37. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + ZINC ALLOYS (Temperature Dependence) (continued)

Data Ref. Set No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Zn	Composition (continued), Specifications, and Remarks
119* 254	Crisp, R.S., Henry, W.G., and Schröder, P.A.	1964	A	4.2, 290		18.89	Calculated composition (18.46 a/o Zn); similar to the above specimen but quenched from 575 C after annealing.
120* 254	Crisp, R.S., et al.	1964	A	4.2, 290		24.35	Calculated composition (23.83 a/o Zn); similar to the above specimen.
121* 255	Muto, Y. and Noto, K.	1964	A	4.2, 293		2.70	Calculated composition (2.63 a/o Zn); 99.9% pure; prepared from 99.998 pure Cu supplied by Furukawa Denki Kogyo Co.; after melting constituent metals together, ingot quenched and annealed in vacuum at 750 C for 6 hr, drawn into wire and annealed in vacuum at 450 C for 5 hr.
122* 255	Muto, Y. and Noto, K.	1964	A	4.2, 293		2.18	Calculated composition (2.12 a/o Zn); similar to the above specimen.
123* 255	Muto, Y. and Noto, K.	1964	A	4.2, 293		1.90	Calculated composition (1.85 a/o Zn); similar to the above specimen.
124* 255	Muto, Y. and Noto, K.	1964	A	4.2, 293		1.31	Calculated composition (1.27 a/o Zn); similar to the above specimen.
125* 255	Muto, Y. and Noto, K.	1964	A	4.2, 293		0.51	Calculated composition (0.50 a/o Zn); similar to the above specimen.
126* 255	Muto, Y. and Noto, K.	1964	A	4.2, 293		0.29	Calculated composition (0.28 a/o Zn); similar to the above specimen.
127 256	Fusfield, H.I.	1953		293	Brass	30	Polycrystalline; rod specimen 15 in. long; cold-worked and pulled in tension to 1 percent elongation; measured after annealed at various temperatures, $T_a$ , ranging from 293 to 533 K.
128* 256	Fusfield, H.I.	1953		293	Brass	30	Similar to the above specimen but pulled to 10% elongation.
129* 256	Fusfield, H.I.	1953		293	Brass	30	Similar to the above specimen but pulled to 20% elongation.
130 257	Ayers, J.D. and Massalski, T.B.	1972	V	637-778		38.67	Calculated composition (38 a/o Zn); $\beta$ -phase; 5 x 0.475 x 0.015 cm; cast; heat-treated and quenched.
131 257	Ayers, J.D. and Massalski, T.B.	1972	V	672-766		38.67	The above specimen pulse-heated at 400 C and cooled to room temperature; $\alpha$ -phase.
132 93	Scala, E. and Robertson, W.D.	1953	A	1385-1522		1.29	Calculated composition (1.25 a/o Zn); impurities: <0.001 each of Fe, Mg, and Si; in liquid state; contained in a refractory insulating tube about 0.6 cm I.D. and 13 cm long.
133* 77	Kapoor, A., Rowlands, J.A., and Woods, S.B.	1974	A	4.2	$\alpha$ -Brass	30.6	Calculated composition (30 a/o Zn); 4 mm diameter x 12 cm long; cast in air, swaged to 0.25 in. thick, and machined to size; cold-worked.
134* 77	Kapoor, A., et al.	1974	A	4.2	$\alpha$ -Brass	30.6	The above specimen annealed in argon at 873 K for 12 hr.
135* 77	Kapoor, A., et al.	1974	A	4.2	$\alpha$ -Brass	30.6	The above specimen reannealed in argon at 973 K for 12 hr.
136* 77	Kapoor, A., et al.	1974	A	4.2	$\alpha$ -Brass	30.6	The above specimen reannealed in argon at 1273 K for 12 hr.
137* 77	Kapoor, A., et al.	1974	A	4.2	Brass	30.6	Single crystal; 3 mm diameter x 15 cm long; supplied by Windsor Metal Crystal Inc., Ltd.
138 130	Sedström, E.	1919		273, 373		7.4	Calculated composition (7.1 a/o Zn); rolled and drawn to 1 mm in diameter; annealed at near melting point for 30 min.
139* 130	Sedström, E.	1919		273, 373		14.4	Calculated composition (14.0 a/o Zn); similar to the above specimen.

\* Not shown in figure.

TABLE 37. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER + ZINC ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Cu Zn	Composition (continued), Specifications, and Remarks
140	130	Sedström, E.	1919		273, 373		27.9	Calculated composition (27.3 a/o Zn); similar to the above specimen.
141	130	Sedström, E.	1919		273, 373		33.0	Calculated composition (32.4 a/o Zn); similar to the above specimen.
142*	258	Broom, T.	1952	B	90.2-373.2	75/25 brass	74.5	Reported composition: 74.5% Cu, 0.1% Fe, 0.05% Sn, <0.01% Pb, <<0.01% Mn; annealed at 600 C for 2 h and furnace cooled; successively drawn from 0.072 in diam to 0.022 in diam in liquid oxygen bath at 90.2 K.
143*	258	Broom, T.	1952	B	90.2-373.2	75/25 brass	74.5	Similar to the above specimen; successively drawn from 0.072 in diam to 0.022 in diam in bath of solid carbon dioxide in acetone at 194.7 K.
144*	258	Broom, T.	1952	B	90.2-373.2	95/25 brass	74.5	Similar to the above specimen; successively drawn from 0.072 in diam to 0.022 in diam in bath of ice and water at 273.2 K.
145*	258	Broom, T.	1952	B	90.2-373.2	75/25 brass	74.5	Similar to the above specimen; successively drawn from 0.072 in diam to 0.022 in diam in bath of boiling water at 373.2 K.

\* Not shown in figure.









TABLE 36. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF COPPER + ZINC ALLOYS (Temperature Dependence) (continued)

T	$\rho$	T	$\rho$	T	$\rho$	$\ln$ ( $\rho_{ann.}/$ $\rho_{def.}$ )
DATA SET 125*						
4.2	0.130	685	13.9*	4.2	3.82	
293.2	1.81	691	14.0*	DATA SET 134*		
DATA SET 126*						
4.2	0.0453	692	14.2*	4.2	3.77	
293.2	1.74	698	14.1*	DATA SET 135*		
DATA SET 127						
$T_a$ (K)	$\rho$	701	14.4	4.2	3.86	
293	5.995	704	14.8*	DATA SET 136*		
366	5.998	708	14.5*	4.2	3.86	
422	5.943	710	14.3*	DATA SET 137*		
478	5.938	722	14.6*	4.2	3.53	
533	5.961	725	14.9	DATA SET 138		
DATA SET 128*						
293	6.003	734	14.8*	273.2	4.39	
366	6.042	739	15.3*	373.2	5.65	
422	5.983	755	15.5*	DATA SET 139*		
478	5.996	778	15.7*	273.2	6.13	
533	6.002	DATA SET 131				
DATA SET 129*						
293	6.003	672	11.9	373.2	7.04	
366	6.042	686	11.9*	DATA SET 140		
422	5.983	696	12.4	273.2	7.63	
478	5.996	703	12.5*	373.2	8.00	
533	6.002	716	12.7*	DATA SET 141		
DATA SET 129*						
293	6.013	720	12.8*	273.2	7.94	
366	6.070	723	12.9*	373.2	8.40	
422	6.059	729	12.8*	$\ln$ ( $\rho_{ann.}/$ $\rho_{def.}$ )		
478	6.033	732	13.0*	DATA SET 142*		
533	6.083	734	13.1*	4.2	3.82	
DATA SET 129*						
293	6.013	738	13.1*	DATA SET 143*		
366	6.070	740	13.0	4.2	3.86	
422	6.059	752	13.2*	DATA SET 144*		
478	6.033	755	13.3	273.2	6.13	
533	6.083	764	13.3*	373.2	7.04	
DATA SET 132						
293	6.013	769	13.6*	DATA SET 145*		
366	6.070	786	14.3	4.2	3.82	
422	6.059	DATA SET 132				
478	6.033	1385	23.83	DATA SET 146*		
533	6.083	1428	24.63	4.2	3.86	
DATA SET 130						
637	13.0	1466	24.75	DATA SET 147*		
644	13.3*	1522	26.70	4.2	3.77	
646	13.4*	DATA SET 133*				
658	13.3*	4.2	4.59	4.2	3.86	
669	13.8*	DATA SET 133*				
670	13.7	DATA SET 133*				
672	13.8*	DATA SET 133*				

\* Not shown in figure.

TABLE 39. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF ZINC + COPPER ALLOYS (Temperature Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Zn Cu	Composition (continued), Specifications, and Remarks
1	228	Yonezumi, K. and Sato, T.	1956		296-789	CuZn	49.29	Calculated composition; 15.0 x 6.0 x 0.34 mm; 99.98% Cu and 99.98% Zn melted together in a Tamman tube and cast into a small plate 2 mm thick, annealed in the argon vapor for 1 hr at 500 C, polished by sand papers; measurements done while heating at a rate of 75 C hr <sup>-1</sup> from room temperature to 440 C and at a rate of 25 C hr <sup>-1</sup> from 440 C to 517 C. The above specimen measured after quenching from 517 C in air.
2	228	Yonezumi, K. and Sato, T.	1956		299	CuZn	49.29	Prepared from 99.95 Zn and 99.98 Cu; melted under a mixture of Sodium Chloride and potassium chloride, sucked up into quartz tubes of 1.2 mm diameter, sealed with argon gas, annealed for 50 hr at 550-600 C, then cooled slowly down to room temperature.
3	259	Sato, T. and Noguchi, S.	1957	A	4.2-293	Gamma Brass	41.6	Similar to the above specimen.
4	259	Sato, T. and Noguchi, S.	1957	A	4.2-293	Gamma Brass	38.0	Similar to the above specimen.
5	259	Sato, T. and Noguchi, S.	1957	A	4.2-293	Gamma Brass	35.4	Similar to the above specimen.
6	259	Sato, T. and Noguchi, S.	1957	A	4.2-293	Gamma Brass	33.3	Similar to the above specimen.
7	259	Sato, T. and Noguchi, S.	1957	A	4.2-293	Gamma Brass	31.5	Similar to the above specimen.
8	260	Martin, M.C.	1963	A	295.2	$\beta$ -brass	49.43	Calculated composition (Cu: Zn a/o = 1.0055); 99.9 pure; single crystal; supplied by Monocrystals Corp., Cleveland; annealed at 400 C for 1 hr, cooled to room temperature in furnace; quenched; measured after quenching at different temperatures ranging from 20 to 680 C.
9	93	Scala, E. and Robertson, W.D.	1953	A	700-1093		0.968	Calculated composition (0.996 a/o Cu); impurities: <0.001 Pb and Si; in liquid state; contained in a refractory insulating tube about 0.6 cm I.D. and 13 cm long.
10*	93	Scala, E. and Robertson, W.D.	1953	A	986, 1037		0.968	The above specimen measured during cooling.
11	109	Materials in Design Engineering	1959		293		0.65-1.25	0.006-0.016 Mg (nominal composition); density 7.17 g cm <sup>-3</sup> .

\* Not shown in figure.

TABLE 40. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF ZINC + COPPER ALLOYS (Temperature Dependence)  
 [Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \cdot m$ ]

T	$\rho$	T	$\rho$	T	$\rho$
DATA SET 1		DATA SET 6		DATA SET 10*	
298.2	4.95	4.2	2.69	986	35.6
324.2	5.34	77.4	3.31	1037	36.3
369.2	6.10	293.2	9.12	DATA SET 11	
422.2	7.11	DATA SET 7		293.2	6.31
472.2	7.75	4.2	3.09		
522.2	8.61	77.4	4.24		
566.2	9.51	293.2	10.33		
618.2	12.02				
620.2	10.72				
712.2	13.01				
720.2	13.83				
726.2	14.17				
733.2	14.45				
739.2	14.67*				
747.2	15.18				
757.2	15.63*				
767.2	15.96*				
780.2	16.18				
703.2	13.67*				
737.2	14.68*				
746.2	15.14*				
759.2	15.74*				
771.2	16.01*				
789.2	16.46*				
DATA SET 2					
299.2	5.00				
DATA SET 3					
4.2	5.10				
77.4	6.37				
293.2	13.04				
DATA SET 4					
4.2	4.95				
77.4	5.56				
293.2	12.09				
DATA SET 5					
4.2	4.01				
77.4	5.34				
293.2	10.72				

\* Not shown in figure.

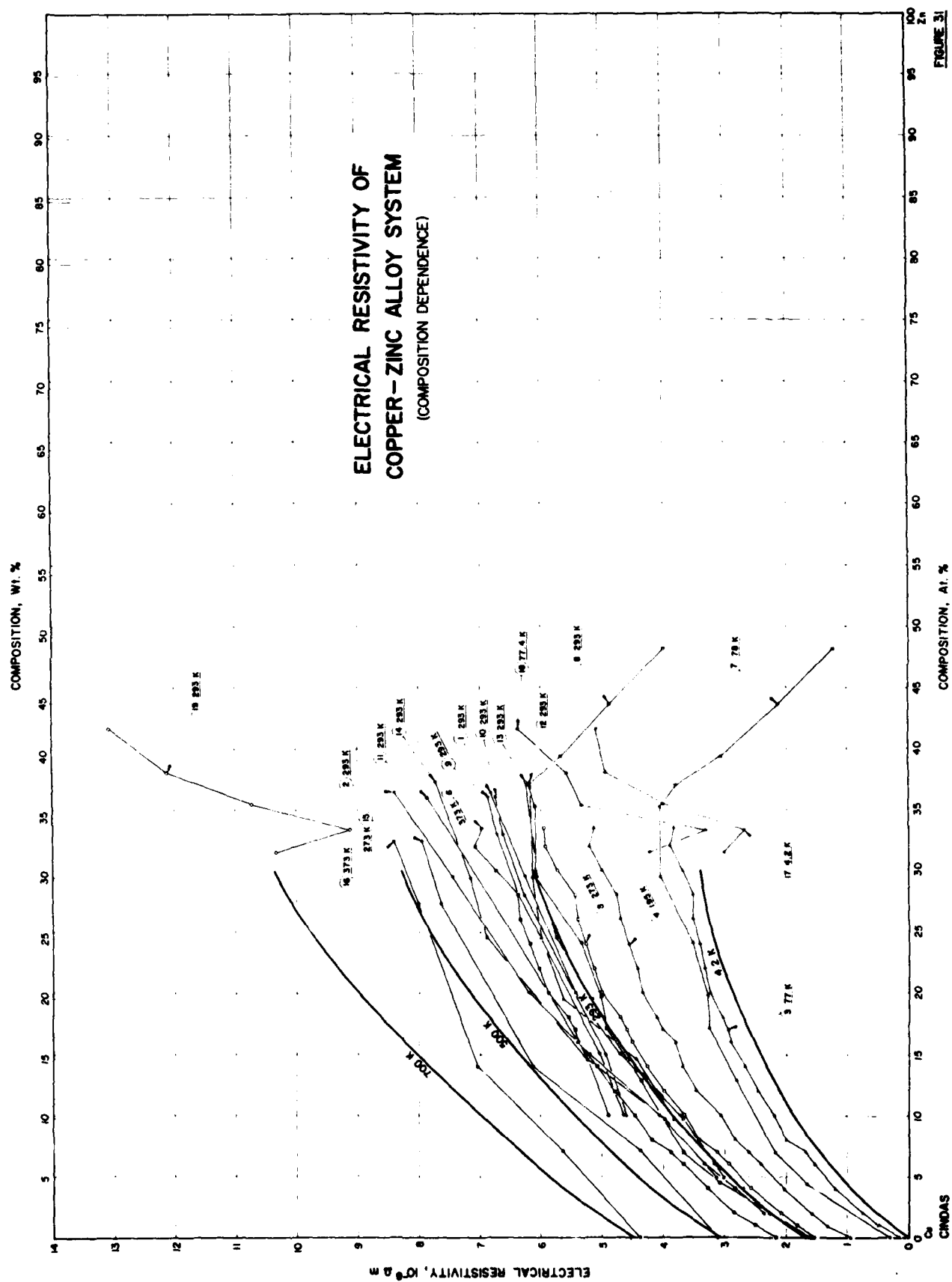


TABLE 41. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF COPPER-ZINC ALLOY SYSTEM (Composition Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp., K	Composition Range (At. % of Zn)	Composition (weight percent), Specifications, and Remarks
1	96	Köster, W. and Rave, H. P.	1964		293.2	2.0-36	Specimens recrystallized; measuring temperature not given, assigned here as 20 C.
2	96	Köster, W. and Rave, H. P.	1964		293.2	2.0-36	The above specimens 90% cold-worked.
3	239	Argent, B. B. and Lee, K. T.	1964	A	77	0-34	Wire specimens 0.039 in. in diameter; prepared from 99.99% pure metals by melting in sealed silica capsules, alloys homogenized at >1023 K for one week, rolled and drawn; annealed at 873 K, reannealed at same temperature for 1 h to give a grain size of 0.027 mm, again annealed at 873 K for 1 h, then air cooled rapidly to room temperature; reported error $\pm 2\%$ .
4	239	Argent, B. B. and Lee, K. T.	1964	A	195	0-34	The above specimens.
5	239	Argent, B. B. and Lee, K. T.	1964	A	273	0-34	The above specimens.
6	239	Argent, B. B. and Lee, K. T.	1964	A	373	0-34	The above specimens.
7	261	Aoyama, S. and Ito, T.	1940		78	0-48	Prepared from 99.967% pure electrolytic copper; annealed in nitrogen at 653 to 673 K for 20 h.
8	261	Aoyama, S. and Ito, T.	1940		293.2	0-48	The above specimens.
9	262	Köster, W. and Schüle, W.	1957		293	10-36	Specimens cold-rolled and recrystallized; measuring temperature not given, assigned here as 20 C.
10	262	Köster, W. and Schüle, W.	1957		293.2	10-36	The above specimens annealed at 773 K and cooled slowly; ordered.
11	262	Köster, W. and Schüle, W.	1957		293.2	10-36	The above specimens 75% cold-worked.
12	263	Presnyakov, A. A., Daukova, L. L., and Klyuchikov, Yu. F.	1961	B	293.2	4.9-37	Wire specimens 2 mm in diameter and 500 mm long wound into spirals; heated at 1073 K for 45 min. and quenched in ice water; measuring temperature not given, assigned here as 20 C; reported error $\pm 0.05\%$ .
13	263	Presnyakov, A. A., et al.	1961	B	293.2	4.9-37	The above specimens annealed at 873 K for 1 h.
14	263	Presnyakov, A. A., et al.	1961	B	293.2	4.9-37	The above specimens forged to 80% cold deformation.
15	130	Sedström, E.	1919		273.2	0-32	1 mm wire specimens; rolled and drawn; annealed at near melting point for 30 min.
16	130	Sedström, E.	1919		373.2	0-32	The above specimens.
17	259	Sato, T. and Noguchi, S.	1957	A	4.2	32-42	Specimens prepared from 99.95 Zn and 99.98 Cu; original materials melted in a flux of mixture of NaCl and KCl, sucked up into quartz tubes 1 to 2 mm in diameter and sealed with argon gas, annealed at 550 to 600 C for 50 h, then cooled slowly to room temperature.
18	259	Sato, T. and Noguchi, S.	1957	A	77.4	32-42	The above specimens.
19	259	Sato, T. and Noguchi, S.	1957	A	293.2	32-42	The above specimens.



### 3.7. Gold-Palladium Alloy System

Twenty data-source references are available for the electrical resistivity of the Au-Pd alloy system. From these references, a total of 93 data sets, six of which are merely single data points, have been extracted along with specimen characterization and measurement information. Forty-three of the data sets are for Au-Pd alloys covering the temperature range from 1.2 to 1400 K, 15 of the data sets are for Pd + Au alloys covering the temperature range from 4 to 1400 K, and 35 of the data sets are isotherms of the electrical resistivity as a function of composition. These experimental data sets are listed in tables 44, 46, and 48 which provide information on specimen characterization and measurement conditions, tabulated in tables 45, 47, and 49, and shown partially in figures 34, 35, and 37. In order to show both functional dependencies of the electrical resistivity, a few of the data sets have been presented in the tables and figures both for temperature dependence and for composition dependence.

Since the electrical resistivity of a specimen depends on the arrangement of its constituent atoms such as ordering, the crystal structure of Au-Pd alloys is briefly discussed here first. The preponderance of data indicates that the Au-Pd alloy system forms a continuous series of solid solutions [264]. However, electron diffraction analyses of thin film specimens by Nagasawa et al. [265] have indicated the existence of long-range ordered structures in Au alloys with 15, 25, and 30 at.% Pd when annealed at temperatures below about 1123 K, the possible existence of such structures near the stoichiometric composition  $\text{AuPd}_3$ , and the absence of such structures in alloys with 10 at.% Pd or 35 to 60 at.% Pd. The authors suggest that the reason why x-ray diffraction studies of bulk specimens have not shown similar evidence of super-lattice structure may be that the time required for homogenization of the alloys is much greater for the bulk specimens than for the thin films.

The existence of short-range order in the Au-Pd alloy system has been established through diffuse x-ray scattering studies of an alloy with 40 at.% Pd by Copeland and Nicholson [266], of alloys with 40, 50, and 60 at.% Pd by Iveronova and Katsnel'son [267,268], and of an alloy with 40 at.% Pd by Lin et al. [269]. These studies further indicated that (1) cold-worked samples can exhibit short-range order, (2) subsequent annealing of cold-worked samples at temperatures from about 373 to 773 K, when followed by quenching, tends to

increase the degree of short-range order, (3) subsequent annealing at temperatures of about 1073 K, when followed by quenching, tends to decrease the degree of short-range order, and (4) subsequent annealing at about 1073 K, when followed by slow cooling, may produce a greater degree of short-range order than any of the other described treatments of cold-worked samples. Devi et al. [270] carried out x-ray investigations of the temperature dependence of the lattice parameter of a 41% Pd alloy, but no deviation from linearity was found up to 873 K, and the authors concluded that measurements of the lattice parameter may not be sensitive to the presence of short-range order in Au-Pd alloys.

The effects of plastic deformation on the electrical resistivity of Au-Pd alloys have been studied by Köster and Halpern [271], Logie et al. [272], and Kim and Flanagan [232]. Normally, the effect of deformation is to increase the electrical resistivity, due to increased scattering from crystal defects introduced by the deformation. However, each of the above studies provided evidence for an anomalous decrease in the resistivity of annealed specimens upon deformation. The anomalous decrease occurs for a wide range of compositions, but it is a maximum for specimens with about 40 at.% Pd. Such specimens, when deformed by about 40% reduction in area, may show a resistivity decrease of nearly 6%. Kim and Flanagan [232], in their consideration of alternative explanations for the anomaly, concluded that cold-work reduces the degree of short-range order present in the specimen. This conclusion, together with the knowledge [232] that short-range ordering causes an increase in the resistivity of Au-Pd alloys, explains the anomaly. Although deformation does produce defects which tend to increase the resistivity, the destruction of short-range order (and the consequent resistivity decrease) overshadows the effect of the defects and results in a net decrease in resistivity.

Studies of the recovery kinetics of cold-worked or quenched specimens upon annealing by Kim and Flanagan [273], Haas and Lücke [274], and Lücke et al. [275] have given further insight into the processes and mechanisms of short-range ordering in Au-Pd alloys. The picture of short-range ordering derived from these studies and ones mentioned earlier is as follows. Above a certain high temperature, short-range ordering does not occur. Below this temperature, the degree of short-range order present at equilibrium increases with decreasing temperature. The formation of short-range order is dependent on the presence of crystal defects (such as vacancies, interstitials, etc.) which allow local



atomic ordering via their migration through the crystal. Such defects may be produced by plastic deformation, quenching, or irradiation. Kim and Flanagan studied the formation of short-range ordering via defects produced by cold-working while Lücke and co-workers primarily studied ordering via defects produced by quenching. As discussed before, the cold-working of annealed specimens may destroy short-range order while at the same time introducing defects which, upon subsequent annealing, may allow the formation of a greater degree of short-range order than was originally present in the annealed specimen. Not only does the formation of short-range order depend on the presence of defects, but the rate of ordering depends on the concentration of defects. The rate of ordering tends to increase with increasing defect concentration and with increasing temperature. In addition, different types of defects may come into play in ordering at different temperatures. Kim and Flanagan, based upon their analysis of ordering rates, described distinct stages of ordering which differ according to the mechanism of ordering and the types of defects involved at different temperatures.

The final state of short-range order achieved at a given temperature may depend on the thermal history of the specimen. For example, Lücke et al. [275] found that equilibrium values of short-range order may be difficult to attain. At low temperatures, the rate of diffusion may be too slow to attain equilibrium. This difficulty may in part be overcome by quenching the specimen from high temperatures, where the vacancy concentration is high, prior to annealing at a lower temperature. The surplus concentration of vacancies 'frozen-in' by the quench then serves to enhance the rate of ordering. In other cases, the excess vacancies may be annealed out before the equilibrium degree of short-range order is attained. With the resultant negligible vacancy concentration, no further ordering can occur and the final degree of short-range order is less than the equilibrium value. A constant resistivity value is thus reached, but it is lower than the resistivity value characteristic of the equilibrium state of short-range order.

The effect of short-range ordering on the magnitude of the electrical resistivity can be sizeable. In their studies of the anomalous decrease in resistivity of annealed alloys upon deformation, Kim and Flanagan [232] found that short-range ordering, even when opposed by the effects of cold-work, may contribute as much as 3% to the electrical resistivity of 25 at.% Pd alloys,

5% to 40 at.% Pd alloys, and 2% to 65 at.% Pd alloys. Similarly, in their studies of the increase of resistivity upon annealing cold-worked specimens, Kim and Flanagan [273] found that short-range ordering may contribute as much as 4 or 5% to the electrical resistivity of 25, 30, and 40 at.% Pd alloys, even when opposed by the effect of the annealing out of crystal defects. Further information was contributed by Lücke et al. [275] who found that the electrical resistivities of specimens in different states of short-range order may differ by as much as 8% for 50 at.% Pd alloys and 1% for 10 at.% Pd alloys.

Lücke et al. [275] (Au + Pd data sets 42 and 43) also investigated the effects of short-range order on the temperature dependence of the electrical resistivity. Using specimens with 50 at.% Pd, they quenched from 1273 K and then annealed at 513 K for different times to establish different degrees of short-range order. The resistivities of the specimens with different degrees of order were then measured from 73 to 473 K. The results showed differences in the magnitude of the resistivity of up to 8% and differences in the magnitude of the resistivity of up to 8% and differences in the temperature derivative of the resistivity (at 273 K) of up to about 3% for the different degrees of short-range order. Rowland et al. [276] (Au + Pd data sets 26-32, Pd + Au data sets 13-15, and Au-Pd data sets 13-28) speculated that a temperature dependence of short-range ordering might be a possible explanation of a very shallow minimum observed around 700 K for alloys with 30 to 60 at.% Pd).

The first step in the generation of recommended electrical resistivity values for this alloy system was the simultaneous examination of the residual resistivity ( $\rho_0$ ), the resistivity at 300 K ( $\rho_{300K}$ ), and the difference between the two, in a manner similar to that used in the analysis of the electrical resistivity of the Ag-Pd alloy system. Only three research documents have reported the residual resistivity of Au-Pd alloys over a wide range of compositions. One of the three documents, by Kim [277] (Au-Pd data set 34), was a thesis unavailable in its original form, so the data was extracted as reported by Kim and Flanagan [278] without specimen characterization or measurement information. Of the three, only Hau [279] (Au + Pd data sets 9-14, Pd + Au data sets 1-3, and Au-Pd data sets 4-6) and Rowland et al. [276] reported the resistivities at 4.2 K and near room temperature. The 1% inaccuracy in the measured resistivities, as reported by Hau, was about 4% better than the inaccuracy

reported by Rowland et al., whose measurement of the specimen dimensions was uncertain. Consequently, the values of Hau were followed rather closely in generating recommended values for the residual resistivity.

The residual resistivities of Hau [279] (Au-Pd data set 4) and Rowland et al. [276] (Au-Pd data set 13) show reasonable agreement over much of the composition range, but the values of Rowland et al. are lower than those of Hau by as much as  $1.3 \times 10^{-8} \Omega \text{m}$  in the 35 to 55 at.% Pd range. As mentioned above, this is the region where short-range ordering is known to have its maximum effect on the resistivity, with greater degrees of short-range order producing greater residual resistivities. The immediate question of whether the discrepancy between the two sets of data is due to a difference in the degree of short-range order present in the respective specimens is difficult to answer conclusively with the known information. The heat treatment of the specimens of Rowland et al., heating at 1073 K followed by slow cooling, could conceivably produce a significant degree of short-range order, as could the heat treatment of the specimens of Hau which consisted of heating at 973 K followed, presumably, by slow cooling. Unfortunately, Rowland et al. had insufficient knowledge of the structural details of their specimens to be certain of the degree of short-range ordering present and Hau reported that short-range order could not be detected in his experiment. Comparison of the data of Hau with the room-temperature resistivities reported by Köster and Halpern [271] (Au-Pd data sets 1-3) and Lücke et al. [275] (Au + Pd data sets 42-43) for alloys with 40 and 50 at.% Pd show that the resistivities reported by Hau are in all cases lower than the resistivities reported by the other authors. Since the specimens of Köster and Halpern and of Lücke et al. were known to possess some degree of short-range order, this information is consistent with the conclusion that the specimens of Hau possessed less short-range ordering than even the least ordered specimens of these other authors. However, the explanation of the discrepancy between the residual resistivity data of Rowland et al. and of Hau is still unclear.

The discrepancy between the residual resistivity data of Hau and of Rowland et al. raises immediate problems for the generation of recommended values of the electrical resistivity at high temperatures. Rowland et al. are the only investigators to report resistivities at high temperatures for a wide range of compositions. Given the discrepancy in residual resistivities, the

low temperature data of Hau, which were followed closely in generating recommended values, cannot simply or easily be joined with the higher temperature data of Rowland et al. This problem is diminished by an examination of  $(\rho_{300K} - \rho_0)$ , the temperature-dependent part of the resistivity at 300 K. While the residual resistivities of Hau and Rowland et al. differ by as much as  $1.3 \times 10^{-8} \Omega m$  in the 35 to 55 at.% Pd range, it was found that the temperature-dependent part of their resistivities differed by less than  $0.1 \times 10^{-8} \Omega m$  at 300 K in this composition range. Apparently, the difference in the specimens of Rowland et al. and Hau is such that only the residual resistivity is affected, and the temperature-dependent part of the resistivity remains largely unaffected. Thus, it becomes possible to synthesize the data of Rowland et al. and Hau if the analysis is carried out primarily in terms of the temperature-dependent part of the resistivity,  $(\rho - \rho_0)$ . This procedure assumes that any differences in the degree of short-range order present in the two sets of specimens have negligible effect on the temperature-dependent parts of the resistivities at all temperatures.

Accordingly, the temperature-dependent part of the resistivity was separated out in all cases where  $\rho_0$  was reported by the authors. In some cases, the authors did not report the residual resistivity, but it still seemed reasonable to estimate  $\rho_0$  for the specimen by using the recommended residual resistivity for the composition in question. In one case, the authors reported the temperature-dependent part of the resistivity directly. Kim and Flanagan [278] reported thermal resistivities based on values of the residual resistivity which were obtained "by extrapolation to the temperature at which the resistivities of pure metals over the same temperature range extrapolate to zero," and not by direct measurement. Comparison of these extrapolations with direct measurements was not possible because neither the extrapolated values nor the total resistivities were reported. Although the gross features of the thermal resistivity as reported by Kim and Flanagan, Hau, and Rowland et al. are similar, the magnitudes of the data of Kim and Flanagan are sometimes inconsistent with the more complete data reported by others. Consequently, the temperature-dependent resistivity data reported by Kim and Flanagan were given little weight in the generation of recommended values.

Smoothed isotherms of the temperature-dependent resistivity as a function of composition were generated by a cross-plotting procedure involving successive

smoothings of the temperature-dependent resistivity as a function of both temperature and composition. The final results for selected temperatures are shown in figure 36. For compositions from 0 to 17 at.% Pd, there is a gradual decrease in the temperature-dependent resistivity from its value for pure gold. This downward slope is supported at low temperatures by the data of Hau and, for dilute alloys up to 700 K by the data of Otter [226] (Au + Pd data sets 1-4). At high temperatures, the data of Rowland et al. also show a negative slope, but one of less magnitude than shown in figure 37. This is because the final smoothed values of the temperature-dependent resistivity pass below the data of Rowland et al. for their 10.5 at.% Pd. These deviations are not inconsistent with an error in sample dimensions, which was suggested as possible by Rowland et al., and result in deviations from the total resistivity reported by Rowland et al. which lie within their experimental inaccuracy of 5%. From a minimum at around 17 at.% Pd, the temperature-dependent resistivity at high temperatures rises to a maximum at about 40 at.% Pd, falls again to a minimum at about 50 at.% Pd, and then rises rapidly to the values for pure palladium.

The final total electrical resistivity values were obtained by adding the residual resistivities to the temperature-dependent resistivities for the selected compositions. The resulting recommended values for Au, Pd, and for 25 Au-Pd binary alloys are presented in table 43 and shown in figures 32 and 33. The recommended values are also shown in figure 37 as a function of composition together with some experimental data. The divergence of the data of Rowland et al. from the recommended values results primarily from differences in the residual resistivity, as discussed above. The recommended values for Au and for Pd are for well-annealed, high-purity specimens, but those values for temperatures below about 100 K are applicable only to Au and Pd having residual electrical resistivities as given at 1 K in table 43. The alloys for which the recommended values are generated have at most a low degree of short-range order and have not been quenched or cold-worked severely. The recommended values cover the temperature range from 1 K to 1400 K and are not corrected for the thermal expansion of the material. The estimated uncertainties in the values for the various alloys and for different temperature ranges are explicitly stated in a footnote to table 43. A few of the values for dilute gold-rich alloys are indicated as provisional because their uncertainties are greater than  $\pm 5\%$ .

TABLE 43. RECOMMENDED ELECTRICAL RESISTIVITY OF GOLD-PALLADIUM ALLOY SYSTEM †

[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Au: 100.00% (100.00 At.%) Pd: 0.00% (0.00 At.%)		Au: 99.50% (99.08 At.%) Pd: 0.50% (0.92 At.%)		Au: 99.00% (98.16 At.%) Pd: 1.00% (1.84 At.%)		Au: 97.00% (94.58 At.%) Pd: 3.00% (5.42 At.%)		Au: 95.00% (91.12 At.%) Pd: 5.00% (8.88 At.%)		Au: 90.00% (82.94 At.%) Pd: 10.00% (17.06 At.%)	
T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
1	0.0220	1	0.35*	1	0.69*	1	2.02*	1	3.31	1	6.17*
4	0.0220	4	0.35	4	0.69	4	2.02	4	3.31	4	6.17
7	0.0221	7	0.35*	7	0.69*	7	2.02	7	3.31	7	6.17
10	0.0226	10	0.35*	10	0.69*	10	2.02	10	3.31	10	6.17
15	0.0258	15	0.35*	15	0.70*	15	2.03	15	3.32	15	6.17
20	0.0346*	20	0.36*	20	0.71*	20	2.04	20	3.33	20	6.18
25	0.0503*	25	0.37*	25	0.73*	25	2.05	25	3.34	25	6.19
30	0.0727*	30	0.40*	30	0.75*	30	2.07	30	3.36	30	6.21
40	0.141*	40	0.46*	40	0.80*	40	2.13	40	3.41	40	6.26
50	0.222	50	0.54*	50	0.88*	50	2.21	50	3.49	50	6.33
60	0.309	60	0.63*	60	0.97*	60	2.29	60	3.57	60	6.41
70	0.396	70	0.72*	70	1.05*	70	2.37	70	3.65	70	6.49
80	0.482	80	0.80*	80	1.14*	80	2.46	80	3.72	80	6.56
90	0.567	90	0.89*	90	1.23*	90	2.54	90	3.80	90	6.63
100	0.652	100	0.97*	100	1.31*	100	2.61	100	3.88	100	6.70
150	1.063	150	1.38	150	1.71	150	2.99	150	4.25	150	7.07
200	1.464	200	1.78	200	2.10	200	3.38	200	4.63	200	7.45
250	1.865	250	2.20	250	2.51	250	3.80	250	5.04	250	7.84
273	2.052	273	2.38	273	2.69	273	3.99	273	5.21	273	8.01
293	2.214	293	2.54	293	2.86	293	4.14	293	5.35	293	8.17
300	2.271	300	2.59	300	2.91	300	4.20	300	5.41	300	8.22
350	2.683	350	3.00	350	3.32	350	4.58	350	5.79	350	8.56
400	3.102	400	3.41	400	3.73	400	4.97	400	6.17	400	8.93
500	3.962	500	4.24	500	4.55	500	5.75	500	6.93	500	9.69
600	4.853	600	5.11	600	5.41	600	6.55	600	7.71	600	10.47
700	5.780	700	6.02	700	6.31	700	7.40	700	8.53	700	11.30
800	6.755	800	6.99*	800	7.25*	800	8.31*	800	9.43	800	12.19
900	7.787	900	8.01*	900	8.26*	900	9.29*	900	10.40	900	13.11
1000	8.884	1000	9.13*	1000	9.36*	1000	10.36*	1000	11.43	1000	14.08
1100	10.057	1100	10.28*	1100	10.52*	1100	11.51*	1100	12.53	1100	15.14
1200	11.312	1200	11.53*	1200	11.75*	1200	11.75*	1200	13.72	1200	16.31
1300	12.632	1300	12.83*	1300	13.07*	1300	13.07*	1300	14.99*	1300	17.58
1337.58	13.146(a)	1348	13.53*	1355	13.87*	1390	15.20*	1400	16.34*	1400	18.96
1338	31.08(c)										
1700	36.26										

† Uncertainties in the electrical resistivity values are as follows:

100.00 Au - 0.00 Pd:  $\pm 1\%$  up to 10 K,  $\pm 2.5\%$  above 10 K to 15 K,  $\pm 6\%$  above 15 K to 40 K,  $\pm 3\%$  above 40 K to 80 K,  $\pm 1\%$  above 80 K to 500 K, and  $\pm 2.5\%$  above 500 K.  
 99.50 Au - 0.50 Pd:  $\pm 5\%$  up to 10 K,  $\pm 5\%$  above 10 K to 100 K, and  $\pm 5\%$  above 100 K.  
 99.00 Au - 1.00 Pd:  $\pm 5\%$  up to 10 K,  $\pm 7\%$  above 10 K to 100 K, and  $\pm 5\%$  above 100 K.  
 97.00 Au - 3.00 Pd:  $\pm 5\%$   
 95.00 Au - 5.00 Pd:  $\pm 5\%$   
 90.00 Au - 10.00 Pd:  $\pm 4\%$

\* Provisional value.

\* In temperature range where no experimental data are available.

TABLE 43. RECOMMENDED ELECTRICAL RESISTIVITY OF GOLD-PALLADIUM ALLOY SYSTEM† (continued)

[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Au 85.00% (75.38 At.%) Pd 15.00% (24.62 At.%)		Au 80.00% (68.36 At.%) Pd 20.00% (31.64 At.%)		Au 75.00% (61.84 At.%) Pd 25.00% (38.16 At.%)		Au 70.00% (55.76 At.%) Pd 30.00% (44.24 At.%)		Au 65.00% (50.98 At.%) Pd 35.00% (49.02 At.%)		Au 60.00% (44.76 At.%) Pd 40.00% (55.24 At.%)	
T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
1	8.57*	1	10.61*	1	12.49	1	15.68*	1	20.57*	1	23.54*
4	8.57*	4	10.61	4	12.49	4	15.68	4	20.57	4	23.54
7	8.57*	7	10.61	7	12.49	7	15.68	7	20.57	7	23.54
10	8.57*	10	10.62	10	12.49	10	15.68	10	20.57	10	23.54
15	8.58*	15	10.63	15	12.51	15	15.69	15	20.58	15	23.56
20	8.59*	20	10.64	20	12.53	20	15.70	20	20.60	20	23.58
25	8.61*	25	10.66	25	12.55	25	15.73	25	20.63	25	23.61
30	8.62*	30	10.68	30	12.57	30	15.76	30	20.67	30	23.65
40	8.67*	40	10.74	40	12.63	40	15.83	40	20.75	40	23.75
50	8.75*	50	10.82	50	12.73	50	15.93	50	20.85	50	23.88
60	8.83*	60	10.91	60	12.82	60	16.03	60	20.97	60	24.03
70	8.91*	70	10.99	70	12.92	70	16.14	70	21.10	70	24.19
80	8.99	80	11.07	80	13.00	80	16.24	80	21.22	80	24.34
90	9.06*	90	11.15	90	13.09	90	16.35	90	21.34	90	24.49
100	9.14*	100	11.23	100	13.19	100	16.44	100	21.45	100	24.64
150	9.52*	150	11.66	150	13.66	150	16.93	150	21.98	150	25.27
200	9.91*	200	12.11	200	14.14	200	17.44	200	22.54	200	25.84
250	10.31*	250	12.55	250	14.65	250	17.97	250	23.07	250	26.42
273	10.50*	273	12.75	273	14.89	273	18.23	273	23.32	273	26.70
293	10.66	293	12.93	293	15.09	293	18.46	293	23.53	293	26.94
300	10.72*	300	12.99	300	15.16	300	18.54	300	23.61	300	27.02
350	11.10*	350	13.45	350	15.73	350	19.10	350	24.14	350	27.63*
400	11.48*	400	13.93	400	16.29	400	19.67	400	24.67	400	28.23*
500	12.28*	500	14.91	500	17.43	500	20.83	500	25.73	500	29.38*
600	13.14*	600	15.91	600	18.61	600	21.92	600	26.74	600	30.41*
700	14.02*	700	16.90	700	19.77	700	22.92	700	27.61	700	31.31*
800	14.93*	800	17.87	800	20.85	800	23.84	800	28.38	800	32.17*
900	15.86*	900	18.87	900	21.88	900	24.75	900	29.17	900	33.05*
1000	16.85*	1000	19.94	1000	22.96	1000	25.69	1000	30.08	1000	34.01*
1100	17.96*	1100	21.12	1100	24.13	1100	26.72	1100	31.13	1100	35.06*
1200	19.19*	1200	22.39	1200	25.38	1200	27.89	1200	32.31	1200	36.20*
1300	20.51*	1300	23.71	1300	26.67	1300	29.16	1300	33.57	1300	37.43*
1400	21.86*	1400	25.02	1400	28.00	1400	30.49	1400	34.84	1400	38.72*

† Uncertainties in the electrical resistivity data are as follows:

85.00 Au - 15.00 Pd:  $\pm 3\%$  up to 400 K and  $\pm 4\%$  above 400 K.80.00 Au - 20.00 Pd:  $\pm 3\%$ 75.00 Au - 25.00 Pd:  $\pm 4\%$ 70.00 Au - 30.00 Pd:  $\pm 5\%$ 65.00 Au - 35.00 Pd:  $\pm 5\%$ 60.00 Au - 40.00 Pd:  $\pm 5\%$  up to 400 K and  $\pm 6\%$  above 400 K.

\* In temperature ranges where no experimental data are available.

TABLE 43. RECOMMENDED ELECTRICAL RESISTIVITY OF GOLD-PALLADIUM ALLOY SYSTEM† (continued)

[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Au: 55.00% (39.77 At.%) Pd: 45.00% (60.23 At.%)		Au: 50.00% (35.07 At.%) Pd: 50.00% (64.93 At.%)		Au: 45.00% (30.65 At.%) Pd: 55.00% (69.35 At.%)		Au: 40.00% (26.48 At.%) Pd: 60.00% (73.52 At.%)		Au: 35.00% (22.53 At.%) Pd: 65.00% (77.47 At.%)		Au: 30.00% (18.80 At.%) Pd: 70.00% (81.20 At.%)	
T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
1	23.18*	1	21.50*	1	19.57*	1	17.41*	1	15.03*	1	12.59*
4	23.18	4	21.50	4	19.57	4	17.41*	4	15.03*	4	12.59
7	23.18	7	21.50	7	19.57	7	17.41*	7	15.03*	7	12.59
10	23.18	10	21.50	10	19.57	10	17.41*	10	15.03*	10	12.59
15	23.19	15	21.51	15	19.59	15	17.42*	15	15.06*	15	12.61
20	23.21	20	21.53	20	19.63	20	17.45*	20	15.09*	20	12.65
25	23.25	25	21.57	25	19.67	25	17.49*	25	15.13*	25	12.69
30	23.30	30	21.62	30	19.72	30	17.54*	30	15.19*	30	12.75
40	23.41	40	21.75	40	19.84	40	17.70*	40	15.34*	40	12.93
50	23.56	50	21.92	50	20.03	50	17.91*	50	15.57*	50	13.18
60	23.74	60	22.13	60	20.27	60	18.17*	60	15.86*	60	13.48
70	23.94	70	22.37	70	20.54	70	18.47*	70	16.19*	70	13.84
80	24.14	80	22.60	80	20.81	80	18.77	80	16.52*	80	14.21
90	24.34	90	22.85	90	21.09	90	19.09	90	16.87*	90	14.58
100	24.54	100	23.09	100	21.36	100	19.40	100	17.21*	100	14.94
150	25.49	150	24.32	150	22.75	150	20.94	150	18.89*	150	16.74
200	26.41	200	25.55	200	24.14	200	22.47	200	20.51*	200	18.45
250	27.24	250	26.74	250	25.49	250	23.98	250	22.08*	250	20.07
273	27.61	273	27.23	273	26.10	273	24.65	273	22.81*	273	20.82
293	27.93	293	27.63	293	26.61	293	25.23	293	23.44*	293	21.49
300	28.04	300	27.76	300	26.79	300	25.42	300	23.66*	300	21.72
350	28.75	350	28.64*	350	27.91	350	26.74	350	25.15*	350	23.35
400	29.41	400	29.42*	400	28.90	400	27.95	400	26.56*	400	24.92
500	30.62	500	30.81*	500	30.03	500	30.08	500	29.09*	500	27.79
600	31.76	600	32.12*	600	32.16	600	31.94	600	31.30*	600	30.32
700	32.84	700	33.38*	700	33.59	700	33.64	700	33.28*	700	32.58
800	33.88	800	34.59*	800	34.95	800	35.19	800	35.06*	800	34.60
900	34.88	900	35.74*	900	36.25	900	36.62	900	36.67*	900	36.45
1000	35.86	1000	36.83*	1000	37.46	1000	37.94	1000	38.19*	1000	38.22
1100	36.91	1100	37.90*	1100	38.63	1100	39.22	1100	39.67*	1100	39.96
1200	38.00	1200	39.01*	1200	39.81	1200	40.49	1200	41.12*	1200	41.65
1300	39.16	1300	40.16*	1300	41.01	1300	41.75*	1300	42.53*	1300	43.22
1400	40.40	1400	41.31*	1400	42.15	1400	43.00*	1400	43.86*	1400	44.63

† Uncertainties in the electrical resistivity data are as follows:

55.00 Au - 45.00 Pd:  $\pm 2\%$  up to 600 K and  $\pm 3\%$  above 600 K.50.00 Au - 50.00 Pd:  $\pm 2\%$ 45.00 Au - 55.00 Pd:  $\pm 2\%$ 40.00 Au - 60.00 Pd:  $\pm 2\%$ 35.00 Au - 65.00 Pd:  $\pm 2\%$ 30.00 Au - 70.00 Pd:  $\pm 3\%$ 

\* In temperature range where no experimental data are available.



TABLE 43. RECOMMENDED ELECTRICAL RESISTIVITY OF GOLD-PALLADIUM ALLOY SYSTEM† (continued)

[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-4} \Omega \text{ m}$ ]

Au: 25.00% (15.26 At.%) Pd: 75.00% (84.74 At.%)		Au: 20.00% (11.90 At.%) Pd: 80.00% (88.10 At.%)		Au: 15.00% (8.70 At.%) Pd: 85.00% (91.30 At.%)		Au: 10.00% (5.68 At.%) Pd: 90.00% (94.34 At.%)		Au: 5.00% (2.76 At.%) Pd: 95.00% (97.24 At.%)		Au: 3.00% (1.64 At.%) Pd: 97.00% (98.36 At.%)	
T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
1	10.22*	1	7.97*	1	5.83*	1	3.79*	1	1.85*	1	1.10*
4	10.22	4	7.97*	4	5.83	4	3.79	4	1.85*	4	1.10
7	10.23	7	7.97*	7	5.83	7	3.79	7	1.85*	7	1.10*
10	10.24	10	7.97*	10	5.85	10	3.80	10	1.86*	10	1.10*
15	10.26	15	8.00*	15	5.87	15	3.83	15	1.88*	15	1.12*
20	10.30	20	8.04*	20	5.91	20	3.87	20	1.93*	20	1.15*
25	10.34	25	8.09*	25	5.96	25	3.92	25	1.98*	25	1.20*
30	10.40	30	8.15*	30	6.03	30	4.00	30	2.05*	30	1.26*
40	10.58	40	8.35*	40	6.22	40	4.18	40	2.23*	40	1.46*
50	10.84	50	8.62*	50	6.49	50	4.46	50	2.50*	50	1.73*
60	11.16	60	8.95*	60	6.84	60	4.82	60	2.85*	60	2.08*
70	11.54	70	9.34*	70	7.24	70	5.22	70	3.26*	70	2.48*
80	11.94	80	9.76*	80	7.66	80	5.65	80	3.69*	80	2.92*
90	12.34	90	10.19*	90	8.11	90	6.09	90	4.12*	90	3.34*
100	12.72	100	10.60*	100	8.54	100	6.54	100	4.59*	100	3.80*
150	14.62	150	12.61*	150	10.65	150	8.68	150	6.72*	150	5.94*
200	16.42	200	14.45*	200	12.56	200	10.65	200	8.75*	200	7.96*
250	18.10	250	16.15*	250	14.30	250	12.45	250	10.65*	250	9.89*
273	18.86	273	16.92	273	15.08	273	13.25	273	11.49*	273	10.75
293	19.53	293	17.60	293	15.77	293	13.95	293	12.21	293	11.51
300	19.77	300	17.85	300	16.01	300	14.20*	300	12.46*	300	11.77*
350	21.51	350	19.64*	350	17.80	350	16.00*	350	14.26*	350	13.67*
400	23.19	400	21.41*	400	19.61	400	17.81*	400	16.07*	400	15.49*
500	26.27	500	24.72*	500	23.05	500	21.31*	500	19.59*	500	18.93*
600	29.08	600	27.72*	600	26.18	600	24.52*	600	22.84*	600	22.14*
700	31.60	700	30.42*	700	28.99	700	27.43*	700	25.82*	700	25.12*
800	33.86	800	32.85*	800	31.56	800	30.09*	800	28.54*	800	27.87*
900	35.94	900	35.12*	900	33.95	900	32.56*	900	31.07*	900	30.43*
1000	37.97	1000	37.31*	1000	36.24	1000	34.88*	1000	33.43*	1000	32.83*
1100	39.95	1100	39.40*	1100	38.41	1100	37.05*	1100	35.64*	1100	35.06*
1200	41.78	1200	41.32*	1200	40.39	1200	39.05*	1200	37.68*	1200	37.12*
1300	43.42	1300	43.02*	1300	42.14	1300	40.85*	1300	39.52*	1300	38.96*
1400	44.89	1400	44.51*	1400	43.65	1400	42.44*	1400	41.15*	1400	40.60*

† Uncertainties in the electrical resistivity values are as follows:

- 25.00 Au - 75.00 Pd:  $\pm 3\%$   
 20.00 Au - 80.00 Pd:  $\pm 4\%$  up to 100 K and  $\pm 2\%$  above 100 K.  
 15.00 Au - 85.00 Pd:  $\pm 4\%$  up to 100 K and  $\pm 2\%$  above 100 K.  
 10.00 Au - 90.00 Pd:  $\pm 4\%$  up to 100 K and  $\pm 2\%$  above 100 K.  
 5.00 Au - 95.00 Pd:  $\pm 5\%$  up to 10 K,  $\pm 4\%$  above 10 K to 100 K, and  $\pm 2\%$  above 100 K.  
 3.00 Au - 97.00 Pd:  $\pm 5\%$  up to 10 K,  $\pm 5\%$  above 10 K to 100 K, and  $\pm 2\%$  above 100 K.

\* In temperature range where no experimental data are available.

TABLE 43. RECOMMENDED ELECTRICAL RESISTIVITY OF GOLD-PALLADIUM ALLOY SYSTEM† (continued)

[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Au: 1.00% (0.54 At.%) Pd: 99.00% (99.46 At.%)		Au: 0.50% (0.27 At.%) Pd: 99.50% (99.73 At.%)		Au: 0.00% (0.00 At.%) Pd: 100.00% (100.00 At.%)	
T	$\rho$	T	$\rho$	T	$\rho$
1	0.36*	1	0.18*	1	0.0200
4	0.36	4	0.18	4	0.0205
7	0.36*	7	0.18*	7	0.0217
10	0.36*	10	0.18*	10	0.0242
15	0.37*	15	0.21*	15	0.0346
20	0.41*	20	0.24*	20	0.0564
25	0.46*	25	0.29*	25	0.0938
30	0.52*	30	0.35*	30	0.151
40	0.70*	40	0.51*	40	0.335
50	0.96*	50	0.78*	50	0.607
60	1.30*	60	1.11*	60	0.940
70	1.70*	70	1.50*	70	1.32
80	2.13*	80	1.93*	80	1.75
90	2.56*	90	2.36*	90	2.19
100	3.01*	100	2.81*	100	2.63
150	5.17*	150	4.98*	150	4.91
200	7.20*	200	7.02*	200	6.99
250	9.17*	250	9.00*	250	8.88
273	10.07	273	9.89	273	9.78
293	10.85*	293	10.67*	293	10.54
300	11.12*	300	10.93*	300	10.80
350	12.99*	350	12.78*	350	12.66
400	14.80*	400	14.59*	400	14.46
500	18.22*	500	18.06*	500	17.89
600	21.43*	600	21.26*	600	21.10
700	24.42*	700	24.25*	700	24.10
800	27.21*	800	27.04*	800	26.89
900	29.79*	900	29.64*	900	29.50
1000	32.19*	1000	32.04*	1000	31.92
1100	34.43*	1100	34.27*	1200	36.21
1200	36.49*	1200	36.34*	1400	39.80
1300	38.37*	1300	38.22*	1600	42.70
1400	40.04*	1400	39.91*	1827	45.14 (s)
				1830	83.0 (t)
				2000	83.0

† Uncertainties in the electrical resistivity values are as follows:

1.00 Au - 99.00 Pd:  $\pm 3\%$  up to 10 K,  $\pm 5\%$  above 10 K to 100 K, and  $\pm 2\%$  above 100 K.0.50 Au - 99.50 Pd:  $\pm 3\%$  up to 10 K,  $\pm 5\%$  above 10 K to 100 K, and  $\pm 2\%$  above 100 K.0.00 Au - 100.00 Pd:  $\pm 2\%$  up to 40 K,  $\pm 1\%$  above 40 K to 350 K,  $\pm 2\%$  above 350 K to 1600 K,  $\pm 2.5\%$  above 1600 K to 1827 K, and  $\pm 5\%$  above 1827 K.

\* In temperature range where no experimental data are available.

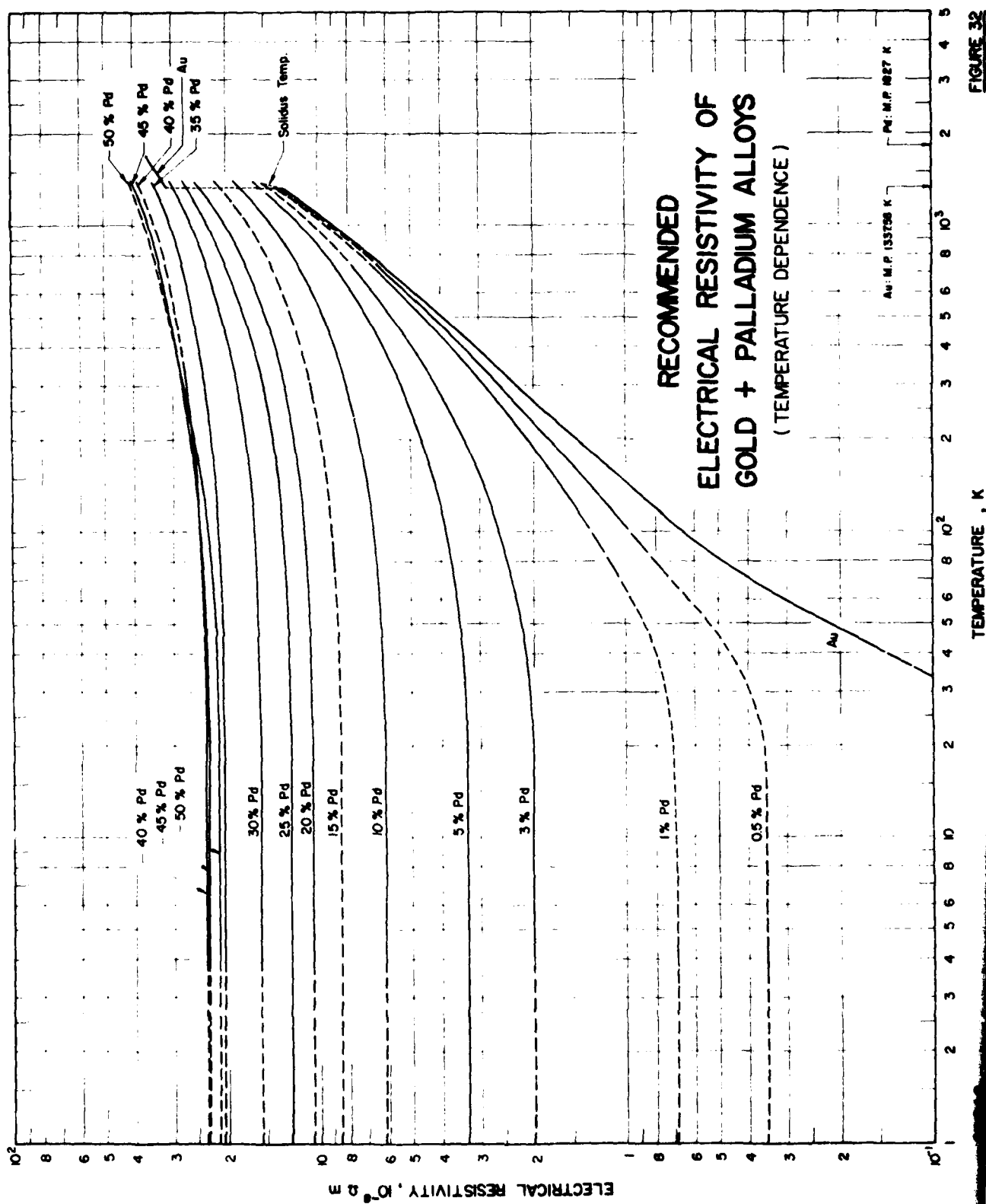


FIGURE 32

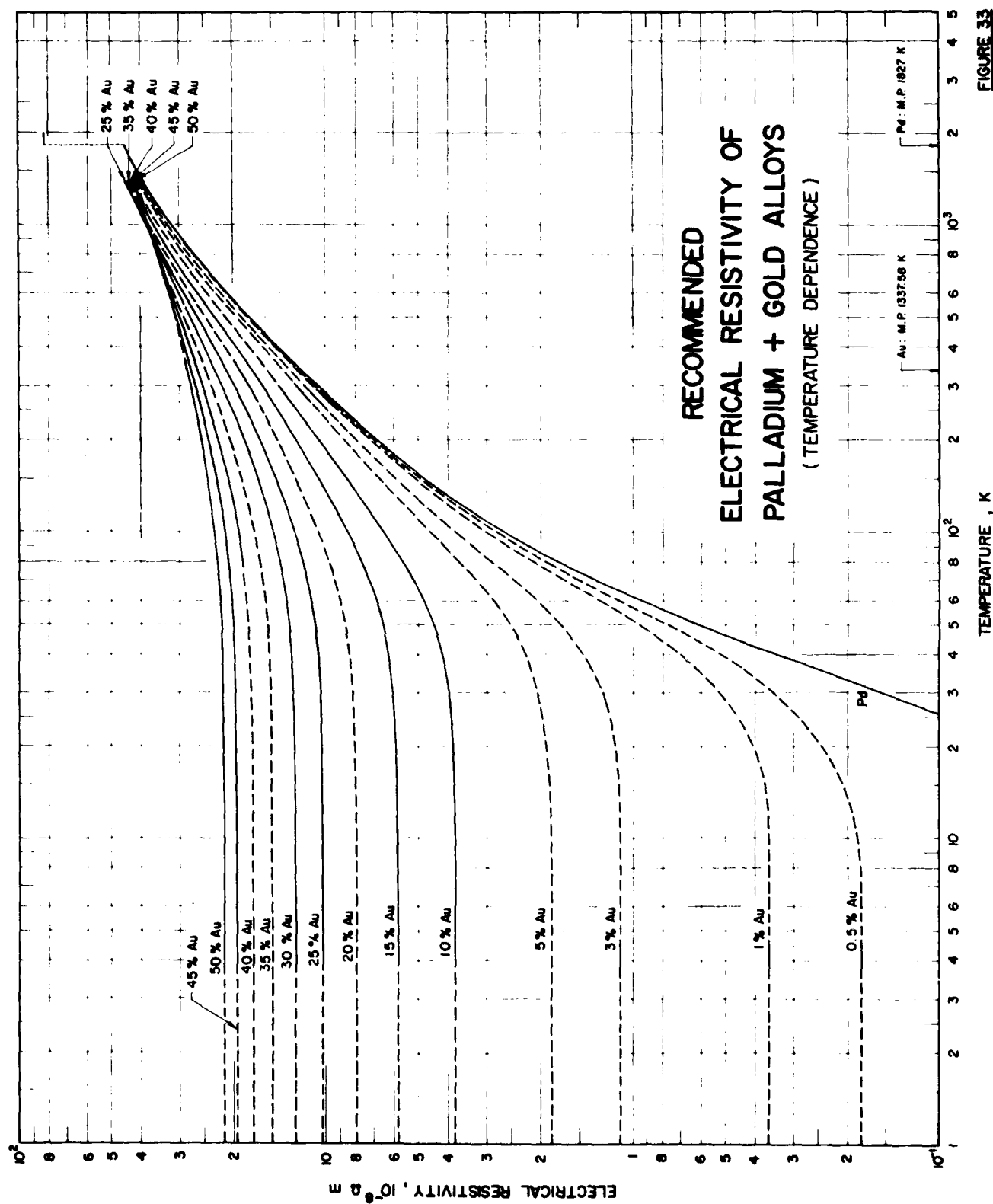


FIGURE 33

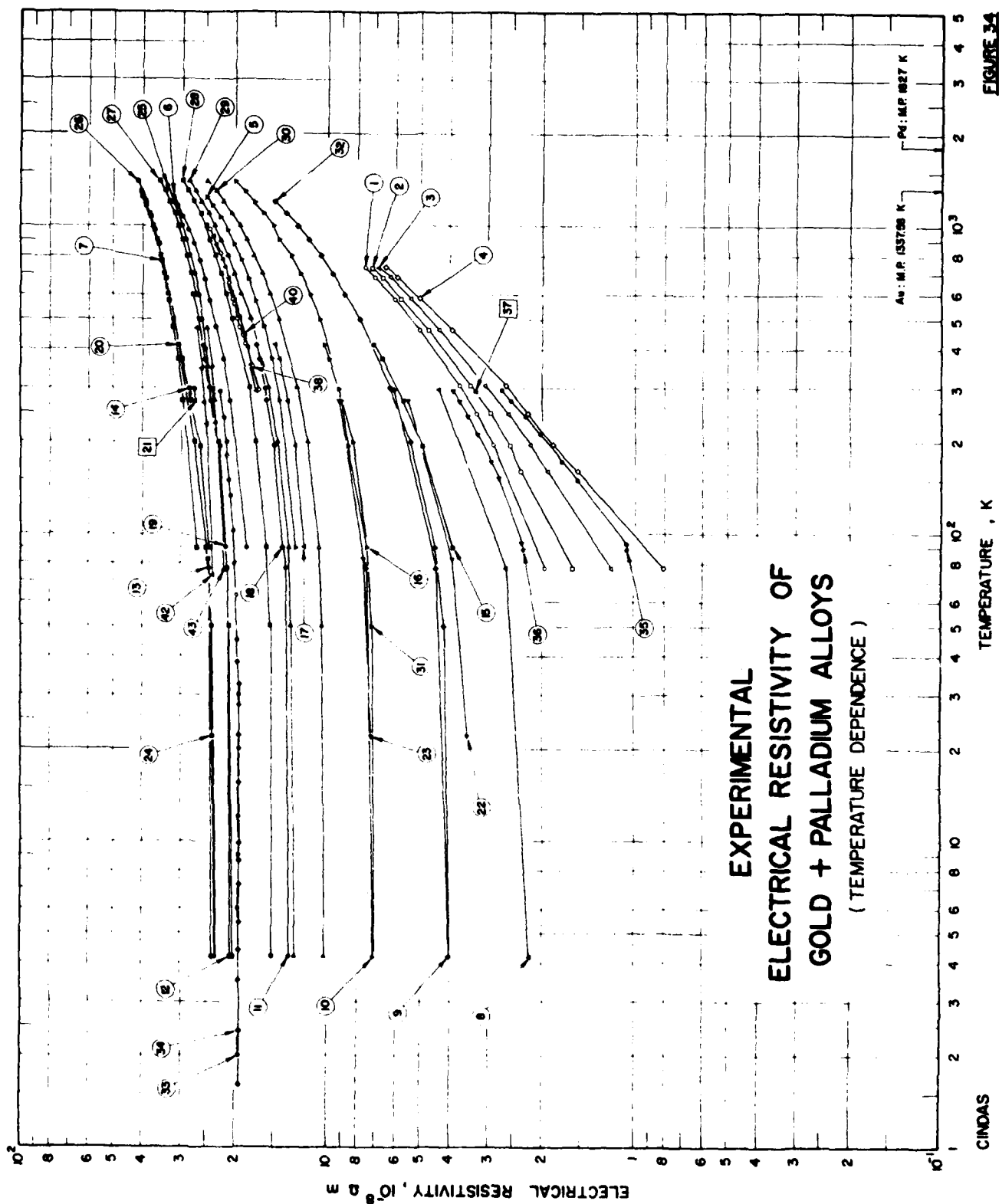


FIGURE 34

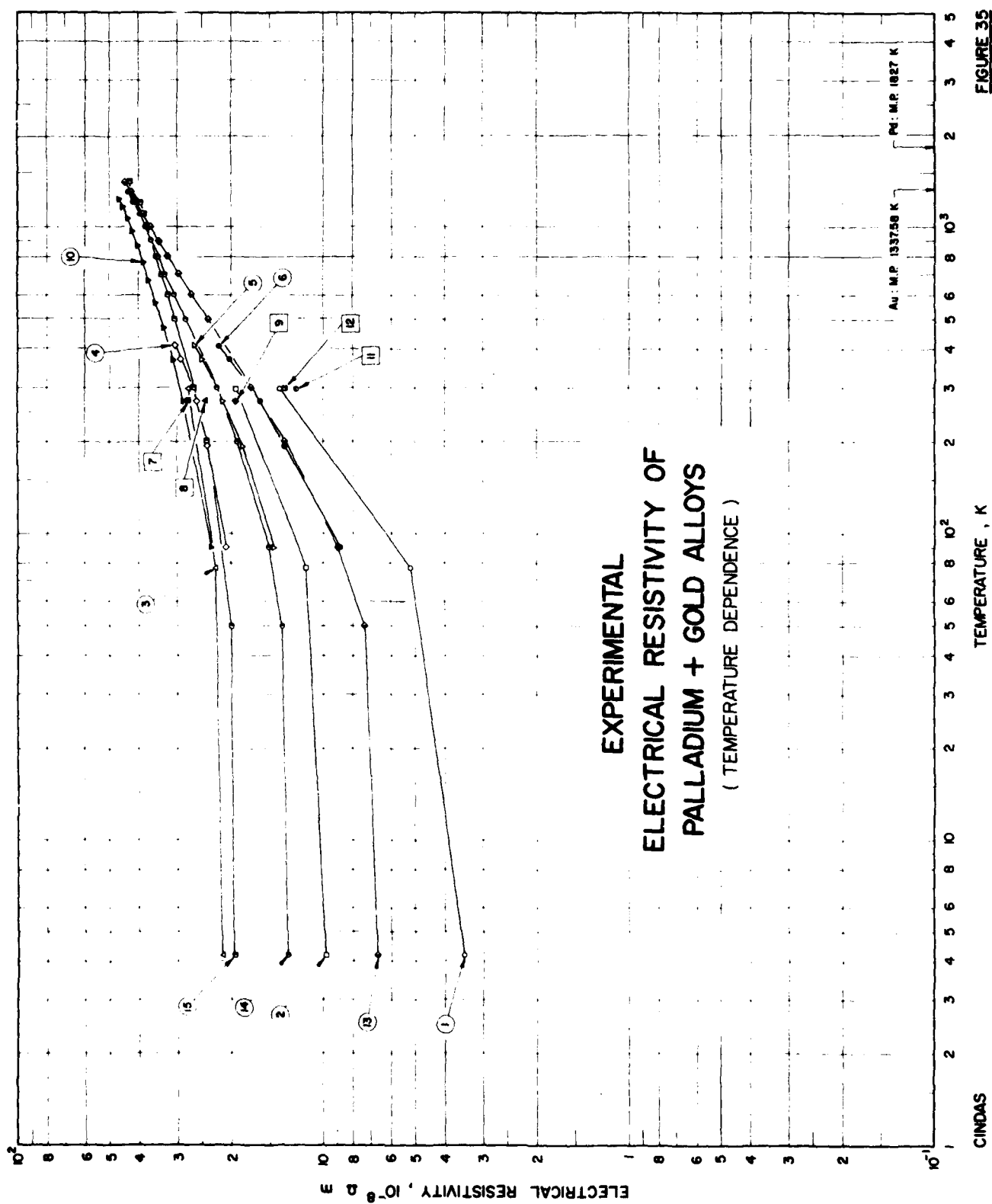


FIGURE 35

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TABLE 44. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD + PALLADIUM ALLOYS (Temperature Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Au Pd	Composition (continued), Specifications, and Remarks
1	225	Otter, F.A., Jr.	1956	A	77-738		2.20	Calculated composition (3.99 a/o Pd); 99.99% gold from Handy and Harman and 99.99% palladium from Bishop Platinum Co. melted in vacuum of less than $10^{-3}$ mm Hg in aluminum crucibles with an induction furnace, super heated approximately 100 K above the melting point, homogenized in vacuum for 24 h at 1173 K, swaged or drawn to wires approximately 30 mils in diam, and annealed for about 1 h at 773 to 823 K; measured in vacuum; data read from figure.
2	226	Otter, F.A., Jr.	1956	A	77-738		1.55	Calculated composition (3.01 a/o Pd); similar to the above specimen.
3	226	Otter, F.A., Jr.	1956	A	77-738		0.90	Calculated composition (1.92 a/o Pd); similar to the above specimen.
4	226	Otter, F.A., Jr.	1956	A	77-738		0.44	Calculated composition (0.81 a/o Pd); similar to the above specimen.
5	280	Combybeare, J. G. G.	1937	V	90-1250		77.5	Calculated composition (65 a/o Au); rod 2 mm in diam and 12 cm long obtained from Johnson Matthey and Co.; annealed until resistance became constant, authors found considerable increase in specific resistance after annealing which was attributed to growth upon annealing of fairly large crystals (observed microscopically) with small fissures between them; measured under vacuum of less than $10^{-6}$ mm Hg except at highest temperatures where pressure increased to $10^{-4}$ mm Hg at times due to evolution of gas; resistance ratio $RT/R_{273}$ and resistivity at 273 K, reported; reproducible to 0.1%; extracted from table and converted to resistivity without correcting for thermal expansion.
6	280	Combybeare, J. G. G.	1937	V	90-1250		69.3	Calculated composition (55 a/o Au); similar to the above specimen.
7	280	Combybeare, J. G. G.	1937	V	90-1250		55.2	Calculated composition (40 a/o Au); similar to the above specimen.
8	379	Hau, N.H.	1966	A	4.2-298	A95	96.767	Calculated composition (94.175 a/o Au), determined by weight and lattice parameter (4.0654 Å); 0.7 mm in diam and about 16 cm long; 99.999% pure Au splatters (0.0002 Ag; <0.0001 Mg, Si, Fe, and Cu) from American Smelting and Refining Co. and 99.99% pure Pd sponge (with no Fe impurity) from Engelhard Industries Inc. melted together in high-purity recrystallized alumina crucible by induction heating in Ar, molded into slugs, etched, homogenized in Ar for 6.8d at 1123 K, slowly cooled, swaged to smaller diam and drawn to final diam with intermediate anneals at 723 K in Ar, and finally annealed in vacuum of $10^{-6}$ mm Hg at 973 K for 24 h; uncertainty in temperature 0.01 K above 16 K and 0.1 K below 16 K, temperature controlled to within 0.01 K, uncertainty in resistivity of about 1%; data extracted from table.
9	379	Hau, N.H.	1966	A	4.2-298	A90	93.878	Calculated composition (89.229 a/o Au); similar to the above specimen but with lattice parameter of 4.0549 Å.
10	279	Hau, N.H.	1966	A	4.2-298	A80	88.296	Calculated composition (80.297 a/o Au); similar to the above specimen but with lattice parameter of 4.0304 Å.
11	279	Hau, N.H.	1966	A	4.2-298	A60	73.538	Calculated composition (60.019 a/o Au); similar to the above specimen but with lattice parameter of 3.9988 Å.
12	379	Hau, N.H.	1966	A	4.2-298	A50	65.066	Calculated composition (50.153 a/o Au); similar to the above specimen but with lattice parameter of 3.9831 Å and with intermediate anneals at 973 K as well as 723 K.

TABLE 44. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD + PALLADIUM ALLOYS (Temperature Dependence) (continued)

Date Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Au Pd	Composition (continued), Specifications, and Remarks
13	279	Hau, N.H.	1966	A	4.2-298	A45	59.835	Calculated composition (44.590 a/o Au); similar to the above specimen but with lattice parameter of 3.9719 Å.
14	279	Hau, N.H.	1966	A	4.2-298	A40	55.148	Calculated composition (39.858 a/o Au); similar to the above specimen but with lattice parameter of 3.9630 Å and with homogenization for 7.7d rather than 6.8d.
15	131	Röhl, H.	1933		90-413	AP1a	4.99	Calculated composition (8.86 a/o Pd); homogenized for 2 h at 1073 K, heat-treated for 2 h at 1073 K and for 10 h at 573 K; data extracted from table.
16	131	Röhl, H.	1933		90-413	AP1	9.96	Calculated composition (17.0 a/o Pd); similar to the above specimen.
17	131	Röhl, H.	1933		90-413	AP2	20.0	Calculated composition (31.6 a/o Pd); similar to the above specimen.
18	131	Röhl, H.	1933		90-413	AP3a	27.5	Calculated composition (41.2 a/o Pd); similar to the above specimen.
19	131	Röhl, H.	1933		90-413	AP3	35.0	Calculated composition (49.9 a/o Pd); similar to the above specimen.
20	131	Röhl, H.	1933		90-413	AP4	45.0	Calculated composition (60.2 a/o Pd); similar to the above specimen.
21	131	Röhl, H.	1933		273.2	AP8	39.8	Calculated composition (55.0 a/o Pd); similar to the above specimen.
22	128	Grüneisen, E. and Reddemann, H.	1934		22-273	22	4.99	Calculated composition (8.86 a/o Pd); annealed at 1073 K for 2 h; this is Röhl's specimen of Au + Pd data set 15; data extracted from table.
23	128	Grüneisen, E. and Reddemann, H.	1934		22-273	23	9.96	Calculated composition (17.0 a/o Pd); annealed at 1073 K for 2 h; this is Röhl's specimen of Au + Pd data set 16; data extracted from table.
24	128	Grüneisen, E. and Reddemann, H.	1934		22-273	24	39.8	Calculated composition (55.0 a/o Pd); annealed at 1073 K for 2 h; this is Röhl's specimen of Au + Pd data set 21; data extracted from table.
25	281	Laubitz, M.J. and Van der Meer, M.P.	1968	A	300-1350	7674	65.05 34.95	Rod specimen of Platinel 1503, 0.178 cm in diam and 20 cm long supplied by Englehard Industries, Inc; thermoelectric power measured as homogeneity check; annealed at 1100 K for 15 h after specimen and been assembled in measuring apparatus; measured in several cycles of increasing and decreasing temperature; 19 data points were fitted by $\rho = 2.236 \times 10^{-7} + 5.48 \times 10^{-11} T + 2.32 \times 10^{-14} T^2 \Omega \text{m}$ with a root mean square deviation of 0.55% for temperatures between 300 K and 1350 K; residual resistivity ratio, $\rho_{300}/\rho_4 = 1.133$ ; data uncorrected for thermal expansion; maximum possible error due to all causes known to author $\pm 0.5\%$ ; data calculated from above equation.
26	276	Rowland, T., Cusack, N.E., and Ross, R.G.	1974	A	4-1400		54.2	Calculated composition (39.0 a/o Au), concentration of main components measured to $\pm 0.3$ a/o by chemical and x-ray analysis, spectrographic analysis revealed major trace impurities of 30 ppm Fe, 20 ppm Ag, and 20 ppm Si; wire specimen between 0.31 and 0.38 mm in diam prepared by author at Johnson Matthey and Co. Ltd. from sponges of spectroscopically pure Pd and 99.999 pure Au by pressing sponge into bars, sintering in vacuum at 1175 K for several hours, induction melting in stream of pure argon, cold rolling ingots to 3 mm square rods, and by finally drawing down to final diam; annealed at 1073 K in vacuum for a few hours and slowly furnace cooled; residual resistivity ratio, $\rho_{300}/\rho_{1400} = 0.8209 \pm 5\%$ uncertainty in resistivity due mainly to uncertain diam; data from tables supplied by author in private communication.



TABLE 44. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD + PALLADIUM ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Au Pd	Composition (continued), Specifications, and Remarks
27	276	Rowland, T., Cusack, N.E., and Ross, R.G.	1974	A	4-1400		63.5	Calculated composition (48.5 a/o Au); similar to the above specimen, but with major trace impurities of 30 ppm Fe and 30 ppm Ag and $\rho_{L,2}/\rho_{T,2} = 0.8780$ .
28	276	Rowland, T., et al.	1974	A	4-1400		69.2	Calculated composition (54.8 a/o Au); similar to the above specimen, but with major trace impurities of 50 ppm Fe and 30 ppm Ag and $\rho_{L,2}/\rho_{T,2} = 0.8476$ .
29	276	Rowland, T., et al.	1974	A	4-1400		73.0	Calculated composition (59.3 a/o Au); similar to the above specimen, but with major trace impurities of 30 ppm Cu, 20 ppm Fe, and 20 ppm Pt and with $\rho_{L,2}/\rho_{T,2} = 0.8318$ .
30	276	Rowland, T., et al.	1974	A	4-1400		80.5	Calculated composition (69.0 a/o Au); similar to the above specimen, but with major trace impurities of 50 ppm Pt, 20 ppm Ag, and 20 ppm Fe and with $\rho_{L,2}/\rho_{T,2} = 0.8273$ .
31	276	Rowland, T., et al.	1974	A	4-1400		87.8	Calculated composition (79.5 a/o Au); similar to the above specimen, but with major trace impurities of 50 ppm Fe and 30 ppm Ag and $\rho_{L,2}/\rho_{T,2} = 0.7921$ .
32	276	Rowland, T., et al.	1974	A	4-1200		94.0	Calculated composition (89.5 a/o Au); similar to the above specimen, but with major trace impurities of 50 ppm Fe and 50 ppm Rh and $\rho_{L,2}/\rho_{T,2} = 0.6611$ .
33	282	Ugodnikova, D.A., Beilin, B.M., and Tsvetkov, Yu. Yu. N.	1974		2-292		73.5 26.5	Calculated composition (40 a/o Pd); metallographic and x-ray analyses showed single phase structure; 0.1 mm diam wire specimens; annealed from 1073 K; data read from figure.
34	292	Ugodnikova, D.A., et al.	1974		1.6-38.2		73.5 26.5	The above specimen; data read from crude, expanded figure.
35	37	Linde, J.O.	1931		89-296		0.625	Calculated composition (1.15 a/o Pd); mint grade Au and extra pure Pd from Heraeus melted together under vacuum of about 0.1 mm Hg in quartz and magnesium oxide powder crucibles, homogenized 10-20 h at temperature near the melting point; formed into square rod of 1.2 mm thickness, drawn through steel or sapphire dies to wire of 0.75 mm diam and 1.2 cm length, and finally annealed at 773 K; diam measured with micrometer to $\pm 0.03$ mm; data extracted from table.
36	37	Linde, J.O.	1931		89-296		2.46	Calculated composition (4.46 a/o Pd); similar to the above specimen.
37	37	Linde, J.O.	1931		291		1.58	Calculated composition (2.88 a/o Pd); similar to the above specimen.
38	271	Köster, W. and Halpern, T.	1961		295-982		26.5	Calculated composition (40 a/o Pd); 99.95 pure Pd and 99.99 pure Au from Fa. Degussa Hanau melted together, homogenized in vacuum at 1073 to 1223 K for 70 h and furnace-cooled; reduced 98.5% by cold-rolling from a thickness of 7 mm to 0.1 mm without intermediate annealing; data read from figure.
39*	271	Köster, W. and Halpern, T.	1961		294-1072		26.5	The above specimen heated at 1373 K and quenched in water.
40	271	Köster, W. and Halpern, T.	1961		293-923		26.5	The above specimen annealed at 1073 K and cooled slowly in furnace.

\* Not shown in figure.

TABLE 44. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD + PALLADIUM ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Au Pd	Composition (continued), Specifications, and Remarks
41*	283	Bücklund, N.	1958	A	1.2-4.2		0.157	Calculated composition (0.29 a/o Pd); spectroscopically pure Pd and Au from Johnson, Matthey and Co. Ltd melted together in evacuated silica tubes in high-frequency furnace or wire-wound furnace; ingots rolled slightly, homogenized at 1073-1173 K for about 2 h, and drawn to 0.2 mm diam wire; the sample was one of several which were annealed at about 773 K when quenching from higher temperatures was not necessary; data read from figure.
42	275	Lücke, K., Haas, H., and Schulze, H. A.	1976		73-469		35.1	Calculated composition (50 a/o Pd); 0.1 mm thick foils prepared from 99.998 pure Au and 99.99 pure Pd; quenched from 1273 K and annealed at 513 K prior to measurement; accuracy of resistivity measurements $3 \times 10^{-3}$ ; data read from figure.
43	275	Lücke, K., et al.	1976		76-472		35.1	The above specimen annealed at 513 K for a different period of time in order to produce a different state of short-range order; several different annealing times were used by the authors, but not specified quantitatively; data sets 42 and 43 represent the highest and lowest resistivities produced by varying the annealing time.

\* Not shown in figure.

TABLE 45. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF GOLD + PALLADIUM ALLOYS (Temperature Dependence)

[illegible]

\* Not shown in figure.



TABLE 46. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF PALLADIUM + GOLD ALLOYS (Temperature Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Pd Au	Composition (continued), Specifications, and Remarks
1	279	Hau, N. H.	1966	A	4.2-298	A5	9.226	Calculated composition (5.206 a/o Au), determined by weight and lattice parameter (3.894 Å); 0.07 mm in diam and about 16 cm long; 99.999% pure Au splatters (0.0002 Ag, < 0.0001 Mg, Si, Fe, and Cu) from American Smelting and Refining Co. and 99.99% pure Pd sponge (with no Fe impurity) from Engelhard Industries Inc. melted together in high-purity recrystallized alumina crucible by induction heating in Ar, molded into slugs, etched, homogenized in Ar for 7.7d at 1123 K, slowly cooled, swaged to smaller diam and drawn down to final diam with intermediate anneals in Ar at 973 and 723 K, and finally annealed in vacuum of $10^{-6}$ mm Hg at 973 K for 24 h; uncertainty in temperature 0.01 K above 16 K and 0.1 K below 16 K; temperature controlled to within 0.01 K, uncertainty in resistivity of about 1%; data extracted from table.
2	279	Hau, N. H.	1966	A	4.2-298	A15	24.265	Calculated composition (14.754 a/o Au); similar to the above specimen but with lattice parameter of 3.9168 Å.
3	279	Hau, N. H.	1966	A	4.2-298	A35	49.999	Calculated composition (35.072 a/o Au); similar to the above specimen but with lattice parameter of 3.9549 Å.
4	131	Röhl, H.	1933		90-413	AP5	56.0	Calculated composition (70.2 a/o Pd); homogenized 2 h at 1073 K, heat treated 2 h at 1073 K and 10 h at 573 K; data extracted from table.
5	131	Röhl, H.	1933		90-413	AP6	70.0	Calculated composition (81.2 a/o Pd); similar to the above specimen.
6	131	Röhl, H.	1933		90-413	AP7	85.0	Calculated composition (91.3 a/o Pd); similar to the above specimen.
7	131	Röhl, H.	1933		273	AP9	50.1	Calculated composition (65.0 a/o Pd); similar to the above specimen.
8	131	Röhl, H.	1933		273	AP10	61.8	Calculated composition (75.0 a/o Pd); similar to the above specimen.
9	131	Röhl, H.	1933		273	AP11	76.8	Calculated composition (86.0 a/o Pd); similar to the above specimen.
10	280	Conybeare, J. G. G.	1937	V	90-1250		38.2	Calculated composition (25 a/o Au); rod 2 mm in diam and 12 cm long obtained from Johnson Matthey and Co.; annealed until resistance became constant; considerable increase in specific resistance upon annealing was attributed to growth of fairly large crystals (observed microscopically) with small fissures between them; measured under vacuum of less than $10^{-4}$ mm Hg except at highest temperatures where pressure increased to $10^{-4}$ mm Hg due to evolution of gas; resistance ratio $R_T/R_{273}$ and resistivity at 273 K were extracted from table and converted by CINDAS to resistivity without correcting for thermal expansion; resistance ratios were reproducible to 0.1%.
11	284	Köster, W. and Hagmann, D.	1961		293		3.63	Calculated composition (2 a/o Au); 0.1 mm diam wire specimen supplied by Fa. Degussa; annealed at 1073 K for 1 h and furnace-cooled; data extracted from table, reported data were multiplied by 10 to account for incorrect placement of the decimal in the table; measurement temperature not stated explicitly, assumed to be 293 K.
12	284	Köster, W. and Hagmann, D.	1961		293		7.07	Calculated composition (4 a/o Au); similar to the above specimen.

TABLE 46. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF PALLADIUM + GOLD ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Pd Au	Composition (continued), Specifications, and Remarks
13	276	Rowland, T., Cusack, N.E., and Reed, R.G.	1974	A	4-1400		16.0	Calculated composition (9.3 a/o Au), concentration of main components measured to $\pm 0.3$ a/o by chemical and x-ray analysis, spectrographic analysis revealed major trace impurities of 10 ppm Fe and 10 ppm Ag; wire specimen between 0.31 and 0.38 mm in diam prepared by author at Johnson Matthey and Co. Ltd. from sponges of spectroscopically pure Pd and 99.999 pure Au by pressing sponge into bars, sintering in vacuum at 1173 K for several hours, induction melting in stream of pure argon, cold-rolling ingots to 3 mm square rods, and by finally drawing down to final diam; annealed at 1073 K in vacuum for a few hours and slowly furnace cooled; residual resistivity ratio, $\rho_{\infty}/\rho_{273} = 0.4106 \pm 5\%$ uncertainty in resistivity, due mainly to uncertain diam; data from tables supplied by author in private communication.
14	276	Rowland, T., et al.	1974	A	4-1400		31.0	Calculated composition (19.5 a/o Au); similar to the above specimen but with major trace impurities of 20 ppm Ag and 30 ppm Fe and $\rho_{\infty}/\rho_{273} = 0.6128$ .
15	276	Rowland, T., et al.	1974	A	4-1400		44.8	Calculated composition (30.5 a/o Au); similar to the above specimen but with major trace impurities of 20 ppm Fe and 40 ppm Ag and $\rho_{\infty}/\rho_{273} = 0.7351$ .

TABLE 47. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF PALLADIUM + GOLD ALLOYS (Temperature Dependence)  
[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega\text{m}$ ]

T	$\rho$	T	$\rho$	T	$\rho$
DATA SET 1		DATA SET 8		DATA SET 14	
4.2	5.480	273.2	24.4	4	13.15
77.3	5.205			50	13.75
298	13.93	DATA SET 9		90	15.13
		273.2	19.5	200	19.05
DATA SET 2				300	22.30
4.2	9.860	DATA SET 10		500	28.20
77.3	11.46	90.2	23.4	600	30.65
298	19.33	273.2	28.7	700	32.79
		373	31.1	800	34.85
DATA SET 3		473	33.4	900	36.60
4.2	21.50	573	35.5	1000	38.40
77.3	22.64	673	37.3	1100	39.98
298	27.76	773	38.9	1200	41.66
		873	40.6	1300	43.10
DATA SET 4		973	42.2	1400	44.60*
90.2	20.9	1073	43.8		
194.7	24.0	1173	45.5	DATA SET 15	
273.2	26.0	1250	46.8	4	19.58
373.2	29.4			50	20.00
413.2	30.7	DATA SET 11		90	20.93*
		293	12.3	200	24.05
DATA SET 5				300	26.60
90.2	14.6	DATA SET 12		500	30.60
194.7	18.4	293	13.4	600	32.28
273.2	21.3			700	33.78
373.2	25.0	DATA SET 13		800	35.00
413.2	26.5	4	6.68	900	36.50*
		50	7.35	1000	37.40
DATA SET 6		90	8.95	1100	38.60
90.2	8.83	200	13.45	1200	39.80
194.7	13.4	300	17.25	1300	42.50*
273.2	16.1	500	23.90	1400	43.00
373.2	20.4	600	27.00		
413.2	22.1	700	29.78		
		800	32.32		
DATA SET 7		900	34.60		
273.2	27.7	1000	36.85		
		1100	39.10		
		1200	41.10		
		1300	42.90		
		1400	44.70		

\* Not shown in figure.

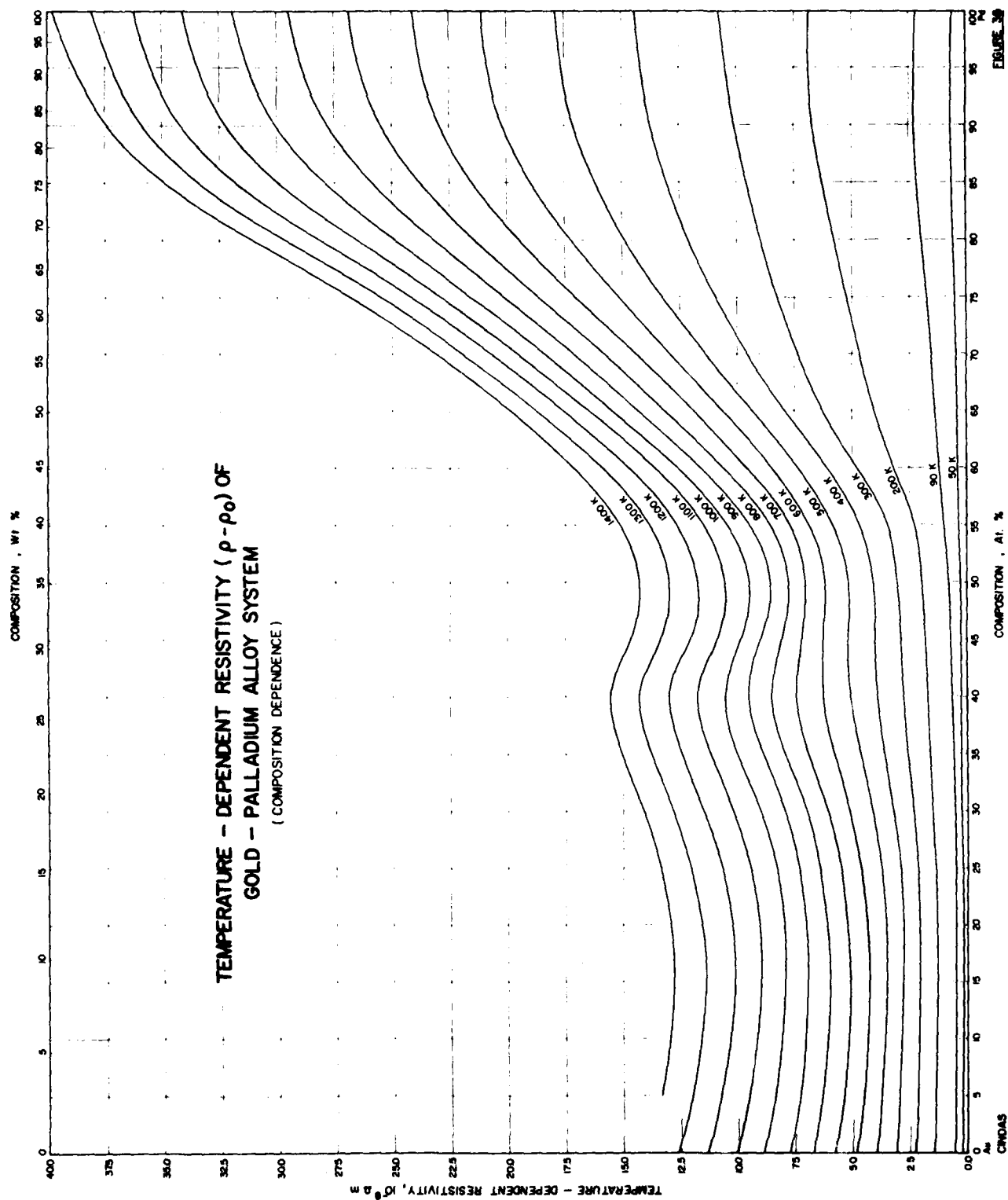


FIGURE 39

COMPOSITION, At. %

CINDAS



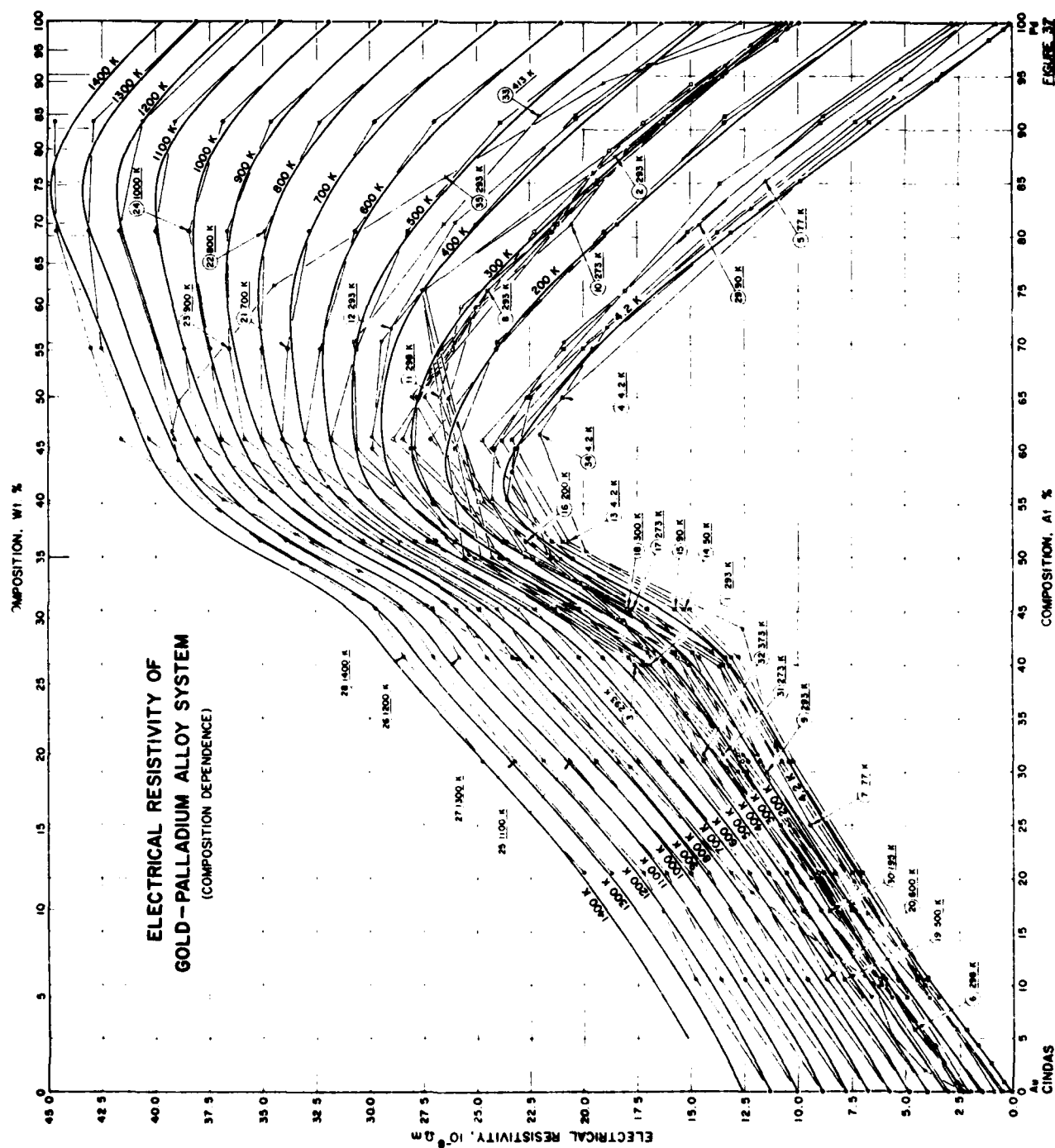


FIGURE 37

TABLE 48. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD-PALLADIUM ALLOY SYSTEM (Composition Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp., K	Composition Range (At. % of Pd)	Composition (weight percent), Specifications, and Remarks
1	271	Köster, W. and Halpern, T.	1961		293	0-100	99.95 pure Pd and 99.99 pure Au from Fa. Degussa Hanau melted together, homogenized in vacuum at 1073 K for 70 h and furnace-cooled; reduced 98.5% by cold-rolling from a thickness of 7 mm to 0.1 mm without intermediate annealing; measurement temperature not stated explicitly, assumed to be 293 K; data extracted from table.
2	271	Köster, W. and Halpern, T.	1961		293	0-100	The above specimens annealed at 973 K and quenched in water.
3	271	Köster, W. and Halpern, T.	1961		293	20-75	The above specimens annealed at 1073 K and cooled slowly in furnace.
4	279	Hau, N.H.	1966	A	4.2	0-100	Compositions determined by weight and lattice parameter; 0.07 mm in diam and about 16 cm long; 99.999% pure Au splatters (0.0002 Ag, <0.0001 Ni, Si, Fe, and Cu) from American Smelting and Refining Co. and 99.99% pure Pd sponge (with no Fe impurity) from Engelhard Industries, Inc. melted together in high-purity recrystallized alumina crucibles by induction heating in Ar, molded into slugs, etched, homogenized in Ar for about 7 d at 1123 K, slowly cooled, swaged to smaller diam and drawn down to final diam with intermediate anneals at 723 and 973 K, and finally annealed in vacuum of $10^{-5}$ mm Hg at 973 K for 24 h; uncertainty in temperature 0.1 K, temperature controlled to within 0.01 K, uncertainty in resistivity of about 1%; data extracted from table.
5	279	Hau, N.H.	1966	A	77.3	0-100	The above specimens; uncertainty in temperature 0.01 K at this higher temperature.
6	279	Hau, N.H.	1966	A	298	0-100	The above specimens.
7	272	Logie, H.J., Jackson, J., Anderson, J.C., and Nabarro, F.R.N.	1961	A	77.4	0-100	Polycrystalline wire specimens 0.2 mm in diam and 100 mm long; 99.998% pure Au and Pd melted together in vacuum, drawn to final diam, annealed at 1173 K for 3 h, and furnace cooled; measured at liquid nitrogen temperature; authors observed small but distinct discontinuity in the slope of the curve at 40 a/o Pd; data read from figure.
8	232	Kim, M.J. and Flanagan, W.F.	1967	P	293	10-95	Clover shaped, polycrystalline foil specimens; 100 mesh powders of 99.99 pure Au and 99.97 pure Pd from Engelhard Industries Inc. melted together in high purity alumina crucibles by induction heating under a vacuum of $10^{-5}$ to $10^{-4}$ Torr; homogenized for 1 week at 1173 K in vacuum of $10^{-5}$ to $10^{-4}$ Torr and surface of ingot ground off to eliminate concentration gradients; specimens, as annealed, were measured in a room temperature silicone oil bath; data read from figure.
9	232	Kim, M.J. and Flanagan, W.F.	1967	P	293	10-95	The above specimens measured after 40% reduction in area.
10	285	Getzel, W.	1911		273	0-100	Specimens 0.1 mm in diam and 50 cm long; supplied by Firma Heraeus; measured at room temperature and corrected to 273 K; data extracted from table.
11	286	Schulze, F.A.	1911		298	17-94	Specimens 1 mm in diam supplied by Firma Heraeus; data extracted from table.
12	287	Ruchitskii, A.A.	1956		293	0-100	Survey report, no measurement information given; measurement temperature not stated explicitly, assumed to be 293 K; data read from figure.



TABLE 49. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF GOLD-PALLADIUM ALLOY SYSTEM (Composition Dependence)

[illegible]

\* Not shown in figure.



### 3.8. Gold-Silver Alloy System

Most of the evidence [264,289] indicates that the gold-silver alloy system forms a continuous series of solid solutions. Grum-Grzhimailo [290] has reported measurements on the electrical resistance, Hall effect, and lattice parameter which indicate the formation of intermetallic compounds  $\text{Ag}_3\text{Au}$ ,  $\text{Ag}_3\text{Au}_3$ , and  $\text{AgAu}_3$ , but this work has been discounted in the present analysis.

The existence of short-range ordering in the gold-silver alloy system is well recognized, though the mechanism of the ordering process and the effects of short-range order on the resistivity are disputed. Lücke and co-workers [291-293,275] have performed isothermal and isochronal annealing investigations of a wide range of silver-gold alloys and have concluded that an increase in the degree of short-range order is associated with an increase in the electrical resistivity and that the short-range order is of the conventional type described by Cowley [294]. On the contrary, Schüle and co-workers [295,296] have reported measurements on specimens very nearly identical with those of Lücke et al. which indicate that an increase in the degree of short-range order is associated with a decrease in the electrical resistivity under certain conditions and that the formation of short-range order proceeds in a non-homogeneous manner via the growth of localized regions of high order, the boundaries of which can cause additional electron scattering and the consequent resistivity increase observed by Lücke et al. The change in the resistivity of quenched gold-silver alloy specimens upon annealing and the formation of short-range order can be as much as 2%.

There are 183 sets of experimental data available for the electrical resistivity of this alloy system, with 17 sets being single data points. These data sets are listed in tables 51, 53, and 55 which provide information on specimen characterization and measurement conditions, tabulated in tables 52, 54, and 56, and shown partially in figures 40, 41, and 42. In order to show both functional dependences of the electrical resistivity, a few of the data sets have been presented in the tables and figures both for temperature dependence and for composition dependence.

In the analysis and synthesis of experimental data to generate recommended values, the dependence of residual resistivity on alloy composition was determined using the abundant data from measurements at 4.2 K and below. A curve

representing the totality of the data was tested and smoothed by examination of second differences. It shows a region of relatively low curvature for atomic fractions  $c$  of silver between 0.65 and 0.85 (between 65 and 85 at.% Ag); the reality of this feature seems to be supported by analysis of the alloy resistivities at higher temperatures.

The analysis of alloy resistivity was based on a study of the quantity  $\Delta(c, T)$ , as given by eq (26), derived from total resistivity values by subtracting from them the residual resistivity for the given  $c$  and the atomic-fraction-weighted average of the pure metal resistivities for the given  $T$ . The quantity  $\Delta$  is only a few percent of the total resistivity, and varies smoothly with  $c$  except for very dilute alloys. The analysis is described in some detail in Appendix 5.1; here a very few comments will suffice.

Examination of  $\Delta$  values makes very evident the presence of relatively large and erratic errors in much of the older data, probably due to inadequate control of the composition and purity of the alloy samples. The analysis began with a study of the data for  $T < 300$  K obtained using samples for which the resistivity was also measured at 4.2 K or below, since knowledge of the latter quantity helped to eliminate or diminish several sources of error. The data of Giauque and Stout [297] (Au + Ag data sets 23-26, Ag + Au data sets 21, 22, and Au-Ag data sets 16-20), Crisp and Rungis [298] (Au + Ag data sets 38-44, Ag + Au data sets 30-39, and Au-Ag data sets 9, 10), Davis and Rayne [299] (Au + Ag data sets 51-57, Ag + Au data sets 48-51, and Au-Ag data sets 11-13), Boes et al. [300] (Au + Ag data sets 6-11, Ag + Au data sets 5-10, and Au-Ag data sets 2, 3), and Huray et al. [135] (Au + Ag data sets 45, 46 and Ag + Au data sets 40, 41) were plotted and cross-plotted to arrive at acceptable smoothed values of  $\Delta$  and  $\Delta/T$  from 100 K to 300 K. The values of  $\Delta$  could be extrapolated to lower temperatures and to more dilute alloys guided by values derived from the data of Stewart and Huebener [301] (Au + Ag data sets 58-60 and Au-Ag data set 14) and of Dugdale and Basinski [29] (Au-Ag data set 15), who studied deviations from Matthiessen's Rule in alloys with less than 1 at.% solute. The analysis was then extended to higher temperatures by using the data of Iyer and Asimow [302] (Au + Ag data sets 2-5, Ag + Au data sets 11-13, and Au-Ag data set 46) on alloys with 5, 10, 30, 50, 70, 90 and 95 at.% silver. These data were smoothed and then partially corrected for errors that have a systematic effect on a plot of  $\Delta$  against  $T$  for a single sample (errors in

solute concentration, in sample dimensions, and in the residual resistivity curve) by comparison with the more accurate low-temperature data. The reasonableness of the corrections was checked, and the values were interpolated and extrapolated to other solute concentrations by use of cross-plots. Finally, the recommended values of the resistivity were derived from the values of  $\Delta$ ,  $\rho_0$ , and the pure-metal resistivities.

The most reliable results were obtained for concentrated alloys (20 to 80 at.% solute), for which extrapolation of  $\Delta$  to low temperature and of  $\Delta/T$  to high temperatures was most reliable. At low temperatures and low solute concentrations the uncertainty (in percent) rises along with the estimated uncertainty in the residual resistance. At high temperatures and low solute concentrations the uncertainty in the extrapolation of  $\Delta$  combines with the uncertainty in the resistivity of the pure metals to produce the estimated  $\pm 4\%$  uncertainty in the result. It is unfortunate that there seem to be no resistivity data on Ag-Au alloys with less than 5 at.% of solute, for  $T > 400$  K.

The resulting recommended electrical resistivity values for Au, Ag, and for 25 Au-Ag binary alloys are presented in table 50 and shown in figures 38, 39, and 42. The recommended values for Au and for Ag are for well-annealed high-purity specimens, but those values for temperatures below about 100 K are applicable only to Au and Ag having residual electrical resistivities as given at 1 K in table 50. As discussed in the beginning of this section, the electrical resistivity of Au-Ag alloys can be varied by changes in their short-range order produced by thermal treatment. This can amount to as much as 2% in the case of concentrated alloys subjected to extremes of quenching and annealing. The data analyzed here were obtained with samples not deliberately subjected to such treatment, and the recommended values may be regarded as applicable to alloys which have not been quenched or cold-worked severely and have the equilibrium degree of short-range order at each temperature. The recommended values cover the temperature range from 1 K to 1200 K and are not corrected for the thermal expansion of the material. The estimated uncertainties in the values for the various alloys and for different temperature ranges are explicitly stated in a footnote to table 50.



TABLE 50. RECOMMENDED ELECTRICAL RESISTIVITY OF GOLD-SILVER ALLOY SYSTEM†

[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Au: 100.00% (100.00 At.%) Ag: 0.00% ( 0.00 At.%)			Au: 99.50% (99.09 At.%) Ag: 0.50% ( 0.91 At.%)			Au: 99.00% (98.19 At.%) Ag: 1.00% ( 1.81 At.%)			Au: 97.00% (94.65 At.%) Ag: 3.00% ( 5.35 At.%)			Au: 95.00% (91.23 At.%) Ag: 5.00% ( 8.77 At.%)			Au: 90.00% (83.13 At.%) Ag: 10.00% (16.87 At.%)		
T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$	
1	0.0220		1	0.272*		1	0.536*		1	1.55*		1	2.47*		1	4.48*	
4	0.0220		4	0.272		4	0.536		4	1.55		4	2.47		4	4.48	
7	0.0221		7	0.272*		7	0.537		7	1.55		7	2.48		7	4.48	
10	0.0226		10	0.274*		10	0.539		10	1.55		10	2.48		10	4.49	
15	0.0258		15	0.280*		15	0.545		15	1.55		15	2.48		15	4.49	
20	0.0346‡		20	0.292		20	0.557		20	1.57		20	2.50		20	4.50	
25	0.0503‡		25	0.312		25	0.577		25	1.59		25	2.52		25	4.52	
30	0.0727‡		30	0.340		30	0.606		30	1.61		30	2.54		30	4.55	
40	0.141‡		40	0.414		40	0.681		40	1.69		40	2.62		40	4.62	
50	0.222		50	0.498		50	0.766		50	1.78		50	2.70		50	4.71	
60	0.309		60	0.583		60	0.855		60	1.87		60	2.80		60	4.80	
70	0.396		70	0.669		70	0.944		70	1.96		70	2.89		70	4.89	
80	0.482		80	0.754		80	1.03		80	2.05		80	2.98		80	4.98	
90	0.507		90	0.838		90	1.12		90	2.14		90	3.07		90	5.07	
100	0.632		100	0.921		100	1.20		100	2.23		100	3.16		100	5.16	
150	1.063		150	1.33		150	1.61		150	2.66		150	3.60		150	5.59	
200	1.464		200	1.73		200	2.00		200	3.07		200	4.00		200	5.99	
250	1.865		250	2.13		250	2.40		250	3.46		250	4.40		250	6.39	
273	2.052		273	2.31		273	2.58		273	3.64		273	4.58		273	6.57	
293	2.214		293	2.47		293	2.75		293	3.80		293	4.74		293	6.73	
300	2.271		300	2.53*		300	2.80*		300	3.86		300	4.79		300	6.79	
350	2.683		350	2.94*		350	3.22*		350	4.26		350	5.19		350	7.19	
400	3.102		400	3.36*		400	3.63*		400	4.66		400	5.59		400	7.55	
500	3.962		500	4.21*		500	4.47*		500	5.48		500	6.40		500	8.39	
600	4.853		600	5.08*		600	5.33*		600	6.32		600	7.25		600	9.23	
700	5.780		700	5.99*		700	6.23*		700	7.20		700	8.14		700	10.12	
800	6.755		800	6.94*		800	7.16*		800	8.13		800	9.09		800	11.06	
900	7.787		900	7.95*		900	8.16*		900	9.13		900	10.09		900	12.05	
1000	8.894		1000	9.04*		1000	9.24*		1000	10.20		1000	11.16		1000	13.10	
1100	10.057		1100	10.19*		1100	10.39*		1100	11.33*		1100	12.30		1100	14.22	
1200	11.312		1200	11.43*		1200	11.61*		1200	12.55*		1200	13.52		1200	15.42	
1300	12.632																
1337.58	13.146 (s)																
1338	31.08 (t)																
1700	38.26																

† Uncertainties in the electrical resistivity values are as follows:

- 100.00 Au - 0.00 Ag:  $\pm 1\%$  up to 10 K,  $\pm 2.5\%$  above 10 K to 15 K,  $\pm 6\%$  above 15 K to 40 K,  $\pm 3\%$  above 40 K to 80 K,  $\pm 1\%$  above 80 K to 500 K, and  $\pm 2.5\%$  above 500 K.  
 99.50 Au - 0.50 Ag:  $\pm 4\%$  below 80 K,  $\pm 3\%$  from 80 to 600 K, and  $\pm 4\%$  above 600 K.  
 99.00 Au - 1.00 Ag:  $\pm 4\%$  below 80 K,  $\pm 3\%$  from 80 to 600 K, and  $\pm 4\%$  above 600 K.  
 97.00 Au - 3.00 Ag:  $\pm 4\%$  below 80 K,  $\pm 3\%$  from 80 to 400 K, and  $\pm 2\%$  above 400 K.  
 95.00 Au - 5.00 Ag:  $\pm 2\%$  up to 600 K and  $\pm 1\%$  above 600 K.  
 90.00 Au - 10.00 Ag:  $\pm 1\%$

‡ Provisional value.

\* In temperature range where no experimental data are available.

TABLE 50. RECOMMENDED ELECTRICAL RESISTIVITY OF GOLD-SILVER ALLOY SYSTEM† (continued)  
 [Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8}$   $\Omega$  m]

Au: 85.00% (75.63 At.%) Ag: 15.00% (24.37 At.%)			Au: 80.00% (68.66 At.%) Ag: 20.00% (31.34 At.%)			Au: 75.00% (62.16 At.%) Ag: 25.00% (37.84 At.%)			Au: 70.00% (56.10 At.%) Ag: 30.00% (43.90 At.%)			Au: 65.00% (50.42 At.%) Ag: 35.00% (49.58 At.%)			Au: 60.00% (45.10 At.%) Ag: 40.00% (54.90 At.%)		
T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$	
1	6.08*		1	7.30*		1	8.18*		1	8.73*		1	9.00*		1	9.01*	
4	6.08		4	7.30		4	8.18		4	8.73*		4	9.00*		4	9.01*	
7	6.08		7	7.30		7	8.18*		7	8.73*		7	9.00		7	9.01*	
10	6.09		10	7.31		10	8.18*		10	8.73*		10	9.00		10	9.01*	
15	6.09		15	7.31		15	8.18*		15	8.74*		15	9.00		15	9.02*	
20	6.10		20	7.32		20	8.19*		20	8.75*		20	9.01		20	9.03*	
25	6.12		25	7.34		25	8.21*		25	8.76*		25	9.03		25	9.04*	
30	6.15		30	7.36		30	8.23*		30	8.78*		30	9.05		30	9.06*	
40	6.22		40	7.43		40	8.30*		40	8.85*		40	9.11		40	9.12*	
50	6.30		50	7.52		50	8.38*		50	8.93*		50	9.19		50	9.20	
60	6.39		60	7.60		60	8.47*		60	9.01*		60	9.28		60	9.28*	
70	6.48		70	7.69		70	8.56*		70	9.10*		70	9.36		70	9.36*	
80	6.57		80	7.78		80	8.64		80	9.19*		80	9.45		80	9.45*	
90	6.66		90	7.87		90	8.73*		90	9.27*		90	9.53		90	9.53*	
100	6.75		100	7.96		100	8.82*		100	9.36*		100	9.61		100	9.61*	
150	7.17		150	8.37		150	9.23*		150	9.76*		150	10.01		150	10.01*	
200	7.37		200	8.77		200	9.62*		200	10.15*		200	10.39		200	10.35*	
250	7.96		250	9.16		250	10.00*		250	10.53*		250	10.77		250	10.75*	
273	8.14		273	9.34		273	10.18		273	10.70		273	10.94		273	10.92	
293	8.30		293	9.50		293	10.34*		293	10.86		293	11.09		293	11.07	
300	8.36*		300	9.55		300	10.39*		300	10.91		300	11.14		300	11.12	
350	8.75*		350	9.94		350	10.78		350	11.29		350	11.52		350	11.50	
400	9.15*		400	10.33		400	11.17*		400	11.68*		400	11.90		400	11.87*	
500	9.95*		500	11.13		500	11.95*		500	12.45*		500	12.67		500	12.62	
600	10.78*		600	11.95		600	12.76*		600	13.24*		600	13.44		600	13.39*	
700	11.66*		700	12.81		700	13.60*		700	14.07*		700	14.26		700	14.19	
800	12.58*		800	13.71		800	14.49*		800	14.94*		800	15.11		800	15.02*	
900	13.55*		900	14.67		900	15.42*		900	15.86*		900	16.01		900	15.90*	
1000	14.58*		1000	15.68		1000	16.41*		1000	16.82*		1000	16.95		1000	16.82*	
1100	15.68*		1100	16.78		1100	17.47*		1100	17.86*		1100	17.96		1100	17.81*	
1200	16.96*		1200	17.90*		1200	18.59*		1200	18.96*		1200	19.04*		1200	18.86*	

† Uncertainties in the electrical resistivity values are as follows:

- 85.00 Au - 15.00 Ag:  $\pm 1\%$ .
- 80.00 Au - 20.00 Ag:  $\pm 1\%$ .
- 75.00 Au - 25.00 Ag:  $\pm 1\%$ .
- 70.00 Au - 30.00 Ag:  $\pm 1\%$ .
- 65.00 Au - 35.00 Ag:  $\pm 1\%$ .
- 60.00 Au - 40.00 Ag:  $\pm 1\%$ .

\* In temperature range where no experimental data are available.

TABLE 50. RECOMMENDED ELECTRICAL RESISTIVITY OF GOLD-SILVER ALLOY SYSTEM† (continued)  
[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Au: 55.00% (40.10 At.%) Ag: 45.00% (59.90 At.%)		Au: 50.00% (35.39 At.%) Ag: 50.00% (64.61 At.%)		Au: 45.00% (30.94 At.%) Ag: 55.00% (69.06 At.%)		Au: 40.00% (26.75 At.%) Ag: 60.00% (73.25 At.%)		Au: 35.00% (22.77 At.%) Ag: 65.00% (77.23 At.%)		Au: 30.00% (19.01 At.%) Ag: 70.00% (80.99 At.%)	
T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
1	8.79*	1	8.37*	1	7.80*	1	7.13*	1	6.40*	1	5.61
4	8.79	4	8.37*	4	7.80	4	7.13	4	6.40*	4	5.61
7	8.79*	7	8.38*	7	7.80	7	7.14	7	6.40*	7	5.61
10	8.79*	10	8.38*	10	7.81	10	7.14	10	6.40*	10	5.61
15	8.80*	15	8.38*	15	7.81	15	7.14	15	6.41*	15	5.61
20	8.81*	20	8.39	20	7.82	20	7.15	20	6.41*	20	5.62
25	8.82*	25	8.40	25	7.83	25	7.16	25	6.43*	25	5.63
30	8.84*	30	8.42	30	7.85	30	7.18	30	6.44*	30	5.65
40	8.90*	40	8.48	40	7.91	40	7.24	40	6.50*	40	5.70
50	8.98*	50	8.55	50	7.98	50	7.30	50	6.56*	50	5.77
60	9.06*	60	8.63	60	8.06	60	7.38	60	6.64*	60	5.84
70	9.14*	70	8.72	70	8.14	70	7.46	70	6.72*	70	5.92
80	9.22	80	8.80	80	8.22	80	7.54	80	6.80*	80	5.99
90	9.31*	90	8.88	90	8.30	90	7.62	90	6.87*	90	6.07
100	9.39*	100	8.96	100	8.37	100	7.69	100	6.95*	100	6.15
150	9.77*	150	9.34	150	8.75	150	8.06	150	7.31*	150	6.50
200	10.14*	200	9.70	200	9.11	200	8.42	200	7.66*	200	6.85
250	10.51*	250	10.06	250	9.46	250	8.76	250	8.00*	250	7.18
273	10.68	273	10.23	273	9.62	273	8.92	273	8.16*	273	7.34
293	10.82	293	10.37	293	9.76	293	9.06	293	8.29	293	7.47
300	10.87	300	10.42	300	9.81	300	9.11	300	8.34	300	7.52
350	11.23	350	10.78	350	10.16	350	9.46*	350	8.68*	350	7.85
400	11.60*	400	11.14*	400	10.52	400	9.81*	400	9.03*	400	8.19*
500	12.34*	500	11.87*	500	11.24	500	10.52*	500	9.73*	500	8.88*
600	13.10*	600	12.62*	600	11.98	600	11.24*	600	10.44*	600	9.58*
700	13.89*	700	13.39*	700	12.74	700	11.99*	700	11.18*	700	10.30*
800	14.71*	800	14.20*	800	13.53	800	12.77*	800	11.94*	800	11.05*
900	15.57*	900	15.03*	900	14.35	900	13.67*	900	12.73*	900	11.82*
1000	16.47*	1000	15.91*	1000	15.21	1000	14.41*	1000	13.55*	1000	12.63*
1100	17.44*	1100	16.86*	1100	16.13	1100	15.31*	1100	14.42*	1100	13.48*
1200	18.47*	1200	17.86*	1200	17.10	1200	16.25*	1200	15.34*	1200	14.38*

† Uncertainties in the electrical resistivity values are as follows:

55.00 Au - 45.00 Ag:  $\pm 1\%$ .

50.00 Au - 50.00 Ag:  $\pm 1\%$ .

45.00 Au - 55.00 Ag:  $\pm 1\%$ .

40.00 Au - 60.00 Ag:  $\pm 1\%$ .

35.00 Au - 65.00 Ag:  $\pm 1\%$ .

30.00 Au - 70.00 Ag:  $\pm 2\%$  up to 400 K and  $\pm 1\%$  above 400 K.

\* In temperature range where no experimental data are available.

TABLE 50. RECOMMENDED ELECTRICAL RESISTIVITY OF GOLD-SILVER ALLOY SYSTEM† (continued)  
 (Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ )

Au: 25.00% (15.44 At. %) Ag: 75.00% (84.56 At. %)		Au: 20.00% (12.04 At. %) Ag: 80.00% (87.96 At. %)		Au: 15.00% ( 8.81 At. %) Ag: 85.00% (91.19 At. %)		Au: 10.00% ( 5.74 At. %) Ag: 90.00% (94.26 At. %)		Au: 5.00% ( 2.80 At. %) Ag: 95.00% (97.20 At. %)		Au: 3.00% ( 1.67 At. %) Ag: 97.00% (98.33 At. %)	
T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
1	4.75*	1	3.85*	1	2.90*	1	1.94*	1	0.964*	1	0.577*
4	4.75*	4	3.85	4	2.90	4	1.94	4	0.964	4	0.577
7	4.76*	7	3.85*	7	2.91	7	1.94*	7	0.965*	7	0.578
10	4.76*	10	3.85*	10	2.91	10	1.94*	10	0.966*	10	0.579
15	4.76*	15	3.86*	15	2.91	15	1.94*	15	0.970*	15	0.583
20	4.77*	20	3.86*	20	2.92	20	1.95*	20	0.976*	20	0.588
25	4.78*	25	3.87*	25	2.93	25	1.96*	25	0.985*	25	0.599
30	4.80*	30	3.89*	30	2.94	30	1.98*	30	1.00*	30	0.614
40	4.85*	40	3.94*	40	2.99	40	2.02*	40	1.05*	40	0.659
50	4.91*	50	4.00*	50	3.05	50	2.08*	50	1.11*	50	0.716
60	4.98*	60	4.07*	60	3.12	60	2.15*	60	1.17*	60	0.779
70	5.06*	70	4.15*	70	3.19	70	2.22*	70	1.24*	70	0.845
80	5.14	80	4.22*	80	3.27	80	2.29*	80	1.31*	80	0.911
90	5.21*	90	4.30*	90	3.34	90	2.37*	90	1.38*	90	0.977
100	5.29*	100	4.37*	100	3.42	100	2.44*	100	1.44	100	1.04
150	5.64*	150	4.72*	150	3.76	150	2.77*	150	1.76	150	1.35
200	5.98*	200	5.05*	200	4.08	200	3.09*	200	2.08	200	1.66
250	6.31*	250	5.37*	250	4.40	250	3.40*	250	2.38	250	1.96
273	6.46	273	5.52	273	4.55	273	3.54	273	2.52	273	2.06
293	6.59	293	5.65*	293	4.67	293	3.66	293	2.64*	293	2.20
300	6.63	300	5.70*	300	4.72	300	3.71	300	2.68*	300	2.24*
350	6.96	350	6.02*	350	5.03	350	4.00	350	2.96*	350	2.52*
400	7.30*	400	6.34*	400	5.34	400	4.31	400	3.25*	400	2.82*
500	7.97*	500	7.00*	500	5.99	500	4.93	500	3.86*	500	3.43*
600	8.66*	600	7.68*	600	6.65	600	5.57	600	4.49*	600	4.07*
700	9.37*	700	8.37*	700	7.32	700	6.24	700	5.15*	700	4.73*
800	10.10*	800	9.09*	800	8.02	800	6.93	800	5.83*	800	5.41*
900	10.85*	900	9.83*	900	8.74	900	7.64	900	6.54*	900	6.11*
1000	11.64*	1000	10.59*	1000	9.50	1000	8.39	1000	7.27*	1000	6.84*
1100	12.47*	1100	11.41*	1100	10.31	1100	9.19	1100	8.06*	1100	7.64*
1200	13.36*	1200	12.28*	1200	11.17*	1200	10.04*	1200	8.90*	1200	8.55*

† Uncertainties in the electrical resistivity values are as follows:

- 25.00 Au - 75.00 Ag:  $\pm 2\%$  up to 400 K and  $\pm 1\%$  above 400 K.
- 20.00 Au - 80.00 Ag:  $\pm 2\%$ .
- 15.00 Au - 85.00 Ag:  $\pm 2\%$ .
- 10.00 Au - 90.00 Ag:  $\pm 2\%$  up to 200 K and  $\pm 1\%$  above 200 K.
- 5.00 Au - 95.00 Ag:  $\pm 1\%$ .
- 3.00 Au - 97.00 Ag:  $\pm 1\%$  up to 200 K and  $\pm 0.5\%$  above 200 K.

\* In temperature range where no experimental data are available.

TABLE 50. RECOMMENDED ELECTRICAL RESISTIVITY OF GOLD-SILVER ALLOY SYSTEM† (continued)  
[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Au: 1.00% (0.55 At.%) Ag: 99.00% (99.45 At.%)		Au: 0.50% (0.27 At.%) Ag: 99.50% (99.73 At.%)		Au: 0.00% (0.00 At.%) Ag: 100.00% (100.00 At.%)	
T	$\rho$	T	$\rho$	T	$\rho$
1	0.193*	1	0.097	1	0.00100
4	0.183	4	0.097	4	0.00100
7	0.194	7	0.098	7	0.00103
10	0.196	10	0.099	10	0.00115
15	0.199	15	0.102	15	0.00130
20	0.205	20	0.107	20	0.00423
25	0.216	25	0.115	25	0.00959
30	0.230	30	0.126	30	0.0195
40	0.269	40	0.162	40	0.0541
50	0.319	50	0.210	50	0.104
60	0.377	60	0.269	60	0.163
70	0.440	70	0.332	70	0.226
80	0.503	80	0.395	80	0.290
90	0.566	90	0.458	90	0.355
100	0.627	100	0.521	100	0.419
150	0.935	150	0.834	150	0.728
200	1.25	200	1.15	200	1.031
250	1.56	250	1.45	250	1.330
273	1.69	273	1.58	273	1.468
293	1.80	293	1.69	293	1.587
300	1.84*	300	1.73*	300	1.629
350	2.12*	350	2.02*	350	1.930
400	2.42*	400	2.33*	400	2.236
500	3.04*	500	2.95*	500	2.863
600	3.69*	600	3.60*	600	3.509
700	4.35*	700	4.26*	700	4.174
800	5.03*	800	4.94*	800	4.860
900	5.73*	900	5.65*	900	5.565
1000	6.47*	1000	6.40*	1000	6.297
1100	7.28*	1100	7.19*	1100	7.085
1200	8.13*	1200	8.03*	1200	7.922
				1235.08	8.233(s)
				1236	17.31(L)
				1400	18.69
				1600	20.38

† Uncertainties in the electrical resistivity values are as follows:

1.00 Au - 99.00 Ag:  $\pm 4\%$  up to 100 K and  $\pm 5\%$  above 100 K.

0.50 Au - 99.50 Ag:  $\pm 4\%$  up to 100 K and  $\pm 5\%$  above 100 K.

0.00 Au - 100.00 Ag:  $\pm 1\%$  up to 10 K,  $\pm 5\%$  above 10 K to 30 K,  $\pm 2\%$  above 30 K to 70 K,  $\pm 1\%$  above 70 K to 400 K,  $\pm 2\%$  above 400 K to 1235.08 K, and  $\pm 4\%$  above 1235.08 K.

\* In temperature range where no experimental data are available.

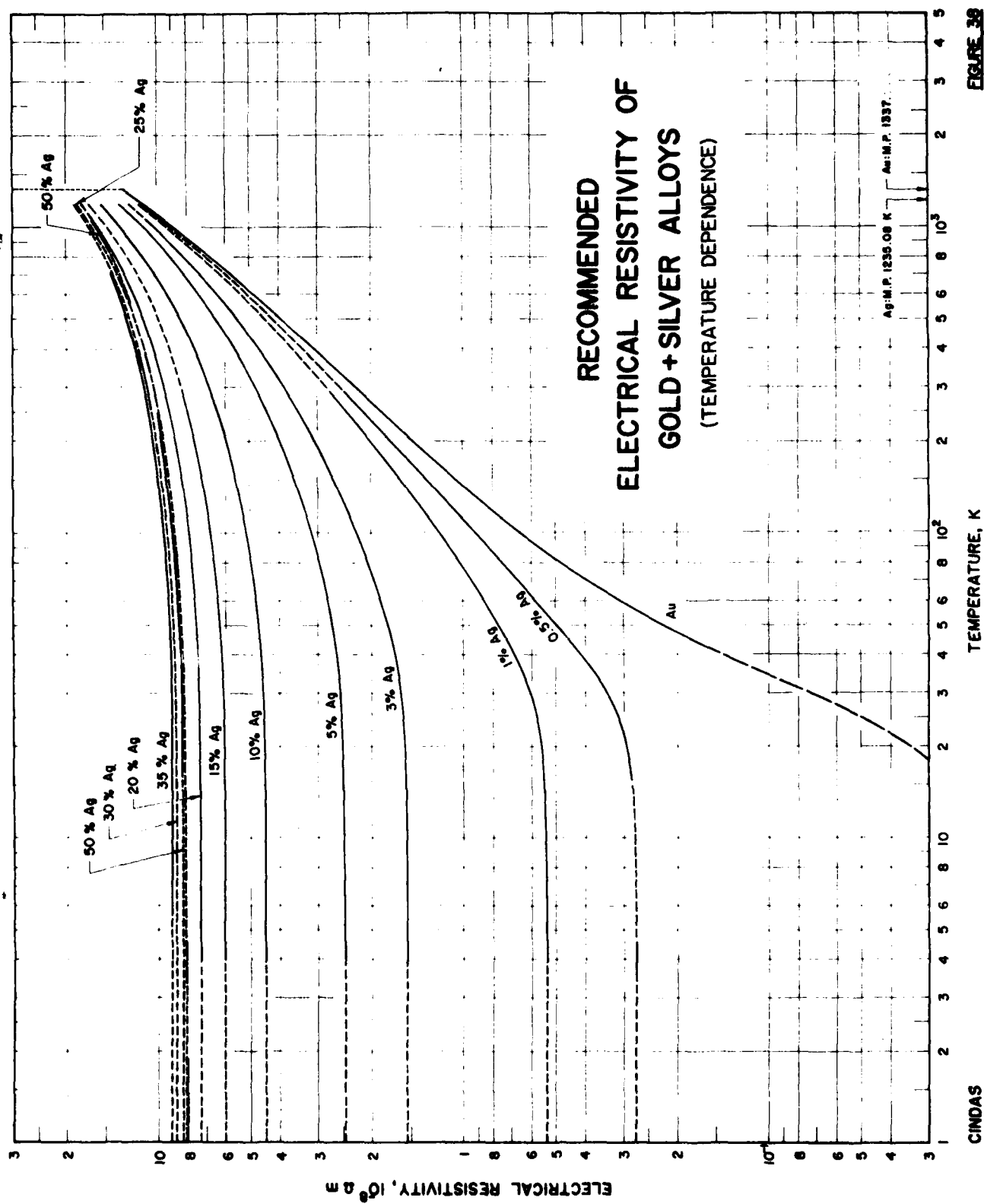
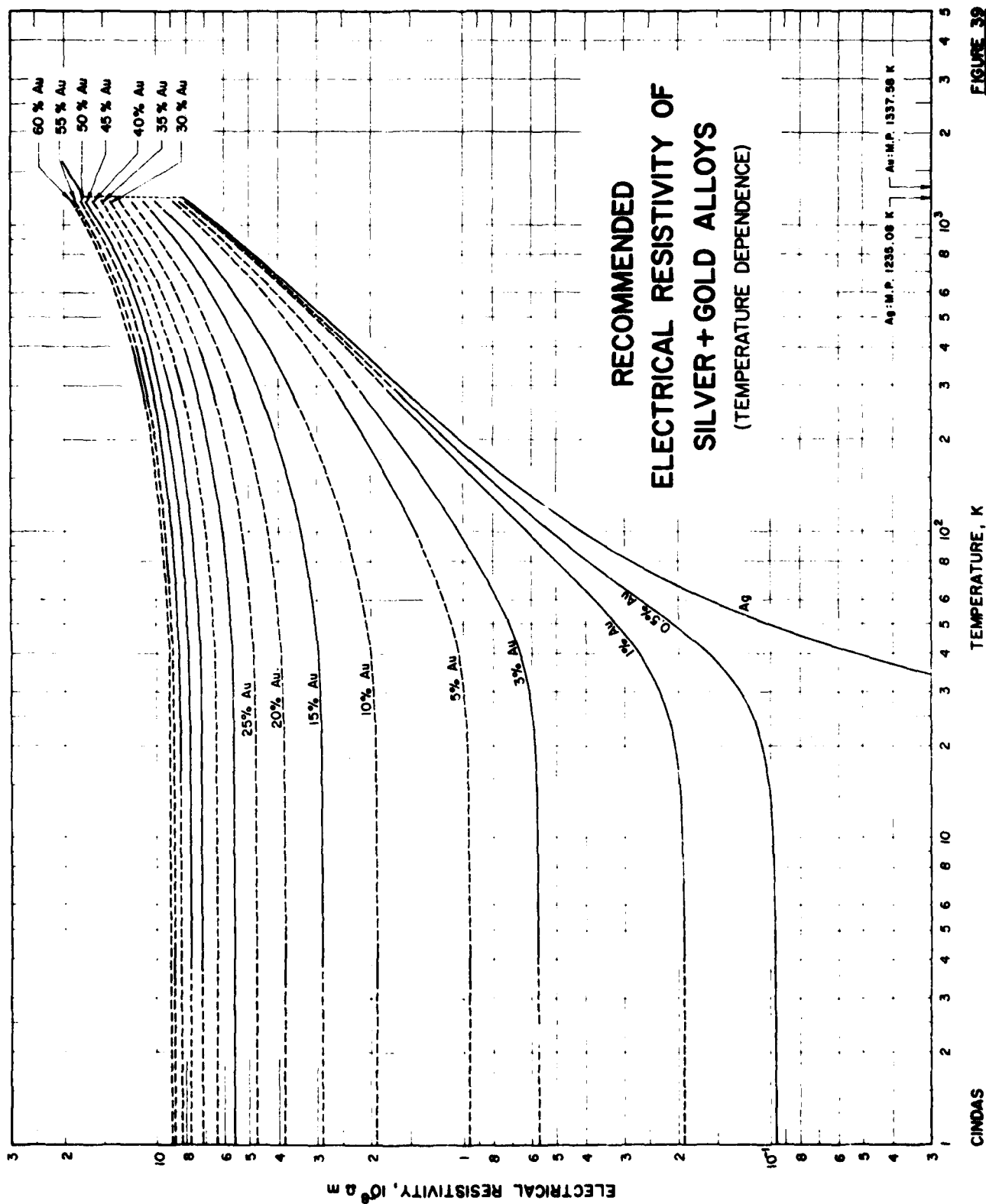


FIGURE 38

CINDAS



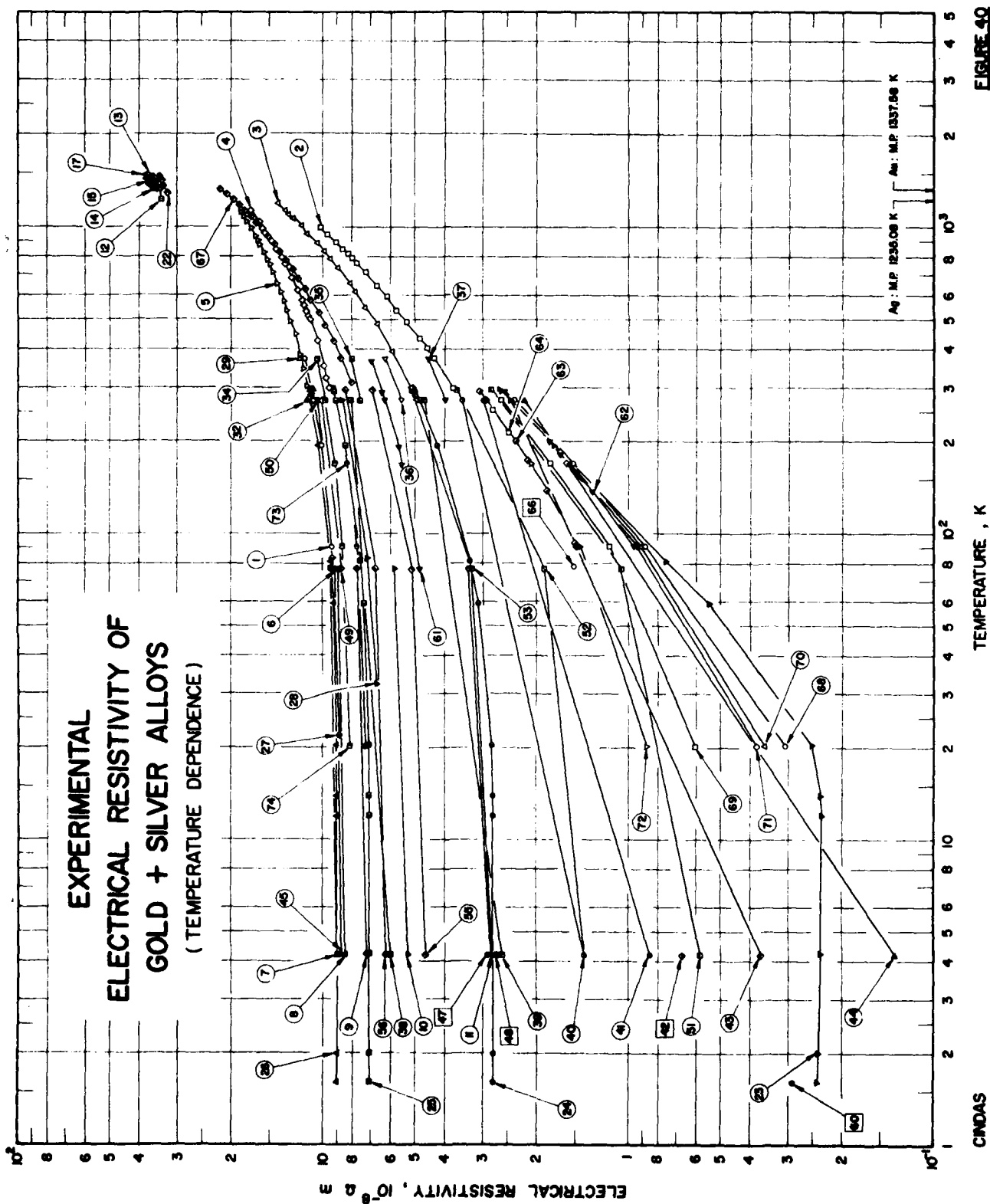


FIGURE 40



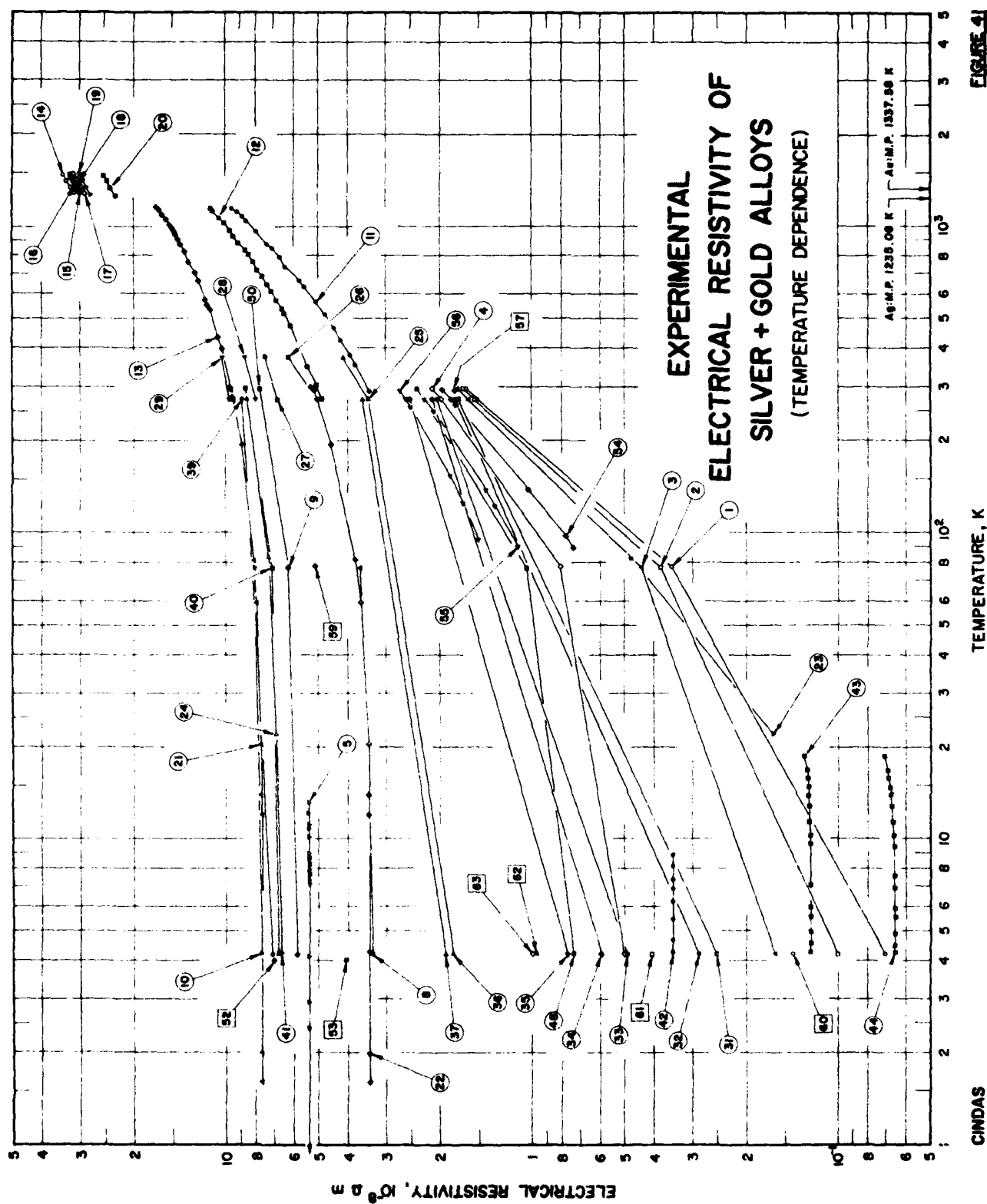


TABLE 51. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD + SILVER ALLOYS (Temperature Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Au Ag	Composition (continued), Specifications, and Remarks
1	258	Broom, T.	1952	B	90-373		66.13	Wire specimen 1.83 mm (0.072 in) in diam; annealed at 873 K for 2 h and furnace cooled; uncertainty in diam $\pm 0.0001$ in; tabular data presentation; resistivity change after 37% deformation by drawing was $0.375 \times 10^{-4} \Omega \text{m}$ at 90 K.
2	302	Iyer, V. K. and Asinow, R. M.	1967		299-998		2.8	Calculated composition (5 a/o Ag), less than 0.1 total impurity with 0.05 Si indicated by spectrographic analysis; 99.998 Ag and Au melted in graphite crucible in Ar; evaporation of Ag resulted in composition change of less than 0.2 a/o; 3.18 mm (0.125 in) diam ingots swaged to 1.02 mm (0.040 in) and drawn to 0.254 mm (0.010 in); annealed in Ar; maintained in Ar above 673 K to prevent evaporation of Ag but measured in vacuum; different specimen measured in 273-303 K range than at higher temperatures, identical values obtained for at least two heating cycles; uncertainty in temperature $\pm 0.3^\circ \text{K}$ (in $^\circ \text{C}$ ), standard deviation of points from best fitted curve less than 0.5% of the mean, resistivity accurate to within 2% with error mainly due to length and density uncertainty; data corrected for thermal expansion; data read from figure.
3	302	Iyer, V. K. and Asinow, R. M.	1967		274-1194		5.74	Calculated composition (10 a/o Ag); similar to the above specimen.
4	302	Iyer, V. K. and Asinow, R. M.	1967		298-1134		19.0	Calculated composition (30 a/o Ag); similar to the above specimen.
5	302	Iyer, V. K. and Asinow, R. M.	1967		276-1123		35.4	Calculated composition (50 a/o Ag); similar to the above specimen.
6	300	Boes, J., Van Dam, A. J., and Bijvoet, J.	1968		4.2, 77		54.9	Calculated composition (40 a/o Au); strip specimen 1 x 1 x 30 mm made by arc melting in Ar and rolling the ingot; data extracted from table.
7	300	Boes, J., et al.	1968		4.2, 77		64.6	Calculated composition (50 a/o Au); similar to the above specimen.
8	300	Boes, J., et al.	1968		4.2, 77		73.3	Calculated composition (60 a/o Au); similar to the above specimen.
9	300	Boes, J., et al.	1968		4.2, 77		81.0	Calculated composition (70 a/o Au); similar to the above specimen.
10	300	Boes, J., et al.	1968		4.2, 77		88.0	Calculated composition (80 a/o Au); similar to the above specimen.
11	300	Boes, J., et al.	1968		4.2, 77		94.3	Calculated composition (90 a/o Au); similar to the above specimen.
12	127	Roll, A. and Motz, H.	1957	E	1332-1462		92.7	Calculated composition (87.5 a/o Au); measured in liquid state; prepared from 99.995 pure Ag and 99.95 pure Au; $d\rho/dT = 0.0159 \times 10^{-4} \Omega \text{m K}^{-1}$ ; $\pm 1\%$ uncertainty in resistivity; data read from figure.
13	127	Roll, A. and Motz, H.	1957	E	1324-1462		84.6	Calculated composition (75 a/o Au); similar to the above specimen but with $d\rho/dT = 0.0168 \times 10^{-4} \Omega \text{m K}^{-1}$ .
14	127	Roll, A. and Motz, H.	1957	E	1322-1462		82.4	Calculated composition (72 a/o Au); similar to the above specimen but with $d\rho/dT = 0.0155 \times 10^{-4} \Omega \text{m K}^{-1}$ .
15	127	Roll, A. and Motz, H.	1957	E	1320-1461		81.0	Calculated composition (70 a/o Au); similar to the above specimen but with $d\rho/dT = 0.0169 \times 10^{-4} \Omega \text{m K}^{-1}$ .
16*	127	Roll, A. and Motz, H.	1957	E	1318-1460		79.5	Calculated composition (68 a/o Au); similar to the above specimen but with $d\rho/dT = 0.0191 \times 10^{-4} \Omega \text{m K}^{-1}$ .

\* Not shown in figure.

TABLE 51. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD + SILVER ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Au Ag	Composition (continued), Specifications, and Remarks
17	127	Roll, A. and Motz, H.	1957	E	1317-1459		78.53	Calculated composition (66.7 a/o Au); similar to the above specimen but with $d\rho/dT = 0.0175 \times 10^{-8} \Omega m K^{-1}$ .
18*	127	Roll, A. and Motz, H.	1957	E	1317-1459		78.0	Calculated composition (66 a/o Au); similar to the above specimen but with $d\rho/dT = 0.0172 \times 10^{-8} \Omega m K^{-1}$ .
19*	127	Roll, A. and Motz, H.	1957	E	1316-1458		77.2	Calculated composition (65 a/o Au); similar to the above specimen but with $d\rho/dT = 0.0145 \times 10^{-8} \Omega m K^{-1}$ .
20*	127	Roll, A. and Motz, H.	1957	E	1313-1460		73.3	Calculated composition (60 a/o Au); similar to the above specimen but with $d\rho/dT = 0.0148 \times 10^{-8} \Omega m K^{-1}$ .
21*	127	Roll, A. and Motz, H.	1957	E	1304-1460		64.6	Calculated composition (50 a/o Au); similar to the above specimen but with $d\rho/dT = 0.0138 \times 10^{-8} \Omega m K^{-1}$ .
22	127	Roll, A. and Motz, H.	1957	E	1290-1461		52.28	Calculated composition (37.5 a/o Au); similar to the above specimen but with $d\rho/dT = 0.0116 \times 10^{-8} \Omega m K^{-1}$ .
23	297	Glaque, W. F. and Stout, J. W.	1938	A	1.6-298	No. 6	0.1	Readily available thermometer-heater wire containing about 0.1% Ag investigated without removal from the spool on which it was supplied; wire had been annealed and then insulated with a double layer of silk but was not re-annealed after winding on spool, however prior to the measurement of the following values the specimen was accidentally heated for 6 to 8 h at 373 K; 0.2408 and $0.2356 \times 10^{-8} \Omega m$ at 1.63 and 4.23 K, respectively; measurement techniques, uncertainties, and treatment of data the same as for curve 24.
24	297	Glaque, W. F. and Stout, J. W.	1938	A	1.6-298	No. 5	5.73	Drawn wire 0.579 mm in diam and about 100 cm long; annealed by heating to redness in a flame during and after drawing, wound loosely on a spool in insulated layers, and again annealed at 383 K for about 15 h; measured in four cooling cycles over period of 2.5 years; resistance ratio as a function of temperature and specific resistance at 273 K reported, from which resistivity was calculated by CINDAS, without correcting for thermal expansion; resistance accurate to 0.01%, resistivity to 1%; data extracted from table.
25	297	Glaque, W. F. and Stout, J. W.	1938	A	1.6-298	No. 4	18.99	Similar to the above specimen.
26	297	Glaque, W. F. and Stout, J. W.	1938	A	1.6-298	No. 3	35.33	Similar to the above specimen.
27	128	Grüneisen, E. and Reddenau, H.	1934		22-273	6	35.4	Calculated composition (50 a/o Ag); single crystal specimen; data extracted from table.
28	128	Grüneisen, E. and Reddenau, H.	1934		32-273	7	15.5	Calculated composition (25 a/o Ag); single crystal; data extracted from table.
29	130	Sedstrom, E.	1919		273, 373		54.62	Calculated composition (39.7 a/o Au); specimen rolled and drawn to thickness of 1 mm and heated 0.5 h at temperature near the melting point; data extracted from table.
30*	130	Sedstrom, E.	1919		273, 373		60.32	Calculated composition (45.4 a/o Au); similar to the above specimen.
31*	130	Sedstrom, E.	1919		273, 373		66.46	Calculated composition (50.9 a/o Au); similar to the above specimen.

\* Not shown in figure.

TABLE 51. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD + SILVER ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Au Ag	Composition (continued), Specifications, and Remarks
32	130	Sedstrom, E.	1919		273, 373		69.17	Calculated composition (55.1 a/o Au); similar to the above specimen.
33*	130	Sedstrom, E.	1919		273, 373		73.19	Calculated composition (59.9 a/o Au); similar to the above specimen.
34	130	Sedstrom, E.	1919		273, 373		81.23	Calculated composition (70.3 a/o Au); similar to the above specimen.
35	130	Sedstrom, E.	1919		273, 373		88.82	Calculated composition (81.3 a/o Au); similar to the above specimen.
36	130	Sedstrom, E.	1919		273, 373		93.84	Calculated composition (89.3 a/o Au); similar to the above specimen.
37	130	Sedstrom, E.	1919		273, 373		97.26	Calculated composition (95.1 a/o Au); similar to the above specimen.
38	298	Crisp, R.S. and Rungis, J.	1970	A	4.2, 273		12.7	Composition (21.0 a/o Ag) estimated by author using Nordheim's rule and the values of residual resistivity per atomic percent solute reported by J.O. Linde (Thesis, University of Stockholm, 1939); 1 mm diam wire specimens prepared from 99.9999 Ag and 99.9999 Au were supplied in three batches by Cambridge Metals Research Ltd; similar specimens analyzed for Fe content by D.S. Russell, Division of Applied Chemistry, N.R.C. of Canada contained 1.7 to 9.0 atomic p.p.m. of Fe; specimen as received was drawn down, etched, washed in distilled water and alcohol, dried, sealed in quartz capsule with 0.33 atm oxygen to minimize Fe contamination, and annealed at 1173 K for 72 h; uncertainty in temperature less than 0.05 K at all temperatures, 1% error in absolute resistivity; measurement temperature of the residual resistivity not stated explicitly, assumed to be 4.2 K; data extracted from table.
39	298	Crisp, R.S. and Rungis, J.	1970	A	4.2, 273		4.43	Estimated composition (7.8 a/o Ag); similar to the above specimen.
40	298	Crisp, R.S. and Rungis, J.	1970	A	4.2, 273		2.29	Estimated composition (4.1 a/o Ag); similar to the above specimen.
41	298	Crisp, R.S. and Rungis, J.	1970	A	4.2, 273		1.33	Estimated composition (2.4 a/o Ag); similar to the above specimen.
42	298	Crisp, R.S. and Rungis, J.	1970	A	4.2, 273		1.05	Estimated composition (1.9 a/o Ag); similar to the above specimen.
43	298	Crisp, R.S. and Rungis, J.	1970	A	4.2, 273		0.47	Estimated composition (0.85 a/o Ag); similar to the above specimen.
44	298	Crisp, R.S. and Rungis, J.	1970	A	4.2, 273		0.203	Estimated composition (0.37 a/o Ag); similar to the above specimen.
45	135	Huray, P.G., Roberts, L.D., and Thomson, J.O.	1971		4.2-296	AgAu	35.39	Calculated composition (50 a/o Ag), chemical analysis by Johnson Matthey Chemicals Ltd. of similar specimens indicated composition uncertainty of 0.5 a/o; foil specimen 0.0508 mm (2 mil) thick and uniform to at least 1%; 99.99 Ag and Au arc-melted and cooled in Ar several times and drop cast (poured) into water-cooled cylindrical mold, cylinders annealed in inert atmosphere for 5 d within 50 K of mp, stripe cut lengthwise from cylinder and rolled to final thickness with intermittent annealing; finally annealed for 100 h at 641 K, furnace cooled to 523 K, maintained at 523 K for 120 h, and quenched into water at room temperature; uncertainty in thickness less than 1%, uncertainties in resistivity $\pm 0.09$ , $\pm 0.08$ , and $\pm 0.11 \times 10^{-8}$ $\Omega$ m at 4.2, 77, and 296 K, respectively; data extracted from table.

\* Not shown in figure.

TABLE 51. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD + SILVER ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Au Ag	Composition (continued), Specifications, and Remarks
46*	135	Huray, P. G., Roberts, L. D., and Thomson, J. O.	1971		4.2-296	AgAu	35.39	Similar to the above specimen except finally suspended in vacuum furnace at 873 K for 30 min or more, then dropped in vacuum into bath of vacuum-pump oil at room temperature.
47	77	Kapoor, A., Rowlands, J. A., and Woods, S. B.	1974		4.2		5.74	Calculated composition (10 a/o Ag); polycrystalline square rod 4 mm <sup>2</sup> in cross section and 10 cm long; 99.999 pure metals induction melted in recrystallized alumina crucibles in Ar and resulting ingot rolled to size; cold worked; resistivity found to be independent of temperature between 1 and 4 K; uncertainty in resistivity $\pm 2\%$ ; data extracted from table.
48	77	Kapoor, A., et al.	1974		4.2		5.74	The above specimen annealed in vacuum for 12 h at 1000 K.
49	303	Van der Sljde, B.	1963		77.273		74.9	Calculated composition (82 a/o Au); wire specimen of 0.3 mm diam; heated at 1148 K for 10 min in air and cooled slowly; data extracted from table.
50	303	Van der Sljde, B.	1963		77.273		64.6	Calculated composition (50 a/o Au); wire specimen of 0.25 mm diam; heated at 873 K for 2 h in vacuum and cooled slowly; data extracted from table.
51	299	Davis, T. H. and Rayne, J. A.	1972	A	4.2-295		1.1	Calculated composition (2.0 a/o Ag); parallelpiped specimen 0.100 x 0.100 x 1.25 in; 99.999 <sup>+</sup> pure Ag and Au induction melted in helium, cast in ingots, cold worked, etched in 10% KCN, homogenized at 1073 K for 72 h, and quenched in cold water; specimen machined from axial section of ingot, etched in KCN, again heated at 1073 K for 72 h, and quenched; KCN etching and use of tungsten carbide tools reduced ferromagnetic impurities; uncertainty in resistivity 1.5%; measurements at ambient temperature were reported as being at $295 \pm 3$ K; data extracted from table.
52	299	Davis, T. H. and Rayne, J. A.	1972	A	4.2-295		2.8	Calculated composition (5.0 a/o Ag); similar to the above specimen.
53	299	Davis, T. H. and Rayne, J. A.	1972	A	4.2-295	a, Sample No. 1	5.7	Calculated composition (10.0 a/o Ag); similar to the above specimen.
54*	299	Davis, T. H. and Rayne, J. A.	1972	A	4.2-295	b, Sample No. 2	5.7	Calculated composition (10.0 a/o Ag); similar to the above specimen.
55	299	Davis, T. H. and Rayne, J. A.	1972	A	4.2-295		10.4	Calculated composition (17.5 a/o Ag); similar to the above specimen.
56	299	Davis, T. H. and Rayne, J. A.	1972	A	4.2-295		15.4	Calculated composition (25.0 a/o Ag); similar to the above specimen.
57*	299	Davis, T. H. and Rayne, J. A.	1972	A	4.2-295		35.4	Calculated composition (50.0 a/o Ag); similar to the above specimen.
58*	301	Stewart, R. G. and Huebener, R. P.	1970	A	1.6		99.86	Calculated composition (0.25 a/o Ag); impurities less than 2 ppm; concentrations determined by wet chemical and neutron activation analyses; polycrystalline wire 10 mil in diameter; obtained from Cominco American, Inc. or Sigmund Cohn, Inc.; rinsed in dilute nitric acid for 20 min, distilled water, and pure alcohol prior to spotwelding to mount; annealed by Joule heating in air starting at 1223 K and ending with 24 h period at 1023 K; cooled to room temperature within 1 h; measured under less than $10^{-4}$ Torr; temperature measurements accurate to within 0.25 K; geometrical factor accurate to $\pm 0.5\%$ ; deviations from Matthiessen's rule reported from 1.6-373 K; data reported as residual resistivity; lowest measurement temperature 1.6 K; data extracted from table.

\* Not shown in figure.

TABLE 51. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD + SILVER ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Au Ag	Composition (continued), Specifications, and Remarks
59*	301	Stewart, R. G. and Huebener, R. P.	1970	A	1.6		99.95 0.05	Calculated composition (0.09 a/o Ag); similar to the above specimen.
60	301	Stewart, R. G. and Huebener, R. P.	1970	A	1.6		99.48 0.52	Calculated composition (0.945 a/o Ag); similar to the above specimen.
61	304	Dewar, J. and Fleming, J. A.	1893	B	76-365		90 10	Wire specimen of the order of 0.08-0.25 mm in diam and 1-3 m in length; wound on vulcanized fiber cylinder, annealed in melted paraffin wax at 473 K, and cooled slowly; measured in boiling oxygen under 760 mm Hg, liquid ethylene, carbon dioxide and ether, melting ice, air, and boiling water; temperatures measured using the resistance of a pure, soft, carefully annealed platinum wire as thermometric property without correcting to absolute thermodynamic scale; no correction for thermal expansion; units of composition not stated, assumed to be weight percent; data extracted from table.
62	37	Linde, J. O.	1931		91-291		0.545	Calculated composition (0.99 a/o Ag); Au from Heraeus and Ag from Kahlbaum melted together under vacuum of about 0.1 mm Hg, homogenized 10-20 h at temperatures near the melting point, formed into square rods of 1.2 mm thickness, drawn through steel or sapphire dies to round wires of 0.75 mm diam and 1.2 cm length, and finally annealed at 773 K; diam measured with micrometer to $\pm 0.003$ mm; data extracted from table.
63	37	Linde, J. O.	1931		91-291		1.59	Calculated composition (2.87 a/o Ag); similar to the above specimen.
64	37	Linde, J. O.	1931		214-242		1.63	Calculated composition (2.93 a/o Ag); similar to the above specimen but annealed at 1073 K.
65*	37	Linde, J. O.	1931		291		1.04	Calculated composition (1.88 a/o Ag); similar to the above specimen but annealing temperature unknown.
66	305	Dawson, H. I.	1965	A	78		98.00 2.00	Calculated composition (3.6 a/o Ag); impurity concentration a few times $10^{-4}$ ; 0.20 or 0.25 mm diam wire; approximate grain size 0.04 mm; annealed at 723 K for 1 h under vacuum of $10^{-5}$ mm Hg; measured in stirred liquid nitrogen; smallest detectable change in resistivity less than $10^{-3}$ $\Omega$ m; data extracted from table.
67	296	Giardina, M. D., Schüle, W., Frank, W., and Seeger, A.	1972		310-1323		9.61	Calculated composition (16.25 a/o Ag); 0.4 mm diam wire specimen; measured during heating (at a rate of about 0.5 K min <sup>-1</sup> ) after slow cooling; data extracted from smooth curve without data points.
68	306	Clay, J.	1908		20-273	No. I	99.604	Although resistivities were reported as being for 0.396% Ag alloy, analysis after measurement actually showed only 0.14% Ag with 0.05% Fe and the rest of the admixture due to other impurities; alloy supplied by Dr. Hofstema, drawn by Heraeus to wires of 0.1 mm diam, annealed, and wound around glass cylinders; resistances measured and reported from 12-170 K; resistivities deduced from resistance measurements and the resistivity at 273 K, calculated according to Matthiessen's observations; data extracted from table.
69	306	Clay, J.	1908		20-273	No. IV	99.132	Similar to the above specimen; resistivities reported as being for 0.986% Ag but analysis showed 0.65% Ag, 0.09% Fe, and the balance other impurities.

\* Not shown in figure.

TABLE 51. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD + SILVER ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Au Ag	Composition (continued), Specifications, and Remarks
70	306	Clay, J.	1908		20-273	No. II	99.524	Similar to the above specimen; total admixture 0.476%.
71	306	Clay, J.	1908		20-273	No. III	99.476	Similar to the above specimen; total admixture 0.524%.
72	306	Clay, J.	1908		20-273	No. V	98.598	Similar to the above specimen; total admixture 1.402%.
73	306	Clay, J.	1908		20-273	No. VI	20.037	Similar to the above specimen, but analysis before drawing showed 20.030% Ag and analysis after measurement showed 20.037% Ag; also this specimen was cleaned with acids each time it was drawn so that impurity content should be less.
74	306	Clay, J.	1908		20-273	No. VII	47.985	Similar to the above specimen; analysis before drawing showed 48.020% Ag, after measurement 47.985% Ag.





TABLE 53. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF GOLD + SILVER ALLOYS (Temperature Dependence) (continued)

DATA SET 24 (cont.)			DATA SET 26 (cont.)			DATA SET 33*			DATA SET 43			DATA SET 53			DATA SET 62		
T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$	
13.92	2.792*		4.23	9.010*		273.2	10.99		4.2	0.370		4.2	2.801*		90.8	0.956	
20.44	2.810		12.0	9.015		373.2	11.76		273.2	2.421*		77	3.268		137.6	1.328	
20.44	2.810*		12.0	9.014*								295	5.120		180.9	1.680	
20.45	2.811*		13.92	9.016		DATA SET 34			DATA SET 44			DATA SET 54*			199.6	1.827	
59.17	3.106		13.92	9.016*		273.2	9.80		4.2	0.135					273	2.429*	
81.36	3.303		20.44	9.030		373.2	10.42		273	2.209		4.2	2.788		290.3	2.565	
81.37	3.304*		20.44	9.030*								77	3.275		291	2.548*	
194.62	4.242		20.45	9.030*		DATA SET 35			DATA SET 45			295	5.113		DATA SET 63		
273.10	4.87		59.17	9.289		273.2	7.58		4.2	8.77		DATA SET 55			90.8	1.483	
293.57	5.035		81.36	9.473		373.2	8.06		77.0	9.18*		4.2	4.627		93	1.502	
296.2	5.055*		81.37	9.473*					296.0	10.84		77	5.111		138.1	1.856	
298.36	5.075		194.62	10.365		DATA SET 36						295	6.911		173	2.149	
DATA SET 25			273.10	10.96		273.2	5.52		DATA SET 46*			DATA SET 56			199.9	2.339	
1.6	7.076		293.57	11.117		373.2	6.29					4.2	6.239		273	2.949	
1.63	7.074*		296.2	11.138*					4.2	8.75		77	6.745		290.3	3.085	
1.98	7.074					DATA SET 37			296.0	10.86		295	8.488		291	3.092*	
1.23	7.074		22	8.85					DATA SET 47			DATA SET 57*			DATA SET 64		
4.24	7.073*		83	9.32		273.2	3.98					4.2	9.020		214.3	2.483	
4.23	7.073*		273	10.8*		373.2	4.55		4.2	2.90		77	9.407		253	2.791	
4.23	7.073*		DATA SET 38									295	11.147		273	2.949*	
12.0	7.078		32	6.69					DATA SET 48			DATA SET 58*			291	3.125*	
12.0	7.077*		83	7.16		4.2	6.038		4.2	2.71		4.2	0.0805		291.6	3.097*	
13.92	7.080		273	8.69		273.2	8.107					1.6	0.0320		DATA SET 65*		
13.92	7.080*		DATA SET 39						DATA SET 49			DATA SET 59*			291	2.823	
20.44	7.096		4.2	7.096*		4.2	2.603		77	8.74		1.6	0.0905		DATA SET 66		
20.44	7.096*		273.2	10.99*		273.2	4.695		273	9.94		1.6	0.0320		78	1.51	
20.45	7.096*		173.2	11.90		DATA SET 40			DATA SET 50			DATA SET 60			DATA SET 67		
59.17	7.374					4.2	1.404		77	9.04*		DATA SET 61			310	8.00	
81.36	7.567		273.2	10.99		4.2	3.517		273	10.53		4.2	0.584		373	8.76	
81.37	7.567*		373.2	11.76		DATA SET 41			295	2.836		76.1	4.817		423	9.25	
194.62	8.482		DATA SET 31*			4.2	0.865					167.1	5.510		473	9.85	
273.10	9.100		273.2	13.89		273.2	2.991		77	1.067		273	4.817		523	10.36	
293.57	9.240		373.2	13.89		DATA SET 42			295	2.836		191.3	5.666		573	10.92	
296.2	9.290*		DATA SET 32			4.2	0.670		DATA SET 52			76.1	4.817		623	11.43	
298.36	9.300		273.2	11.24					273	1.401*		273	6.280		673	12.03	
DATA SET 26			373.2	11.90*		4.2	1.401*		273	1.890		274.1	6.293*		723	12.59	
1.6	9.009					4.2	1.890		295	3.673		289.6	6.417		773	13.34	
1.63	9.011*											364.8	6.997		823	13.90	
1.98	9.008														873	14.41	
4.23	9.009*																
4.24	9.009*																
4.23	9.010*																

\* Not shown in figure.

TABLE 52. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF GOLD + SILVER ALLOYS (Temperature Dependence) (continued)

T	$\rho$	T	$\rho$
DATA SET 67 (cont.)		DATA SET 73	
923	15.11	20.24	7.2119
973	13.58*	90.43	7.7563
1023	16.62	169.33	8.3980
1073	17.36	273	9.1978*
1123	18.18	DATA SET 74	
1173	18.97	20.24	8.1686
1223	19.64	90.43	8.6426
1273	20.72	169.33	9.1374
1323	21.60	273	9.9378*
DATA SET 68			
20.24	0.3077		
90.43	0.8619		
169.33	1.5307		
273	2.3713		
DATA SET 69			
20.24	0.6061		
90.43	1.1611		
169.33	1.8063		
273	2.6371		
DATA SET 70			
20.24	0.3591		
90.43	0.9164		
169.33	1.5709		
273	2.4167*		
DATA SET 71			
20.24	0.3809		
90.43	0.9406		
169.33	1.5961		
273	2.4441*		
DATA SET 72			
20.24	0.8763		
90.43	1.4434		
169.33	2.0930		
273	2.9302*		

\* Not shown in figure.

TABLE 53. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF SILVER + GOLD ALLOYS (Temperature Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Ag Au	Composition (continued), Specifications, and Remarks
1	307	Weinberg, I.	1967	A	4.2-295		0.38	Calculated composition (0.21 a/o Au), solute concentration determined by chemical analysis; 99.999 Ag from Cominco and 99.999 Au from American Smelting and Refining Co. melted in previously outgassed, high purity graphite crucibles by induction heating under a dynamic vacuum of $10^{-6}$ Torr; shaken vigorously for 1 h, billets inverted and a similar melting cycle repeated, homogenized at 873 K for 6 d under $10^{-6}$ Torr vacuum, swaged to 0.070 in and drawn through diamond die to final diam of 0.010 in with frequent etchings with mixture of ammonium hydroxide and hydrogen peroxide, and finally annealed at 953 K for 24 h; data extracted from table.
2	307	Weinberg, I.	1967	A	4.2-295		0.51	Calculated composition (0.28 a/o Au); similar to the above specimen.
3	307	Weinberg, I.	1967	A	4.2-295		0.84	Calculated composition (0.46 a/o Au); similar to the above specimen.
4	307	Weinberg, I.	1967	A	4.2-295		2.7	Calculated composition (1.5 a/o Au); similar to the above specimen.
5	300	Boes, J., Van Dam, A.J., and Bijvoet, J.	1968		0.94-13		28.6	Calculated composition (18 a/o Au); strip specimen 30 x 1 x 1 mm; prepared by arc melting in argon and rolling; annealed for 1 h at 1073 K in vacuum and quenched; data read from figure.
6*	300	Boes, J., et al.	1968		1.0-13			Similar to the above specimen but 0.10 Yb (0.07 a/o) added.
7*	300	Boes, J., et al.	1968		1.1-14			The above specimen annealed at 1073 K and quenched.
8	300	Boes, J., et al.	1968		4.2, 77		16.9	Calculated composition (10 a/o Au); similar to the specimen of data set number 5 above but data extracted from table.
9	300	Boes, J., et al.	1968		4.2, 77		31.3	Calculated composition (20 a/o Au); similar to the above specimen.
10	300	Boes, J., et al.	1968		4.2, 77		43.9	Calculated composition (30 a/o Au); similar to the above specimen.
11	302	Iyer, V.K. and Astumow, R.M.	1967		290-1147		8.8	Calculated composition (5 a/o Au), less than 0.1 total impurity with 0.05 Si indicated by spectrographic analysis; 99.999 Ag and Au melted in graphite crucible in Ar, evaporation of Ag resulted in composition change of less than 0.2 a/o; 0.125 in diam ingots swaged to 0.040 in and drawn to 0.010 in; annealed in Ar; maintained in Ar above 673 K to prevent evaporation of Ag but measured in vacuum, different specimen measured in 273-303 K range than at higher temperatures, identical values obtained for at least two heating cycles; uncertainty in temperature $\pm 0.3^\circ$ T ( $^\circ$ C), standard deviation of points from best fitted curve less than 0.5% of the mean, resistivity accurate to within 2% with error mainly due to length and density uncertainty; data corrected for thermal expansion; data read from figure.
12	302	Iyer, V.K. and Astumow, R.M.	1967		275-1147		16.9	Calculated composition (10 a/o Au); similar to the above specimen.
13	302	Iyer, V.K. and Astumow, R.M.	1967		276-1158		43.9	Calculated composition (30 a/o Au); similar to the above specimen.
14	127	Roll, A. and Motz, H.	1957	E	1286-1461		47.7	Calculated composition (33.3 a/o Au); measured in liquid state; prepared from 99.995 pure Ag and 99.95 pure Au; $d\rho/dT = 0.0154 \times 10^{-6} \Omega$ m K $^{-1}$ ; $\pm 1\%$ uncertainty in resistivity; data read from figure.
15	127	Roll, A. and Motz, H.	1957	E	1282-1461		43.9	Calculated composition (30 a/o Au); similar to the above specimen but with $d\rho/dT = 0.0137 \times 10^{-6} \Omega$ m K $^{-1}$ .

\* Not shown in figure.

TABLE 53. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF SILVER + GOLD ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Ag Au	Composition (continued), Specifications, and Remarks
16	127	Roll, A. and Motz, H.	1957	E	1279-1463		41.5	Calculated composition (28 a/o Au); similar to the above specimen but with $d\rho/dT = 0.0134 \times 10^{-8} \Omega \text{m K}^{-1}$ .
17	127	Roll, A. and Motz, H.	1957	E	1277-1463		39.1	Calculated composition (26 a/o Au); similar to the above specimen but with $d\rho/dT = 0.0153 \times 10^{-8} \Omega \text{m K}^{-1}$ .
18	127	Roll, A. and Motz, H.	1957	E	1275-1463		37.2	Calculated composition (24.5 a/o Au); similar to the above specimen but with $d\rho/dT = 0.0138 \times 10^{-8} \Omega \text{m K}^{-1}$ .
19	127	Roll, A. and Motz, H.	1957	E	1273-1462		35.3	Calculated composition (23 a/o Au); similar to the above specimen but with $d\rho/dT = 0.0126 \times 10^{-8} \Omega \text{m K}^{-1}$ .
20	127	Roll, A. and Motz, H.	1957	E	1257-1462		20.7	Calculated composition (12.5 a/o Au); similar to the above specimen but with $d\rho/dT = 0.0100 \times 10^{-8} \Omega \text{m K}^{-1}$ .
21	297	Glaesche, W. F. and Stout, I. W.	1938	A	1.6-298	No. 2	56.06	Drawn wire 0.579 mm in diam and about 100 cm long; annealed by heating to redness in a flame during and after drawing, wound loosely on a spool in insulated layers, and again annealed at 383 K for about 15 h; measured in four cooling cycles over period of 2.5 years; resistance ratio as a function of temperature and specific resistance at 273 K reported, from which resistivity was calculated by CINDAS, without correcting for thermal expansion; resistance accurate to 0.01%; data extracted from table.
22	297	Glaesche, W. F. and Stout, I. W.	1938	A	1.6-298	No. 1	83.12	Similar to the above specimen.
23	128	Grüneisen, E. and Reddemann, H.	1934		22-273	4	0.87	Calculated composition (0.37 a/o Au); wire specimen; data extracted from table.
24	128	Grüneisen, E. and Reddemann, H.	1934		22-273	5	37.8	Calculated composition (25 a/o Au); single crystal wire specimen; data extracted from table.
25	130	Sedstrom, E.	1919		273, 373		8.78	Calculated composition (5.0 a/o Au); wire specimen rolled and drawn to thickness of 1 mm and annealed at temperature near the melting point for 0.5 h; data extracted from table.
26	130	Sedstrom, E.	1919		273, 373		16.4	Calculated composition (9.7 a/o Au);
27	130	Sedstrom, E.	1919		273, 373		23.66	Calculated composition (14.5 a/o Au);
28	130	Sedstrom, E.	1919		273, 373		31.37	Calculated composition (20.0 a/o Au);
29	130	Sedstrom, E.	1919		273, 373		44.16	Calculated composition (30.2 a/o Au);
30*	296	Cripp, R. S. and Rungla, J.	1970	A	4.2, 273		0.164	Composition (0.09 a/o Au) estimated by using Nordheim's rule and the values of residual resistivity per atomic percent solute reported by J. O. Linde (Ph.D. Thesis, University of Stockholm, 1939); 1 mm diam wire specimens prepared from 99.9999 Ag and 99.999 or 99.9999 Au were supplied in three batches by Cambridge Metals Research Ltd; similar specimens analyzed for Fe content by D. S. Russell, Division of Applied Chemistry, N.R.C. of Canada showed 1.7 to 9.0 atomic ppm of Fe; specimen as received was drawn down, etched, washed in distilled water and alcohol, dried, sealed in quartz capsule with 0.33 atm oxygen to minimize Fe contamination, and annealed at 1173 K for 72 h; uncertainty in temperature less than 0.05 K at all temperatures, 1% error in absolute resistivity; measurement temperature of residual resistivity not stated, assumed to be 4.2 K; data extracted from table.

\* Not shown in figure.

TABLE 53. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF SILVER + GOLD ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Ag Au	Composition (continued), Specifications, and Remarks
31	298	Crisp, R.S. and Rungis, J.	1970	A	4.2-273		1.25	Estimated composition (0.69 a/o Au); similar to the above specimen.
32	298	Crisp, R.S. and Rungis, J.	1970	A	4.2-273		1.43	Estimated composition (0.79 a/o Au); similar to the above specimen.
33	298	Crisp, R.S. and Rungis, J.	1970	A	4.2-273		2.47	Estimated composition (1.37 a/o Au); similar to the above specimen.
34	298	Crisp, R.S. and Rungis, J.	1970	A	4.2-273		2.97	Estimated composition (1.65 a/o Au); similar to the above specimen.
35	293	Crisp, R.S. and Rungis, J.	1970	A	4.2-273		3.95	Estimated composition (2.2 a/o Au); similar to the above specimen.
36	298	Crisp, R.S. and Rungis, J.	1970	A	4.2-273		9.27	Estimated composition (5.3 a/o Au); similar to the above specimen.
37	298	Crisp, R.S. and Rungis, J.	1970	A	4.2-273		9.94	Estimated composition (5.7 a/o Au); similar to the above specimen.
38*	298	Crisp, R.S. and Rungis, J.	1970	A	4.2-273		16.9	Estimated composition (10 a/o Au); similar to the above specimen.
39	298	Crisp, R.S. and Rungis, J.	1970	A	4.2-273		40.3	Estimated composition (27 a/o Au); similar to the above specimen.
40	135	Huray, P.G., Roberts, L.D., and Thomson, J.O.	1971		4.2-296	Ag <sub>2</sub> Au	62.16 37.84	Calculated composition (25 a/o Au). Chemical analysis by Johnson Matthey Chemicals Ltd. of similar specimens indicated composition uncertainty of 0.5 a/o; foil specimen 0.0508 mm (2 mil) thick and uniform to at least 1%; 99.99 Ag and Au arc-melted and cooled in Ar several times and drop cast into water cooled cylindrical mold, cylinders annealed in inert atmosphere for 5 d within 50 K of melting point, strips cut lengthwise from cylinder and rolled to final thickness with intermittent annealing; finally annealed for 100 h at 641 K, furnace cooled to 523 K, maintained at 523 K for 120 h, and quenched in room temperature water; uncertainty in thickness less than 1%; uncertainties in resistivity $\pm 0.07$ , $\pm 0.07$ , and $\pm 0.08 \times 10^{-8} \Omega m$ at 4.2, 77, and 296 K respectively; data extracted from table.
41	135	Huray, P.G., et al.	1971		4.2-296	Ag <sub>2</sub> Au	62.16 37.84	Similar to the above specimen except finally suspended in vacuum furnace at 873 K for 30 min or more, then dropped in vacuum into room temperature bath of vacuum-pump oil.
42	308	Barber, A.J. and Caplin, A.D.	1975	A	4.26-8.83	Sample 63	1.67	Calculated, nominal composition (0.92 a/o Au); wire specimen 0.45 mm in diam; high purity Ag and Au melted together in quartz ampoules, homogenized, and rolled and drawn into wire; annealed at about 873 K for 6-10 h in a sealed capsule containing up to 10 Torr oxygen or in a continuous flow of oxygen at $10^{-1}$ Torr in order to precipitate Fe and remove resistance minimum; measured in helium environments residual resistivity $0.3470 \times 10^{-8} \Omega m$ ; uncertainty in temperature less than $\pm 200$ mK below 20 K and $\pm 100$ mK below 10 K, less than 1% uncertainty in geometrical factor; data read from figure.

\* Not shown in figure.

TABLE 53. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF SILVER + GOLD ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Ag Au	Composition (continued), Specifications, and Remarks
43	308	Barber, A.J. and Caplin, A.D.	1975	A	4.25-18.75	Sample 38	0.64	Similar to the above specimen; calculated, nominal composition (0.35 a/o Au); (0.62 mm diam); residual resistivity $0.1230 \times 10^{-8} \Omega\text{m}$ .
44	308	Barber, A.J. and Caplin, A.D.	1975	A	4.24-18.71	Sample 37	0.33	Similar to the above specimen; calculated, nominal composition (0.18 a/o Au); (0.45 mm diam); residual resistivity $0.0650 \times 10^{-8} \Omega\text{m}$ .
45*	308	Barber, A.J. and Caplin, A.D.	1975	A	4.27-18.92	Sample 36	0.13	Similar to the above specimen; calculated, nominal composition (0.07 a/o Au); (0.40 mm diam); residual resistivity $0.0314 \times 10^{-8} \Omega\text{m}$ .
46*	308	Barber, A.J. and Caplin, A.D.	1975	A	4.27-19.05	Sample 35	0.064	Similar to the above specimen; calculated, nominal composition (0.035 a/o Au); (0.38 mm diam); residual resistivity $0.0156 \times 10^{-8} \Omega\text{m}$ .
47*	308	Barber, A.J. and Caplin, A.D.	1975	A	3.13-19.59	Sample 51	0.033	Similar to the above specimen; calculated, nominal composition (0.018 a/o Au); (0.15 mm diam); residual resistivity $0.00858 \times 10^{-8} \Omega\text{m}$ .
48	299	Davis, T.H. and Rayne, J.A.	1972	A	4.2-295		96.41 3.59	Calculated composition (2.0 a/o Au); parallelepiped specimen $0.100 \times 0.100 \times 1.25 \text{ in}$ ; 99.999% pure Ag and Au induction melted in helium; cast in ingots, cold worked, etched in 10% KCN, homogenized at 1073 K for 72 h, and quenched in cold water; specimen machined from axial section of ingot using tungsten carbide tools and KCN etching to avoid ferromagnetic contaminants; finally heated at 1073 K for 72 h and quenched; measurements at ambient temperature were reported as being at $295 \pm 3 \text{ K}$ ; uncertainty in resistivity 1.5%; data extracted from table.
49*	299	Davis, T.H. and Rayne, J.A.	1972	A	4.2-295		83.13 16.87	Calculated composition (10.0 a/o Au); similar to the above specimen.
50	299	Davis, T.H. and Rayne, J.A.	1972	A	4.2-295		68.66 31.34	Calculated composition (20.0 a/o Au); similar to the above specimen.
51*	299	Davis, T.H. and Rayne, J.A.	1972	A	4.2-295		56.10 43.90	Calculated composition (30.0 a/o Au); similar to the above specimen.
52	309	Sparks, L.L. and Rust, J.G.	1972		4		41.5	0.00020 Fe, 0.00020 Si, 0.00005 Al, 0.00005 Ca, 0.00005 Ni, 0.00003 Pb, and 0.00002 Ni, calculated composition (28 a/o Au); cast in billets 2.5 cm in diam, swaged to 0.6 cm diam, drawn to 0.06 cm with carbon dies, and finally drawn to 0.020 cm wire with diamond dies; no lubrication, annealing, or etching during drawing process; annealed at 873 K for 1 h; $\pm 3\%$ uncertainty in resistivity; data read from figure.
53	309	Sparks, L.L. and Rust, J.G.	1972		4		22.2	Calculated composition (13.5 a/o Au); no details reported.
54	38	Linde, J.O.	1932		89.3-293		1.65	Calculated composition (0.91 a/o Au); Kahlbaum Ag and Heraeus Au melted together in 3 x 40 mm quartz tubes, homogenized, drawn through sapphire die to about 0.75 mm diam, and annealed in vacuum at 773 K for 5-10 h; data extracted from table.
55	38	Linde, J.O.	1932		89.6-293		3.63	Calculated composition (2.02 a/o Au); similar to the above specimen.
56	38	Linde, J.O.	1932		94.5-290		5.38	Calculated composition (3.02 a/o Au); similar to the above specimen.
57	38	Linde, J.O.	1932		291		0.960	Calculated composition (0.528 a/o Au); similar to the above specimen.
58*	38	Linde, J.O.	1932		291		2.83	Calculated composition (1.57 a/o Au); similar to the above specimen.

\* Not shown in figure.

TABLE 53. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF SILVER + GOLD ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Ag Au	Composition (continued), Specifications, and Remarks
59	305	Dawson, H.I.	1965	A	78			Calculated composition (15.4 a/o Au), impurity concentration a few times $10^{-4}$ ; 0.20 or 0.25 mm diam wire; approximate grain size 0.02 mm; annealed at 723 K for 1 h under vacuum of $10^{-5}$ mm Hg; measured in stirred liquid nitrogen; smallest detectable change in resistivity less than $10^{-12}$ $\Omega$ m; data extracted from table.
60	310	Van Baarle, C., Cortier, F. W., and Winemilus, P.	1967		4.2		0.75	Calculated composition (0.41 a/o Au); prepared from Tadanac High Purity Silver from Consolidated Mining and Smelting Company of Canada, Ltd of 99.9999 nominal purity; spectrographic analysis by supplier showed 1 ppm Si and less than 0.1 ppm Fe, Ca, and Ag; ingots of diam 0.3 cm and several centimeters long prepared with slow solidification and low cooling rate; axial homogeneity checked by measuring resistivity of samples from both ends of ingot; annealed in vacuum at 823 K for 24 h; resistivities reported as $\rho_0$ , assumed to be measured at 4.2 K; data extracted from table.
61	310	Van Baarle, C., et al.	1967		4.2		2.28	Similar to the above specimen; calculated composition (1.26 a/o Au); annealed at 848 K in vacuum for 48 h.
62	310	Van Baarle, C., et al.	1967		4.2		5.21	Similar to the above specimen; calculated composition (2.92 a/o Au); annealed at 848 K in vacuum for 48 h.
63	310	Van Baarle, C., et al.	1967		4.2			The above specimen strained 10%.

TABLE 54. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF SILVER + GOLD ALLOYS (Temperature Dependence)

[Temperature, T, K; Electrical Resistivity, $\rho$ , $10^{-8} \Omega \text{ m}$ ]																	
T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$	
DATA SET 1			DATA SET 5 (cont.)			DATA SET 7 (cont.)*			DATA SET 11 (cont.)			DATA SET 13 (cont.)			DATA SET 16 (cont.)		
4.2	0.07		11.10	5.36324		6.54	5.7310		632	5.54		707	12.70		1363	31.4	
77.3	0.35		11.26	5.36364*		6.77	5.7306		663	5.75		763	13.25		1383	31.5*	
273.0	1.52		12.26	5.36422		6.95	5.7306		733	6.40		827	13.68		1404	31.7*	
295.0	1.65		13.20	5.36489		7.15	5.7305		792	6.66		877	14.10		1423	32.3*	
DATA SET 2			DATA SET 6*			7.38	5.7304		846	7.09		903	14.33		1442	32.5	
4.2	0.10		1.02	5.6194		7.59	5.7303		875	7.31		931	14.56		1463	32.8*	
77.3	0.38		2.42	5.6170		7.84	5.7302		964	7.96		965	14.88		DATA SET 17		
273.0	1.54		2.92	5.6165		8.16	5.7303		1044	8.60		1011	15.24		1277	28.9	
295.0	1.67		4.13	5.6157		8.45	5.7303		1074	8.85		1054	15.74		1285	28.9*	
DATA SET 3			4.13	5.6155		8.86	5.7303		1116	9.19		1096	16.13		1304	29.4*	
4.2	0.16		7.54	5.6144		9.34	5.7303		1147	9.58		1141	16.61		1324	29.6*	
77.3	0.44		7.66	5.6143		10.32	5.7301		DATA SET 12			1158	16.83		1342	29.9	
273.0	1.62		7.90	5.6142		10.66	5.7299		275	4.89		DATA SET 14			1363	30.3*	
295.0	1.76		8.06	5.6143		11.28	5.7300		304	5.08		1286	31.3		1383	30.5*	
DATA SET 4			8.24	5.6143		11.79	5.7302		359	5.40		1321	31.7*		1402	30.8	
4.2	0.50		8.49	5.6142		12.91	5.7306		471	6.17		1341	37.2		1424	31.2*	
77.3	0.81		8.65	5.6141		13.46	5.7310		519	6.48		1360	32.6*		1443	31.5*	
273.0	1.99		8.88	5.6141		13.85	5.7313		537	6.56		1381	32.8*		1463	31.8	
295.0	2.13		10.07	5.6143		14.32	5.7315		572	6.85		1401	33.2		DATA SET 18		
DATA SET 5			10.49	5.6144		DATA SET 8			612	7.11		1420	33.4*		1275	28.9*	
0.94	5.36345*		11.00	5.6144		4.2	3.32		642	7.35		1440	33.6*		1285	29.0*	
2.40	5.36325		12.63	5.6151		77	3.66		719	7.96		1461	34.1		1304	29.3*	
2.40	5.36304*		DATA SET 7*			DATA SET 9			766	8.17		DATA SET 15			1323	29.5	
2.91	5.36298		1.07	5.7447		DATA SET 10			801	8.49		1282	30.0		1343	29.9*	
4.13	5.36290		1.92	5.7404		4.2	5.82		839	8.79		1302	30.0*		1363	30.4*	
4.13	5.36260*		2.95	5.7367		77	6.22		923	9.49		1322	30.3*		1403	30.7*	
7.16	5.36298		3.60	5.7356		DATA SET 11			961	9.71		1341	30.6		1422	31.1*	
7.24	5.36260		4.21	5.7338		290	3.40		1024	10.21		1360	30.9*		1442	31.3	
7.60	5.36260		4.21	5.7335		352	3.77		1065	10.52		1381	31.2*		1463	31.7*	
7.71	5.36268		4.48	5.7335		424	4.21		1127	11.11		1400	31.5		DATA SET 19		
7.92	5.36257		4.99	5.7335		464	4.44		1147	11.19		1420	31.7*		1273	27.6	
8.13	5.36283		4.71	5.7331		508	4.74		DATA SET 13			1441	32.0*		1285	27.8*	
8.33	5.36275		5.13	5.7324		568	5.09		276	9.50		1461	32.3		1303	28.1*	
8.57	5.36275		5.38	5.7321		DATA SET 16			297	9.62		DATA SET 16			1323	28.3*	
8.72	5.36274		5.49	5.7320		376	3.91		399	10.39		1279	30.1*		1342	28.7	
8.96	5.36275		5.49	5.7318		424	4.21		435	10.64		1294	30.2*		1363	28.9*	
10.14	5.36329		5.75	5.7317		464	4.44		535	11.37		1304	30.5		1383	29.2*	
10.81	5.36343		5.98	5.7315		512	4.74		580	11.54		1323	30.7*		1402	29.3	
			6.38	5.7311		568	5.09		667	12.35		1343	31.1*		1422	29.6*	

\* Not shown in figure.



[illegible]

\* Not shown in figure.

TABLE 54. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF SILVER + GOLD ALLOYS (Temperature Dependence) (continued)

T	$\rho$	T	$\rho$
<u>DATA SET 47 (cont.)*</u>		<u>DATA SET 55</u>	
14.32	0.009822	89.6	1.113
15.35	0.01018	122.1	1.322
16.41	0.01066	138.7	1.423
17.26	0.01117	248.5	2.100
18.75	0.01198	273	2.250
19.59	0.01268	292.6	2.363*
<u>DATA SET 48</u>		<u>DATA SET 56</u>	
4.2	0.736	94.5	1.493
77	1.047	125.0	1.678
295	2.375	133.0	1.856
<u>DATA SET 49</u>		256.9	2.494
4.2	3.360	273	2.594
77	3.668	290.2	2.697
295	5.135	<u>DATA SET 57</u>	
<u>DATA SET 50</u>		291	1.781
4.2	5.835*	<u>DATA SET 58*</u>	
77	6.199*	291	2.158
295	7.723	<u>DATA SET 59</u>	
<u>DATA SET 51</u>		78	5.13
4.2	7.774	<u>DATA SET 60</u>	
77	8.127	4.2	0.140
295	9.747	<u>DATA SET 61</u>	
<u>DATA SET 52</u>		4.2	0.408
4	6.97	<u>DATA SET 62</u>	
<u>DATA SET 53</u>		4.2	0.97
4	4.05	<u>DATA SET 63</u>	
<u>DATA SET 54</u>		4.2	0.99
89.3	0.731		
97.4	0.771		
138.1	1.026		
262.5	1.784		
273	1.827		
292.6	1.962		

\* Not shown in figure.

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ELECTRICAL RESISTIVITY OF TEN SELECTED BINARY ALLOY  
SYSTEMS(U) THERMOPHYSICAL AND ELECTRONIC PROPERTIES  
INFORMATION ANALYSIS CENTER LAFAYETTE IN C Y HO ET AL.

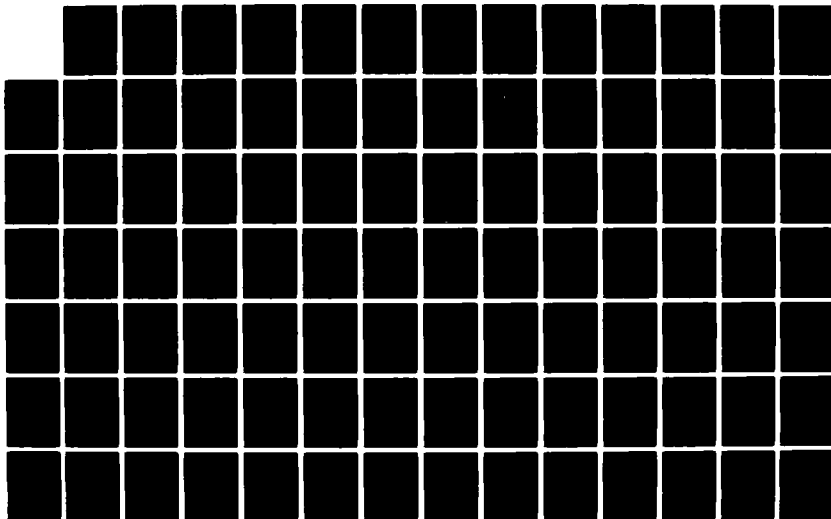
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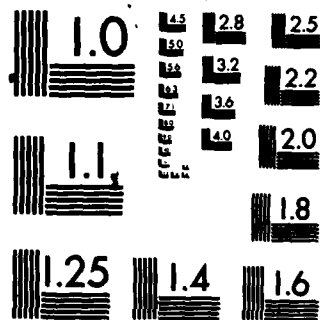
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MICROCOPY RESOLUTION TEST CHART  
NATIONAL BUREAU OF STANDARDS-1963-A

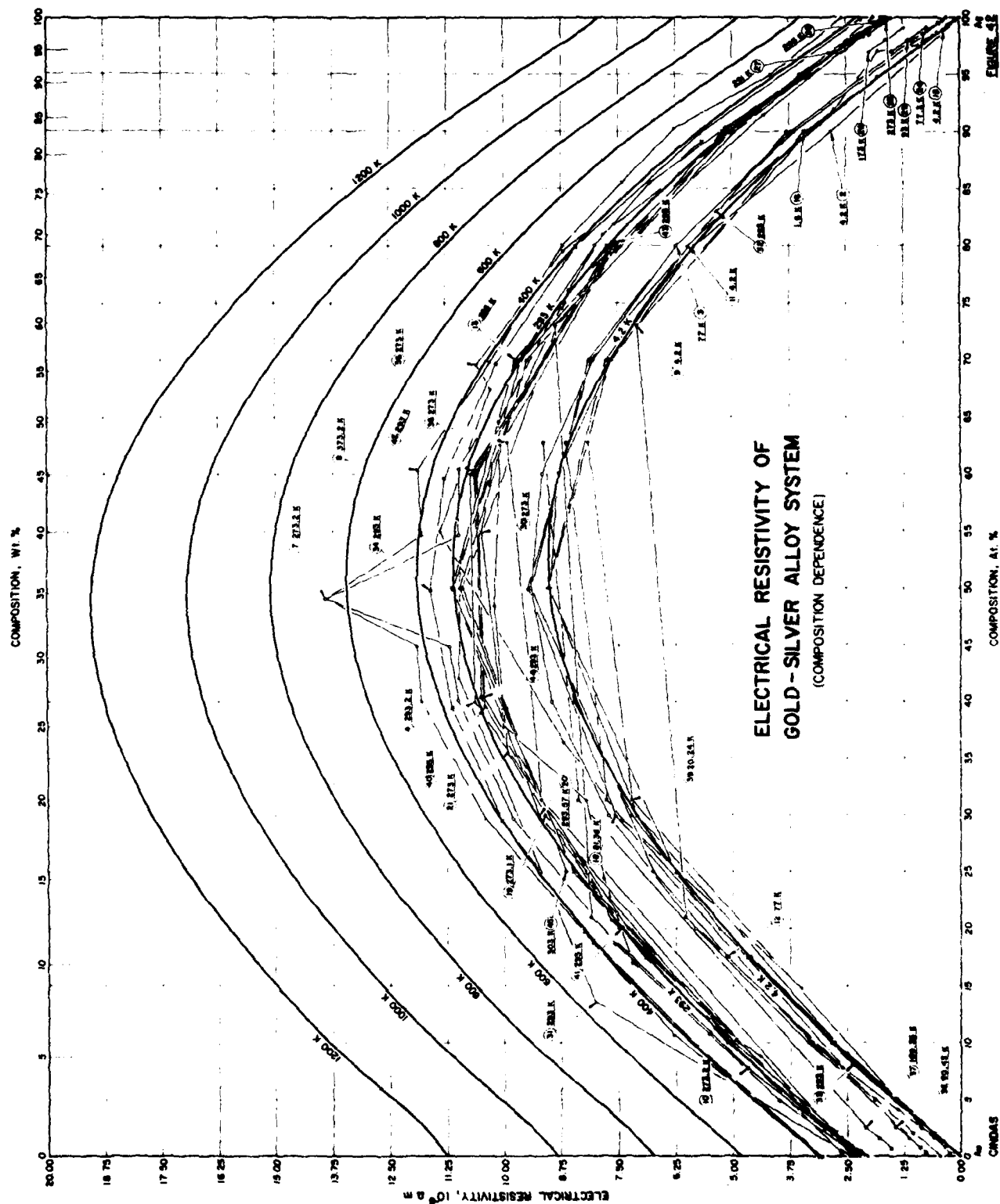


TABLE 55. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD-SILVER ALLOY SYSTEM (Composition Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp., K	Composition Range (At. % of Ag)	Composition (weight percent), Specifications, and Remarks
1*	311	Hove, R.A. and Enderby, J.E.	1967	A	1408	0-100	Liquid alloys prepared from 99.999 pure Ag and Au supplied by Koch-Light by melting under Ar; fused quartz sample holders, 0.003 in Mo foil contacts coated with colloidal graphite to prevent contamination; no more than 0.01 a/o Mo was dissolved by sample and there was no visual evidence of the quartz cells being attacked by the samples except in case of the 90 a/o Ag alloy which was heated to above 1453 K; simultaneous thermopower and resistivity measurements; data read from figure.
2	300	Boes, J., Van Dam, A.J., and Bijvoet, J.	1968		4.2	0-100	Strip specimens 30 x 1 x 1 mm; prepared by arc-melting in Ar and rolling; annealed for 1 h at 1073 K in vacuum and quenched; data extracted from table.
3	300	Boes, J., et al.	1968		77	0-100	The above specimens.
4	312	Richter, T.	1963	V	293.2	0-100	Alloys prepared from 99.99 <sup>+</sup> pure metals by arc-melting; homogenized at 873 K for 1 h; data extracted from table.
5*	127	Roll, A. and Motz, H.	1957	R	1373	0-100	Alloys in liquid state; prepared from 99.985 pure silver and 99.95 pure gold; $\pm 1\%$ uncertainty in resistivity; data read from figure.
6*	127	Roll, A. and Motz, H.	1957	R	1473	0-100	The above specimens.
7	130	Sodström, E.	1919		273.1	0-100	Wire specimens 1 mm thick; rolled and drawn; heated at near melting point for 30 min; data extracted from table.
8	130	Sodström, E.	1919		373.2	0-100	The above specimens.
9	298	Crisp, R.S. and Rungis, J.	1970	A	4.2	0.15-100	Compositions estimated by using Nordheim's rule and the values of residual resistivity per atomic percent solute reported by J.O. Linde (Ph.D. Thesis, University of Stockholm, 1939); 1 mm diam wire specimens prepared from 99.9999 Ag and 99.999 or 99.9999 Au were supplied in three batches by Cambridge Metals Research Ltd.; some specimens analyzed for Fe content showed 1.7 to 9.0 atomic ppm of Fe; specimens as received were drawn down, etched, washed in distilled water and alcohol, dried, sealed in quartz capsules with 0.33 atm oxygen to minimize Fe contamination, and annealed at 1173 K for 72 h; uncertainty in temperature less than 0.05 K, 1% error in absolute resistivity; measurement temperature of residual resistivity not stated, assumed to be 4.2 K; data extracted from table.
10	298	Crisp, R.S. and Rungis, J.	1970	A	273.2	0.15-100	The above specimens.
11	299	Davis, T.H. and Rayne, J.A.	1972	A	4.2	0-100	Parallelogram specimens 0.100 x 0.100 x 1.25 in; 99.999 <sup>+</sup> pure Ag and Au induction melted in helium; cast in ingots, cold worked, etched in 10% KCN, homogenized at 1073 K for 72 h and quenched in cold water; specimen machined from axial section of ingot using tungsten carbide tools and KCN etching to avoid ferromagnetic contaminants; finally heated at 1073 K for 72 h and quenched; uncertainty in resistivity $\pm 1.5\%$ ; authors reported residual resistivity of 0.298 x 10 <sup>-4</sup> $\Omega$ m for 1 a/o Ag in Au and of 0.375 x 10 <sup>-4</sup> $\Omega$ m for 1 a/o Au in Ag; data extracted from table.
12	299	Davis, T.H. and Rayne, J.A.	1972	A	77	0-100	The above specimens measured at 77 K.
13	299	Davis, T.H. and Rayne, J.A.	1972	A	295	0-100	The above specimens measured at 295 $\pm$ 3 K.

\* Not shown in figure.

TABLE 55. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD-SILVER ALLOY SYSTEM (Composition Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp., K	Composition Range (At. % of Ag)	Composition (weight percent), Specifications, and Remarks
14*	301	Stewart, R. G. and Huebner, R. P.	1970	A	1.6	0.25-0.94	Polycrystalline wire specimens 10 mil in diam; concentrations determined by wet chemical and neutron activation analyses; minor constituent impurities about 2 ppm; supplied by Cominco American, Inc. and Sigmund Cohn, Inc.; rinsed in dilute nitric acid prior to spot welding to sample holder; annealed by Joule heating in air starting at 1223 K and ending with 24 h at 1023 K; cooled to room temperature within 1 h; temperatures accurate to within 0.25 K, geometric factor to 0.5%; thermal expansion correction neglected; data extracted from table.
15	29	Dugdale, J. S. and Eastinaki, Z. S.	1967	A	4.2	99.0-99.9	Rod specimens 1 mm in diam and 15 cm long; 99.999 Ag from American Smelting and Refining Co. and 99.99* Au from Canadian Mint melted together in sealed quartz tube and quenched; homogenized at 1213 K for 3 weeks; ingots drawn to required diam and annealed at 923 K for 18 h in close fitting Pyrex tubes; 0.365 x 10 <sup>-8</sup> Ωm residual resistivity per atomic percent solute reported for Au in Ag, accurate to better than 1%; deviations from Matthiessen's rule reported from 4.2 K to room temperature; resistivity uncertainty ± 0.1%; data read from figure.
16	297	Glaucque, W. F. and Stout, J. W.	1938	A	1.6	0.2-90	Drawn wires 0.579 mm in diam and about 1 m long; annealed by heating to redness in a flame during and after drawing, wound loosely on spool in insulated layers, and again annealed at 383 K for about 15 h; measured in four cooling cycles over period of 2.5 years; resistance ratio as function of temperature and specific resistance at 273 K reported, from which resistivity was calculated by CINDAS, without correcting for thermal expansion; resistance accurate to 0.01%, resistivity to 1%; data extracted from table.
17*	297	Glaucque, W. F. and Stout, J. W.	1938	A	12.0	0.2-90	The above specimens.
18	297	Glaucque, W. F. and Stout, J. W.	1938	A	81.36	0.2-90	The above specimens.
19	297	Glaucque, W. F. and Stout, J. W.	1938	A	273.1	0.2-90	The above specimens.
20	297	Glaucque, W. F. and Stout, J. W.	1938	A	293.57	0.2-90	The above specimens.
21*	313	Beckmann, B.	1911		273	2.1-98.05	Original document not available; data, without measurement information, extracted from International Critical Tables of Numerical Data, Physics, Chemistry and Technology, Vol. 6, McGraw-Hill Book Co., p. 160, 1929.
22*	308	Barber, A. J. and Caplin, A. D.	1975	A	0	99.08-99.982	Wire specimens: high purity Ag and Au melted together in quartz ampoules, homogenized, and rolled and drawn into wire; annealed at 873 K for 6-10 h in a sealed capsule containing up to 10 Torr oxygen or in a continuous flow of oxygen at 10 <sup>-1</sup> Torr in order to precipitate Fe; uncertainty in temperature less than ± 200 mK below 20 K and ± 100 mK below 10 K; uncertainty in geometrical factor less than 1%; resistivity measured down to 1.5 K and extrapolated to 0 K if necessary; data extracted from table.
23*	307	Weinberg, I.	1967	A	4.2	99.5-99.95	Compositions determined by chemical analysis; wire specimens 0.010 in. in diam; 99.999 Ag from Cominco and 99.999 Au from American Smelting and Refining Co. melted under dynamic vacuum of 10 <sup>-5</sup> Torr in previously outgassed, high purity graphite crucibles by induction heating, shaken vigorously for 1 h; billets inverted and a similar melting cycle repeated; ingot given a homogenizing anneal at 873 K for 6 days under vacuum of 10 <sup>-1</sup> Torr, swaged to 0.070 in and drawn through diamond die to final diameter; etched frequently during drawing and swaging with mixture of ammonium hydroxide and hydrogen peroxide; final anneal at 953 K for 24 h; temperature and voltage measurements accurate to 0.1 K and 0.01 μV respectively; data extracted from table.

\* Not shown in figure.

TABLE 55. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD-SILVER ALLOY SYSTEM (Composition Dependence) (continued)

Data Ref. Set No.	Author(s)	Year	Method Used	Temp., K	Composition Range (At. % of Ag)	Composition (weight percent), Specifications, and Remarks
24 307 Weinberg, I.		1967	A	77.3	98.5-99.95	The above specimens.
25 307 Weinberg, I.		1967	A	273	98.5-99.95	The above specimens.
26 307 Weinberg, I.		1967	A	295	98.5-99.95	The above specimens.
27 38 Linde, J.O.		1932		291	96.98-99.09	Kahlbaum Ag and Heraeus Au melted together in 3 x 40 mm quartz tubes, homogenized, drawn through sapphire dies to about 0.75 mm diam, and annealed in vacuum at 773 K for 5-10 h; data extracted from table.
28 38 Linde, J.O.		1932		173	96.98-99.09	The above specimens.
29 38 Linde, J.O.		1932		93	96.98-99.09	The above specimens.
30 314 Broniewski, W. and Wesolowski, K.		1932		273	0-99.2	Components melted in vacuum in high frequency furnace; ingots tempered at 1073 K for 50 h, reduced to 5 mm diam wires and tempered at 823 K for 1 h; conductivity reported in figure in incorrect units of $\Omega\text{cm}^{-2}$ .
31 315 Shimizu, Y.		1932		293	0-100	Purest metals from Kahlbaum melted in silica tube in electric furnace; mixed and cast in rods 8 mm thick and 8 cm long in an iron mold the inner surface of which was coated with caolin; then melted and annealed in vacuo in silica tubes; to avoid iron contamination, surfaces treated with dilute hydrochloric or sulphuric acid, washed with water, and dried in absolute alcohol; drawn to wire 0.82 mm thick and again annealed before measurement; measurement temperature not stated explicitly, assumed to be 293 K; densities before and after expelling absorbed gases also measured; data extracted from table.
32 316 Linde, J.O.		1968	B		83-99	Components melted in sealed silica tubes, slightly cold-rolled, and homogenized by annealing for some hours at high temperatures; drawn into wires and measured in cold worked state as well as after annealing in the 673-873 K range; measured at 293 K; data reported as $\Delta\rho/c$ where $\Delta\rho$ is the difference between the resistivity values of the alloys and that of the pure base metal at 293 K and $c$ is the atomic concentration; the $\Delta\rho$ values are plotted in the figure; the dilute alloy limit of $\Delta\rho/c$ was reported as $0.36 \times 10^{-8} \Omega\text{m}$ per atomic percent Au in Ag; data read from figure.
33 316 Linde, J.O.		1968	B		1.3-17.5	Similar to the above specimens; the dilute alloy limit of $\Delta\rho/c$ was reported as $0.32 \times 10^{-8} \Omega\text{m}$ per atomic percent Ag in Au.
34 287 Rudnikski, A.A.		1946		293	0-100	Measurement temperature not stated explicitly, assumed to be 293 K; data extracted from table.
35 317 Strohhal, V. and Barus, C.		1883		273	14.9-89.1	Original paper not available; data as reported by Borelius, G., [Ann. d. Phys., 4, 77(2), 109-37, 1925]; data read from figure.
36 318 Clay, J.		1911		273	0.721-62.75	Reported compositions represent the total impurity of the sample and not necessarily the Ag content alone (for details, see data sets 68-74 of gold + silver alloys); alloys supplied by Dr. Holtsma, drawn by Heraeus to wires of 0.1 mm diam, annealed, and wound around glass cylinders; resistivities deduced from resistance measurements in the 15-170 K range and the resistivity at 273 K, calculated according to Matthiessen's observations; data extracted from table.
37 318 Clay, J.		1911		169.33	0.721-62.75	The above specimens measured at 169.33 K.
38 318 Clay, J.		1911		90.43	0.721-62.75	The above specimens measured at 90.43 K.



TABLE 55. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF GOLD-SILVER ALLOY SYSTEM (Composition Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp., K	Composition Range (At. % of Ag)	Composition (weight percent), Specifications, and Remarks
39	318	Clay, J.	1911		20.24	0.731-43.75	The above specimens measured at 20.24 K.
40	319	Auer, H., Riedl, E., and Seemann, H. J.	1934		293	0-100	Spectrographic analysis showed less than $10^{-4}$ Fe; Fe content primarily superficial rather than bulk and reduced to minimum by successive acid treatments; final shape of specimens cylindrical rods 4 mm in diam and 120 mm long; material initially cast in square forms 8 mm thick and then rolled to intermediate 5 mm thick stage of square cross section; rinsed with dilute hydrochloric acid, annealed at 1073 K and drawn to 4 mm; rinsed in dilute hydrochloric acid again and measured in this initial state; alloys designated as Series A; data extracted by author from table.
41	319	Auer, H., et al.	1934		293	0-100	The above specimens annealed at 1073 K for 4 h in vacuum to remove gas content and air cooled.
42	319	Auer, H., et al.	1934		293	0-100	The above specimens remelted in vacuum, reduced to 2.5 to 3.5 mm diam and annealed several hours in vacuum before measuring.
43	319	Auer, H., et al.	1934		293	0-100	Cylindrical specimens 4 mm in diam and 100 mm long and measured in initial state; alloys designated as series B by author; data extracted from table.
44	319	Auer, H., et al.	1934		293	20.7-90.3	The above specimens measured after annealed in vacuum at 1073 K.
45	320	Bridgman, P. W.	1933		303	25-75	Compositions determined by weighing component elements; initial material prepared by Mehl, R., Superintendent of Department of Metallurgy at Naval Research Laboratory by melting weighed amounts of components in vacuum and annealing at 1193 K for 40 h; forged into bars 3 mm in diam and about 2.5 cm long and annealed again; bars lengthened by forging an additional 15%, annealed at 1173 K for 30 min and turned in Jeweller's lathe to true cylinder; drawn out through steel dies, with frequent annealings in open gas flame, to final diam of 0.048 cm; final annealing at 1173 K in electric furnace, protected by quartz tubing; tabular data presentation; the average temperature coefficients of resistance at atmospheric pressure between 0 and 100 °C for the 25, 50, and 75 at. % Ag alloys were 0.001048, 0.000774, and 0.000888 ( $\Delta\Omega/\Omega \text{ K}^{-1}$ ) respectively.
46*	302	Iyer, V. K. and Asimow, R. M.	1967		300	0-100	Spectrographic analysis showed total impurities less than 0.1% with 0.05% Si; 99.998 pure Ag and Au melted together in graphite crucible in Ar, 0.125 in ingots swaged to 0.040 in diam wire and then drawn to 0.010 in diam; measured in vacuum, values are averages of measurements on two specimens for each composition; temperature accurate to $\pm 0.3\%$ , resistivity to 5% with error mainly due to uncertainty in length and density; corrected for thermal expansion; data read from figure.

\* Not shown in figure.

TABLE 54.  
EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF GOLD-SILVER ALLOY SYSTEM (Composition Dependence)

[Composition, C. at/o Ag; Electrical Resistivity,  $\rho$ ,  $10^{-8}$   $\Omega$ m]

C	$\rho$	DATA SET 1* (T = 14.6 K)	C	$\rho$	DATA SET 4 (cont.) (T = 293.2 K)	C	$\rho$	DATA SET 5 (cont.) (T = 1473 K)	C	$\rho$	DATA SET 8 (cont.) (T = 373.2 K)	C	$\rho$	DATA SET 10 (cont.) (T = 273.2 K)	C	$\rho$	DATA SET 13 (cont.) (T = 295 K)
0	32.0	10	4.98	34	38.2	54.6	11.8	98.35	2.126	17.5	6.911						
10	35.7	20	7.30	35	37.6	60.3	11.9	98.63	2.052	25.0	8.488						
20	37.1	40	10.64	40	37.4	69.8	10.3	99.21	1.768	50.0	11.147						
30	38.6	60	16.75	60	36.1	80.7	8.70	99.31	1.758	70.0	9.747						
40	39.4	80	23.7	80	34.6	85.5	7.41	99.81	1.532	90.0	7.723						
50	40.5	90	5.03*	90	34.3	90.3	6.25	99.0	1.532	90.0	5.135						
60	41.5	95	3.50	95	32.5	95.0	4.13	99.0	2.375	100	1.618						
70	42.7	100	2.49	100	32.9	100	2.57										
80	44.0	0	2.01	74	31.9	74	31.9	2.0	0.584								
90	45.4	10	1.61	75.5	31.6	75.5	31.6	5.0	1.401								
100	46.9	20	1.29	77	30.3	77	30.3	10.0	2.801*								
0	0	30	0.82	87.5	25.1	87.5	25.1	10.0	2.788	0.09	0.0320						
10	2.83	40	0.47	100	19.4	100	19.4	17.5	4.627	0.25	0.0605						
20	5.29	50	0.28	DATA SET 7 (T = 273.2 K)	1.9	0.670	25.0	6.239	0.09	0.292							
30	7.20	60	0.18	0	2.27	7.8	2.603	80.0	5.835								
40	8.47	70	0.12	4.9	3.98	21.0	6.038	90.0	3.360								
50	9.96	80	0.08	10.7	5.52	73.0	7.084	98.0	0.738								
60	11.6	90	0.05	18.7	7.58	90.0	3.303	100	0.000*								
70	13.5	100	0.03	29.7	9.80	94.3	1.923										
80	15.6	0	0	40.1	11.0	94.7	1.813										
90	17.9	10	35.8	44.9	11.2	97.6	0.768										
100	20.4	20	35.7	49.1	13.9	98.35	0.593										
0	0.45	30	32.7	54.6	11.0	98.63	0.493										
10	3.36	40	31.1	60.3	11.0	99.21	0.285										
20	5.81	50	31.5	69.8	9.71	99.31	0.249										
30	7.72	60	30.4	80.0	8.00	99.81	0.033										
40	9.95	70	30.2	85.5	6.80												
50	12.4	80	29.0	90.3	5.13												
60	15.1	90	24.2	95.0	3.41												
70	17.9	100	18.5	100	1.88												
80	20.8	0	33.2	DATA SET 8 (T = 373.2 K)	0.15	2.126											
90	23.8	10	36.3	0	3.12	0.37	2.209										
100	26.9	20	37.2	4.9	4.65	0.85	2.421										
0	2.23	30	38.0	10.7	6.29	2.4	2.991										
1	2.47	40	38.5	18.7	8.06	4.1	3.517										
2	2.71	50	39.0	29.7	10.4	7.6	4.685										
3	2.95	60	39.5	40.1	11.8	21.0	8.107										
4	3.19	70	40.0	49.1	13.9*	73.0	8.874										
5	3.43	80	40.5	49.1	13.9*	90.0	9.958										
6	3.67	90	41.0	49.1	13.9*	94.3	3.581										
7	3.91	100	41.5	49.1	13.9*	97.6	2.507										
8	4.15	0	44.0	49.1	13.9*												
9	4.39	10	44.5														
10	4.63	20	45.0														
11	4.87	30	45.5														
12	5.11	40	46.0														
13	5.35	50	46.5														
14	5.59	60	47.0														
15	5.83	70	47.5														
16	6.07	80	48.0														
17	6.31	90	48.5														
18	6.55	100	49.0														
19	6.79	0	49.5														
20	7.03	10	50.0														
21	7.27	20	50.5														
22	7.51	30	51.0														
23	7.75	40	51.5														
24	7.99	50	52.0														
25	8.23	60	52.5														
26	8.47	70	53.0														
27	8.71	80	53.5														
28	8.95	90	54.0														
29	9.19	100	54.5														
30	9.43	0	55.0														
31	9.67	10	55.5														
32	9.91	20	56.0														
33	10.15	30	56.5														
34	10.39	40	57.0														
35	10.63	50	57.5														
36	10.87	60	58.0														
37	11.11	70	58.5														
38	11.35	80	59.0														
39	11.59	90	59.5														
40	11.83	100	60.0														
41	12.07	0	60.5														
42	12.31	10	61.0														
43	12.55	20	61.5														
44	12.79	30	62.0														
45	13.03	40	62.5														
46	13.27	50	63.0														
47	13.51	60	63.5														
48	13.75	70	64.0														
49	13.99	80	64.5														
50	14.23	90	65.0														
51	14.47	100	65.5														
52	14.71	0	66.0														
53	14.95	10	66.5														
54	15.19	20	67.0														
55	15.43	30	67.5														
56	15.67	40	68.0														
57	15.91	50	68.5														
58	16.15	60	69.0														
59	16.39	70	69.5														
60	16.63	80	70.0														
61	16.87	90	70.5														
62	17.11	100	71.0														
63	17.35	0	71.5														
64	17.59	10	72.0														
65	17.83	20	72.5														
66	18.07	30	73.0														
67	18.31	40	73.5														
68	18.55	50	74.0														
69	18.79	60	74.5														
70	19.03	70	75.0														
71	19.27	80	75.5														
72	19.51	90	76.0														
73	19.75	100	76.5														
74	19.99	0	77.0														
75	20.23	10	77.5														
76	20.47	20	78.0														
77	20.71	30	78.5														
78	20.95	40	79.0														
79	21.19	50	79.5														
80	21.43	60	80.0														
81	21.67	70	80.5														
82	21.91	80	81.0														
83	22.15	90	81.5														
84	22.39	100	82.0														
85	22.63	0	82.5														
86	22.87	10	83.0														
87	23.11	20	83.5														
88	23.35	30	84.0														
89	23.59	40	84.5														
90	23.83	50	85.0														
91	24.07	60	85.5														
92	24.31	70	86.0														
93	24.55	80	86.5														
94	24.79	90	87.0														
95	25.03	100	87.5														
96	25.27	0	88.0														
97	25.51	10	88.5														
98	25.75	20	89.0														
99	25.99	30	89.5														
100	26.23	40	90.0														
101	26.47	50	90.5														
102	26.71	60	91.0														
103	26.95	70	91.5														
104	27.19	80	92.0														
105	27.43	90	92.5														
106	27.67	100	93.0														
107	27.91	0	93.5														
108	28.15	10	94.0														
109	28.39	20	94.5														
110	28.63	30	95.0														
111	28.87	40	95.5														
112	29.11	50	96.0														
113	29.35	60	96.5														
114	29.59	70	97.0														
115	29.83	80	97.5														
116	30.07	90	98.0														
117	30.31	100	98.5														
118	30.55	0	99.0														
119	30.79	10	99.5														
120	31.03	20	100.0														
121	31.27	30	100.5														
122	31.51	40	101.0														
123	31.75	50	101.5														
124	31.99	60	102.0														
125	32.23	70	102.5														
126	32.47	80	103.0														
127	32.71	90	103.5														

\* Not shown in figure.



TABLE 54. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF GOLD-SILVER ALLOY SYSTEM (Composition Dependence) (continued)

C	P
DATA SET 48 (cont.) <sup>a</sup>	
$(T = 300 \text{ K})$	
10.0	5.12
30.0	9.54
50.0	11.26
70.0	9.76
90.0	5.09
95.0	3.43
100.0	1.69

<sup>a</sup> Not shown in figure.

### 3.9. Iron-Nickel Alloy System

The electrical resistivity of the iron-nickel alloy system has received a considerable amount of attention, no doubt due in large measure to both elements being present in a large number of technologically important alloys. There are 45 data-source references available in the literature on the electrical resistivity of Fe-Ni alloys. From these references, 198 sets of experimental data were obtained spanning a temperature range 0.4 to 1995 K. These experimental data sets are listed in tables 58, 60, and 62 which provide information on specimen characterization and measurement conditions, tabulated in tables 59, 61, and 63, and shown partially in figures 47, 48, and 50.

The iron-nickel alloy system does not form a continuous series of solid solutions at moderate and low temperatures (below 1183 K). The maximum solid solubility of nickel in iron is 6.81 wt.% (6.5 at.%) at 618 K and the solubility decreases at higher and lower temperatures [321]. For nickel-rich alloys the solubility of iron in nickel is uncertain due to the formation of FeNi<sub>3</sub> ordered structure; it may be below 3 wt.% around room temperature. In addition, there is a martensitic transformation from metastable austenite in alloys containing up to 30 wt.% Ni resulting in a metastable  $\alpha_2$  phase. The phase diagram is further complicated by magnetic transitions: at about 1040 K in the  $\alpha$  phase, at about 673 K in the  $\alpha + \gamma$  equilibrium phase mixture, and on a curve reaching a maximum of about 885 K at about 66 wt.% Ni in the  $\gamma$  phase [264]. Finally, there is an order-disorder transformation due to the formation of FeNi<sub>3</sub>, covering a wide range of composition, from perhaps 50 to 85 wt.% Ni, which has a maximum transition temperature of about 776 K.

Regarding the metastable states in the iron-rich region, early work identified the transformation of metastable austenite as being of the diffusionless, martensitic type [322]. Later, effects of strain and grain size upon martensite formation were studied from changes in the electrical resistivity [323,324]. The results indicate that the transformation is essentially independent of heating/cooling rate, the transformation may occur over an interval of temperatures spanning perhaps 100 degrees for a 30% Ni alloy, and the amount of martensite formed upon cooling depends upon specimen history.

The electrical resistivity of dilute alloys with either iron or nickel as base metal has been measured during investigations of the deviations from

the Matthiessen's rule. At the lowest temperatures the deviations follow a roughly  $T^2$  behavior attributed to spin-mixing as reported by Fert and Campbell [325] (Fe + Ni data set 93) for 1% or 2% Ni in Fe and by Farrell and Greig [326] (Ni + Fe data sets 3-5) for up to 5% Fe in Ni. Further, the variation with composition is quite weak at these temperatures. Near room temperature the rapidly rising deviations have leveled off as observed by Farrell and Greig [326], Schwerer and Conroy [327] (Ni + Fe data sets 47-51), and Schwerer and Cuddy [328] (Fe + Ni data set 94). Above 300 K Schwerer and Cuddy [329] (Fe + Ni data sets 95, 96 and Ni + Fe data sets 64, 65) find evidence for spin-disorder resistivity in the generally decreasing temperature dependence of the deviations for both 1% Ni in Fe and 1% Fe in Ni at temperatures up to the Curie point.

Electrical resistivity of alloys with compositions in the  $\leq 30\%$  Ni region is well documented. The extensive measurements of Shirakawa [330] (Fe + Ni data sets 65-76) from 78 to 1123 K map out the temperature hysteresis loops for alloys with compositions between 4 and 30% Ni. It is unfortunate that the resistivity data reported on the lower Ni content specimens are atypically small in value, increasingly so at elevated temperatures; this is perhaps the result of problems inherent with a two-probe measurement technique. These findings were confirmed by the work of Ascher [331] (Fe + Ni data sets 90, 91) in which a 30% Ni specimen had been chilled in liquid air prior to measurement. The specimen was subsequently subjected to a complete temperature cycle with the resulting electrical resistivity in both martensitic and austenitic states being in close agreement with that of Shirakawa [330]. The change in electrical resistivity upon transforming the austenite to martensite is quite striking for this composition, amounting to a reduction of about 60%. A number of alloy compositions had previously been measured by Ingersoll [332] (Fe + Ni data sets 1-8) at these temperatures, but with only heating curves reported. In retrospect it may be noted that these smoothed data apparently do show the onset of the inverse transformation at higher temperatures. Recent work by Reed et al. [333] (Fe + Ni data sets 36-39) on an alloy with 29% Ni show the effects of a low-temperature anneal upon the electrical resistivity. The findings include the following: annealing of point defects affects only the residual resistivity such that the observed  $3.1 \times 10^{-8} \Omega\text{m}$  residual resistivity recovery corresponds to an estimated 0.5 at.% in point defect concentration, the reduction in electrical resistivity which accompanies the martensitic

transformation amounts to an average of  $0.46 \times 10^{-8} \Omega \text{m}$ /percent martensite at 293 K, and the effects of dislocations upon electrical resistivity is negligible at room temperature, with estimates being down at the  $0.1 \times 10^{-8} \Omega \text{m}$  level. Also are the findings of Livingston and Mukherjee [334] (Fe + Ni data set 42) on a slow-cooled specimen with 29.7 at.% Ni which show a very rapid onset to the decrease in resistivity at the transformation temperature.

The Invar composition range of about 35% to 50% Ni has received recent attention. The residual resistivity is found to increase sharply with increasing iron content. An explanation in terms of a weak ferromagnet model was suggested both by Armstrong and Fletcher [335] and by Gautier and Loegel [336]. Further, the temperature dependent resistivity is found to increase with temperature as  $T^n$ , where  $n$  is reported as  $n = 2$  below 60 K [336] and  $n = 1.8$  to 2.0 up to 120 K [335]. Generally, the residual resistivity data from [335] (Fe-Ni data set 34) along with those of Mikhailova [337] (Fe + Ni data set 48) and Clark et al. [103] (Fe + Ni data sets 31, 32, and Ni + Fe data set 32) are for commercial grade materials. Consequently, these data lie somewhat above those [336] (Fe + Ni data sets 50-56), of Larikov et al. [338] (Fe-Ni data set 27), and of Kondorskii and Sedov [339] (Fe-Ni data set 26). Near room temperature are the data of Jellinghaus and Andres [340] (Fe-Ni data set 30), Kalinin et al. [341] (Fe-Ni data set 25), and Window [342] (Fe-Ni data set 29), all of which are in reasonably good agreement. At higher temperatures are the data of Larikov et al. [338] (Fe + Ni data sets 80-83 and Ni + Fe data set 61) up to 1300 K and of Somura [343] (Fe + Ni data sets 84-89) up to about 800 K. Both of them reported a concentration dependence that varies rapidly, but diminishes toward the highest temperatures, and perhaps even disappears above 700 K [343]. Also in this connection are the weakly concentration-dependent data reported by Window [342] (Fe-Ni data set 28) at 873 K. If, as has been suggested, alloys in the Invar range are describable in terms of a mixed phase model (see, for instance ref. [344]), then specimen thermal history becomes very important. For instance, the cooling data reported by Larikov et al. [338] deviate toward slightly smaller resistivity values, but only below the Curie temperature. This may indicate that some phase segregation is taking place.

In the nickel-rich portion of the Fe-Ni system, the formation of ordering associated with  $\text{FeNi}_3$  has been intensively studied. Basically two techniques

have been reported to achieve the order. Following the initial high-temperature annealing a sequence of anneals below the critical temperature may be carried out. Or, a constant cooling rate may be applied. The first method was used in the early work of Källback [345] who reported electrical resistivity of the disordered (quenched) state (Ni + Fe data set 29) and also of states with partial, but undetermined amounts of order (Ni + Fe data set 28). Recently Moore et al. [346] (Ni + Fe data sets 39-42) subjected an alloy to a series of extended anneals designed to achieve essentially complete ordering. The electrical resistivity was reduced some 35% for temperatures between 4.2 K and 400 K by this procedure. The second method of producing order was reported by Wakelin and Yates [347] (Ni + Fe data sets 9-27) who applied cooling rates from 100 deg hr<sup>-1</sup> to 0.1 deg hr<sup>-1</sup> on alloys containing 50%-80% Ni. The essentially linear nature of the cooling curves at the slower cooling rates suggested that the fully ordered state was achieved. However, the cooling curves for the slower rates were relatively displaced, perhaps indicating that an extra resistivity mechanism was present. This was tentatively identified with dislocations being frozen in during the rapid cooldowns. Overall, this method resulted in alloy specimens having varying degrees of order. Consequently, some ambiguities arise when making comparisons of the electrical resistivity data on these alloy specimens with other data.

Data which fall between the two preceding composition ranges are those reported by Szentirmay [348] (Ni + Fe data sets 62, 63) on alloys with 58% and 65% Ni. The data are for quenched alloys at temperatures between 150 K and 350 K. The 58% Ni specimen was later subjected to an anneal following which the electrical resistivity was shown to be reduced by some 2%-5%.

The electrical resistivity of the molten alloys shows a linear temperature dependence as reported by Ono and Yagi [349] (Fe + Ni data sets 40, 41 and Ni + Fe data sets 34-36), by Baum et al. [350] (Fe + Ni data sets 57-64, Ni + Fe data sets 52-55, and Fe-Ni data set 24), and by Epin et al. [351] (Fe-Ni data set 33). The slope values reported in ref. [349,351] agree rather well for the nickel-rich compositions while those from ref. [350] are lower by a factor of ten. The ratio of resistivities, liquid to solid, for compositions throughout the alloy system was also reported in ref. [351]. An alloy with 75% Ni was reported by Ono and Yagi [349] to show an anomalously low resistivity. This was attributed to possible FeNi<sub>3</sub> clusters. Temperature hysteresis was



reported by Baum et al. [350] upon cycling through the melting range. This may be an indication of the tendency for structures in the Fe-Ni system to be retained into the melting range as suggested by Filippov and Kretonikov [352].

Much early work on the electrical resistivity of the iron-nickel system was devoted to mapping out the composition dependence. This information, usually obtained at or near room temperature, helped to form initial insights into the inner workings of this system. The early efforts of Burgess and Aston [353] (Fe-Ni data set 10) directed toward technical application were followed by the work of Honda [354] (Fe-Ni data sets 7, 8) showing the effects of cold-chilling on alloys having  $\leq 30\%$  Ni, by the combined composition and temperature dependence measurements of Ingersoll [332] (Fe-Ni data sets 14-22), and by the careful pressure dependence measurements of Bridgman [355] (Fe-Ni data set 12).

Each of the experimental areas discussed above formed the basis for a set of recommendations. These were subsequently merged smoothly together to form the recommendations for the entire Fe-Ni system.

Recommendations for the residual resistivity were based upon a number of low temperature data sets which generally included a value at 4.2 K. These are the data of Fert and Campbell [325], Schwerer and Cuddy [329], and Reed et al. [333] for low nickel compositions. Also based upon were the temperature-dependent data of Bäcklund [356] (Fe + Ni data sets 13, 14) and Soffer et al. [357] (Fe + Ni data sets 20-24), from which  $\rho_0$  was extracted. At higher nickel compositions the recommendations were based on the data of Cadeville and Loegel [358] (Fe-Ni data set 9), Larikov et al. [338], and Kondorskii and Sedov [339] for 35-60% Ni, of Van Elst and Gorter [220] (Ni + Fe data set 6) and Moore et al. [346] (disordered state only) for 75% Ni, of Berger and Rivier [359] (Ni + Fe data set 2) for 85% Ni, and of Farrell and Greig [326], Schwerer and Conroy [327], Fert and Campbell [325], and Schwerer and Cuddy [329] for  $\geq 95\%$  Ni.

At low temperatures, the temperature-dependent resistivity follows a  $T^2$  dependence. The scattering strength is quite strong in the Invar region, in which the recommendations follow the data of Gautier and Loegel [336], and diminishes with increasing base metal compositions, for which the recommendations follow the data of Soffer et al. [357], Fert and Campbell [325], Schwerer and Cuddy [328], Farrell and Greig [326], and Schwerer and Conroy [327].

At higher temperatures, a cross-correlation involving both temperature and composition was applied. The data used for this were those of Schwerer and Cuddy [328,329], Soffer et al. [357], Bäcklund [356], Reed et al. [333], Livingston and Mukherjee [334], Ascher [331], Shirakawa [330], Ingersoll [332], Larikov et al. [338], Szentirmay [348], Moore et al. [346], Farrell and Greig [326], and of Schwerer and Conroy [327]. Curie temperatures were taken from Hansen [264].

For the martensitic and austenitic metastable states, recommendations are given for compositions between 3 and 30% Ni. The transformation temperatures are the median, 50% transformation completion, values from Hansen [264]. Values applicable to a 30% Ni composition follow the data of Reed et al. [333] and Livingston and Mukherjee [334] up to 300 K. The temperature hysteresis loops for all compositions are based upon the data of Ascher [331] and of Shirakawa [330].

Recommendations at high temperatures include values for the liquid state. Solidus and liquidus temperatures are taken from Hansen [264]. Values for the solid state are based upon the data of Baum et al. [350], and for the step-up resistivity increase across the melting range upon the  $\rho_l/\rho_s$  data from Epin et al. [351]. The linear temperature dependence of the electrical resistivity in the molten region follows the slope values reported by Epin et al. [351], and for nickel-rich compositions also follows the slope values reported by Ono and Yagi [349].

It may be worthwhile to point out that the data from ref. [349-351] were measured with a rotating field method. Since the electrical resistivity in this case is proportional to specimen volume (see for instance ref. [349]), a significant thermal expansion correction arises, amounting to 7-8% for pure nickel or iron at the melting point. Therefore, inasmuch as ref. [349] reports expansion corrections having been applied to the data, it is thought that actually all three references, [349-351], are reporting corrected data. As a result, the recommended values for alloys below 1300 K are not corrected for thermal expansion, while those at higher temperatures, up to and beyond the melting range, do in fact include the corrections.

The resulting recommended electrical resistivity values for Fe, Ni, and for 25 Fe-Ni binary alloys are presented in table 57 and shown in figures 43-46, and 49. The recommended values for Fe and for Ni are for well-annealed

high-purity specimens, but those values for temperatures below about 100 K are applicable only to Fe and Ni having residual electrical resistivities as given at 1 K in table 57. The alloys for which the recommended values are generated are those having the stated compositions and being in the appropriate metallurgical state. The latter includes the fully martensitic or austenitic metastable states for 3% to 30% Ni and the completely disordered state for other compositions. The recommended values cover a full range of temperature from 1 K to 2000 K for both solid and molten states. The estimated uncertainties in the values for the various alloys and for different temperature ranges are explicitly stated in a footnote to table 57. Some of the values in table 57 are indicated as provisional because their uncertainties are greater than  $\pm 5\%$ .

TABLE 57. RECOMMENDED ELECTRICAL RESISTIVITY OF IRON-NICKEL ALLOY SYSTEM†

[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Fe: 100.00% (100.00 At.%) Ni: 0.00% (0.00 At.%)		Fe: 99.50% (99.52 At.%) Ni: 0.50% (0.48 At.%)		Fe: 99.00% (99.05 At.%) Ni: 1.00% (0.95 At.%)		Fe: 97.00% (97.14 At.%) Ni: 3.00% (2.86 At.%)		Fe: 95.00% (95.23 At.%) Ni: 5.00% (4.77 At.%)		Fe: 90.00% (90.44 At.%) Ni: 10.00% (9.56 At.%)	
T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho_M$	T	$\rho_M$	T	$\rho_M$
1	0.0225	1	0.95	1	1.90	1	5.60*	1	8.15*	1	12.4*
4	0.0227	4	0.95	4	1.90	4	5.60*	4	8.15*	4	12.4*
7	0.0231	7	0.95	7	1.90	7	5.60*	7	8.15*	7	12.4*
10	0.0238	10	0.95	10	1.91	10	5.61*	10	8.16*	10	12.4*
15	0.0257	15	0.96	15	1.91	15	5.62*	15	8.18*	15	12.4*
20	0.0287	20	0.96	20	1.92	20	5.64*	20	8.20*	20	12.4*
30	0.0421	30	0.98	30	1.95	30	5.69*	30	8.27*	30	12.5*
40	0.0758	40	1.03	40	2.00	40	5.76*	40	8.36*	40	12.6*
50	0.148	50	1.11	50	2.09	50	5.87*	50	8.49*	50	12.8*
60	0.271	60	1.23	60	2.23	60	6.04*	60	8.66*	60	13.0*
70	0.452	70	1.42	70	2.42	70	6.28*	70	8.91*	70	13.3*
80	0.693	80	1.67	80	2.67	80	6.54*	80	9.22*	80	13.7*
90	0.967	90	1.96	90	2.97	90	6.89	90	9.60*	90	14.1*
100	1.28	100	2.30	100	3.32	100	7.29	100	10.0	100	14.5
150	3.15	150	4.18	150	5.15	150	9.34	150	12.1	150	17.0
200	5.20	200	6.24	200	7.36	200	11.7	200	14.7	200	19.8
250	7.44	250	8.56	250	9.70	250	14.1	250	17.3	250	22.7
273	8.57	273	9.72	273	10.9	273	15.4	273	18.7	273	24.2
293	9.61	293	10.8	293	12.0	293	16.5	293	19.9	293	25.5
300	9.98	300	11.1	300	12.4	300	16.9	300	20.2	300	25.9
400	16.1	400	17.4	400	18.7	400	23.3	400	26.8	400	33.2
500	23.7	500	25.0	500	26.3	500	31.0	500	34.6	500	41.6
600	32.9	600	34.2	600	35.4	600	40.1	600	43.8	600	51.5
700	44.0	700	45.3	700	46.4	700	51.2	700	55.2	700	63.1
800	57.1	800	58.3	800	59.6	800	64.7	800	68.5	800	72.5
900	90.8	900	73.7	900	75.1	900	80.5	900	84.7	900	97.3
1043	101.1	1000	91.9	1000	93.1	995	97.4*	925	88.9	900	92.6
1185	111.9( $\alpha$ )	1040	101.3	1037	101.6	1000	98.3*	1000	102.3	965	106.2
1185	111.0( $\gamma$ )	1100	107.4	1100	107.8	1075	108.0*	1035	107.4	1000	107.8
1400	117.9	1200	111.8*	1200	112.0	1200	113.2	1200	114.1	1200	115.9*
1667	124.4( $\gamma$ )	1400	118.2*	1400	118.6	1400	119.9	1400	120.8	1400	122.7*
1667	124.6( $\delta$ )	1600	123.4*	1600	123.8	1600	125.2	1600	126.3	1600	128.6*
1811	126.2( $\delta$ )	1808	128.3*(s)	1803	128.4*(s)	1791	129.5*(s)	1782	130.7*(s)	1762	132.6*(s)
1811	135.2( $\delta$ )	1810	135.4*(s)	1808	135.6*(s)	1800	136.4*(s)	1793	137.4*(s)	1777	139.0*(s)
2000	138.2	2000	138.4*	2000	138.6*	2000	139.7*	2000	140.6	2000	142.4*

† Uncertainties in the electrical resistivity values are as follows:

100.00 Fe - 0.00 Ni:  $\pm 5\%$  from 50 K to 100 K,  $\pm 3\%$  above 100 K to 200 K,  $\pm 2\%$  above 200 K to 1811 K, and  $\pm 5\%$  above 1811 K.99.50 Fe - 0.50 Ni:  $\pm 5\%$  below 100 K,  $\pm 3\%$  from 100 K to 200 K,  $\pm 2\%$  above 200 K to 1808 K, and  $\pm 5\%$  above 1808 K.99.00 Fe - 1.00 Ni:  $\pm 5\%$  below 100 K,  $\pm 3\%$  from 100 K to 200 K,  $\pm 2\%$  above 200 K to 1803 K, and  $\pm 5\%$  above 1803 K.97.00 Fe - 3.00 Ni:  $\pm 5\%$  below 1075 K (martensite),  $\pm 3\%$  from 995 K to 1791 K (austenite), and  $\pm 5\%$  above 1791 K.95.00 Fe - 5.00 Ni:  $\pm 7\%$  below 100 K,  $\pm 5\%$  from 100 K to 1035 K (martensite),  $\pm 3\%$  from 925 K to 1782 K (austenite), and  $\pm 5\%$  above 1782 K.90.00 Fe - 10.00 Ni:  $\pm 7\%$  below 100 K,  $\pm 5\%$  from 100 K to 965 K (martensite),  $\pm 3\%$  from 770 K to 1763 K (austenite), and  $\pm 5\%$  above 1763 K. $\rho_M$ ,  $\rho_A$  Electrical resistivity for martensite, austenite respectively.

\* Provisional value.

\* In temperature range where no experimental data are available.

TABLE 87. RECOMMENDED ELECTRICAL RESISTIVITY OF IRON-NICKEL ALLOY SYSTEM† (continued)

[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Fe: 85.00% (85.63 At.%) Ni: 15.00% (14.37 At.%)			Fe: 80.00% (80.79 At.%) Ni: 20.00% (19.21 At.%)			Fe: 75.00% (75.93 At.%) Ni: 25.00% (24.07 At.%)			Fe: 70.00% (71.04 At.%) Ni: 30.00% (28.96 At.%)			Fe: 65.00% (66.13 At.%) Ni: 35.00% (33.87 At.%)			Fe: 60.00% (61.19 At.%) Ni: 40.00% (38.81 At.%)		
T	$\rho_M$	$\rho_A$	T	$\rho_M$	$\rho_A$	T	$\rho_M$	$\rho_A$	T	$\rho_M$	$\rho_A$	T	$\rho$	T	$\rho$	T	$\rho$
1	15.1 <sup>±</sup>		1	16.8 <sup>±</sup>		1	17.8 <sup>±</sup>		1	18.2 <sup>±</sup>		1	47.9 <sup>±</sup>	1	22.9 <sup>±</sup>		
4	15.1 <sup>±</sup>		4	16.8 <sup>±</sup>		4	17.8		4	18.2		4	47.9 <sup>±</sup>	4	22.9 <sup>±</sup>		
7	15.1 <sup>±</sup>		7	16.8 <sup>±</sup>		7	17.8		7	18.2		7	47.9 <sup>±</sup>	7	22.9 <sup>±</sup>		
10	15.1 <sup>±</sup>		10	16.8 <sup>±</sup>		10	17.8		10	18.2		10	48.0 <sup>±</sup>	10	23.0 <sup>±</sup>		
15	15.1 <sup>±</sup>		15	16.8 <sup>±</sup>		15	17.8		15	18.3		15	48.1 <sup>±</sup>	15	23.0 <sup>±</sup>		
20	15.1 <sup>±</sup>		20	16.8 <sup>±</sup>		20	17.9		20	18.3		20	48.2 <sup>±</sup>	20	23.2 <sup>±</sup>		
30	15.2 <sup>±</sup>		30	17.0 <sup>±</sup>		30	18.0		30	18.4		30	48.6 <sup>±</sup>	30	23.5 <sup>±</sup>		
40	15.4 <sup>±</sup>		40	17.1 <sup>±</sup>		40	18.2		40	18.6		40	49.1 <sup>±</sup>	40	23.9 <sup>±</sup>		
50	15.6 <sup>±</sup>		50	17.3 <sup>±</sup>		50	18.4		50	18.9		50	49.7 <sup>±</sup>	50	24.3 <sup>±</sup>		
60	15.8 <sup>±</sup>		60	17.6 <sup>±</sup>		60	18.7		60	19.2		60	50.5 <sup>±</sup>	60	25.2 <sup>±</sup>		
70	16.2 <sup>±</sup>		70	18.0 <sup>±</sup>		70	19.1		70	19.6		70	51.4 <sup>±</sup>	70	25.9 <sup>±</sup>		
80	16.6 <sup>±</sup>		80	18.4 <sup>±</sup>		80	19.5		80	20.0		80	52.5 <sup>±</sup>	80	26.1 <sup>±</sup>		
90	17.0 <sup>±</sup>		90	18.9 <sup>±</sup>		90	19.9		90	20.4		90	53.5 <sup>±</sup>	90	27.7 <sup>±</sup>		
100	17.5		100	19.3		100	20.4		100	20.9		100	54.7 <sup>±</sup>	100	28.6 <sup>±</sup>		
150	20.0		150	22.1		150	23.3		150	23.7		150	61.7 <sup>±</sup>	150	34.5 <sup>±</sup>		
200	23.0		200	25.2		200	26.4		200	27.0		200	68.9 <sup>±</sup>	200	41.8		
250	26.2		250	28.5		250	29.9		250	30.6	76.8	250	76.2 <sup>±</sup>	250	50.0		
273	27.8		273	30.1		273	31.6		273	32.3	78.9	273	79.4 <sup>±</sup>	273	53.8		
293	29.2		293	31.6		293	33.0		293	33.9	80.9	293	82.0 <sup>±</sup>	293	57.1		
300	29.7		300	32.2		300	33.5		300	34.4	81.4	300	83.0	300	58.2		
400	37.3		400	40.0		395	41.1	85.4	400	42.4	90.5	400	94.3	400	73.9		
500	46.2		500	49.0		400	41.5	85.9	500	51.8	97.5	482	101.6	500	88.5		
600	56.7		535	52.6	90.6	500	50.8	93.2	600	62.8	102.7	500	102.4	600	101.1		
635	60.9	92.0	600	59.8	94.6	600	61.7	98.9	700	75.2	106.7	600	106.4	604	101.4		
700	68.6	95.8	700	72.0	99.8	700	73.9	103.4	740	80.4	106.1	700	109.9	700	105.6		
800	83.0	100.9	800	86.2	104.2	800	87.9	107.1	800	110.1	110.1	800	113.2	800	109.1		
900	97.0	105.6	875	98.2	107.2	810	89.4	107.4	900	113.5	113.5	900	116.2	900	112.3		
929	102.9	106.4	900	106.1	108.1	900	110.4	110.4	1000	116.5	116.5	1000	119.1	1000	115.5		
1000	109.7		1000	111.5	109.7	1000	113.7	113.7	1100	119.4	119.4	1100	121.8	1100	118.3		
1200	116.9 <sup>*</sup>		1200	118.3 <sup>*</sup>	118.3 <sup>*</sup>	1200	118.9 <sup>*</sup>	118.9 <sup>*</sup>	1200	122.1 <sup>*</sup>	122.1 <sup>*</sup>	1200	124.3	1200	121.1		
1400	123.5 <sup>*</sup>		1400	124.7 <sup>*</sup>	124.7 <sup>*</sup>	1400	125.7 <sup>*</sup>	125.7 <sup>*</sup>	1400	127.1 <sup>*</sup>	127.1 <sup>*</sup>	1400	128.5 <sup>*</sup>	1400	126.0 <sup>*</sup>		
1600	129.7 <sup>*</sup>		1600	130.5 <sup>*</sup>	130.5 <sup>*</sup>	1600	131.0 <sup>*</sup>	131.0 <sup>*</sup>	1600	131.5	131.5	1600	132.0 <sup>*</sup>	1600	130.2 <sup>*</sup>		
1748	133.8 <sup>*</sup> (s)		1737	134.2 <sup>*</sup> (s)	134.2 <sup>*</sup> (s)	1728	134.2 <sup>*</sup> (s)	134.2 <sup>*</sup> (s)	1720	134.0(s)	134.0(s)	1715	133.7 <sup>*</sup> (s)	1710	132.4 <sup>*</sup> (s)		
1763	140.1 (L)		1751	140.6 (L)	140.6 (L)	1741	140.8 <sup>*</sup> (L)	140.8 <sup>*</sup> (L)	1732	141.0(L)	141.0(L)	1724	141.3 <sup>*</sup> (L)	1718	140.6 (L)		
2000	143.7		2000	144.3	144.3	2000	144.8 <sup>*</sup>	144.8 <sup>*</sup>	2000	144.9	144.9	2000	145.0 <sup>*</sup>	2000	144.9		

† Uncertainties in the electrical resistivity are as follows:

- 85.00 Fe - 15.00 Ni:  $\pm 7\%$  below 100 K,  $\pm 5\%$  from 100 K to 920 K (martensite),  $\pm 3\%$  from 920 K to 1748 K (austenite), and  $\pm 5\%$  above 1748 K.  
 80.00 Fe - 20.00 Ni:  $\pm 7\%$  below 100 K,  $\pm 5\%$  from 100 K to 875 K (martensite),  $\pm 3\%$  from 875 K to 1737 K (austenite), and  $\pm 5\%$  above 1737 K.  
 75.00 Fe - 25.00 Ni:  $\pm 5\%$  below 810 K (martensite),  $\pm 3\%$  from 810 K to 1728 K (austenite), and  $\pm 5\%$  above 1728 K.  
 70.00 Fe - 30.00 Ni:  $\pm 3\%$  below 740 K (martensite),  $\pm 3\%$  from 740 K to 1720 K (austenite), and  $\pm 5\%$  above 1720 K.  
 65.00 Fe - 35.00 Ni:  $\pm 15\%$  below 150 K,  $\pm 10\%$  above 150 K to 300 K, and  $\pm 5\%$  above 300 K.  
 60.00 Fe - 40.00 Ni:  $\pm 10\%$  below 200 K,  $\pm 5\%$  from 200 K to 604 K,  $\pm 3\%$  above 604 K to 1710 K, and  $\pm 5\%$  above 1710 K.

 $\rho_M$ ,  $\rho_A$  Electrical resistivity for martensite, austenite respectively.

\* Provisional value.

\* In temperature range where no experimental data are available.

TABLE 57. RECOMMENDED ELECTRICAL RESISTIVITY OF IRON-NICKEL ALLOY SYSTEM† (continued)  
[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Fe: 55.00% (56.23 At.%) Ni: 45.00% (43.77 At.%)			Fe: 50.00% (51.25 At.%) Ni: 50.00% (48.75 At.%)			Fe: 45.00% (46.24 At.%) Ni: 55.00% (53.76 At.%)			Fe: 40.00% (41.21 At.%) Ni: 60.00% (58.79 At.%)			Fe: 35.00% (36.15 At.%) Ni: 65.00% (63.85 At.%)			Fe: 30.00% (31.06 At.%) Ni: 70.00% (68.94 At.%)		
T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
1	13.1 <sup>†</sup>	1	8.93 <sup>†</sup>	1	7.00 <sup>†</sup>	1	5.74 <sup>†</sup>	1	4.93 <sup>†</sup>	1	4.93 <sup>†</sup>	1	4.46 <sup>†</sup>				
4	13.1 <sup>†</sup>	4	8.94 <sup>†</sup>	4	7.00 <sup>†</sup>	4	5.74 <sup>†</sup>	4	4.93 <sup>†</sup>	4	4.93 <sup>†</sup>	4	4.46 <sup>†</sup>				
7	13.1 <sup>†</sup>	7	8.95 <sup>†</sup>	7	7.01 <sup>†</sup>	7	5.75 <sup>†</sup>	7	4.94 <sup>†</sup>	7	4.94 <sup>†</sup>	7	4.47 <sup>†</sup>				
10	13.1 <sup>†</sup>	10	8.96 <sup>†</sup>	10	7.03 <sup>†</sup>	10	5.76 <sup>†</sup>	10	4.95 <sup>†</sup>	10	4.95 <sup>†</sup>	10	4.47 <sup>†</sup>				
15	13.2 <sup>†</sup>	15	9.00 <sup>†</sup>	15	7.06 <sup>†</sup>	15	5.78 <sup>†</sup>	15	4.97 <sup>†</sup>	15	4.97 <sup>†</sup>	15	4.49 <sup>†</sup>				
20	13.3 <sup>†</sup>	20	9.06 <sup>†</sup>	20	7.10 <sup>†</sup>	20	5.82 <sup>†</sup>	20	4.99 <sup>†</sup>	20	4.99 <sup>†</sup>	20	4.52 <sup>†</sup>				
30	13.5 <sup>†</sup>	30	9.23 <sup>†</sup>	30	7.23 <sup>†</sup>	30	5.91 <sup>†</sup>	30	5.07 <sup>†</sup>	30	5.07 <sup>†</sup>	30	4.59 <sup>†</sup>				
40	13.9 <sup>†</sup>	40	9.46 <sup>†</sup>	40	7.40 <sup>†</sup>	40	6.05 <sup>†</sup>	40	5.19 <sup>†</sup>	40	5.19 <sup>†</sup>	40	4.69 <sup>†</sup>				
50	14.3 <sup>†</sup>	50	9.76 <sup>†</sup>	50	7.63 <sup>†</sup>	50	6.22 <sup>†</sup>	50	5.34 <sup>†</sup>	50	5.34 <sup>†</sup>	50	4.83 <sup>†</sup>				
60	14.8 <sup>†</sup>	60	10.1 <sup>†</sup>	60	7.90 <sup>†</sup>	60	6.43 <sup>†</sup>	60	5.52 <sup>†</sup>	60	5.52 <sup>†</sup>	60	5.00 <sup>†</sup>				
70	15.4 <sup>†</sup>	70	10.6 <sup>†</sup>	70	8.25 <sup>†</sup>	70	6.69 <sup>†</sup>	70	5.73 <sup>†</sup>	70	5.73 <sup>†</sup>	70	5.20 <sup>†</sup>				
80	16.1 <sup>†</sup>	80	11.0 <sup>†</sup>	80	8.65 <sup>†</sup>	80	6.99 <sup>†</sup>	80	5.98 <sup>†</sup>	80	5.98 <sup>†</sup>	80	5.43 <sup>†</sup>				
90	16.8 <sup>†</sup>	90	11.6 <sup>†</sup>	90	9.11 <sup>†</sup>	90	7.35 <sup>†</sup>	90	6.28 <sup>†</sup>	90	6.28 <sup>†</sup>	90	5.69 <sup>†</sup>				
100	17.6 <sup>†</sup>	100	12.3 <sup>†</sup>	100	9.63 <sup>†</sup>	100	7.73 <sup>†</sup>	100	6.55 <sup>†</sup>	100	6.55 <sup>†</sup>	100	5.97 <sup>†</sup>				
150	22.1 <sup>†</sup>	150	16.2 <sup>†</sup>	150	12.6 <sup>†</sup>	150	10.2 <sup>†</sup>	150	8.49 <sup>†</sup>	150	8.49 <sup>†</sup>	150	7.71 <sup>†</sup>				
200	27.6 <sup>†</sup>	200	20.8 <sup>†</sup>	200	16.3 <sup>†</sup>	200	13.3 <sup>†</sup>	200	11.3 <sup>†</sup>	200	11.3 <sup>†</sup>	200	10.2 <sup>†</sup>				
250	33.7 <sup>†</sup>	250	25.9 <sup>†</sup>	250	21.0 <sup>†</sup>	250	17.4 <sup>†</sup>	250	14.9 <sup>†</sup>	250	14.9 <sup>†</sup>	250	13.5 <sup>†</sup>				
273	36.6 <sup>†</sup>	273	28.4 <sup>†</sup>	273	23.4 <sup>†</sup>	273	19.6 <sup>†</sup>	273	17.0 <sup>†</sup>	273	17.0 <sup>†</sup>	273	15.3 <sup>†</sup>				
293	39.2 <sup>†</sup>	293	30.6 <sup>†</sup>	293	25.6 <sup>†</sup>	293	21.6 <sup>†</sup>	293	18.8 <sup>†</sup>	293	18.8 <sup>†</sup>	293	17.1 <sup>†</sup>				
300	40.2	300	31.4	300	26.4 <sup>†</sup>	300	22.5	300	19.6	300	19.6	300	17.7				
400	53.9	400	43.7	400	38.3 <sup>†</sup>	400	34.0	400	30.4 <sup>†</sup>	400	30.4 <sup>†</sup>	400	27.4				
500	68.9	500	57.5	500	51.4 <sup>†</sup>	500	46.6	500	42.3 <sup>†</sup>	500	42.3 <sup>†</sup>	500	38.5				
600	84.7	600	72.6	600	65.4 <sup>†</sup>	600	60.2	600	55.6 <sup>†</sup>	600	55.6 <sup>†</sup>	600	50.7				
700	101.0	700	88.4	700	80.2 <sup>†</sup>	700	74.7	700	69.6 <sup>†</sup>	700	69.6 <sup>†</sup>	700	64.1				
703	101.2	703	101.0	703	80.2 <sup>†</sup>	703	74.7	703	69.6 <sup>†</sup>	703	69.6 <sup>†</sup>	703	64.1				
800	104.7	800	101.8	800	100.8 <sup>†</sup>	800	90.1	800	84.7 <sup>†</sup>	800	84.7 <sup>†</sup>	800	79.1				
900	109.1	900	105.8	900	103.9 <sup>†</sup>	900	100.0	900	96.2 <sup>†</sup>	900	96.2 <sup>†</sup>	900	91.2				
1000	111.3	1000	109.2	1000	107.6 <sup>†</sup>	1000	102.0	1000	98.1 <sup>†</sup>	1000	98.1 <sup>†</sup>	1000	92.5				
1100	114.3	1100	112.3	1100	110.9 <sup>†</sup>	1100	105.6	1100	102.1 <sup>†</sup>	1100	102.1 <sup>†</sup>	1100	96.7				
1200	117.2	1200	115.4	1200	114.1 <sup>†</sup>	1200	109.0	1200	105.9 <sup>†</sup>	1200	105.9 <sup>†</sup>	1200	100.8				
1400	122.7 <sup>†</sup>	1400	121.1 <sup>†</sup>	1400	120.1 <sup>†</sup>	1400	112.3 <sup>†</sup>	1400	109.9 <sup>†</sup>	1400	109.9 <sup>†</sup>	1400	104.7 <sup>†</sup>				
1600	127.6	1600	126.3 <sup>†</sup>	1600	125.7 <sup>†</sup>	1600	118.7 <sup>†</sup>	1600	116.4 <sup>†</sup>	1600	116.4 <sup>†</sup>	1600	112.5 <sup>†</sup>				
1706	130.0(s)	1706	128.7 <sup>†</sup> (s)	1706	128.4 <sup>†</sup> (s)	1706	124.9 <sup>†</sup>	1706	123.0 <sup>†</sup>	1706	123.0 <sup>†</sup>	1706	120.1 <sup>†</sup>				
1713	138.8(L)	1708	138.4 <sup>†</sup> (L)	1708	138.4 <sup>†</sup> (L)	1708	127.9 <sup>†</sup> (s)	1708	126.4 <sup>†</sup> (s)	1708	126.4 <sup>†</sup> (s)	1708	123.8 <sup>†</sup> (s)				
2000	144.7	2000	144.5 <sup>†</sup>	2000	144.2 <sup>†</sup>	2000	143.6 <sup>†</sup>	2000	143.3 <sup>†</sup>	2000	143.3 <sup>†</sup>	2000	143.3 <sup>†</sup>				

† Uncertainties in the electrical resistivity are as follows:

55.00 Fe - 45.00 Ni:  $\pm 7\%$  below 300 K,  $\pm 5\%$  from 300 K to 703 K,  $\pm 3\%$  above 703 K to 1706 K, and  $\pm 5\%$  above 1706 K.  
 50.00 Fe - 50.00 Ni:  $\pm 7\%$  below 300 K,  $\pm 5\%$  from 300 K to 782 K,  $\pm 3\%$  above 782 K to 1704 K, and  $\pm 5\%$  above 1704 K.  
 45.00 Fe - 55.00 Ni:  $\pm 7\%$  below 300 K,  $\pm 5\%$  from 300 K to 833 K,  $\pm 3\%$  above 833 K to 1703 K, and  $\pm 5\%$  above 1703 K.  
 40.00 Fe - 60.00 Ni:  $\pm 7\%$  below 300 K,  $\pm 5\%$  from 300 K to 859 K,  $\pm 3\%$  above 859 K to 1702 K, and  $\pm 10\%$  above 1702 K.  
 35.00 Fe - 65.00 Ni:  $\pm 7\%$  below 300 K,  $\pm 5\%$  from 300 K to 871 K,  $\pm 3\%$  above 871 K to 1702 K, and  $\pm 10\%$  above 1702 K.  
 30.00 Fe - 70.00 Ni:  $\pm 5\%$  below 1702 K and  $\pm 10\%$  above.

<sup>†</sup> Provisional value.

\* In temperature range where no experimental data are available.

TABLE 57. RECOMMENDED ELECTRICAL RESISTIVITY OF IRON-NICKEL ALLOY SYSTEM† (continued)  
[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-3} \Omega \text{ m}$ ]

Fe: 25.00% (25.95 At.%) Ni: 75.00% (74.05 At.%)	Fe: 20.00% (20.81 At.%) Ni: 80.00% (79.19 At.%)	Fe: 15.00% (15.65 At.%) Ni: 85.00% (84.35 At.%)	Fe: 10.00% (10.46 At.%) Ni: 90.00% (89.54 At.%)	Fe: 5.00% (5.24 At.%) Ni: 95.00% (94.76 At.%)	Fe: 3.00% (3.15 At.%) Ni: 97.00% (96.85 At.%)
T	T	T	T	T	T
1 4 7 10 15 20 30 40 50 60 70 80 90 100 150 200 250 273 293 300 400 500 600 700 800 865 900 1000 1100 1200 1400 1500 1703 1706 2000	1 4 7 10 15 20 30 40 50 60 70 80 90 100 150 200 250 273 293 300 400 500 600 700 800 833 900 1000 1100 1200 1400 1600 1705 1710 2000	1 4 7 10 15 20 30 40 50 60 70 80 90 100 150 200 250 273 293 300 400 500 600 700 786 800 833 900 1000 1100 1200 1400 1600 1707 1715 2000	1 4 7 10 15 20 30 40 50 60 70 80 90 100 150 200 250 273 293 300 400 500 600 700 733 800 900 1000 1100 1200 1400 1600 1719 2000	1 4 7 10 15 20 30 40 50 60 70 80 90 100 150 200 250 273 293 300 400 500 600 656 700 800 900 1000 1100 1200 1400 1600 1720 1726 2000	1 4 7 10 15 20 30 40 50 60 70 80 90 100 150 200 250 273 293 300 400 500 600 656 700 800 900 1000 1100 1200 1400 1600 1720 1726 2000
4.26* 4.26 4.27 4.27 4.29 4.31 4.39 4.49 4.61 4.77 4.94 5.14 5.37 5.62 7.30 9.71 12.8 14.3 15.9 16.4 25.1 35.1* 46.4* 59.0 73.2 83.4 85.3* 89.9* 94.4* 98.6*	4.08* 4.08 4.09 4.09 4.11 4.13 4.20 4.29 4.42 4.56 4.73 4.92 5.14 5.39 7.05 9.27 12.1 13.6 14.9 15.4 23.1 32.1 42.3 54.3 68.7 74.1 77.6 82.3 86.7 91.1* 99.6* 107.8* 112.0*(g) 121.5*(t) 135.0*	3.71* 3.71* 3.72* 3.72* 3.74* 3.76* 3.83* 3.92* 4.03* 4.17* 4.33* 4.52 4.64 4.97 6.39 8.64 11.2 12.6 13.8 14.2 21.1 29.1 38.6 50.5 63.6 64.5 69.1 73.5 77.8 82.0* 90.2* 98.3 102.5(g) 124.3*(t) 127.6*	2.95* 2.95* 2.96* 2.96* 2.98* 3.00* 3.06* 3.15* 3.27* 3.40* 3.56* 3.74 3.95 4.20 5.79 7.77 10.3 11.4 12.5 12.9 18.9 26.3 35.4 47.1 51.7* 55.1* 59.2* 63.3* 67.3 71.3* 79.1* 86.6* 90.6*(g) 114.1*(t) 117.3* 1600 1710 1719 2000	1.73 1.73 1.74 1.74 1.76 1.78 1.84 1.94 2.06 2.27 2.50 2.76 3.04 3.34 4.93 6.66 8.65 9.66 10.6 10.9 16.1* 22.9* 31.6* 40.6* 41.8* 45.6* 49.2* 52.6* 55.9* 59.3* 65.9* 72.3* 76.0*(g) 98.7*(t) 102.8* 1600 1716 1724 2000	1.10 1.10 1.11 1.11 1.13 1.15 1.21 1.30 1.44 1.61 1.62 2.05 2.31 2.59 4.09 5.78 7.64 8.53 9.32 9.59 14.6* 20.9* 29.6* 35.8* 38.1* 41.6* 45.0* 48.1* 51.3* 54.3* 60.3* 66.1* 69.7*(g) 93.0*(t) 96.0* 1600 1720 1726 2000

† Uncertainties in the electrical resistivity are as follows:

25.00 Fe - 75.00 Ni:  $\pm 5\%$  below 100 K,  $\pm 3\%$  from 100 K to 400 K,  $\pm 5\%$  above 400 K to 1703 K, and  $\pm 10\%$  above 1703 K.

20.00 Fe - 80.00 Ni:  $\pm 5\%$  below 1705 K and  $\pm 10\%$  above.

15.00 Fe - 85.00 Ni:  $\pm 5\%$  below 1707 K and  $\pm 10\%$  above.

10.00 Fe - 90.00 Ni:  $\pm 5\%$  below 733 K,  $\pm 7\%$  from 733 K to 1710 K, and  $\pm 10\%$  above 1710 K.

5.00 Fe - 95.00 Ni:  $\pm 5\%$  below 679 K,  $\pm 7\%$  from 679 K to 1716 K, and  $\pm 10\%$  above 1716 K.

3.00 Fe - 97.00 Ni:  $\pm 5\%$  below 656 K,  $\pm 7\%$  from 656 K to 1720 K, and  $\pm 10\%$  above 1720 K.

\* Provisional value.

† In temperature range where no experimental data are available.

TABLE 57. RECOMMENDED ELECTRICAL RESISTIVITY OF IRON-NICKEL ALLOY SYSTEM† (continued)

[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \cdot \text{cm}$ ]

Fe: 1.00% (1.05 AL%) Ni: 99.00% (98.95 AL%)		Fe: 0.50% (0.53 AL%) Ni: 99.50% (99.47 AL%)		Fe: 0.00% (0.00 AL%) Ni: 100.00% (100.00 AL%)			
T	$\rho$	T	$\rho$	T	$\rho$		
1	0.40*	1	0.21*	1	0.0032		
4	0.40	4	0.21	4	0.0036		
7	0.41	7	0.22	7	0.0044		
10	0.41	10	0.22	10	0.0057		
15	0.42	15	0.23	15	0.0090		
20	0.44	20	0.24	20	0.0140		
30	0.50	30	0.27	30	0.0317		
40	0.58	40	0.33	40	0.0678		
50	0.68	50	0.42	50	0.135		
60	0.83	60	0.55	60	0.242		
70	1.01	70	0.71	70	0.377		
80	1.21	80	0.90	80	0.545		
90	1.43	90	1.10	90	0.741		
100	1.66	100	1.32	100	0.959		
150	2.99	150	2.60	150	2.21		
200	4.57	200	4.08	200	3.67		
250	6.31	250	5.78	250	5.32		
273	7.17	273	6.65	273	6.16		
293	7.94	293	7.43	293	6.93		
300	8.12	300	7.72	300	7.20		
400	12.8	400	12.3	400	11.8		
500	19.0	500	18.5	500	17.7		
600	27.4	600	26.7	600	25.5		
637	31.2	634	30.4	630	28.7		
700	34.1	700	33.2	700	32.1		
800	37.5	800	36.5	800	35.5		
900	40.7	900	39.7	900	38.6		
1000	43.7	1000	42.5	1000	41.4		
1100	46.5*	1100	45.3*	1100	44.1		
1200	49.2*	1200	48.0*	1200	46.6		
1400	54.7*	1400	53.1*	1400	51.7		
1600	60.1*	1600	58.5*	1600	56.9		
1725	63.3*(s)	1727	61.6*(s)	1728	60.2*(s)		
1727	65.8*(t)	1728	64.0*(t)	1728	62.2*(t)		
2000	88.8*	2000	87.0*	2000	85.2*		

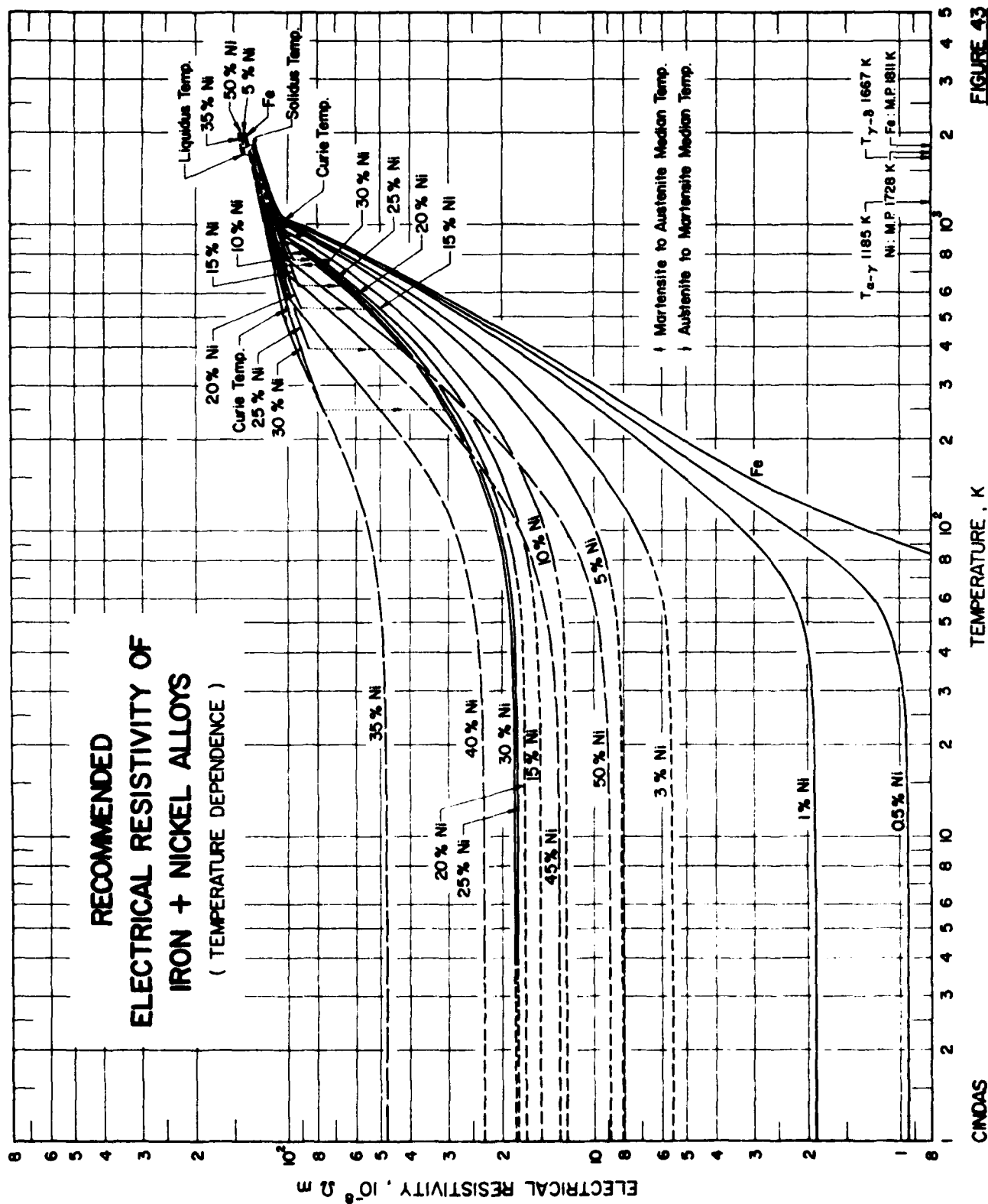
† Uncertainties in the electrical resistivity are as follows:

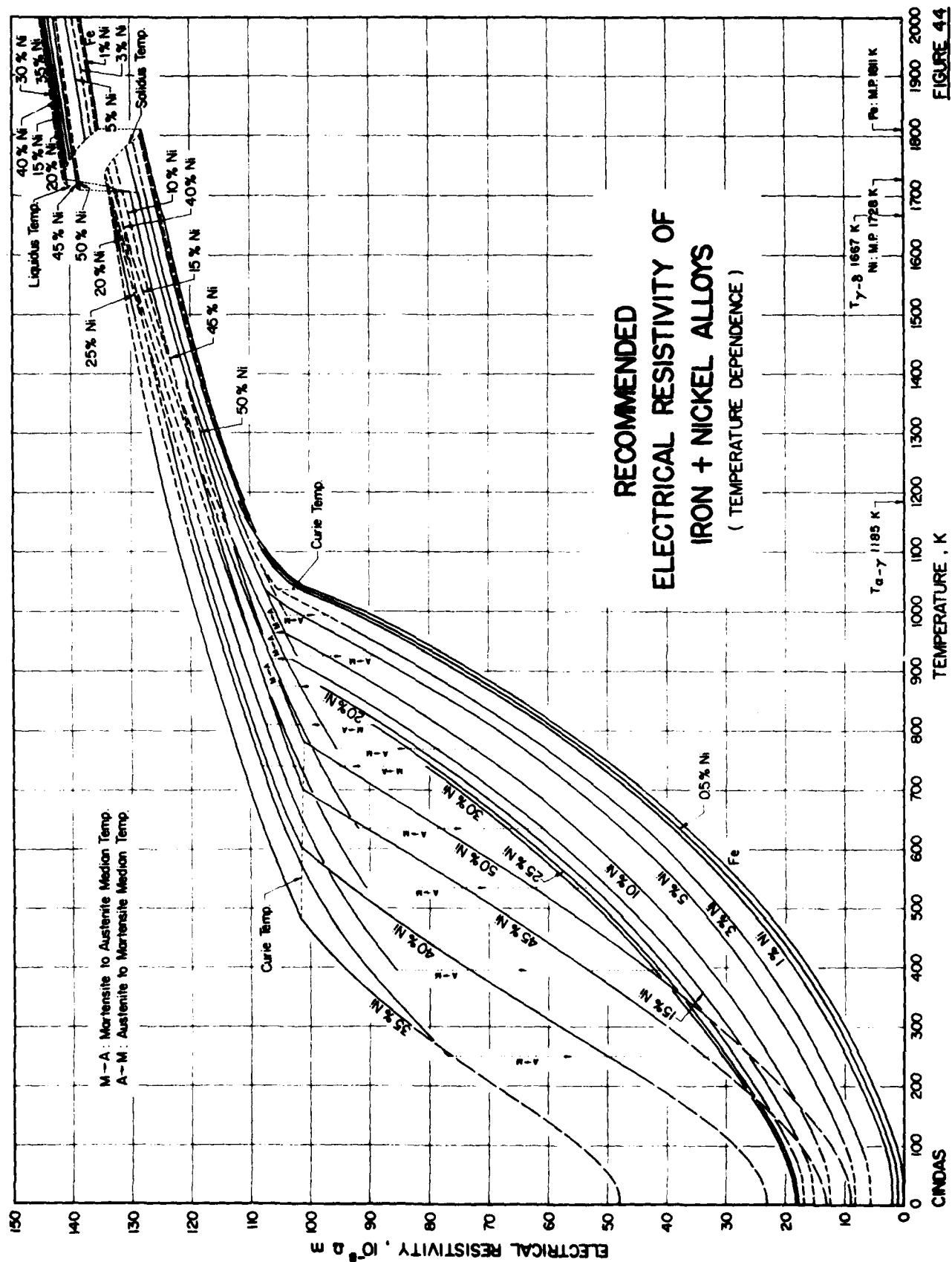
1.00 Fe - 99.00 Ni:  $\pm 5\%$  below 150 K,  $\pm 3\%$  from 150 K to 1000 K, and  $\pm 10\%$  above 1725 K.0.50 Fe - 99.50 Ni:  $\pm 5\%$  below 150 K,  $\pm 3\%$  from 150 K to 1000 K,  $\pm 5\%$  above 1000 K to 1727 K, and  $\pm 10\%$  above 1727 K.0.00 Fe - 100.00 Ni:  $\pm 5\%$  from 50 K to 150 K,  $\pm 5\%$  above 150 K to 1300 K,  $\pm 5\%$  above 1300 K to 1728 K, and  $\pm 10\%$  above 1728 K.

\* Provisional value.

\* In temperature range where no experimental data are available.







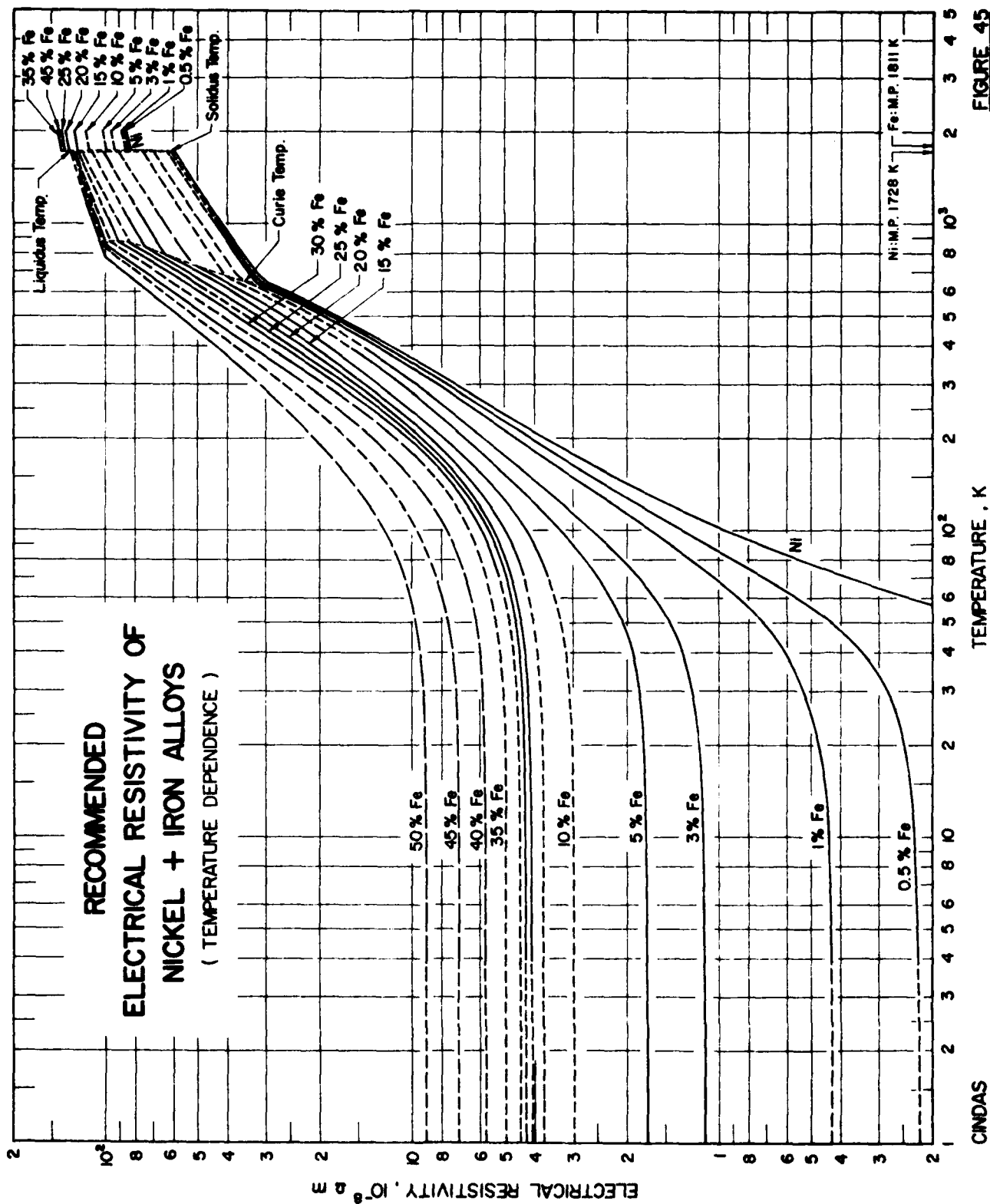


FIGURE 45

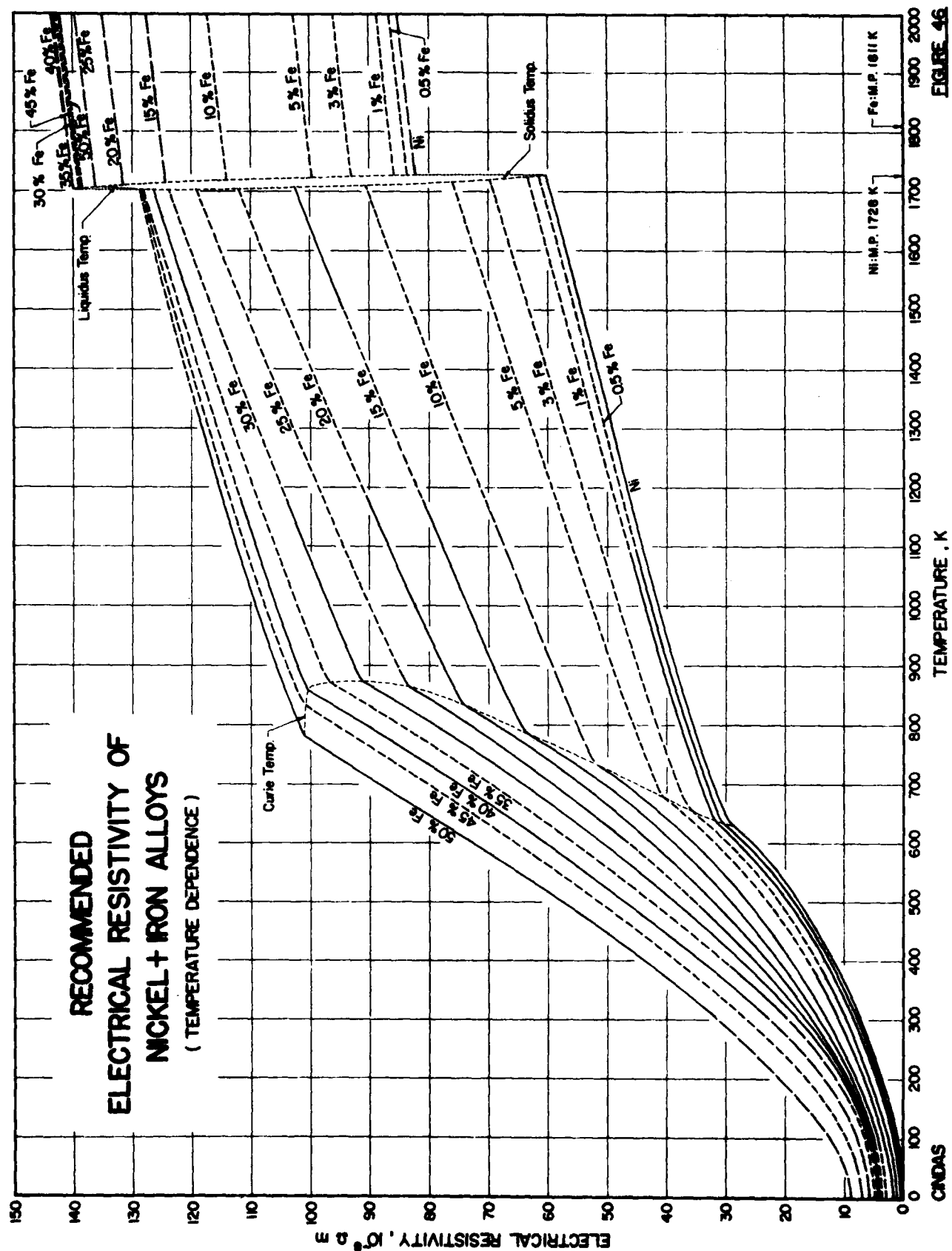


FIGURE 46

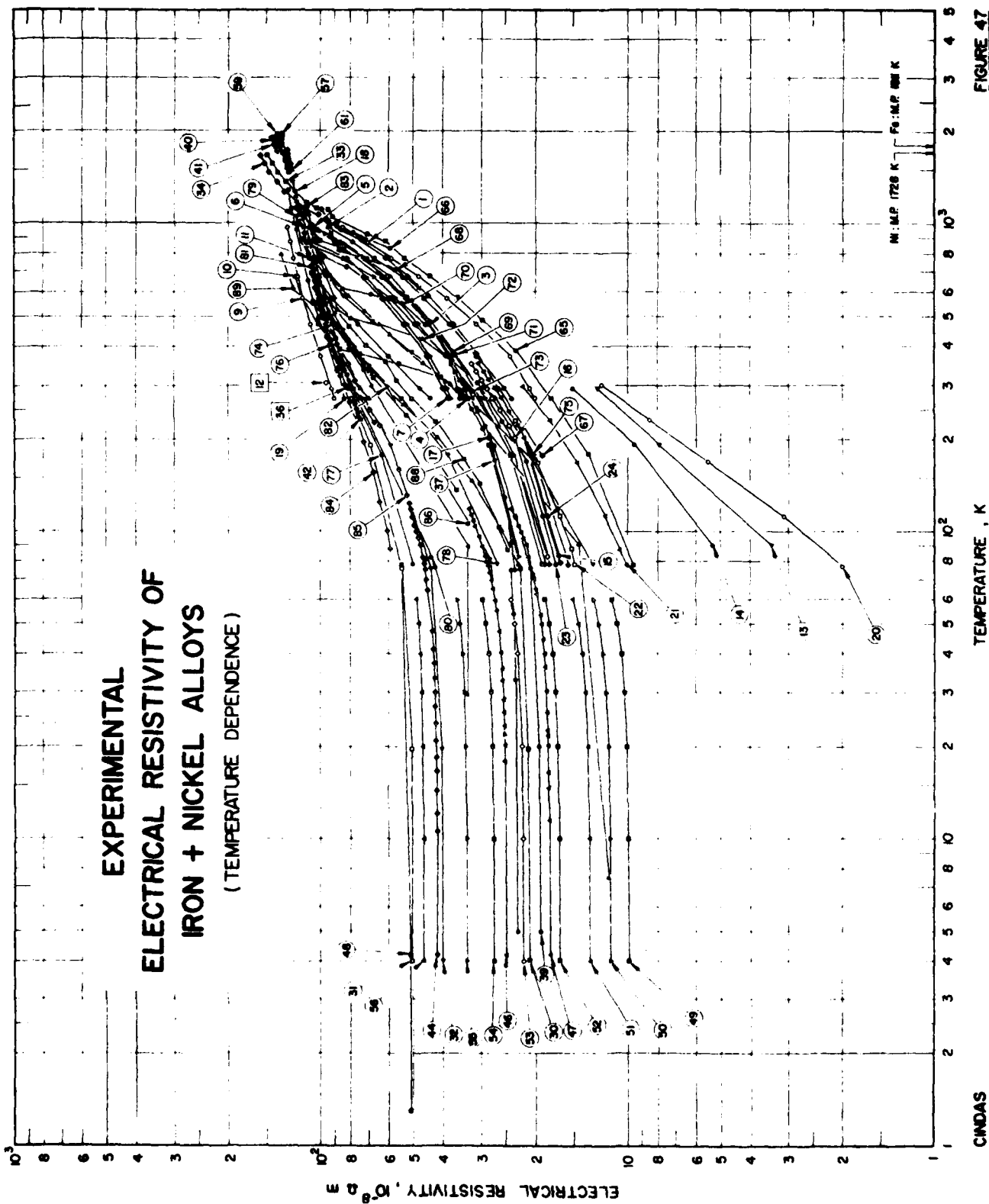


FIGURE 47

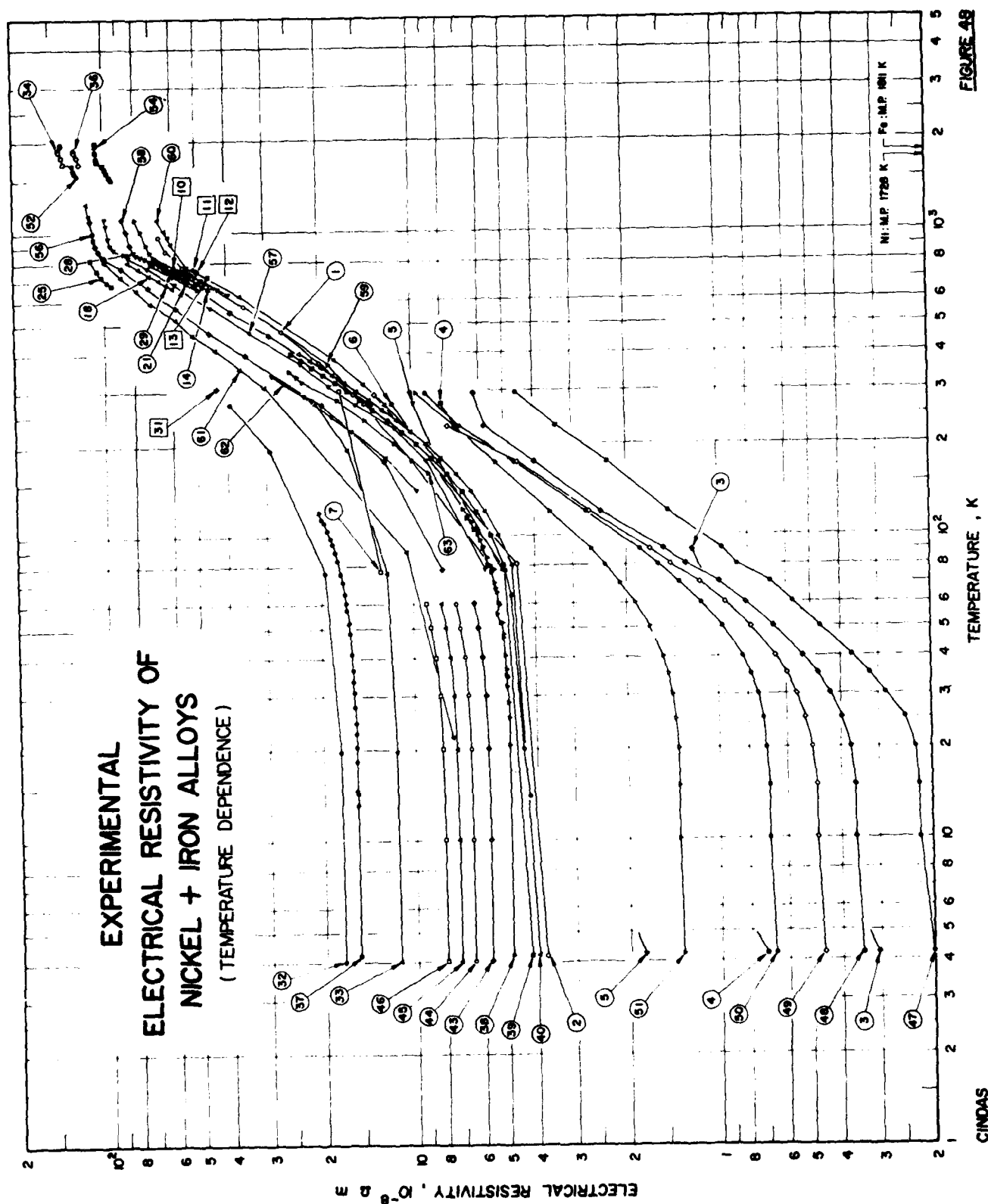


FIGURE 48

TABLE 5a. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF IRON + NICKEL ALLOYS (Temperature Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Fe Ni	Composition (continued), Specifications, and Remarks
1	332	Ingersoll, L. R.	1920	A	273-973	150 J	4.0	Composition includes an estimated <0.10 C; alloys prepared from doubly electro-deposited iron of 99.97 purity and electrolytic nickel of high purity; ingots formed from weighed amounts by melting in a magnesia crucible heated within a resistor furnace and subsequently forging into bars from which 1 cm bars were then machined; a measuring current of some 7.5 amps was applied; temperatures were measured with a copper-constantan thermocouple; smoothed values were reported by the author at 100 deg. intervals in figure format along with a tabulated point at 293 K; data apparently taken with increasing temp only, possible transformation to austenite at higher temperatures was not discussed.
2	332	Ingersoll, L. R.	1920	A	273-973	150 L	7.0	Similar to the above specimen.
3	332	Ingersoll, L. R.	1920	A	273-973	166 A	13.0	Similar to the above specimen.
4	332	Ingersoll, L. R.	1920	A	273-973	166 B	14.0	Similar to the above specimen.
5	332	Ingersoll, L. R.	1920	A	273-973	166 E	18.0	Similar to the above specimen.
6	332	Ingersoll, L. R.	1920	A	273-973	150 S	21.0	Similar to the above specimen.
7	332	Ingersoll, L. R.	1920	A	273-973	166 G	22.11	Similar to the above specimen except specimen composition determined by chemical analysis.
8*	332	Ingersoll, L. R.	1920	A	273-973	166 I	26.40	Similar to the above specimen.
9	332	Ingersoll, L. R.	1920	A	273-973	166 L	35.09	Similar to the above specimen.
10	332	Ingersoll, L. R.	1920	A	273-973	166 M	40.0	Similar to the above specimen except composition not determined.
11	332	Ingersoll, L. R.	1920	A	273-973	166 O	47.08	Similar to the above except specimen composition determined chemically.
12	196	Ellis, W.C., Morgan, F.L., and Sager, F.G.	1928		305	Climax	70 30	Specimen dimensions apparently about 35 cm in length x 2.5 mm in dia., electrical resistivity measurement not discussed.
13	356	Böcklund, N.G.	1961	A	90, 193, 293		0.95	Composition reported is 0.90 at. % Ni; alloys originally supplied (circa 1920) by Heracus, Hanau, Germany, in wire form and were kindly given for this investigation by G. Borclius; the wire was remelted and subsequently rolled into bars of 15 mm <sup>2</sup> cross-section x 100 mm length; the remelting was carried out in a sealed silica tube containing an argon atmosphere under 120 mm Hg pressure at room temp and placed within a graphite crucible in an induction furnace, the rolling was followed by an anneal at 1373 K for 2 hrs in an evacuated silica tube to remove any inhomogeneities; subsequently the bars were rolled to final form and given a final anneal at 773 K for 10 hrs to relieve cold-work effects; separation of steel edge potential probes was measured by an optical comparator and specimen cross-section was determined from length, mass, and density; specimen and mount immersed directly into temperature baths (liquid oxygen, solid carbon dioxide, or ice water); electrical measurements carried out with a high-precision thermoelectric potentiometer of Desselhorst type.

\* Not shown in figure.

TABLE 58. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF IRON + NICKEL ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Fe Ni	Composition (continued), Specifications, and Remarks
14	356	Bicklund, N.G.	1961	A	90, 193, 293		1.90	Composition reported is 1.81 at. % Ni; preparation and measurement similar to that of the above specimen.
15	360	Kohlhaas, R. and Kierspe, W.	1965	A	83-353	10 Ni 14	3.74	Composition reported for this steel includes: 0.06 wt. % C, 0.32 wt. % Si, and 0.45 wt. % Mn; specimens heat-treated in two stages: 0.5 hr at 1123 K followed by air-cooling to 873 K for a 2 hr anneal and air-cooling to room temp; electrical resistivity measurements apparently performed in companion fashion to thermal conductivity.
16	360	Kohlhaas, R. and Kierspe, W.	1965	A	83-353	12 Ni 19	4.75	Composition reported for this steel includes: 0.086 wt. % C, 0.35 wt. % Si, and 0.40 wt. % Mn; preparation and measurement similar to the above specimen.
17	360	Kohlhaas, R. and Kierspe, W.	1965	A	83-353	X 8 Ni 9	8.35	Composition reported for this steel includes: 0.051 wt. % C, 0.29 wt. % Si, 0.74 wt. % Mn, 0.016 wt. % P, 0.006 wt. % S, 0.009 wt. % N, and 0.093 wt. % Cr; specimen heat treated in two stages: 0.5 hr at 1063 K followed by air-cooling to 943 K for a 3.5 hr anneal followed again by an air-cool to room temperature; similar in other respects to the above specimen.
18	363	Bungard, K. and Spyra, W.	1965	A	293-1373	Ni 36	37	Composition reported for this steel includes: 0.02 wt. % C, 0.05 wt. % Si, 0.32 wt. % Mn, 0.012 wt. % P, 0.008 wt. % S, 0.08 wt. % Al, 0.05 wt. % Co, and 0.06 wt. % Mo; specimen heat treated at 1273 K for 24 hrs and water-quenched; electrical resistivity apparently measured in companion fashion along with thermal conductivity.
19	361	Griffiths, E., Powell, R.W., and Hickman, M.J.	1939	A	273-623		28, 37	Composition reported for this high-alloy steel includes: 0.29 wt. % C, 0.15 wt. % Si, 0.89 wt. % Mn, 0.003 wt. % S, 0.009 wt. % P, 0.030 wt. % Cu, 0.012 wt. % Al, 0.027 wt. % As, and trace of Cr; heat treatment consisted of heating to 1223 K and water quenching; electrical resistivity measured in companion fashion with thermal conductivity; specimen current used was 10 amp, thermocouple leads serve as potential probes.
20	357	Soffer, S., Dreesen, J.A., and Pugh, E.M.	1965	A	78-302		0.42	Composition reported as nominally 0.4 at. % Ni; alloys prepared by melting weighed amounts in an induction furnace, cold-rolling to thickness and machining plates approximately 10 cm x 1 cm x 1 mm; specimens were subsequently annealed to minimize strains; probes of same composition material held in place by phosphor bronze springs on a Micarta sample holder; two thick copper sheets with a Mylar spacer acted to reduce thermal gradients; copper to alloy junctions were enclosed within copper sheaths; the electrical measurements were taken with a Wenner potentiometer and a dc null detector; sample assembly immersed directly into baths of liquid nitrogen (77K), methane (112K), ethylene (169K), propane (231K), silicone oil (room temp), and occasionally liquid argon (87K).
21	357	Soffer, S., Dreesen, J.A., and Pugh, E.M.	1965	A	78-299		5.2	Composition reported as nominal 5 at. % Ni; similar to the above specimen.



TABLE 58. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF IRON + NICKEL ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Fe Ni	Composition (continued), Specifications, and Remarks
22	357	Soffer, S., Dreessen, J.A., and Pugh, E.M.	1965	A	78-303		10.5	Composition reported as a nominal 10 at. % Ni; similar to the above specimen.
23	357	Soffer, S., et al.	1965	A	78-303		15.7	Composition reported as a nominal 15 at. % Ni; similar to the above specimen except that hot-rolling was found necessary; this was thought to be due to mixed-phase makeup of the alloy which is consistent with accepted phase diagrams, reduced electrical resistivity of this alloy thought to be due to segregation of mixed phases.
24	357	Soffer, S., et al.	1965	A	78-297		20.8	Composition reported as a nominal 20 at. % Ni; similar to the above specimen.
25*	364	Wilbur B. Driver Co.	1966		298	Niromet 48	52 48	Nominal composition; data from manufacturer's product literature.
26*	364	Wilbur B. Driver Co.	1966		298	Niromet 46	54 46	Nominal composition; data from manufacturer's product literature.
27*	364	Wilbur B. Driver Co.	1966		298	Niromet 44	56 44	Nominal composition; data from manufacturer's product literature.
28*	364	Wilbur B. Driver Co.	1966		298	Niromet 42	58 42	Nominal composition; data reported from manufacturer's product literature.
29*	364	Wilbur B. Driver Co.	1966		298	Niromet 36	64 36	Nominal composition; data reported from manufacturer's product literature.
30	103	Clark, A.F., Chida, G.E., and Wallace, G.H.	1970	A	4-273	Fe-29% Ni	~69 28.45	Composition reported for this nickel steel includes: 0.91 wt. % Si, 0.51 wt. % Mn, 0.20 wt. % Cu, and lesser amounts of S and C; alloy described as being fully austenitic and having an average grain size of 0.025 mm; specimen rough-machined from stock materials to dimensions of about 152 mm long x 6.35 mm dia, and subsequently ground flat with sides parallel to $\pm 0.005$ mm; specimen mount forms part of a dip probe to be lowered into temperature baths of liquid helium (4K), liquid hydrogen (20K), liquid nitrogen (76K), solid carbon dioxide (192K), and ice water (273K); data acquisition was fully automated to take and average nineteen successive readings; for the lower resistivity specimens a potentiometer and null detector were available enabling a resolution of a few $\times 10^{-4}$ volt.
31	103	Clark, A.F., et al.	1970	A	4-273	Invar 36	~62 35.99	Composition reported for this technical alloy includes: 0.81 wt. % Mn, 0.35 wt. % Si, 0.17 wt. % Se, and lesser amounts of C, S, and P; alloy described as being 12-15% cold-drawn and having an average grain size of 0.02 mm; similar in other respects to the above specimen.
32	103	Clark, A.F., et al.	1970	A	4-273	LR 35	~63 36.04	Composition reported for this technical alloy includes small but unspecified amounts of C, Co, Si, Cu, Al, Mn, Cr, P, S, N, and O; alloy described as being hot-rolled and having an average grain size of 0.04 mm; similar in other respects to the above specimen.
33	365	Smolin, M.D.	1966		1123-1673		2.1	Composition reported is 2 at. % Ni; no other experimental details reported.
34	365	Smolin, M.D.	1966		1073-1673		4.2	Composition reported is 4 at. % Ni; no other experimental details reported.
35*	363	Wotruba, K.	1967		294	Hiperduk	51.34 48.24	Composition reported for this alloy includes 0.38 wt. % Mn and 0.04 wt. % C, as determined by the Research Inst. of V.I. Lenin Enterprise at P'zen; wire specimens were formed 10.0 cm x 0.05 cm in dimension; specimens annealed at 1173 K in an atmosphere of technical hydrogen.

\* Not shown in figure.

TABLE 58. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF IRON + NICKEL ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Fe Ni	Composition (continued), Specifications, and Remarks
36	333	Reed, R. P., Clark, A. F., and Schramm, R. E.	1971		295	Fe-49 Ni	29	Alloy heat treatment consisted of a vacuum anneal at 1323 K for 1 hr followed by an air cool to 295 K; structure reported to be 100% austenitic; a defect annealing study is reported by authors.
37	333	Reed, R. P., et al.	1971		4-295			Above specimen cooled quickly to 4 K; structure reported to be about 97% bcc martensite; electrical resistivity data collected continuously from 4 K to room temp with holds at 77 K of 6.7 hrs and at 193 K of 2 hrs; electrical resistivity values read from figure showing continuous record.
38*	333	Reed, R. P., et al.	1971		4-295			Above specimen quickly re-cooled to specimen 4 K; electrical resistivity apparently re-measured at both temperatures of liquid helium (4K), liquid nitrogen (77K), solid carbon dioxide (193K), and room temperature (295K).
39	333	Reed, R. P., et al.	1971		4-273			Above specimen aged 10 wks at room temp and again recooled to 4 K; remeasurement includes both temperatures of liquid hydrogen (20K) and ice water (273K); specimen subsequently etched and found to contain 98% martensite; total electrical resistivity change at room temp attributed to martensite only; resistivity at room temp due to dislocations estimated to be only 0.1 $\mu\Omega\text{cm}$ ; resistivity recovery from 4 K up to room temp of 3.09 $\mu\Omega\text{cm}$ suggests a concentration of point defects amounting to about 0.5 at.% being produced by the transformation.
40	349	Ozo, Y. and Yagi, T.	1972	R	1749-1898		20.8	Composition reported as 20.0 at.% Ni; in liquid state; alloy specimens prepared from weighed amounts of metals having purities of 99.9% or more by melting in a vacuum induction furnace; a specimen of about 10 g contained in a recrystallized alumina crucible, 10 mm ID x 62 mm height, was suspended by a tungsten wire 0.12 mm dia; specimen heated under reduced pressure of 0.05 to 0.1 mm Hg by a graphite heating element of coaxial design in order that counter-flowing currents would contribute no net magnetic field within the specimen; density data from the literature were used to correct for volume thermal expansion; data fit to a linear temperature dependence the coefficients of which are reported as $\rho = 0.0403 T + 78.33$ in $\mu\Omega\text{cm}$ and $C^\circ$ respectively; overall probable error is stated to be $\pm 0.5\%$ .
41	349	Ozo, Y. and Yagi, T.	1972	R	1717-1898		40.8	Composition reported as 39.6 at.% Ni; similar to the above specimen except $\rho = 0.0352 T + 88.01$ in $\mu\Omega\text{cm}$ and $C^\circ$ respectively.
42	334	Livingston, H. and Mukherjee, K.	1972	A	273-234		30.8	Composition reported as 29.7 at.% Ni; polycrystalline alloy prepared by arc melting 99.999% pure Fe (5-10 ppm C) and Ni in argon atmosphere; samples annealed at 1273 K for 24 hrs under vacuum; average grain size was 0.2 $\mu\text{m}$ ; flat specimen carefully cut to $1.2 \times 0.5 \times 0.08 \text{ cm}^3$ ; electrical circuitry achieved sensitivity of $8 \times 10^{-4}$ volt; specimen temperature bath of dry ice and acetone was used; cooling data only are reported with a martensitic transformation temp of 234 K being reported.
43*	334	Livingston, H. and Mukherjee, K.	1972	A	263-234		30.8	Above specimen again measured at decreasing temps but in an 8.2 kOe transverse magnetic field.

\* Not shown in figure.

TABLE 58. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF IRON + NICKEL ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Fe Ni	Composition (continued), Specifications, and Remarks
44	335	Armstrong, B. E. and Fletcher, R.	1972	A	4-125		62.2 37.1	Composition analysis originated from supplier and reported as 35.8 ± 1 at. % Ni, 0.4 at. % Mn, 0.3 at. % C, 0.13 at. % Co, 0.1 at. % Si, 0.07 at. % Cu, and balance Fe; alloys commercial grade and supplied in form of cold-rolled rods by Henry Wiggin and Co., Ltd., Hereford, England; specimens measured in as-received condition; electrical resistivity measured along with thermoelectric power using thermocouple leads as potential probes; data scatter at 0.2% level while overall uncertainty estimated to be ±2%; data read from figures as temperature independent and dependent resistivities which were subsequently combined.
45*	335	Armstrong, B. E. and Fletcher, R.	1972	A	4-122			Above specimen heat-treated by holding at 1373 K for 1 hr and water quenching, then annealing at 573 K for 12 hrs and at 373 K for 2 days.
46	335	Armstrong, B. E. and Fletcher, R.	1972	A	4-121		56.0 43.7	Composition analysis originated from supplier and reported as 42.4 ± 1 at. % Ni, 0.3 at. % C, 0.1 at. % Si, 0.04 at. % Cu, 0.03 at. % Mn, and balance Fe; similar to above specimen but with no heat treatment.
47	335	Armstrong, B. E. and Fletcher, R.	1972	A	4-120		49.8 49.5	Composition analysis originated from supplier and reported as 48.1 ± 1 at. % Ni, 0.5 at. % Mn, 0.2 at. % Si, 0.2 at. % C, 0.06 at. % Co, 0.05 at. % Cu, and balance Fe; similar to above specimen.
48	337	Mikhailova, G. N.	1971	A	0.4-293	Invar	35 37	Composition reported for this technical alloy includes: 0.7 wt. % Mn, 0.6 wt. % Si, and 0.25 wt. % C; the alloy was annealed in hydrogen at 633 K.
49	336	Gautier, F. and Loegel, B.	1972	A	4-60		51.8	Composition reported as 53 at. % Fe; alloy preparation reported by Cadville and Loegel [358]; alloys prepared by melting 4N8 starting materials from Koch-light provenance in an induction furnace under argon atmosphere, ingots subsequently annealed 1 day at 1323 K and quenched, atomic structure at room temp is fcc as determined from x-ray measurements; data obtained from the expression $\rho = \rho_0 + 0.00038 T^2$ ( $\rho$ in $\mu\Omega\text{cm}$ , $T \leq 60$ K) reported by authors.
50	336	Gautier, F. and Loegel, B.	1972	A	4-60		53.8	Composition reported as 55 at. % Fe; alloy preparation and measurement similar to above specimen; data obtained from the expression $\rho = \rho_0 + 0.00043 T^2$ ( $\rho$ in $\mu\Omega\text{cm}$ , $T \leq 60$ K) reported by authors.
51	336	Gautier, F. and Loegel, B.	1972	A	4-60		55.8	Composition reported as 57 at. % Fe; alloy preparation and measurement similar to above specimen; data obtained from the expression $\rho = \rho_0 + 0.000502 T^2$ ( $\rho$ in $\mu\Omega\text{cm}$ , $T \leq 60$ K) reported by authors.
52	336	Gautier, F. and Loegel, B.	1972	A	4-60		57.8	Composition reported as 59 at. % Fe; alloy preparation and measurement similar to above specimen; data obtained from the expression $\rho = \rho_0 + 0.000557 T^2$ ( $\rho$ in $\mu\Omega\text{cm}$ , $T \leq 60$ K) reported by authors.
53	336	Gautier, F. and Loegel, B.	1972	A	4-60		59.8	Composition reported at 61 at. % Fe; alloy preparation and measurement similar to above specimen; data obtained from the expression $\rho = \rho_0 + 0.000623 T^2$ ( $\rho$ in $\mu\Omega\text{cm}$ , $T \leq 60$ K) reported by authors.
54	336	Gautier, F. and Loegel, B.	1972	A	4-60		61.8	Composition reported as 62.5 at. % Fe; alloy preparation and measurement similar to above specimen; data obtained from the expression $\rho = \rho_0 + 0.000668 T^2$ ( $\rho$ in $\mu\Omega\text{cm}$ , $T \leq 60$ K) reported by authors.

\* Not shown in figure.

TABLE 56. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF IRON + NICKEL ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Fe Ni	Composition (continued), Specifications, and Remarks
55	336	Gautier, F. and Loegel, B.	1972	A	4-60		62.8	Composition reported as 64 at. % Fe; alloy preparation and measurement similar to above specimen; data obtained from the expression, $\rho = \rho_0 + 0.000725 T^2$ ( $\rho$ in $\mu\Omega\text{cm}$ , $T \leq 60$ K) reported by authors.
56	336	Gautier, F. and Loegel, B.	1972	A	4-60		64.9	Composition reported as 66 at. % Fe; alloy preparation and measurement similar to above specimen; data obtained from the expression $\rho = \rho_0 + 0.000735 T^2$ ( $\rho$ in $\mu\Omega\text{cm}$ , $T \leq 60$ K) reported by authors.
57	350	Baum, B.A., Tyuganov, G.V., Cel'd, P.V., and Khasin, G.A.	1971	R	1564-1961		5.2	Composition reported as 5 at. % Ni; data reported for both solid and liquid states in figure format; $d\rho/dT = 0.0072 \mu\Omega\text{cm K}^{-1}$ in liquid state; data in increasing temperature.
58*	350	Baum, B.A., et al.	1972	R	1940-1535			Above specimen measured at decreasing temperatures.
59	350	Baum, B.A., et al.	1972	R	1473-1960		15.6	Composition reported as 15 at. % Ni; data reported for both solid and liquid states; $d\rho/dT = 0.0034 \mu\Omega\text{cm K}^{-1}$ in liquid state; increasing temperature data.
60*	350	Baum, B.A., et al.	1972	R	1887-1499			Above specimen measured at decreasing temperatures.
61	350	Baum, B.A., et al.	1972	R	1491-1948		31.1	Composition reported as 30 at. % Ni; data reported for both solid and liquid states; $d\rho/dT = 0.0011 \mu\Omega\text{cm K}^{-1}$ in liquid state; increasing temperature data.
62*	350	Baum, B.A., et al.	1972	R	1878-1483			Above specimen measured at decreasing temperatures.
63*	350	Baum, B.A., et al.	1972	R	1527-1976		46.2	Composition reported as 45 at. % Ni; data reported for both solid and liquid states; $d\rho/dT = 0.0044 \mu\Omega\text{cm K}^{-1}$ in liquid state; increasing temperature data.
64*	350	Baum, B.A., et al.	1972	R	1919-1502			Above specimen measured at decreasing temperatures.
65	330	Shirakawa, Y.	1939	B	76-1126	No. 1	95.66	Alloys prepared from electrolytic iron and nickel supplied by Nippon-Denkai-Seitetsusho and by Monson and Co., respectively; chemical analysis of starting materials include, for iron: 0.05% Si, 0.05% P, 0.04% C, 0.02% Co, 0.02% Mn, 0.01% Al, and 0.003% S, and for nickel: 0.02% Fe, 0.01% C, 0.003% S, 0.002% Si, 0.001% P, and 0.001% Mn; weighed amounts were mixed and then melted under hydrogen atmosphere in an alumina crucible within a Tamman furnace; wire formed by drawing molten alloy up into silica capillary which was subsequently carefully broken; specimen annealed at 1273 K for 1 hr under vacuum and cooled slowly with length oriented along east-west direction; specimen dimensions 5.36 cm length x 0.0297 cm diameter; a nickel lead wire soldered to each end of specimen with pure silver; specimen plus lead wires annealed again at 1123 K under high vacuum for 1 hr and slowly cooled; specimen orientation during measurement always east-west; magneto-resistance in longitudinal fields up to 1.6 kOe also measured; electrical resistivity measurement technique is of 2-probe type; data reported in tabular form; measurements reported for increasing temperatures.
66	330	Shirakawa, Y.	1939	B	1063-632	No. 2	90.07	Above specimen with cooling data reported.
67	330	Shirakawa, Y.	1939	B	76-1123			Similar to above specimen except dimensions were 5.45 cm length x 0.0319 cm diameter; heating data reported.

\* Not shown in figure.

TABLE 58. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF IRON + NICKEL ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Fe Ni	Composition (continued), Specifications, and Remarks
68	330	Shirakawa, Y.	1939	B	1025-615	No. 3	86.04	Above specimen with cooling data reported.
69	330	Shirakawa, Y.	1939	B	78-1123	No. 3	86.04	Similar to above specimen except dimensions were 5.40 cm length x 0.0347 cm diameter; heating data reported.
70	330	Shirakawa, Y.	1939	B	1076-373	No. 4	80.44	Above specimen with cooling data reported.
71	330	Shirakawa, Y.	1939	B	78-1123	No. 4	80.44	Similar to above specimen except dimensions were 6.79 cm length x 0.0445 cm diameter; heating data reported.
72	330	Shirakawa, Y.	1939	B	1121-364	No. 5	75.31	Above specimen with cooling data reported.
73	330	Shirakawa, Y.	1939	B	78-1121	No. 5	75.31	Similar to above specimen except dimensions were 6.37 cm length x 0.0467 cm diameter; heating data reported.
74	330	Shirakawa, Y.	1939	B	1074-273	No. 6	70.13	Above specimen with cooling data reported.
75	330	Shirakawa, Y.	1939	B	78-1123	No. 6	70.13	Similar to above specimen except dimensions were 5.02 cm length x 0.0890 cm diameter; heating data reported.
76	330	Shirakawa, Y.	1939	B	1033-178	No. 7	64.96	Above specimen with cooling data reported.
77	330	Shirakawa, Y.	1939	B	78-1127	No. 7	64.96	Similar to above specimen except dimensions were 3.43 cm length x 0.0342 cm diameter.
78	330	Shirakawa, Y.	1939	B	78-1143	No. 8	60.00	Similar to above specimen except dimensions were 6.67 cm length x 0.0447 cm diameter.
79	330	Shirakawa, Y.	1939	B	78-1122	No. 9	50.11	Similar to above specimen except dimensions were 5.46 cm length x 0.0326 cm diameter.
80	338	Larikov, L.N., Ussov, Yu.V., and Boychev, I.N.	1977		4-1289		35.68	Measurement included data readings at both increasing and decreasing temperatures; data reported in figure format.
81	338	Larikov, L.N., et al.	1977		4-1281		36.9	Similar to above specimen.
82	338	Larikov, L.N., et al.	1977		4-1280		37.64	Similar to above specimen.
83	338	Larikov, L.N., et al.	1977		4-1290		44.86	Similar to above specimen.
84	343	Somura, T.	1977	C	88-672		34.1	Composition reported as 33 at. % Ni; alloys prepared by arc-melting iron and nickel rods of 99.99% purity; ingots cut into rectangular plates of dimension 0.8 x 0.4 x 0.03 cm; specimens annealed at 1273 K under vacuum for 5 hrs; thin copper leads spot welded to specimens; electrical measurements performed using four-probe AC technique with primary current of 0.4 amp at 30 Hz; specimen temp controlled to $\pm 0.2$ deg; data reported in combined figure format; Curie temperature reported as 456 K.
85	343	Somura, T.	1977	C	82-701		37.2	Composition reported as 36 at. % Ni; similar to above specimen with Curie temperature reported to be 511 K.
86	343	Somura, T.	1977	C	106-719		39.2	Composition reported as 38 at. % Ni; similar to above specimen with Curie temperature reported to be 598 K.
87	343	Somura, T.	1977	C	132-630			Above specimen with $\rho^{-1} d\rho/dT$ determined directly from incremental measurement over 2 to 3 deg. interval.

\* Not shown in figure.

TABLE 58. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF IRON + NICKEL ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Fe Ni	Composition (continued), Specifications, and Remarks
88	343	Somura, T.	1977	C	88-772		41.2	Composition reported as 40 at. % Ni; similar to above specimen except $\rho_0/dT$ not measured; Curie temperature reported to be 630 K.
89	343	Somura, T.	1977	C	78-838		43.2	Composition reported as 42 at. % Ni; similar to above specimen with Curie temperature reported to be 663 K.
90*	331	Ascher, E. A.	1955	A	301-876	Ni 30	30.1	Composition as reported includes $\leq 0.02\%$ Si; alloy prepared from weighed amounts of starting materials by melting and rolling to thin plate; specimen heat-treated at 1273 K for 2 hrs under hydrogen atmosphere; furnace quenched to 903 K, and air-cooled to room temperature; specimen temperatures measured with thermocouples and indicate temperature gradients of no more than 1.5 deg during measurements; electrical measurements performed with Dieselhorst potentiometer using shielded components; specimen immersed in liquid air for 4 hrs; data presented in tabular form.
91*	331	Ascher, E. A.	1955	A	295-763			Same specimen heated to 923 K for 50 hrs and remeasured.
92*	331	Ascher, E. A.	1955	A	303-323	Ni 31	31.4	Composition as reported includes 0.28% Si; alloy prepared as above except rolling followed by anneal at 873 K for 2 hrs and furnace cooling for 4 hrs; data reported in tabular form.
93*	325	Port, A. and Campbell, L. A.	1976	A	1.3-44.8		1.1	Composition reported as 1 at. % Ni; alloy prepared at the CEN, Grenoble in a levitation furnace from starting materials obtained by zone melting (RRR = 700 and 25 for Ni and Fe respectively); alloy drawn into wire of 1 mm diameter and subsequently annealed at 1073 K; absence of precipitates checked by microscopy; alloy analyzed by calorimetry or atomic absorption spectroscopy; accuracy generally limited by current stability ( $\pm 2 \times 10^{-5}$ ); temperatures in helium range measured with germanium resistor to an accuracy of $\pm 0.1$ deg; total electrical resistivity recovered from tabulated residual resistivity values for both 1 at. % Ni and 2 at. % Ni combined with temperature dependent resistivity deduced from measurements on both alloys and presented as a single curve in figure format.
94*	328	Schwerer, F. C. and Cuddy, L. J.	1970	A	4-1097		0.99	Composition reported as 0.94 at. % Ni; alloys preparation not reported; specimen heated in a flowing He-2% $\text{H}_2$ atmosphere within a noninductively wound furnace; temperature measurement by means of thermocouple with accuracy of few degrees; measurement procedure involved simultaneous measurement of specimen and iron standard ( $\rho_0 \approx 0.04 \mu\Omega\text{cm}$ ) thus obviating more precise temperature determination; temperature increased at about 1 deg/min during measurement; chief source of errors was geometrical factor for which precision was $\pm 3\%$ giving errors at high temperature of $\pm 0.5 \mu\Omega\text{cm}$ ; data reported as solute resistivity in figure format.

\* Not shown in figure.

TABLE 58. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF IRON + NICKEL ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Fe Ni	Composition (continued), Specifications, and Remarks
95*	329	Schwerer, F. C. and Cuddy, L. J.	1970	A	4.2-1140		0.46	Composition reported as 0.44 at. % Ni; specimen preparation not reported; specimen preparation not reported; solute concentration determined by chemical analysis; current and potential leads of iron spot welded with gauge length measured by traveling microscope; average cross-section deduced from mass, length, and density; largest sources of error thought to be geometrical factors for which uncertainties are thought to be < 0.2%; errors in solute resistivity certainly less than 0.5% resistivity of iron standard; high temperature measurements involved mounting specimen and iron standard ( $\rho_0 \approx 0.04 \mu\Omega\text{cm}$ ) within non-inductively wound furnace in thermal proximity of each other and of thermocouple well, atmosphere of He-2% H <sub>2</sub> flowed past during measurements; temperature monitored with Pt/Pt 10% Rh thermocouple; current of 0.5 amp stabilized to $1:10^4$ was used, measurements made with temperature changing at rate of $1 \text{ deg min}^{-1}$ ; voltages amplified with Keithley 140 and measured with digital voltmeter; complete set of measurements (including specimen, standard, and reversal) could be made within temperature change of $\leq 0.2 \text{ deg}$ ; data reported as solute resistivity ( $= \rho_{\text{alloy}} - \rho_{\text{standard}}$ ) in figure format.
96*	329	Schwerer, F. C. and Cuddy, L. J.	1970	A	4.2-1101		0.99	Composition reported as 0.94 at. % Ni; preparation and measurement similar to above specimen.

\* Not shown in figure.

TABLE 50. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF IRON + NICKEL ALLOYS (Temperature Dependence)

[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-4}$   $\Omega$  m]

[illegible]

\* Not shown in figure.



[illegible]

**Not shown in figure.**





TABLE 59. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF IRON + NICKEL ALLOYS (Temperature Dependence) (continued)

DATA SET 83 (cont.)			DATA SET 84 (cont.)			DATA SET 85 (cont.)			DATA SET 86 (cont.)			DATA SET 87 (cont.)*			DATA SET 88 (cont.)		
T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho^{-1} \rho_0 / T$		T	$\rho$	
550	71*		506	99.7		498	96.2*		431	84.4*		608	0.011		552	93.3*	
505	53*		519	99.7*		517	96.9*		444	85.7*		630	0.011		565	95.2*	
445	54*		531	101.0*		529	98.1*		447	87.0*		T	$\rho$		580	95.9*	
385	42*		544	101.0*		542	96.8*		463	89.5*		DATA SET 88			596	97.8*	
339	36*		556	102.2		548	99.4		481	92.1*		88	24.8		615	99.0*	
302	32*		569	102.2*		561	99.4*		494	93.3		94	25.4*		621	99.7*	
DATA SET 84			594	103.5*		573	100.0*		506	94.6*		104	26.0		630	101.0*	
88	59.7		606	103.3*		586	100.6*		512	95.2*		113	27.3*		649	101.6	
100	61.0		619	104.8*		601	101.3*		522	95.9*		125	27.3*		668	102.9*	
106	62.2*		641	105.4*		617	101.9*		538	97.1*		144	30.5		681	104.1*	
119	63.5*		650	105.4*		639	102.5*		544	98.4*		154	31.7*		696	104.1*	
125	64.8		659	106.7*		658	103.1*		556	99.7*		163	33.0*		703	104.1*	
138	65.4*		672	106.7*		673	104.4*		566	100.3*		173	34.3		712	104.1*	
DATA SET 85			701	105.0		DATA SET 86			575	101.6*		185	35.6*		718	105.4*	
156	66.0*		DATA SET 86			606	102.9*		594	102.2		194	37.5*		743	105.4	
166	68.6*		82	46.9		616	104.1*		606	104.1*		204	39.4*		762	107.3*	
178	69.8*		84	48.2*		106	33.0		616	104.1*		216	41.3*		772	107.3*	
184	71.1*		104	48.8*		125	34.9*		628	105.4*		229	42.5		DATA SET 89		
197	72.4		116	50.7*		138	36.2		641	105.4*		242	45.1*		78	22.8	
203	73.7*		132	52.6		150	37.5*		647	104.8*		254	47.6*		88	24.1*	
216	74.3*		148	53.3*		163	40.0*		659	105.4*		263	48.9*		100	25.9	
228	76.2*		160	55.8		172	41.3*		672	106.0*		273	50.2		112	27.2*	
234	76.8*		179	58.3*		178	42.5*		691	107.3		279	50.6*		131	29.7*	
247	78.1*		191	59.6*		191	43.8*		694	107.3*		289	52.1*		147	32.3*	
259	79.4*		207	62.1*		197	46.3*		709	107.9*		295	53.3*		172	35.4*	
266	80.0		223	64.6		209	47.6*		719	107.9*		311	56.5*		188	38.0*	
278	81.3*		235	66.5*		222	49.5*		DATA SET 87*			326	59.7*		203	41.8	
284	82.5*		251	69.1		228	51.4*		T	$\rho^{-1} \rho_0 / T$		332	61.0*		222	44.9*	
300	83.8		263	71.0*		241	52.7		132	0.310		345	62.2*		244	48.7*	
325	86.3*		273	72.2*		253	55.2*		147	0.314		358	64.1*		266	53.8	
331	87.6*		285	73.5*		259	56.5*		161	0.315		364	65.4*		288	58.2*	
344	87.6		298	76.0*		272	57.8*		207	0.308		376	67.9*		312	63.3*	
356	88.9*		310	77.3		278	59.7*		236	0.302		386	69.2*		331	69.6	
366	89.5*		323	78.2*		288	61.6*		261	0.291		395	70.5*		353	74.1*	
381	90.8*		335	81.1*		297	62.9*		287	0.278		405	71.7		372	78.5*	
388	91.4		351	81.7		303	65.4*		303	0.273		414	73.7*		403	84.8*	
397	92.1*		360	83.6*		319	67.3		354	0.238		427	75.6*		422	90.5*	
409	93.3*		373	84.9*		331	68.6*		372	0.227		433	76.8*		444	94.3*	
419	93.3*		389	86.1*		338	70.5*		390	0.201		449	79.4*		481	101.9*	
425	94.6*		401	87.4*		350	72.4		407	0.201		464	80.6		500	105.7	
438	95.2*		407	88.7		359	73.7*		425	0.168		477	83.2*		534	111.4*	
450	96.5		426	89.9*		372	75.6*		468	0.141		489	84.4*		563	115.8*	
463	97.1*		454	91.2		381	77.5*		506	0.097		496	86.3*		591	119.6*	
475	97.8*		470	92.5*		388	76.7*		535	0.068		508	87.6*		618	123.4	
484	98.4*		479	93.7*		403	80.0*		556	0.044		514	89.5*		628	124.7*	
494	99.0*		489	94.3*		409	81.9		586	0.031		530	90.8*		641	125.9*	
				96.0*		422	83.2*										

\* Not shown in figure.



TABLE 60. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF NICKEL + IRON ALLOYS (Temperature Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Ni Fe	Composition (continued), Specifications, and Remarks
1	332	Ingersoll, L. R.	1920	A	273-293	173W	90.0	Composition includes an estimated <0.10% C; alloys prepared from doubly electro-deposited iron of 99.97% purity and electrolytic nickel of high purity; ingots formed from weighed amounts by melting in a magnesia crucible heated within a resistor furnace and subsequently forged into bars from which 1 cm bars were machined; a measuring current of some 7.5 amps was applied; temperatures were measured with a copper-constantan thermocouple; smoothed values were reported at 100 deg. intervals in figure format along with a tabulated room temp value.
2	350	Berger, L. and Rivier, D.	1962	A	4, 80, 293		85.2 14.8	Alloy prepared from starting materials obtained from Johnson-Matthey by melting weighed amounts in a high-frequency furnace while under a vacuum of $10^{-3}$ torr in an alumina crucible; melting procedure repeated to improve alloy homogeneity; specimen cut to dimensions of 51 mm long x 2 mm dia and subsequently annealed at 1173 K for 2 hrs under vacuum followed by a furnace-cool of 150 deg hr <sup>-1</sup> down to 433 K, a treatment sufficient to assure a disordered state; composition determined by x-ray fluorescence; specimen immersed into temperature baths of liquid helium (4K) and liquid air (80K); electrical measurements performed with Desselhorst potentiometer.
3	326	Farrell, T. and Greig, D.	1968	A	4-89		0.8	Composition reported as 0.8 at. % Fe; alloys prepared by chill-casting under vacuum; iron content obtained from x-ray fluorescence measurements; length/area ratio determined from length, density, and mass to better than $\pm 0.6\%$ ; data reported as temperature independent and temperature dependent resistivities presented in table and figure formats respectively; subsequent to data reading the total resistivity was formed and is given here.
4	326	Farrell, T. and Greig, D.	1968	A	4-273		1.7	Composition reported as 1.8 at. % Fe; similar to above specimen.
5	326	Farrell, T. and Greig, D.	1968	A	4-273		4.4	Composition reported to be 4.6 at. % Fe was obtained from residual resistivity measurement; similar in other respects to specimen above.
6	220	Van Elst, H.C. and Gortier, C.J.	1954	A	14-273		24	Alloys kindly prepared and given by Laboratories of the Philips Lampworks, Eindhoven; polycrystalline rod specimens about 20 mm length x 1 mm <sup>2</sup> area; electrical measuring instrumentation included a Desselhorst compensator and a Zernike galvanometer; relative accuracy of resistance measurement was 1 part in $10^4$ ; data reported in tabular form.
7	366	Matsumoto, G., Satoh, T., and Iida, S.	1966		77, 293	Permalloy	80 20	Thin film specimens kindly prepared in Japan's Overseas Radio and Cable System Laboratory; films of thickness 600 Å, 1200 Å, and 3000 Å were prepared during the same evaporation run; the electrical resistivity of the three films agreed within $\pm 5\%$ over the measured temperature range.
8*	366	Matsumoto, G. et al.	1966		293	Permalloy	80 20	Bulk specimen prepared from Permalloy material.

\* Not shown in figure.

TABLE 60. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF NICKEL + IRON ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Ni Fe	Composition (continued), Specifications, and Remarks
9*	347	Wakelin, R.J. and Yates, E.L.	1953	A	763		75.9	Composition reported is 75 at. % Ni; alloys prepared from 99.94% pure nickel and iron containing 0.7% oxygen by melting weighed amounts in pure alumina crucible under vacuum within a high-frequency induction furnace; rapid melting under pressure too low for an electrodeless discharge to occur resulted in minimal impurity uptake as evidenced by chemical analyses typically showing $\leq 0.2\%$ total impurities; alloys were hot-forged and then homogenized at 1473 K for 18 hrs; subsequent photomicrographs showed uniform composition throughout; specimens were machined to dimensions of 10 cm x 4 mm x 1 mm with cross-section uniform to $\pm 0.02$ mm in either dimension; reproducibility of measurements indicated $\pm 0.5\%$ uncertainty; specimen chamber temperature was controlled to within 0.5 deg; specimen disordered by heating to 873 K for 12 hrs followed by rapid cooling to measuring temperature and holding for anneal times up to 48 hrs.
10	347	Wakelin, R.J. and Yates, E.L.	1953	A	773			Similar to above specimen except annealing temperature is lower and measurements span up to 50 hrs.
11	347	Wakelin, R.J. and Yates, E.L.	1953	A	763			Similar to above specimen except annealing temperature is lower and measurements span up to 164 hrs.
12	347	Wakelin, R.J. and Yates, E.L.	1953	A	753			Similar to above specimen except annealing temperature is lower and measurements span up to 164 hrs.
13	347	Wakelin, R.J. and Yates, E.L.	1953	A	722			Similar to above specimen except annealing temperature is lower and measurements span up to 117 hrs.
14	347	Wakelin, R.J. and Yates, E.L.	1953	A	662-828		75.9	Similar to above specimen through the 873 K anneal; measurement sequence involved cooling rate of 70 deg hr <sup>-1</sup> .
15*	347	Wakelin, R.J. and Yates, E.L.	1953	A	675-803			Similar to above specimen except cooling rate was 1.3 deg hr <sup>-1</sup> .
16*	347	Wakelin, R.J. and Yates, E.L.	1953	A	691-814			Similar to above specimen except cooling rate was 0.29 deg hr <sup>-1</sup> .
17*	347	Wakelin, R.J. and Yates, E.L.	1953	A	666-783			Similar to above specimen except cooling rate was 0.10 deg hr <sup>-1</sup> .
18	347	Wakelin, R.J. and Yates, E.L.	1953	A	667-806		71.0	Composition reported as 70 at. % Ni with total impurity content of $\leq 0.20\%$ ; preparation and measurement sequence similar to above specimen with cooling rate of 70 deg hr <sup>-1</sup> .
19*	347	Wakelin, R.J. and Yates, E.L.	1953	A	694-804			Similar to above specimen except cooling rate was 1.3 deg hr <sup>-1</sup> .
20*	347	Wakelin, R.J. and Yates, E.L.	1953	A	682-799			Similar to above specimen except cooling rate was 0.29 deg hr <sup>-1</sup> .
21*	347	Wakelin, R.J. and Yates, E.L.	1953	A	663-762		80.8	Composition reported as 80 at. % Ni with total impurity content of $\leq 0.20\%$ ; preparation and measurement sequence similar to above specimen with cooling rate of 70 deg hr <sup>-1</sup> .
22*	347	Wakelin, R.J. and Yates, E.L.	1953	A	659-796			Similar to above specimen except cooling rate was 1.3 deg hr <sup>-1</sup> .

\* Not shown in figure.

TABLE 60. MEASUREMENT INFORMATION OF THE ELECTRICAL RESISTIVITY OF NICKEL + IRON ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Ni Fe	Composition (continued), Specifications, and Remarks
23*	347	Wakelin, R.J. and Yates, E.L.	1953	A	676-801			Similar to above specimen except cooling rate was 0.29 deg hr <sup>-1</sup> .
24*	347	Wakelin, R.J. and Yates, E.L.	1953	A	671-775			Similar to above specimen except cooling rate was 0.10 deg hr <sup>-1</sup> .
25*	347	Wakelin, R.J. and Yates, E.L.	1953	A	683-831		51.3	Composition reported as 50 at.-% Ni with total impurity content $\leq 0.20\%$ ; preparation and measurement sequence similar to above specimen with cooling rate of 60 deg hr <sup>-1</sup> .
26*	347	Wakelin, R.J. and Yates, E.L.	1953	A	704-863			Similar to above specimen except cooling rate was 1.4 deg hr <sup>-1</sup> .
27*	347	Wakelin, R.J. and Yates, E.L.	1953	A	685-801			Similar to above specimen except cooling rate was 0.29 deg hr <sup>-1</sup> .
28	345	Källback, O.	1947	A	631-877		75.93	Composition reported as 75 at.-% Ni; alloy prepared at Kungl. Tekn. Högskolan, Stockholm, by a method typically resulting in a carbon-free nickel; weighed amounts of metals melted in a high-frequency furnace in an evacuated and sealed quartz tube which had been placed within a graphite cylinder heater element; weighing accuracy of $\pm 0.02\%$ and a preparation loss of 0.3% result in composition uncertainty of about 0.1%; spectrographic analysis carried out after all measurements showed 0.04% Cu, 0.01% Ca, 0.01% Mg, $< 0.01\%$ Si, and $< 0.01\%$ Co; ingot cold-rolled subsequent to remelting in quartz from dimensions of 16 mm length x 2 mm diameter down to an 8 mm square bar following which a homogenization at 1223 K for 18 hrs was carried out; specimen dimensions: 25 mm length x 0.8 mm x 0.8 mm; Pt/Pt - 10% Rh thermocouple leads served as potential probes; for measurement, the specimen was sealed into a glass tube which was kept diffusion pumped; uncertainty in resistivity estimated at $\pm 2$ to $\pm 3\%$ ; temperature differences along specimen $\leq 0.2$ deg and specimen enclosure temperature to within a degree; thermal history consisted of heating to 833 K, cooling to 763 for 24 hrs, then to 753 K for 4 days, to 723 K for 2 days and to 623 K for a time, all the while taking data for the ordered state; a critical point at 729 K and a Curie point at 859 K were deduced from the data.
29	345	Källback, O.	1947	A	703-877		75.93	Same specimen reheated to 783 K followed by cooling at a rate of 1 deg/min down to 703 K for measurements on the disordered state.
30*	364	Wilbur B. Driver Co.	1966		298	Balco	70	Nominal composition; data reported from manufacturer's product literature.
31	364	Wilbur B. Driver Co.	1966		298	Niron 52	51	Nominal composition; data reported from manufacturer's product literature.

\* Not shown in figure.



TABLE 60. MEASUREMENT INFORMATION OF THE ELECTRICAL RESISTIVITY OF NICKEL + IRON ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Ni Fe	Composition (continued), Specifications, and Remarks
32	103	Clark, A. F., Childs, G. E., and Wallace, G. H.	1970	A	4-273	Fe-50 Ni	50.39 48	Composition reported for this nickel steel includes: 0.4 wt. % Mn, 0.28 wt. % Si, and lesser amounts of Cr, Co, C, Mo, P, and S; alloy described as being annealed and having an average grain size of 0.07 mm; specimen rough-machined from stock materials to dimensions of about 152 mm long x 6.35 mm diameter and subsequently ground flat with sides parallel to $\pm 0.005$ mm; specimen mount formed part of a dip probe to be lowered into temperature baths of liquid helium (4K), liquid hydrogen (20K), liquid nitrogen (76K), solid carbon dioxide (192K), and ice water (273K); data acquisition fully automated to record and average nineteen successive readings; for lower resistivity specimens a potentiometer and null detector were available enabling a resolution of a few $\times 10^{-8}$ volt.
33	103	Clark, A. F., et al.	1970	A	4-273	Fe-29 Ni	78.63 20	Composition reported for this nickel steel includes: 0.51 wt. % Mn, 0.19 wt. % Si, and lesser amounts of C, Mo, P, and S; alloy described as being hot-rolled and having an average grain size of 0.015 mm.
34	349	Omo, Y., and Yagi, T.	1972	R	1702-1898		61.0	Composition reported as 59.8 at. % Ni; in liquid state; alloy specimens prepared from weighed amounts of metals having purities 99.9% or better by melting in a vacuum-induction furnace; a specimen of about 10 g contained in a recrystallized alumina crucible, 10 mm in I.D. x 62 mm in height, was suspended by a tungsten wire 0.12 mm in diameter; specimen heated under reduced pressure of 0.05 to 0.1 mm Hg by a graphite heating element of coaxial design such that counter-flowing currents would contribute no net magnetic field within the specimen; density data from literature used to correct for volume thermal expansion; data fit to a linear temperature dependence the coefficients of which are reported as $\rho = 0.0232 T + 102.28$ in $\mu\Omega\text{cm}$ and $C^\circ$ respectively; overall probable error stated to be $\pm 0.5\%$ .
35*	349	Omo, Y., and Yagi, T.	1972	R	1707-1898		76.3	Composition reported as 74.4 at. % Ni; similar to above specimen except $\rho = 0.0259 T + 83.67$ in $\mu\Omega\text{cm}$ and $C^\circ$ respectively.
36	349	Omo, Y., and Yagi, T.	1972	R	1710-1898		80.3	Composition reported as 79.5 at. % Ni; similar to above specimen except $\rho = 0.0396 T + 61.46$ in $\mu\Omega\text{cm}$ and $C^\circ$ respectively.
37	335	Armstrong, B. E. and Fletcher, R.	1972	A	4-122		52.1 47.2	Composition originated from supplier and reported as 50.7 $\pm 0.6$ at. % Ni, 0.5 at. % Mn, 0.3 at. % C, 0.1 at. % Si, 0.07 at. % Co, 0.04 at. % Cu, and balance Fe; commercial grade alloy supplied in form of cold-rolled rods by Henry Wiggin and Co., Ltd., Hereford, England; specimen measured in as-received condition; electrical resistivity measured along with thermoelectric power using thermocouple leads as potential probes; data scatter at 0.5% level while overall uncertainty estimated to be $\pm 2\%$ ; data read from figures as temperature independent and dependent resistivities which were subsequently combined.
38	335	Armstrong, B. E. and Fletcher, R.	1972	A	4-121		78.9 24.1	Composition reported as 75.0 $\pm 1$ at. % Ni and balance Fe; alloy prepared from 99.9% starting materials; specimen heat-treatment consisted of heating to 1273 K for 1 hr and cooling to 873 K at 100 deg $\text{hr}^{-1}$ followed by a hold at 873 K for 8 hrs and water quenching; measurement procedure similar to that for above specimens.

\* Not shown in figure.

TABLE 80. MEASUREMENT INFORMATION OF THE ELECTRICAL RESISTIVITY OF NICKEL + IRON ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Ni Fe	Composition (continued), Specifications, and Remarks
30	346	Moore, J. P., Kollie, T. G., Graves, R. S., and McElroy, D. L.	1971	A	4-400	Ni <sub>4</sub> Fe:O1	75.4 24.2	Composition as reported includes <0.05 wt. % Al, 0.02 wt. % Cu, <0.02 wt. % Cr, <0.02 wt. % Mn, <0.02 wt. % V, <0.01 wt. % Si, <0.01 wt. % Ti, 0.0088 wt. % O <sub>2</sub> , 0.0023 wt. % N <sub>2</sub> , 0.0020 wt. % C, and 0.0002 wt. % H <sub>2</sub> ; alloy prepared from stock materials with contaminant levels no higher than 450 ppm (nickel stock) and 1050 ppm (iron stock); weighed amounts of metals melted repeatedly to insure homogenization and drop-cast into a 2.5 cm diameter, water-cooled copper mold; casting received surface machining and subsequently swaged to about 0.33 cm; 7.5 cm section of rod was cut and lapped to diameter tolerance of $\pm 0.00025$ ; rod wrapped in tantalum foil, encapsulated in a quartz tube under $10^{-4}$ mm Hg vacuum and heat-treated by annealing at 1375 K for 24 hrs followed by an ice-water quench; specimen considered to be completely disordered; most probable uncertainty in measured electrical resistivity was $\pm 0.3\%$ ; data presented in tabular form.
40	346	Moore, J. P., et al.	1971	A	4-400	Ni <sub>4</sub> Fe:O1		Above specimen heated to 1350 K under $10^{-4}$ mmHg vacuum during experiment and subsequently cooled to room temp at a rate of about 5 deg/min; specimen considered to have high degree of local order but no long-range order.
41*	346	Moore, T. P., et al.	1971	A	767-602	Ni <sub>4</sub> Fe:O2		Specimen cut from remaining half of ingot with composition being quite similar to above specimen; heat treatment consisted of heating to 1360 K under $10^{-4}$ mmHg vacuum and cooling to a soak temperature (767K) over 34-day period; cooling to soak temperature carried out such that no measurable ( $\pm 0.1\%$ ) change in electrical resistivity occurred during 24 hrs following 4 hr soak at a given temp; soaks applied at several temperatures below the order-disorder critical temperature of 711 K; soak times for the seven soak temperatures given here are 47, 28, 19, 25, 28, 12, and 21 days, respectively.
42*	346	Moore, T. P., et al.	1971	A	4-400			Above specimen measured following achievement of an essentially completely ordered state.
43	336	Gautier, F., and Loegel, B.	1972	A	4-60		39.8	Composition reported as 41 at. % Fe; alloy preparation reported by Cadeville and Loegel [35]; alloys prepared by melting 4X8 starting materials from Koch-light provenance in an induction furnace under argon atmosphere with the ingots subsequently annealed 1 day at 1323 K and quenched; crystalline structure at room temp is fcc as determined from x-ray measurements; data obtained from the expression $\rho = \rho_0 + 0.00019 T^2$ ( $\rho$ in $\mu\Omega\text{-cm}$ , $T \leq 60\text{K}$ ) reported by authors.
44	336	Gautier, F., and Loegel, B.	1972	A	4-60		42.8	Composition reported as 44 at. % Fe; preparation and measurement similar to above specimen; data obtained from the expression $\rho = \rho_0 + 0.00022 T^2$ ( $\rho$ in $\mu\Omega\text{-cm}$ , $T \leq 60\text{K}$ ) reported by authors.
45	336	Gautier, F., and Loegel, B.	1972	A	4-60		45.8	Composition reported as 47 at. % Fe; preparation and measurement similar to above specimen; data obtained from the expression $\rho = \rho_0 + 0.000265 T^2$ ( $\rho$ in $\mu\Omega\text{-cm}$ , $T \leq 60\text{K}$ ) reported by authors.
46	336	Gautier, F., and Loegel, B.	1972	A	4-60		48.8	Composition reported as 50 at. % Fe; preparation and measurement similar to above specimen; data obtained from the expression $\rho = \rho_0 + 0.00037 T^2$ ( $\rho$ in $\mu\Omega\text{-cm}$ , $T \leq 60\text{K}$ ) reported by authors.

\* Not shown in figure.

TABLE 60. MEASUREMENT INFORMATION OF THE ELECTRICAL RESISTIVITY OF NICKEL + IRON ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Ni Fe	Composition (continued), Specifications, and Remarks
47	327	Schwerer, F. C. and Conroy, J. W.	1971	A	4-297		0.48	Composition reported as 0.50 at. % Fe (premelting value); alloy prepared by arc-melting nickel and iron, each supplied by United Mineral and Chemical Corp., with nominal purity of 99.997%; ingot cold-rolled and machined/ground to cylinder 3.81 cm long x 0.25 cm diameter; specimens annealed at 1273 K for 1 wk; mass spectrographic analysis indicated impurity levels of starting materials and of alloy specimens were similar and with principal impurities being 20 ppm Cu and 0.003 wt. % C; residual resistivity of nickel standard of 0.013 $\mu\Omega\text{cm}$ thought due to dissolved carbon; geometrical factor determined from mass, density and length measurement to $\pm 0.2\%$ accuracy; data scatter reduced by cycling specimen through longitudinal magnetic field of $\sim 1$ kOe prior to measurement at each temperature; data reported as residual resistivity and deviations from Matthiessen's rule in figure format; following a reading of the figures the total resistivity was recovered and is given here.
48	327	Schwerer, F. C. and Conroy, J. W.	1971	A	4-297		0.85	Composition reported as 0.89 at. % Fe (premelting value); similar to the above specimen.
49	327	Schwerer, F. C. and Conroy, J. W.	1971	A	4-297		1.14	Composition reported as 1.20 at. % Fe (premelting value); similar to the above specimen.
50	327	Schwerer, F. C. and Conroy, J. W.	1971	A	4-302		1.7	Composition reported at 1.8 at. % Fe (average over two specimens); similar to the above specimen.
51	327	Schwerer, F. C. and Conroy, J. W.	1971	A	4-300		3.66	Composition reported as 3.85 at. % Fe (premelting value); similar to above specimen.
52	350	Baum, B. A., Tyagunov, G. V., Gel'd, P. V., and Khasin, G. A.	1971	R	1560-1979		66.1	Composition reported as 65 at. % Ni; data reported in both solid and liquid states in figure format; $d\rho/dT = 0.0020 \mu\Omega\text{cm K}^{-1}$ in liquid state; increasing temperature data.
53*	350	Baum, B. A., et al.	1971	R	1943-1481			Data show temperature hysteresis across melting range; above specimen measured at decreasing temperatures.
54	350	Baum, B. A., et al.	1971	R	1514-1995		85.6	Composition reported as 85 at. % Ni; data reported in both solid and liquid states; $d\rho/dT = 0.0032 \mu\Omega\text{cm K}^{-1}$ in liquid state; increasing temperature data.
55*	350	Baum, B. A., et al.	1971	R	1953-1423			Data show temperature hysteresis across melting range; above specimen measured at decreasing temperature.

\* Not shown in figure.

TABLE 60. MEASUREMENT INFORMATION OF THE ELECTRICAL RESISTIVITY OF NICKEL + IRON ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Ni	Fe	Composition (continued), Specifications, and Remarks
56	330	Shirakawa, Y.	1939	B	78-1123	No. 10		39.76	Alloys prepared from electrolytic iron and nickel supplied by Nippon-Denkai-Saitetsusho and by Monson and Co. respectively; chemical analysis of starting materials include, for iron: 0.03% Si, 0.03% P, 0.04% C, 0.02% Co, 0.02% Mn, 0.01% Al, 0.003% S, and for nickel: 0.02% Fe, 0.01% C, 0.003% S, 0.002% Si, 0.001% P, and 0.001% Ni; weighed amounts were mixed and then melted under hydrogen atm. in an alumina crucible within a Tamman furnace; wire formed by drawing molten alloy up into a silica capillary which was subsequently carefully broken; specimen annealed at 1273 K for 1 hr under vacuum and cooled slowly with length oriented along east-west direction; specimen dimensions were 6.22 cm length x 0.0320 cm dia; nickel lead wire soldered to each end of specimen with pure silver; specimen plus leads annealed again at 1123 K under high vacuum for 1 hr and slowly cooled; specimen orientation during measurement always east-west; magnetoresistance in longitudinal fields up to 1.6 kOe also measured; resistivity measurement technique is of two-probe type; data reported in tabular form.
57	330	Shirakawa, Y.	1939	B	78-1123	No. 11		29.85	Similar to above specimen except dimensions were 7.02 cm length x 0.0220 cm diameter.
58	330	Shirakawa, Y.	1939	B	78-1123	No. 12		19.93	Similar to above specimen except dimensions were 6.92 cm length x 0.0239 cm diameter.
59	330	Shirakawa, Y.	1939	B	78-1123	No. 13		18.15	Similar to above specimen except dimensions were 2.60 cm length x 0.0416 cm diameter.
60	330	Shirakawa, Y.	1939	B	78-1123	No. 14		10.21	Similar to above specimen except dimensions were 3.62 cm length x 0.0355 cm diameter.
61	338	Larkov, L.N., Usov, Yu.V., and Boychuk, I.N.	1977		4-1275		50.84		Measurement included both increasing and decreasing temperatures; data reported in figure format.
62	346	Szostkirmay, Za.	1976	A	143-340	FN-O	58.2		Composition reported for material designated FN-O includes: 0.01 wt.-% Si, <0.01 wt.-% Mn, 0.0174 wt.-% O, 0.0044 wt.-% C, and 0.0008 wt.-% S; alloys made at Caspel Metalworks, Budapest, from 4N nickel and 3.5 N iron starting materials by induction melting under vacuum, re-melting in an electron-beam furnace, and lastly cold-rolling with 95% reduction; compositions determined after final rolling; specimens cut parallel to rolling direction into dimensions of 43 x 5 x 0.22 mm <sup>3</sup> and subjected to a heat treatment sequence consisting of heating to 1273 K for 3 hrs, annealing at 973 K for 1 hr in vacuum, and oil-quenching; an annealing investigation was also carried out involving an anneal at 973K for 1 hr (T > T <sub>c</sub> ) and thereafter at 703 K for 3 hr (T > T <sub>c</sub> ordering), and lastly furnace cooled at rate of 100 deg hr <sup>-1</sup> ; three specimens from each alloy composition were annealed, one ea with zero magnetic field, or with saturation magnetic field applied either parallel or perpendicular to rolling direction; specimen strip mounted between two copper blocks which in turn were enclosed within a cryostat thermostated by circulating liquid nitrogen vapors for temps below room temp or by water for above; specimen temp cooling rate varied between 0.5 to 1.5 deg/min; uncertainty estimated to be ±0.04; quenched specimens data.

TABLE 60. MEASUREMENT INFORMATION OF ELECTRICAL RESISTIVITY OF NICKEL + IRON ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Ni Fe	Composition (continued), Specifications, and Remarks
63	348	Szentirmai, Zs.	1976	A	163-351	FN-L1	64.5	Composition reported includes 0.01 wt. % Si, <0.01 wt. % Mn, 0.0008 wt. % O, 0.0032 wt. % C, 0.0005 wt. % S, and 0.0022 wt. % Li; similar in other respects to above specimen.
64*	329	Schwerer, F.C. and Cuddy, L.J.	1970	A	4.2-942		0.52	Composition reported as 0.55 at. % Fe; alloys prepared from high-purity (99.997% nominal as supplied by United Mineral and Chemical Corp., N.Y.) by arc-melting under argon atmosphere and cold-rolling to 60% reduction; alloy sealed in quartz capsule under nitrogen atmosphere and annealed at 1273 K for 1 wk; specimen machined as 3.8 cm x 0.25 cm diameter cylinder and reannealed for 2 hr at 1073 K; solute concentration determined by chemical analysis; current and potential leads of nickel spot welded with gauge length measured by traveling microscope; average cross-section determined from length, density, and mass; largest sources of error thought to be geometrical factors for which uncertainties are <0.2%; errors in solute resistivity certainly less than $\pm 0.5\%$ of iron standard; high temperature measurements involved mounting specimen and nickel standard ( $\rho_0$ 0.024 $\mu\Omega\text{cm}$ ) in noninductively wound furnace in thermal proximity of each other and thermocouple well; specimen environment He-2% H <sub>2</sub> flowing atmosphere; temperature monitored with Pt/Pt-10% Rh thermocouple; electric current of 0.5 amp stabilized to $1:10^5$ applied; temperature changed at rate of 1 deg min <sup>-1</sup> ; voltages measured with Keithley 140 amplifier and digital voltmeter; complete set of measurements (including specimen, standard, and reversal) could be made within temperature change of $\pm 0.2$ deg; data reported as solute resistivity ( $= \rho_{\text{alloy}} - \rho_{\text{standard}}$ ) in figure format.
65*	329	Schwerer, F.C. and Cuddy, L.J.	1970	A	4.2-906		1.01	Composition reported as 1.06 at. % Fe; preparation and measurement similar to above specimen.

\* Not shown in figure.

TABLE 61 EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF NICKEL + IRON ALLOYS (Temperature Dependence)

Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$

T	$\rho$	T	$\rho$	Time(hrs)	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
<u>DATA SET 1</u>											
293	14.3	70.2	1.46*	0	59.2*	662	45.1	769	57.9	659	47.3
273	15.5	77.9	1.64*	25	61.0	696	49.0	783	61.2	691	51.8
373	20.0	89.1	1.92*	48	61.0	727	52.9			700	52.9
261	26.4	261	7.42*			758	57.4	<u>DATA SET 18</u>			
273	7.99					775	59.7*			719	56.3
<u>DATA SET 5</u>											
773	45.0					789	62.0*			748	61.1
773	55.2					828	68.0			778	66.1
673	62.8									798	69.7
973	66.3										
<u>DATA SET 2</u>											
4.18	3.78										
80.5	4.60										
92.7	13.22										
<u>DATA SET 3</u>											
4.2	0.307										
11.4	0.319*										
13.3	0.324*										
14.5	0.326*										
20.1	0.340*										
22.3	0.348*										
27.1	0.370*										
31.0	0.390*										
36.0	0.421*										
41.7	0.465*										
53.0	0.592*										
64	0.743*										
63.3	0.743*										
70.3	0.850*										
81.6	1.07*										
89.2	1.22										
<u>DATA SET 4</u>											
4.2	0.713										
12.5	0.723*										
14.5	0.731*										
16.3	0.737*										
22.0	0.762*										
30.5	0.815*										
35.4	0.868*										
40.5	0.916*										
43.6	0.964*										
52.9	1.13*										
59.2	1.31*										
<u>DATA SET 22*</u>											
675	48.6										
691	51.2										
700	52.6										
707	53.9										
716	55.2										
730	57.6										
739	58.9										
752	61.4										
756	62.0										
764	63.5										
772	64.8										
784	66.9										
801	70.0										
<u>DATA SET 23*</u>											
675	48.6										
691	51.2										
700	52.6										
707	53.9										
716	55.2										
730	57.6										
739	58.9										
752	61.4										
756	62.0										
764	63.5										
772	64.8										
784	66.9										
801	70.0										
<u>DATA SET 24*</u>											
671	47.1										
682	48.9										
691	50.2										
701	52.0										
708	53.5										
720	55.6										
736	58.6										
754	61.5										
761	62.6										
769	64.2										
775	65.1										
<u>DATA SET 25*</u>											
683	94.5										
701	97.3										
707	97.8										
730	101.2										
734	101.9										
749	104.0										
761	105.5										
<u>DATA SET 17*</u>											
666	35.0										
678	36.3										
686	37.4										
696	38.6										
706	39.8										
718	41.7										
726	42.9										
735	44.5										
744	46.6										
753	48.3										
763	50.6										
765	54.7										
<u>DATA SET 19*</u>											
675	42.2										
709	47.5										
727	50.2										
741	53.2										
771	59.0										
783	61.4										
803	64.2										
<u>DATA SET 20*</u>											
691	39.3										
707	41.4										
717	43.3										
726	44.7										
735	46.4										
750	50.8										
759	55.1										
775	59.8										
779	60.5										
796	63.1										
809	65.1										
814	66.0										
<u>DATA SET 21*</u>											
663	49.2										
696	53.7										
730	58.5										
762	63.2										
801	70.0										
<u>DATA SET 16*</u>											
691	39.3										
707	41.4										
717	43.3										
726	44.7										
735	46.4										
750	50.8										
759	55.1										
775	59.8										
779	60.5										
796	63.1										
809	65.1										
814	66.0										
<u>DATA SET 9*</u>											
0	56.7*										
16	55.2*										
42	53.6*										
67	51.7*										
113	50.7*										
164	49.6										
<u>DATA SET 13</u>											
0	52.0*										
27	49.0*										
46	47.9*										
94	46.3*										
117	46.8										
<u>DATA SET 7</u>											
77	12.8										
293	17.2										
<u>DATA SET 8*</u>											
293	15.8										

\* Not shown in figure.

**TABLE 61.**

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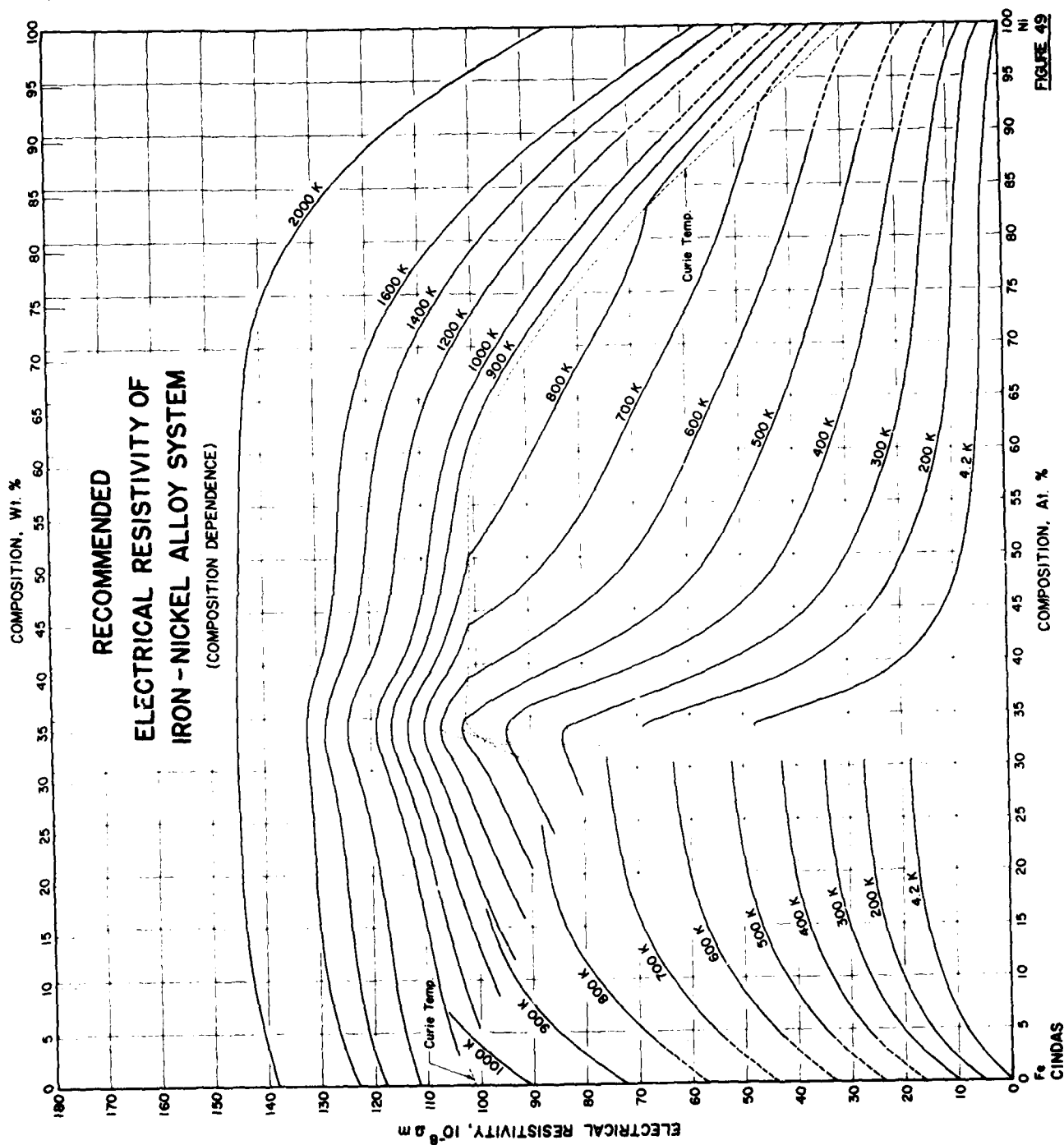
\* Not shown in figure.





TABLE 61. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF NICKEL + IRON ALLOYS (Temperature Dependence) (continued)

T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
DATA SET 59 (cont.)				DATA SET 61 (cont.)			
372	16.6	462	51	182	12.4	4.2	0.23
474	26.4*	492	56*	191	13.2*	300	0.51
573	34.9*	522	59	203	13.9*	331	0.50
672	45.0*	552	62*	212	14.8*	372	0.49
773	58.0*	574	68*	222	15.7	407	0.47
803	63.4	596	71	232	16.7*	445	0.45
815	65.9*	626	74*	242	17.5*	480	0.41
824	67.4*	656	78	251	18.2	509	0.39
833	67.9*	686	82*	261	19.4*	544	0.34
843	68.8	723	88	272	20.8*	576	0.33
873	70.2	753	94*	281	21.4*	603	0.30
921	72.1	775	97*	291	22.5	632	0.10
1023	75.8	812	100	300	23.7*	637	0.13
1123	78.4	842	103*	310	24.7*	640	0.36
DATA SET 60				321	26.2	649	0.53
78	5.11	909	107	330	27.2*	686	0.61
179	8.11	1029	110*	340	28.6	701	0.65
273	12.3	1096	112*	DATA SET 63			
290	12.8*	1155	112	163	8.9	754	0.66
319	14.4	1222	113*	173	9.5*	806	0.68
385	17.9	1275	113	183	10.2*	847	0.69
504	25.4	1352	113*	193	10.8*	885	0.70
629	36.0	1425	112*	204	11.5	942	0.72
723	46.3*	1466	112*	DATA SET 65*†			
752	50.4*	1506	109*	4.2	0.51		
763	52.4*	1552	107*	301	1.23		
768	53.6*	1606	106*	342	1.19		
773	53.9	1666	104*	380	1.17		
792	55.0*	1730	103*	429	1.19		
823	56.1*	1775	98*	478	1.15		
873	58.1	1848	91*	543	1.15		
923	60.0	1899	86*	601	1.03		
973	61.6	1952	77*	619	0.87		
1023	63.2	2006	71*	630	0.85		
1073	64.9*	2066	65*	636	1.22		
1123	66.5	2123	57*	642	1.48		
DATA SET 61				659	1.76		
4.2	9.1	395	41*	682	1.82		
89	11	336	32*	726	1.90		
313	30	DATA SET 62				781	1.95
358	36	143	9.8	834	2.04		
417	44	154	10.4*	906	2.14		
440	47*	163	11.1*				
		173	11.7*				



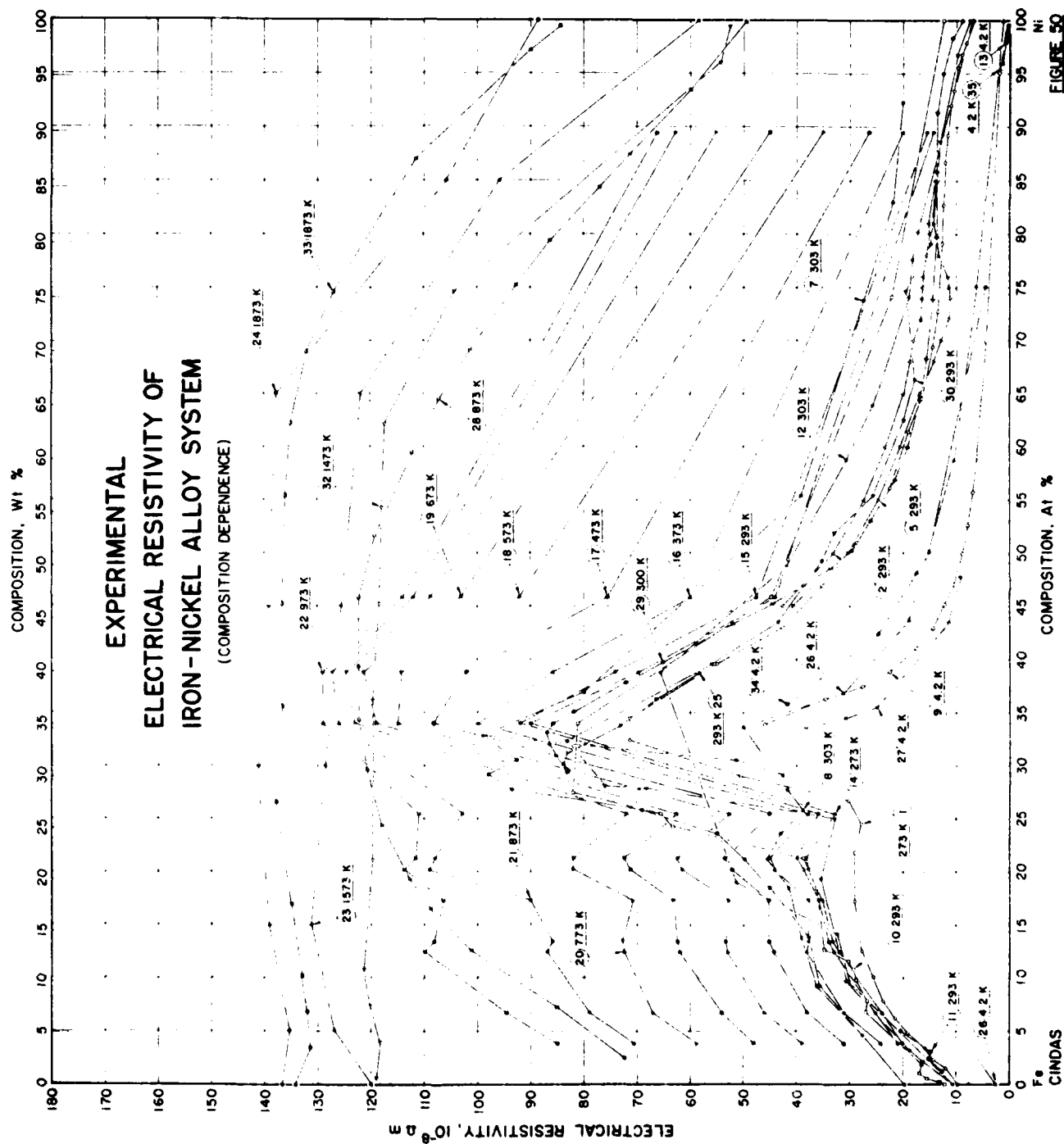


FIGURE 50

TABLE 62. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF IRON-NICKEL ALLOY SYSTEM (Composition Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp., K	Composition Range (At. % of Ni)	Composition (weight percent), Specifications, and Remarks
1	367	Brodeur, W. and Smolin, J.	1933		273	0-100	Alloys prepared from weighed amounts of electrolytic iron and nickel by melting under vacuum in 5 mm silica tube within high frequency furnace; specimen annealed 100 hrs at 1223 K in vacuum and subsequently slow-cooled to room temp in 24 hrs; photomicrographs indicated mixed phases occurring between the 28 and 32 percent nickel compositions; data presented as a smoothed curve of electrical conductivity in figure format.
2	347	Wakelin, R.J. and Yates, E.L.	1953	A	293	45-85	Alloys prepared from 99.94% pure nickel and iron containing 0.7% oxygen by melting weighed amounts in pure alumina crucibles under vacuum within a high-frequency furnace; rapid melting under pressure too low for an electrodeless discharge to occur resulted in minimal impurity uptake as evidenced by chemical analysis typically showing < 0.2% total impurity content; alloys hot-forged and homogenized at 1473 for 18 hrs; subsequent photomicrographs showed uniform composition throughout; specimens machined to dimensions of 10 cm x 4 cm x 1.1 mm with cross-section uniform to $\pm 0.02$ mm; reproducibility of measurements indicated $\pm 0.5\%$ uncertainty; specimen chamber temperature controlled to $\pm 0.5$ deg; specimen disordered by heating to 973 K and quenching.
3*	347	Wakelin, R.J. and Yates, E.L.	1953	A	293	45-85	Similar to above specimens except cooling rate from 973 K was 100 deg hr <sup>-1</sup> .
4*	347	Wakelin, R.J. and Yates, E.L.	1953	A	293	45-85	Similar to above specimens except cooling rate was 1.3 deg hr <sup>-1</sup> .
5	347	Wakelin, R.J. and Yates, E.L.	1953	A	293	45-85	Similar to above specimens except cooling rate was 0.28 deg hr <sup>-1</sup> .
6*	347	Wakelin, R.J. and Yates, E.L.	1953	A	293	75	Similar to above specimens except cooling rate was 0.09 deg hr <sup>-1</sup> .
7	354	Honda, K.	1918		303	0-92	Composition of commercial grade starting materials was for iron: 0.31% Mn, 0.25% C, 0.35% Mn, 0.29% C, 0.25% Co, 0.15% Si, and 0.017% S; alloys prepared by melting weighed amounts in a porcelain crucible within a Tamman electric furnace; procedure repeated if lengthwise microscopic examination revealed inhomogeneities; alloys forged, annealed, and filed into rod form of dimension 20 cm x 5 cm; three alloy samples chosen for nickel content determination which subsequently confirmed initial weight readings; specimen annealed at 1173 K and cooled to room temperature before measurement; data reported in tabular format.
8	354	Honda, K.	1918		303	0-33	Similar to above specimens except cooled in liquid air prior to measurement.
9	358	Cadeville, M.C. and Loegel, B.	1973	A	4.2	34-96	Alloys prepared by melting nickel and iron 4N8 from Koch-light provenance in an induction furnace under argon atmosphere; specimens subsequently annealed at 1323 K for 1 day and quenched; room temperature x-ray measurements indicate only fcc structure being present.
10	353	Burgess, C.F. and Aston, J.	1911		297	0.27-75.06	Alloys prepared "with every precaution to avoid contamination"; composition determined from chemical analysis; bars 1 cm in diameter were quenched from 1173K; knife edge potential probes held in place by spring clips and spaced 7.854 cm apart which makes the electrical resistivity numerically equal to the measured electrical resistance; data presented in tabular form and carried units of "microhms/cm <sup>2</sup> "; electrical measurements performed using a voltmeter-ammeter method to facilitate an extensive measurement program, admittedly less accurate than a bridge technique.
11	62	Gulyaev, A.P. and Trusova, E.F.	1950		293	1.2-5.0	Review article data reported in tabular format; temperature presumably a nominal room temperature.

\* Not shown in figure.

TABLE 62. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF IRON-NICKEL ALLOY SYSTEM (Composition Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp., K	Composition Range (At. % of Ni)	Composition (weight percent), Specifications, and Remarks
12	355	Bridgman, P. W.	1928	A	303	0-100	Alloys kindly given by McKeehan of Bell Telephone Laboratories; composition checks indicate actual contents are within $\pm 0.5\%$ of nominal composition; alloys in wire form having been hard drawn to 0.25 mm diameter; specimens given an anneal at 1273 K for 1 hr under vacuum and cooled slowly (several hrs) to room temperature; effects of pressure upon electrical resistance also reported; data presented in table and figure format giving electrical resistivity and temperature and pressure coefficients; uncertainty due largely to wire diameter measurement and estimated to be $\pm 1\%$ to $\pm 2\%$ .
13	327	Schwerer, F. C. and Courroy, J. W.	1971	A	4.2	96-100	Alloys prepared by arc-melting nickel and iron, each supplied by United Mineral and Chemical Corp. and of nominal purity 99.997%; ingots cold-rolled and machined to cylinders 3.81 cm long x 0.25 cm diameter; specimens sealed in quartz capsules under purified nitrogen atmospheres and annealed at 1273 K for 1 wk; mass spectrographic analyses indicated that impurity levels were similar for starting materials and alloy specimens; principal metallic impurities were Cu at 0.0020 wt.%, level and C at 0.003 wt.%; compositions given here are premelt values; residual resistivity of nickel standard thought due to dissolved carbon; geometrical factors determined from density, mass, and length measurements to about $\pm 0.2\%$ ; data scatter reduced by cycling specimens through longitudinal magnetic fields of $\sim 1\text{ kOe}$ at each temperature prior to measurement; uncertainty thought to be about $\pm 0.002 \mu\Omega\text{cm}$ .
14	333	Ingersoll, L. R.	1920	A	273	3.8-69.5	Composition includes an estimated $< 0.10\%$ C; alloys prepared from doubly electrodeposited iron of 99.97% purity and electrolytic nickel of high purity; ingots formed from weighed amounts by melting in a magnesia crucible heated within a resistor furnace and subsequently forging and machining into 1 cm bars; a measuring current of some 7.5 amps was applied; temperatures were measured with a copper-constantan thermocouple; smoothed values were reported in figure format; data apparently taken with increasing temperature only; possible transformation to austenite at highest temperatures for the $\leq 30$ at. % Ni alloys was not discussed.
15	332	Ingersoll, L. R.	1920	A	293	3.8-69.5	The above specimens.
16	332	Ingersoll, L. R.	1920	A	373	3.8-69.5	The above specimens.
17	332	Ingersoll, L. R.	1920	A	473	3.8-69.5	The above specimens.
18	332	Ingersoll, L. R.	1920	A	573	3.8-69.5	The above specimens.
19	332	Ingersoll, L. R.	1920	A	673	3.8-69.5	The above specimens.
20	332	Ingersoll, L. R.	1920	A	773	3.8-69.5	The above specimens.
21	332	Ingersoll, L. R.	1920	A	873	3.8-69.5	The above specimens.
22	332	Ingersoll, L. R.	1920	A	973	3.8-69.5	The above specimens.
23	350	Baum, B. A., Tygunov, G. V., Gel'd, P. V., and Khasin, G. A.	1971	R	1573	5-100	Specimens in solid state.
24	350	Baum, B. A., et al.	1971	R	1873	0-100	Specimens in liquid state.
25	341	Kalinin, V. M., Damllov, M. A., Komarova, L. K., and Tseytla, A. M.	1973		293	29.5-36.7	Alloy specimens prepared as cylinders with dimensions 150 mm long x 12 mm diameter; specimens annealed at 1173 K for 2 hrs under $10^{-4}$ mmHg vacuum prior to measurement; temperature presumably a nominal room temperature.

TABLE 63. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF IRON-NICKEL ALLOY SYSTEM (Composition Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp., K	Composition Range (At. % of Ni)	Composition (weight percent), Specifications, and Remarks
26	339	Kondorakii, E. I., and Sedov, V. L.	1960		4.2	0-100	Specimen rods of dimension 112 mm long x 5.9 mm diameter fabricated from technically pure starting materials; prior to measurement specimens vacuum annealed at 1273 K for 8 hrs and furnace cooled; pressure dependence of resist- dual resistance also reported in hydrostatic pressure range; data presented in tabular form.
27	338	Larikov, I. N., Usoy, Yu. V., and Boychuk, I. N.	1977		4.2	34.64-49.59	Compositions reported as 35.68, 36.9, 37.64, 44.86, 50.84 wt.%; data reported in figure format.
28	342	Window, B.	1973		873	2.5-99.9	No experimental details reported; smoothed data presented in figure format.
29	342	Window, B.	1973		300	29.2-100	Similar to above specimen.
30	340	Jellinghaus, W., and de Andr�s, M. P.	1960	B	293	0-100	Alloys prepared from carbonyl iron and carbonyl nickel powder; composition of starting materials included $\leq 0.01\%$ each C, Si, and Mn; ingot cold worked to 0.3 mm thickness and subsequently annealed under hydrogen atmosphere at 1423 K for 2 hrs; specimen cut to 150 mm length x 15 mm width; instrumentation incorporated a Thomson bridge and a standard resistance; data presented in tabular form.
31*	340	Jellinghaus, W. and de Andr�s, M. P.	1960	B	293	0-100	Selected specimens from those above subjected to 80%-90% cold-work.
32	351	Epia, V. N., Baum, B. A., and Tyagunov, G. V.	1977	R	1473	0-100	No experimental details reported; specimens in solid state; data presented in figure format.
33	351	Epia, V. N., et al.	1977	R	1873	0-100	No experimental details reported; specimens in liquid state; data presented in figure format along with $dp/dT$ for liquid state; $\rho_{liquid}/\rho_{solid}$ for these same specimens.
34	335	Armstrong, B. E. and Fletcher, R.	1972	A	4.2	35.8-75.0	Alloys of commercial grade (with exception of 75 at. % Ni) and supplied by Henry Wiggin and Co., Hereford, England, in form of cold-rolled rods sub- sequently measured in as-received condition; 75 at. % Ni alloy prepared from 99.9% pure starting materials and subsequently heat-treated to 1273 K for 1 hr and cooled to 873 K at 100 deg hr <sup>-1</sup> followed by a hold at 873 K for 8 hrs and water quenching; data scatter at 0.2% level while overall uncertainty esti- mated to be $\pm 2\%$ ; data presented in figure format.
35	326	Farrell, T. and Greig, D.	1968	A	4.2	95.4-99.2	Alloys prepared by chill-casting under vacuum; iron content obtained from x-ray fluorescence measurement upon the two more dilute alloys and from residual resistivity for the other; geometrical factor determined to better than $\pm 0.6\%$ from length, density, and mass measurements; overall uncertainty estimated to be no more than $\pm 1\%$ ; data presented in tabular form.

\* Not shown in figure.

TABLE 62. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF IRON-NICKEL ALLOY SYSTEM (Composition Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp., K	Composition Range (At. % of Ni)	Composition (weight percent), Specifications, and Remarks
36*	325	Fert, A. and Campbell, I.A.	1976	A	4.2	1, 2, 97-100	Alloys prepared at the CEN, Grenoble, in a levitation furnace from base metals obtained by zone melting (RRR = 700 and 25 for Ni and Fe respectively); alloys drawn into wires of 1 mm diameter and subsequently annealed at 1173 K for the nickel based alloys or 1073 K for the iron based alloys; absence of precipitates checked by micrography; most alloys analyzed by calorimetry or atomic absorption spectroscopy; resistivity measured between 1.3 K and 77 K and at several temperatures between 77 K and 300 K; accuracy generally limited (except for less concentrated alloys) by current stability of $\pm 2 \times 10^{-4}$ ; temperature in helium range measured with a germanium resistor with an accuracy of $\pm 0.1$ deg; data presented in tabular form except for the nickel base alloys for which residual resistivities were recovered by reading compositions from a figure and combined with a tabulated $\rho_0$ /at. % Fe value.
37*	325	Fert, A. and Campbell, I.A.	1976	A	24	97-100	Specimens same as the above; total electrical resistivities recovered from residual resistivities above combined with temperature dependent values at 24 K as read from a figure.

\* Not shown in figure.

TABLE 63. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF IRON-NICKEL ALLOY SYSTEM (Composition Dependence)  
[Composition, C, at/o Ni; Electrical Resistivity,  $\rho$ ,  $10^{-8}$  Qm]

DATA SET 1 (T = 273 K)		DATA SET 2 (cont.) (T = 293 K)		DATA SET 4 (cont.) (T = 293 K)		DATA SET 7 (cont.) (T = 303 K)		DATA SET 10 (cont.) (T = 297 K)		DATA SET 14 (T = 273 K)	
C	$\rho$	C	$\rho$	C	$\rho$	C	$\rho$	C	$\rho$	C	$\rho$
0.0	9	70	18.1	81	14.2	83.0	21.8	6.73	26.9	3.8	20.1
1.5	11.9	72	18.7	85	13.5	92.3	20.0	7.80	26.7	6.7	23.9
2.3	14.3	74	16.5					9.75	28.6	12.4	32.0
3.4	15.9	75	16.4	DATA SET 5 (T = 293 K)		DATA SET 8 (T = 303 K)		10.80	29.4	13.4	33.2
4.8	19.1	76	16.2					11.55	30.3	17.3	35.2
6.2	21.4	78	15.5	45	40.6	0	19.8	12.55	34.8	20.2	37.8
7.8	23.7	80	15.0	50	30.3	4.6	27.5	16.45	36.2	21.26	38.3
10.1	25.6	81	15.1	55	24.6	9.2	36.2	21.26	38.7	25.44	32.8
12.5	27.7	85	14.0	60	20.5	13.8	39.1	24.27	33.2	33.96	90.0
17.2	29.1			65	16.9	18.5	41.3	25.44	65.5	38.8	69.7
21.9	29.1	DATA SET 3 <sup>a</sup> (T = 293 K)		68	14.2	81.2	45.5	27.41	82.0	45.84	44.0
24.5	27.8			70	12.9	23.6	42.9	45.84	44.7	89.5	14.3
26.7	30.2			72	11.4	25.8	38.8	74.11	22.1		
29.0	50.5	45	41.4	74	11.1	27.7	41.7			DATA SET 15 (T = 293 K)	
32.4	71.4	50	31.6	75	11.3	29.1	42.6	DATA SET 11 (T = 293 K)		3.8	20.9
34.5	71.4	55	25.4	76	11.5	30.5	51.3	1.2	12.6	6.7	25.2
38.8	54.9	60	21.3	78	13.2	32.8	99.0	2.4	15.3	12.4	33.0
43.0	42.6	65	18.2	80	13.7			3.0	14.7	17.3	35.9
46.9	35.3	68	16.9	81	14.2	DATA SET 9 (T = 4.2 K)		5.0	20.4	20.2	38.8
51.0	28.7	70	16.3	85	13.5	33.9	46.0	DATA SET 12 (T = 303 K)		21.26	40.0
56.1	22.4	72	15.3			36.2	34.5	25.44	35.9	33.96	92.0
61.4	19.0	74	15.5	DATA SET 6 <sup>a</sup> (T = 293 K)		37.4	27.5	38.8	65.6	38.8	74.1
64.5	16.5	75	15.0	75	8.86	38.9	22.3	45.84	47.5	45.84	47.5
69.4	14.4	78	14.8	DATA SET 7 (T = 303 K)		41.0	17.4	0	10.5	89.5	15.5
72.7	13.4	80	14.2			42.9	14.3	24.1	32.4		
79.1	12.6	81	14.4			45.1	12.1	28.0	51.4	DATA SET 16 (T = 373 K)	
82.7	12.3	85	13.6	0	19.8	47.0	10.4	38.8	65.6	3.8	24.1
88.2	11.6			4.6	27.6	49.9	8.8	48.8	41.9	6.7	31.1
92.0	11.0	DATA SET 4 <sup>a</sup> (T = 293 K)		9.2	35.6	52.5	7.7	58.8	30.7	12.4	38.2
93.4	10.4	45	40.7	13.8	37.7	55.7	6.8	63.9	25.7	13.4	37.8
95.6	9.7	50	30.4	18.5	45.0	59.0	6.1	74.1	18.9	17.3	37.8
96.8	8.9	55	24.6	21.2	49.8	69.9	4.4	80.2	17.1	20.2	44.1
97.9	8.0	60	20.6	23.6	54.9	83.1	0.6	100	7.0	21.26	45.0
100.0	6.7	65	17.2	25.8	69.0			DATA SET 13 (T = 4.2 K)		33.96	100.0
		68	15.7	27.7	93.5	0.26	13.1	96.15	1.328	38.8	85.8
		70	14.4	29.1	98.0	0.53	15.4	98.03	0.683	45.84	60.0
		72	13.0	30.5	92.6	1.02	16.9	98.80	0.473	89.5	20.0
		74	13.0	32.8	99.0	1.64	16.4	99.11	0.349		
		75	13.0	36.9	80.0			99.50	0.202		
		76	13.3	55.4	39.1						
		78	13.8	73.8	27.4						
		80	13.7								

<sup>a</sup> Not shown in figure.



\* Not shown in figure.

TABLE 63. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF IRON-NICKEL ALLOY SYSTEM (Composition Dependence) (continued)

C	$\rho$
<u>DATA SET 34</u> ( $T = 4.2$ K)	
35.8	41.7
42.4	24.6
48.1	17.2
50.7	15.1
75.0	4.6
<u>DATA SET 35</u> ( $T = 4.2$ K)	
85.4	1.80
96.2	0.713
99.2	0.307
<u>DATA SET 36*</u> ( $T = 4.2$ K)	
1	1.9
2	2.6
97.05	1.032
98.94	0.370
99.80	0.070
<u>DATA SET 37*</u> ( $T = 24$ K)	
97.05	1.056
98.94	0.424
99.80	0.128

\* Not shown in figure.

### 3.10. Silver-Palladium Alloy System

The preponderance of available evidence [264,289,321] suggests that the silver-palladium alloy system forms a continuous series of solid solutions without the formation of long-range ordered structure. However, the results of electrical resistivity measurements by Savitskii and Pravoverov [368,369] have suggested to the contrary, which will be discussed later in detail as it bears directly on the evaluation of electrical resistivity data for these alloys.

Investigations of the effect of plastic deformation on the electrical resistivity of silver-palladium alloys have been undertaken by Aarts and Houston-MacMillan [370], Rao [371], Nicholson and co-workers [372-374], and others and have shown an anomalous decrease of resistivity with increasing deformation. These investigations, together with isochromal and isothermal annealing studies of recovery processes and Hall constant measurements, have been interpreted (see, for example, ref. [373]) as showing the existence of short-range ordering in the silver-palladium alloy system. No x-ray evidence for the existence of short-range order in these alloys exists, probably because the scattering factors for silver and palladium atoms are nearly the same. In an x-ray investigation of a range of alloy compositions, Rao and Rao [375] found that the variation of lattice parameter with temperature deviates from a straight line, but concluded that their results could not be explained by the formation of short-range order.

There are 123 sets of data available for the electrical resistivity of this alloy system. These experimental data sets are listed in tables 65, 67, and 69 which provide information on specimen characterization and measurement conditions, tabulated in tables 66, 68, and 70, and shown partially in figures 53, 54, and 56.

Resistivity values reported for Ag-Pd alloys show striking discrepancies. Residual resistivities given for alloys with 50% silver differ by as much as 13%, and the room-temperature resistivities by 35%. Large differences occur even within the work of single authors; for instance, Kemp et al. [376] (Ag + Pd data sets 41 and 42) report for two samples with 60% silver (wire and rod, respectively) residual resistivities that differ by over 5%, and changes in resistivity to 300 K that differ by 25%. The discrepancies are such as to

indicate that the preparation of Ag-Pd alloys of known composition, good homogeneity, and predictable resistivity offers serious (and perhaps unexpected) difficulties.

Pravoverov and Savitskii [368,369] (Ag + Pd data sets 58-61, Pd + Ag data sets 32-35, and Ag-Pd data set 23) have reported measurements on Ag-Pd alloys subjected to very prolonged annealing (124 hours at 1273 K, followed by 200 hours at 1073 K), with results very different from those of other workers. For alloys with over 70% silver they report essentially the same room-temperature resistivities as do other workers, but for alloys richer in palladium their values are much lower. A plot of their resistivity values against the atomic fraction of silver clearly indicates a downward cusp at 40% silver, where they obtain  $\rho_{300} = 28 \times 10^{-8} \Omega\text{m}$ , about 2/3 as large as the values obtained by others; in addition, their value for 50% silver is quite low, suggesting the presence of a kink, if not a cusp, at this composition. Plots of microhardness against composition show well-defined downward cusps at both compositions. Pravoverov and Savitskii attribute these features to the existence of ordered structures corresponding to the compounds  $\text{Pd}_3\text{Ag}_2$  and  $\text{PdAg}$ . Samples quenched from 1123 K, 1273 K, and 1373 K show progressive lessening of these features, and they are entirely absent from a plot of the resistivities of samples quenched from 1473 K, as they are from a plot of microhardness against composition. The results of Pravoverov and Savitskii provide a clear indication that the order in Pd + Ag alloys can increase slowly during prolonged anneals, becoming quite high for certain composition, and that the heat treatment of these alloys can markedly affect their electrical resistivity. The thermal histories of samples used by other workers in this field are often inadequately described, but they seem to be diverse enough to account for a good deal of the scatter in the reported results.

The difference of the results of Pravoverov and Savitskii from those of all other workers with Pd + Ag alloys is so great as to suggest that they are dealing with an essentially different, more ordered, material. Unfortunately, there is also some question as to the accuracy of their measurements. In the case of the alloy with 15% palladium, their room-temperature resistivity fits in well with the results of others, but their reported increase in resistivity with temperature is less than half that reported by a number of other workers for Ag-rich alloys. Their experimental arrangement for measurements at high

temperatures, though inadequately described, appears to be unconventional, and may be responsible for the fact that their high-temperature resistivities are consistently lower than those of other workers. It is not clear whether this can have any relation to the low values that they find for Pd-rich alloys at room temperature. At any rate, their data are not used in our analysis.

The Ag-Pd system is of great theoretical interest. Mott and Jones [6, p. 297ff.] pointed out that the d-states of the transition metal Pd have little effect on electrical conduction in the silver-rich alloys, but that electron scattering into these states comes to be the dominating factor, and produces a striking change in the behavior of the alloys as the concentration of Pd increases. A useful survey of the problem is given by Mott [377, p. 372ff.]. A number of calculations have been based on the model of Mott and Jones, but their success has been mainly qualitative rather than quantitative. It is now evident why the behavior of the system is so complex, but the theory has suggested no useful analytic form for representation of the residual resistivities of the system, or of the even more striking peculiarities of the temperature-dependent part of the resistivities,  $\rho - \rho_0$ , which are illustrated in figure 55.

The difficulty of making a rational choice between divergent data without the assistance of a suitable theory has made it difficult (despite the relative abundance of data) to arrive at recommended values for the electrical resistivity that have small limits of error. The division of the total resistivity, for purposes of analysis, into residual and temperature-dependent parts is less useful for the attainment of accuracy in the total than it is when the parts exhibit a simpler behavior. This approach has been used, however, to obtain recommended values that reflect, as well as may be, those features in the behavior of the temperature-dependent part that are reliably indicated by the data. It is believed that the trends in the temperature-dependent part of the resistivity are more accurately represented by the recommended values than might be suggested by the error limits (absolute, not fractional) assigned to the total resistivities.

The greater part of the available data relates to residual and room-temperature resistivities. The difference between these quantities can best be determined from measurements of the two quantities made on a single sample, since this minimizes the effect of many errors, including those of sample

characterization. In our analysis the data of Kemp et al. [376] (Ag + Pd data sets 29-44, Pd + Ag data sets 13-18, and Ag-Pd data sets 10-13), Ricker and Pflüger [378] (Ag + Pd data sets 11-15 and Pd + Ag data sets 3-6), and Coles and Taylor [45] (Pd + Ag data set 1 and Ag-Pd data set 3) were used in determining values for  $\rho(c, 300 \text{ K}) - \rho(c, 0 \text{ K})$  over the entire range of  $c$ . (The wire data of Kemp et al. were given preference over their rod data, since only the wire data extended to intermediate temperatures.) The individual contours for  $\rho(c, 0 \text{ K})$  and  $\rho(c, 300 \text{ K})$  were then determined from the entire mass of data at liquid-helium and room temperatures, subject to the constraint of their established separation. This correlation of low- and room-temperature data in determining  $\rho(c, 0 \text{ K})$  resolved some problems of choice, and appeared to give a significant improvement in the accuracy with which  $\rho(c, 0 \text{ K})$  could be determined.

Below room temperature, the data of Kemp et al. [376] gives a reasonable coverage of the silver-rich alloys, but leaves serious gaps between 5% and 30% silver and between 30% and 50% silver. The second of these gaps is filled by the data of Ahmad and Greig [175] (Pd + Ag data set 22) and of Coles and Taylor [45]. In the first gap one must use the data of Ricker and Pflüger [378], which is scantier and less well presented. Ricker and Pflüger made no measurements below liquid-air temperature, and they apparently extrapolated their curves to 0 K in rather rough fashion. Their data for each of their samples, as read from their curves, have been adjusted to fit our recommended values for  $\rho(c, 300 \text{ K})$  by addition of an appropriate constant, with results that are useful down to about 100 K. At temperatures below 40 K the work of Murani [379] (Ag + Pd data sets 52-56 and Pd + Ag data sets 19-21), Edwards et al. [380] (Ag + Pd data sets 23-25 and Pd + Ag data sets 7-10), and Chen et al. [381] (Ag + Pd data sets 7-10) has provided values of  $\rho_1$  (not always concordant) for a wide range of compositions. (The values given by Murani for alloys with 62% and 70% silver are quite out of line with other results, and have been ignored.) The work of Greig and Rowlands [382, 383] (Pd + Ag data sets 23-26) provides values of  $\rho_1$  for dilute alloys with up to 2% Ag in Pd, and the work of Schroeder et al. [180] (Ag + Pd data sets 1-4) gives values of  $\rho_1$  for up to 10% Pd in Ag.

In general, the analysis of these data began with the construction of smoothed temperature plots of  $\rho_1$  for the individual samples. In the case of

dilute alloys, interpolation and smoothing was facilitated by working with the ratio of the reported  $\rho_1$  to that of the pure metal, or its differences from that of the pure metal - that is, the deviation from Matthiessen's rule (DMR). Values read from these curves were then used in constructing a family of isotherms of  $\rho_1$  over the whole range of composition, from which were read values for the compositions to appear in the tables. Finally, the values thus obtained were used in constructing smoothed plots against  $T$  of  $\rho_1$  (or of a ratio or difference), from which values were read for the temperatures to appear in the table.

It may be mentioned that Kemp et al. considered the deviations from Matthiessen's rule to be insignificant for up to 30% Pd in Ag. Schroeder et al., however, found small but measurable values for the DMR appearing rather abruptly in their sample with 10% palladium; they attribute this to disengagement of the Fermi surface from the zone boundary. Since the data of Coles and Taylor support the conclusion of Kemp et al. that the DMR remains very small for up to 30% palladium, we have used the data of Schroeder et al. for alloys with up to 10% palladium (which indicates that the DMR is negligible for alloys with less than 5% palladium) and have taken the DMR to be constant thereafter for alloys with up to 30% palladium, except at temperatures below 100 K, where the very-low-temperature isotherms indicate some variation in this practically negligible quantity.

The extension of the analysis to temperatures above 300 K had to be based primarily on the data of Ricker and Pflüger, normalized to the recommended values for 300 K, as noted above. For concentrated alloys there exist also the data of Coles and Taylor [45] (Pd + Ag data set 1) for an alloy with 39% silver, and the data of Ahmad and Greig [15] (Pd + Ag data set 22) for an alloy with 40% silver. Their results are in reasonable agreement with those of Ricker and Pflüger, and the results of Ahmad and Greig were actually given preference for the 40% Ag alloys, since they provide a value of  $\rho_0$  measured on the same sample. Measurements of Araj et al. [384] (Ag + Pd data set 57, Pd + Ag data sets 27-31, and Ag-Pd data set 25), still unpublished when this analysis was made, are in reasonable agreement with other results at room temperature, but indicate a much greater increase in resistivity at high temperatures.

No data exists for dilute solutions of Ag in Pd above room temperature. The interpolation between results for pure palladium and an alloy with 10% Ag was made by assuming that there would be a continuation of the tendency, indicated by the low-T data, for the range of concentration in which the DMR is positive to decrease to insignificance as T rises.

The data for dilute (<30%) solutions of Pd in Ag offers some problems. The data of Schroeder et al. [180] indicate that the DMR of a 10 percent alloy is positive and essentially constant from 100 K to 250 K. On the other hand, the data of Ricker and Pflüger for  $T > 300$  K yield  $\rho_1$ -plots for alloys with 10, 20, and 30% palladium that sag markedly below the pure-silver plot as temperature rises, yielding negative DMR but at the same time indicating that the DMR is increasing algebraically over this range. If this were true, it would mean that above 300 K the DMR becomes negative and fairly large (over  $0.5 \times 10^{-8} \Omega m$ ) with increasing palladium concentration, but that this tendency is sharply reversed before 10% palladium is reached, with the DMR increasing thereafter. The data of Otter [226] (Ag + Pd data sets 19-22) for alloys with 1, 2, 3, and 4% palladium, though not obtainable from his figures with high accuracy, is inconsistent with this idea. We have accepted as working hypothesis the idea that through some systematic error the resistivity values of Ricker and Pflüger tend to be too low at high T, with the error greater the lower the total resistivity of the specimen. We have therefore followed the suggestion of the low-T data in assuming that the DMR of these alloys is positive and constant for all  $T > 150$  K. A corresponding small modification, well within the limits of error, has been made in the high-T data of Ricker and Pflüger for alloys with 40% and 50% palladium.

From this point, the analysis followed the lines indicated above for the low-T data. The 300 K isotherm of  $\rho_1$ , which equals  $(\rho - \rho_0)$ , derived from low-temperature data and illustrated in figure 55, shows a minimum for about 40% silver and a maximum for about 52% silver, as was noted earlier by Coles and Taylor. As the temperature rises, these features of the isotherms are increasingly marked. Addition of  $\rho_0$  to  $\rho_1$  yields isotherms of  $\rho$  that are increasingly flattened for silver concentrations around 40%, and the 1100 K isotherm actually has a slight dimple in this region. Changes in the analysis of the existing data can decrease the depth of this dimple, but the existence of the deep minima in the isotherms of  $\rho_1$ , and of the flattening in the isotherms of  $\rho$ , is indicated



by all data. This unexpected form of the isotherms of  $\rho$  would certainly make trouble in an analysis based on direct consideration of  $\rho$ , and might well lead to some smoothing out of a real feature of the isotherms.

The resulting recommended electrical resistivity values for Ag, Pd, and for 25 Ag-Pd binary alloys are presented in table 64 and shown in figures 51, 52, and 56. The recommended values for Ag and for Pd are for well-annealed high-purity specimens, but those values for temperatures below about 100 K are applicable only to Ag and Pd having residual electrical resistivities as given at 1 K in table 64. The recommended values for Ag-Pd alloys are intended to apply to samples that are well homogenized and are annealed to remove the effects of cold work, but that have not been subjected to very prolonged annealing such as that which seems to have produced the relatively ordered alloys measured by Pravoverov and Savitskii. This is hardly an ideal characterization of a material, but no more precise one can be given for the materials from which our recommended values have been derived. The recommended values cover the temperature range from 1 K to 1100 K, and are not corrected for the thermal expansion of the material. The uncertainties assigned to the recommended values, as indicated in a footnote to table 64, are intended to represent the possible effects of measurement errors and of other difficulties in our analysis, and also include some allowance for differences in the structure of different carefully prepared samples. This latter allowance is based on the scatter in existing data but can not be precisely defined; certainly it does not define a range that includes all existing results, or all future ones. A better characterization of these alloys, and some precise reference values, are much to be desired.

TABLE 64. RECOMMENDED ELECTRICAL RESISTIVITY OF SILVER-PALLADIUM ALLOY SYSTEM†  
[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Ag: 100.00% (100.00 At. %) Pd: 0.00% (0.00 At. %)		Ag: 99.50% (99.49 At. %) Pd: 0.50% (0.51 At. %)		Ag: 99.00% (98.99 At. %) Pd: 1.00% (1.01 At. %)		Ag: 97.00% (96.96 At. %) Pd: 3.00% (3.04 At. %)		Ag: 95.00% (94.93 At. %) Pd: 5.00% (5.07 At. %)		Ag: 90.00% (89.88 At. %) Pd: 10.00% (10.12 At. %)	
T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
1	0.00100	1	0.210	1	0.421	1	1.262*	1	2.102*	1	4.20
4	0.00100	4	0.210	4	0.421	4	1.262	4	2.102	4	4.20
7	0.00103	7	0.210	7	0.421	7	1.262	7	2.102	7	4.20
10	0.00115	10	0.210	10	0.421	10	1.262	10	2.103	10	4.20
15	0.00190	15	0.211	15	0.422	15	1.263	15	2.105	15	4.21
20	0.00423	20	0.213	20	0.424	20	1.265	20	2.110	20	4.22
25	0.00959	25	0.218	25	0.429	25	1.270	25	2.118	25	4.23
30	0.0195	30	0.229	30	0.440	30	1.281	30	2.129	30	4.24
40	0.0541	40	0.263	40	0.474	40	1.315	40	2.164	40	4.29
50	0.104	50	0.313	50	0.524	50	1.365	50	2.213	50	4.36
60	0.163	60	0.372	60	0.583	60	1.424	60	2.269	60	4.43
70	0.226	70	0.435	70	0.646	70	1.487	70	2.329	70	4.50
80	0.290	80	0.499	80	0.710	80	1.551	80	2.396	80	4.57
90	0.355	90	0.564	90	0.775	90	1.616	90	2.463	90	4.65
100	0.419	100	0.628	100	0.839	100	1.680	100	2.528	100	4.72
150	0.728	150	0.938	150	1.149	150	1.890	150	2.838	150	5.07
200	1.031	200	1.240	200	1.451	200	2.292	200	3.140	200	5.38
250	1.330	250	1.539	250	1.750	250	2.59	250	3.44	250	5.68
273	1.468	273	1.676	273	1.891	273	2.73	273	3.58	273	5.82
293	1.587	293	1.796	293	2.007	293	2.85	293	3.70	293	5.94
300	1.629	300	1.838	300	2.049	300	2.89	300	3.74	300	5.98
350	1.930	350	2.14	350	2.35	350	3.19	350	4.04	350	6.28
400	2.236	400	2.45*	400	2.66	400	3.50	400	4.34	400	6.59
500	2.863	500	3.07*	500	3.28	500	4.12	500	4.97	500	7.21
600	3.509	600	3.72*	600	3.93	600	4.77	600	5.62	600	7.86
700	4.174	700	4.38*	700	4.59	700	5.43	700	6.28	700	8.52
800	4.860	800	5.07*	800	5.28*	800	6.12*	800	6.96*	800	9.21
900	5.565	900	5.77*	900	5.98*	900	6.83*	900	7.64*	900	9.91
1000	6.297	1000	6.51*	1000	6.72*	1000	7.56*	1000	8.38*	1000	10.64
1100	7.085	1100	7.29*	1100	7.50*	1100	8.35*	1100	9.20*	1100	11.43
1200	7.922										
1235.08	8.233(s)										
1236	17.31(t)										
1400	18.69										
1600	20.38										

† Uncertainties in the electrical resistivity values are as follows:

100.00 Ag - 0.00 Pd:  $\pm 1\%$  up to 10 K,  $\pm 5\%$  above 10 K to 30 K,  $\pm 2\%$  above 30 K to 70 K,  $\pm 1\%$  above 70 K to 400 K,  $\pm 2\%$  above 400 K to 1235.08 K, and  $\pm 4\%$  above 1235.08 K.  
 99.50 Ag - 0.50 Pd:  $\pm 5\%$  below 70 K,  $\pm 3\%$  from 70 to 300 K, and  $\pm 2\%$  above 300 K.  
 99.00 Ag - 1.00 Pd:  $\pm 8\%$  below 70 K,  $\pm 3\%$  from 70 to 300 K, and  $\pm 2\%$  above 300 K.  
 97.00 Ag - 3.00 Pd:  $\pm 4\%$  up to 100 K and  $\pm 3\%$  above 100 K.  
 95.00 Ag - 5.00 Pd:  $\pm 4\%$  up to 100 K and  $\pm 3\%$  above 100 K.  
 90.00 Ag - 10.00 Pd:  $\pm 5\%$  up to 500 K and  $\pm 4\%$  above 500 K.

\* In temperature range where no experimental data are available.

TABLE 64. RECOMMENDED ELECTRICAL RESISTIVITY OF SILVER-PALLADIUM ALLOY SYSTEM† (continued)  
[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Ag: 85.00% (84.82 At. %) Pd: 15.00% (15.18 At. %)		Ag: 80.00% (79.78 At. %) Pd: 20.00% (20.22 At. %)		Ag: 75.00% (74.74 At. %) Pd: 25.00% (25.26 At. %)		Ag: 70.00% (69.71 At. %) Pd: 30.00% (30.29 At. %)		Ag: 65.00% (64.69 At. %) Pd: 35.00% (35.31 At. %)		Ag: 60.00% (59.67 At. %) Pd: 40.00% (40.33 At. %)	
T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho$
1	6.30*	1	8.39	1	10.52	1	12.91	1	15.7	1	18.8
4	6.30*	4	8.39	4	10.52	4	12.91	4	15.7	4	18.8
7	6.30*	7	8.39	7	10.52	7	12.91	7	15.7	7	18.8
10	6.30*	10	8.39	10	10.52	10	12.91	10	15.7	10	18.8
15	6.31*	15	8.40	15	10.52	15	12.91	15	15.7	15	18.8
20	6.31*	20	8.40	20	10.53	20	12.91	20	15.7	20	18.8
25	6.32*	25	8.41	25	10.53	25	12.92	25	15.7	25	18.8
30	6.34*	30	8.43	30	10.55	30	12.93	30	15.7	30	18.8
40	6.39*	40	8.48	40	10.59*	40	12.98	40	15.8*	40	18.9
50	6.45*	50	8.54	50	10.65*	50	13.04	50	15.8*	50	18.9
60	6.52*	60	8.61	60	10.72*	60	13.11	60	15.9*	60	19.0
70	6.60*	70	8.68	70	10.80*	70	13.19	70	16.0*	70	19.1
80	6.67*	80	8.76	80	10.88*	80	13.27	80	16.1*	80	19.2
90	6.75*	90	8.83	90	10.96*	90	13.35	90	16.1*	90	19.3
100	6.82*	100	8.91	100	11.04*	100	13.43	100	16.2*	100	19.4
150	7.17*	150	9.26	150	11.39*	150	13.78	150	16.6*	150	19.9
200	7.43*	200	9.57	200	11.70*	200	14.08	200	17.0*	200	20.3
250	7.78*	250	9.87	250	12.00*	250	14.39	250	17.3*	250	20.7
273	7.92*	273	10.01	273	12.14*	273	14.53	273	17.4*	273	20.9
293	8.04*	293	10.13	293	12.26*	293	14.65	293	17.6*	293	21.1
300	8.08*	300	10.17	300	12.30*	300	14.69	300	17.6*	300	21.2
350	8.38*	350	10.47	350	12.60*	350	14.99	350	18.0*	350	21.6
400	8.68*	400	10.78	400	12.91*	400	15.30	400	18.3*	400	22.0
500	9.31*	500	11.40	500	13.53*	500	15.92	500	18.9*	500	22.8
600	9.96*	600	12.05	600	14.18*	600	16.57	600	19.6*	600	23.6
700	10.63*	700	12.71	700	14.84*	700	17.23	700	20.3*	700	24.4
800	11.31*	800	13.40	800	15.53*	800	17.92	800	21.0*	800	25.2
900	12.00*	900	14.09	900	16.22*	900	18.61	900	21.7*	900	26.0
1000	12.73*	1000	14.83	1000	16.95*	1000	19.34	1000	22.5*	1000	26.8
1100	13.52*	1100	15.62	1100	17.75*	1100	20.14	1100	23.3*	1100	27.7

† Uncertainties in the electrical resistivity values are as follows:

- 85.00 Ag - 15.00 Pd:  $\pm 3\%$  up to 300 K and  $\pm 4\%$  above 300 K.
- 80.00 Ag - 20.00 Pd:  $\pm 3\%$  up to 300 K and  $\pm 4\%$  above 300 K.
- 75.00 Ag - 25.00 Pd:  $\pm 3\%$  up to 300 K and  $\pm 4\%$  above 300 K.
- 70.00 Ag - 30.00 Pd:  $\pm 3\%$  up to 300 K and  $\pm 4\%$  above 300 K.
- 65.00 Ag - 35.00 Pd:  $\pm 4\%$ .
- 60.00 Ag - 40.00 Pd:  $\pm 4\%$ .

\* In temperature range where no experimental data are available.

TABLE 64. RECOMMENDED ELECTRICAL RESISTIVITY OF SILVER-PALLADIUM ALLOY SYSTEM† (continued)  
[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Ag: 55.00% (54.66 At. %) Pd: 45.00% (45.34 At. %)			Ag: 50.00% (49.66 At. %) Pd: 50.00% (50.34 At. %)			Ag: 45.00% (44.66 At. %) Pd: 55.00% (55.34 At. %)			Ag: 40.00% (39.67 At. %) Pd: 60.00% (60.33 At. %)			Ag: 35.00% (34.69 At. %) Pd: 65.00% (65.31 At. %)			Ag: 30.00% (29.71 At. %) Pd: 70.00% (70.29 At. %)		
T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$	
1	22.5		1	28.4		1	35.7		1	39.7		1	39.1		1	35.5*	
4	22.5		4	28.4		4	35.7		4	39.7		4	39.1		4	35.5	
7	22.5		7	28.4		7	35.7		7	39.7		7	39.1		7	35.5	
10	22.5		10	28.4		10	35.7		10	39.7		10	39.1		10	35.5	
15	22.5		15	28.4		15	35.7		15	39.7		15	39.1		15	35.5	
20	22.5		20	28.4		20	35.7		20	39.7		20	39.1		20	35.5	
25	22.5		25	28.4		25	35.7		25	39.7		25	39.1*		25	35.5	
30	22.6*		30	28.4		30	35.7		30	39.7		30	39.1*		30	35.5	
40	22.6*		40	28.5		40	35.8*		40	39.7		40	39.2*		40	35.7	
50	22.7*		50	28.6		50	35.9*		50	39.9		50	39.4*		50	35.8	
60	22.8*		60	28.7		60	36.1*		60	40.1		60	39.6*		60	36.1	
70	22.9*		70	28.9		70	36.2*		70	40.3		70	39.8*		70	36.3	
80	23.0		80	29.0		80	36.4*		80	40.4		80	40.0*		80	36.6	
90	23.2*		90	29.1		90	36.5*		90	40.6		90	40.2*		90	36.8	
100	23.3*		100	29.3		100	36.7*		100	40.8		100	40.4*		100	37.1	
150	23.9*		150	29.9		150	37.3*		150	41.4		150	41.2*		150	38.3	
200	24.4*		200	30.5		200	37.8*		200	41.9		200	42.0*		200	39.3	
250	25.0*		250	31.0		250	38.1*		250	42.1		250	42.5*		250	40.1	
273	25.3*		273	31.2		273	38.2*		273	42.2		273	42.7*		273	40.4	
293	25.5		293	31.4		293	38.3		293	42.2		293	42.8		293	40.6	
300	25.6		300	31.5		300	38.3		300	42.2		300	42.9		300	40.7	
350	26.1*		350	32.0		350	38.4*		350	42.3		350	43.1*		350	41.3	
400	26.6*		400	32.4		400	38.5*		400	42.3		400	43.3*		400	41.7	
500	27.6*		500	33.2		500	38.7*		500	42.2		500	43.5*		500	42.5	
600	28.5*		600	34.0		600	39.0*		600	42.2		600	43.7*		600	43.1	
700	29.4*		700	34.8		700	39.4*		700	42.1		700	43.8*		700	43.6	
800	30.2*		800	35.5		800	39.7*		800	42.2		800	43.9*		800	44.1	
900	31.0*		900	36.3		900	40.2*		900	42.3		900	44.1*		900	44.6	
1000	31.7*		1000	36.9		1000	40.8*		1000	42.4		1000	44.4*		1000	45.1	
1100	32.4*		1100	37.6		1100	41.1*		1100	42.6		1100	44.8*		1100	45.7	

† Uncertainties in the electrical resistivity values are as follows:

- 55.00 Ag - 45.00 Pd:  $\pm 4\%$  up to 300 K and  $\pm 5\%$  above 300 K.
- 50.00 Ag - 50.00 Pd:  $\pm 3\%$  up to 300 K and  $\pm 4\%$  above 300 K.
- 45.00 Ag - 55.00 Pd:  $\pm 2\%$  up to 300 K and  $\pm 4\%$  above 300 K.
- 40.00 Ag - 60.00 Pd:  $\pm 2\%$  up to 300 K and  $\pm 4\%$  above 300 K.
- 35.00 Ag - 65.00 Pd:  $\pm 2\%$  up to 300 K and  $\pm 3\%$  above 300 K.
- 30.00 Ag - 70.00 Pd:  $\pm 2\%$  up to 250 K and  $\pm 3\%$  above 250 K.

\* In temperature range where no experimental data are available.

TABLE 64. RECOMMENDED ELECTRICAL RESISTIVITY OF SILVER-PALLADIUM ALLOY SYSTEM† (continued)  
[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Ag: 25.00% (24.74 At.%) Pd: 75.00% (75.26 At.%)			Ag: 20.00% (19.78 At.%) Pd: 80.00% (80.22 At.%)			Ag: 15.00% (14.83 At.%) Pd: 85.00% (85.17 At.%)			Ag: 10.00% (9.88 At.%) Pd: 90.00% (90.12 At.%)			Ag: 5.00% (4.94 At.%) Pd: 95.00% (95.06 At.%)			Ag: 3.00% (2.96 At.%) Pd: 97.00% (97.04 At.%)		
T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$		T	$\rho$	
1	30.47		1	24.72		1	18.56*		1	12.37		1	6.18*		1	3.71	
4	30.47		4	24.72		4	18.56		4	12.37		4	6.18		4	3.71	
7	30.47		7	24.72		7	18.56*		7	12.37		7	6.18		7	3.71	
10	30.47		10	24.72		10	18.56*		10	12.37		10	6.18		10	3.71	
15	30.47		15	24.72		15	18.57*		15	12.38		15	6.20		15	3.72	
20	30.47		20	24.74		20	18.59*		20	12.40		20	6.22		20	3.75	
25	30.51		25	24.77		25	18.63*		25	12.45		25	6.27		25	3.80	
30	30.56		30	24.83		30	18.69*		30	12.51		30	6.34		30	3.88	
40	30.71		40	24.99		40	18.87*		40	12.71		40	6.56		40	4.11	
50	30.90*		50	25.21		50	19.11*		50	12.97		50	6.85		50	4.41	
60	31.15*		60	25.49		60	19.41*		60	13.29		60	7.19		60	4.77	
70	31.43*		70	25.81		70	19.76*		70	13.67		70	7.59		70	5.17	
80	31.75*		80	26.16		80	20.14*		80	14.08		80	8.02		80	5.60	
90	32.08*		90	26.53		90	20.55*		90	14.52		90	8.46		90	6.05	
100	32.40*		100	26.90		100	20.97*		100	14.95		100	8.91		100	6.50	
150	33.86*		150	28.67		150	22.97*		150	17.06		150	11.09		150	8.67	
200	35.13*		200	30.20		200	24.75*		200	19.03		200	13.15		200	10.72	
250	36.20*		250	31.56		250	26.38*		250	20.88		250	15.11		250	12.69	
273	36.67*		273	32.15		273	27.08*		273	21.69		273	15.98		273	13.56	
293	37.06		293	32.64		293	27.68		293	22.39		293	16.72		293	14.30	
300	37.19		300	32.8		300	27.89*		300	22.63		300	16.98		300	14.56	
350	38.1*		350	33.9		350	28.3*		350	24.3		350	18.8*		350	16.38*	
400	38.8*		400	34.9		400	30.6*		400	25.9		400	20.5*		400	18.14*	
500	40.1*		500	36.8		500	33.1*		500	28.9		500	23.8*		500	21.50*	
600	41.2*		600	38.5		600	35.3*		600	31.6		600	26.9*		600	24.64*	
700	42.1*		700	39.9		700	37.3*		700	34.2		700	29.8*		700	27.6*	
800	43.0*		800	41.2		800	39.1*		800	36.6		800	32.5*		800	30.4*	
900	43.9*		900	42.5		900	40.8*		900	38.8		900	35.0*		900	32.9*	
1000	44.8*		1000	43.7		1000	42.3*		1000	40.8		1000	37.2*		1000	35.3*	
1100	45.6*		1100	44.8		1100	43.7*		1100	42.6		1100	39.2*		1100	37.3*	

† Uncertainties in the electrical resistivity values are as follows:

25.00 Ag - 75.00 Pd:  $\pm 5\%$  up to 300 K and  $\pm 4\%$  above 300 K.  
20.00 Ag - 80.00 Pd:  $\pm 5\%$  up to 300 K and  $\pm 4\%$  above 300 K.  
15.00 Ag - 85.00 Pd:  $\pm 5\%$ .

10.00 Ag - 90.00 Pd:  $\pm 6\%$ .

5.00 Ag - 95.00 Pd:  $\pm 6\%$  up to 100 K and  $\pm 5\%$  above 100 K.  
3.00 Ag - 97.00 Pd:  $\pm 6\%$  below 50 K,  $\pm 3\%$  from 50 to 200 K, and  $\pm 2\%$  above 200 K.

\* In temperature range where no experimental data are available.

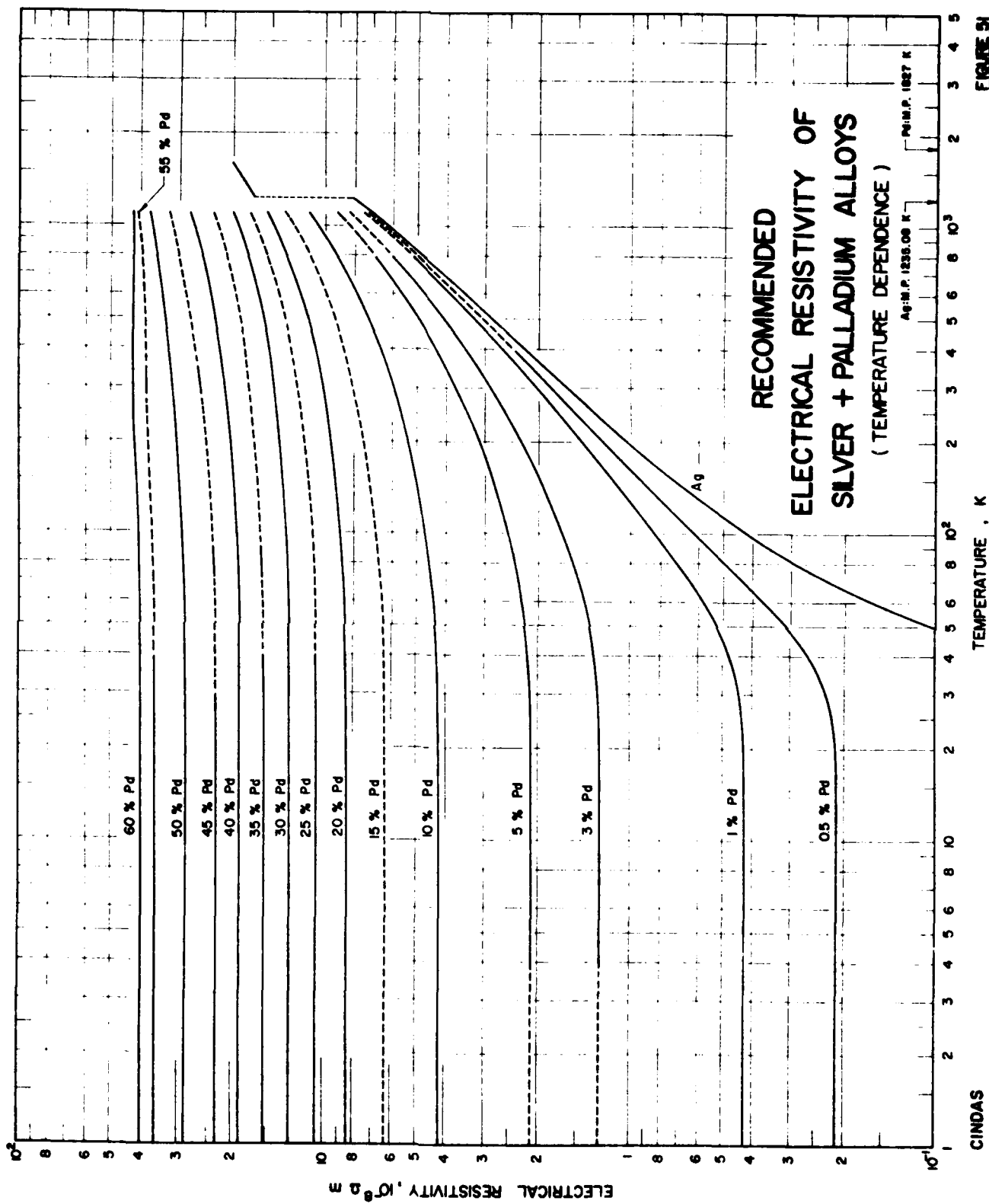
TABLE 64. RECOMMENDED ELECTRICAL RESISTIVITY OF SILVER-PALLADIUM ALLOY SYSTEM† (continued)  
[Temperature, T, K; Electrical Resistivity,  $\rho$ ,  $10^{-8} \Omega \text{ m}$ ]

Ag: 1.00% (0.99 At. %) Pd: 99.00% (99.01 At. %)		Ag: 0.50% (0.49 At. %) Pd: 99.50% (99.51 At. %)		Ag: 0.00% (0.00 At. %) Pd: 100.00% (100.00 At. %)			
T	$\rho$	T	$\rho$	T	$\rho$		
1	1.236	1	0.617	1	0.0200		
4	1.236	4	0.617	4	0.0205		
7	1.236	7	0.618	7	0.0217		
10	1.240	10	0.622	10	0.0242		
15	1.252	15	0.633	15	0.0346		
20	1.280	20	0.659	20	0.0564		
25	1.33	25	0.705	25	0.0938		
30	1.40	30	0.773	30	0.151		
40	1.62	40	0.976	40	0.335		
50	1.90	50	1.253	50	0.607		
60	2.25	60	1.592	60	0.940		
70	2.64	70	1.98	70	1.32		
80	3.07	80	2.40	80	1.75		
90	3.52	90	2.85	90	2.19		
100	3.97	100	3.29	100	2.63		
150	6.14	150	5.47	150	4.81		
200	8.21	200	7.54	200	6.89		
250	10.18	250	9.52	250	8.88		
273	11.06	273	10.41	273	9.78		
293	11.32	293	11.17	293	10.54		
300	12.08*	300	11.43*	300	10.80		
350	13.92*	350	13.28*	350	12.66		
400	15.70*	400	15.07*	400	14.46		
500	19.13*	500	18.50*	500	17.89		
600	22.32*	600	21.70*	600	21.10		
700	25.3*	700	24.7*	700	24.10		
800	28.1*	800	27.5*	800	26.3		
900	30.7*	900	30.1*	900	29.50		
1000	33.1*	1000	32.5*	1000	31.92		
1100	35.2*	1100	34.7*	1200	36.21		
				1400	39.80		
				1600	42.70		
				1827	45.14 (s)		
				1830	83.0 (t)		
				2000	83.0		

† Uncertainties in the electrical resistivity values are as follows:

1.00 Ag - 99.00 Pd:  $\pm 4\%$  up to 50 K and  $\pm 2\%$  above 50 K.  
0.50 Ag - 99.50 Pd:  $\pm 4\%$  up to 50 K and  $\pm 2\%$  above 50 K.  
0.00 Ag - 100.00 Pd:  $\pm 2\%$  up to 40 K,  $\pm 1\%$  above 40 K to 350 K,  $\pm 2.5\%$  above 350 K to 1600 K, and  $\pm 5\%$  above 1600 K.

\* In temperature range where no experimental data are available.



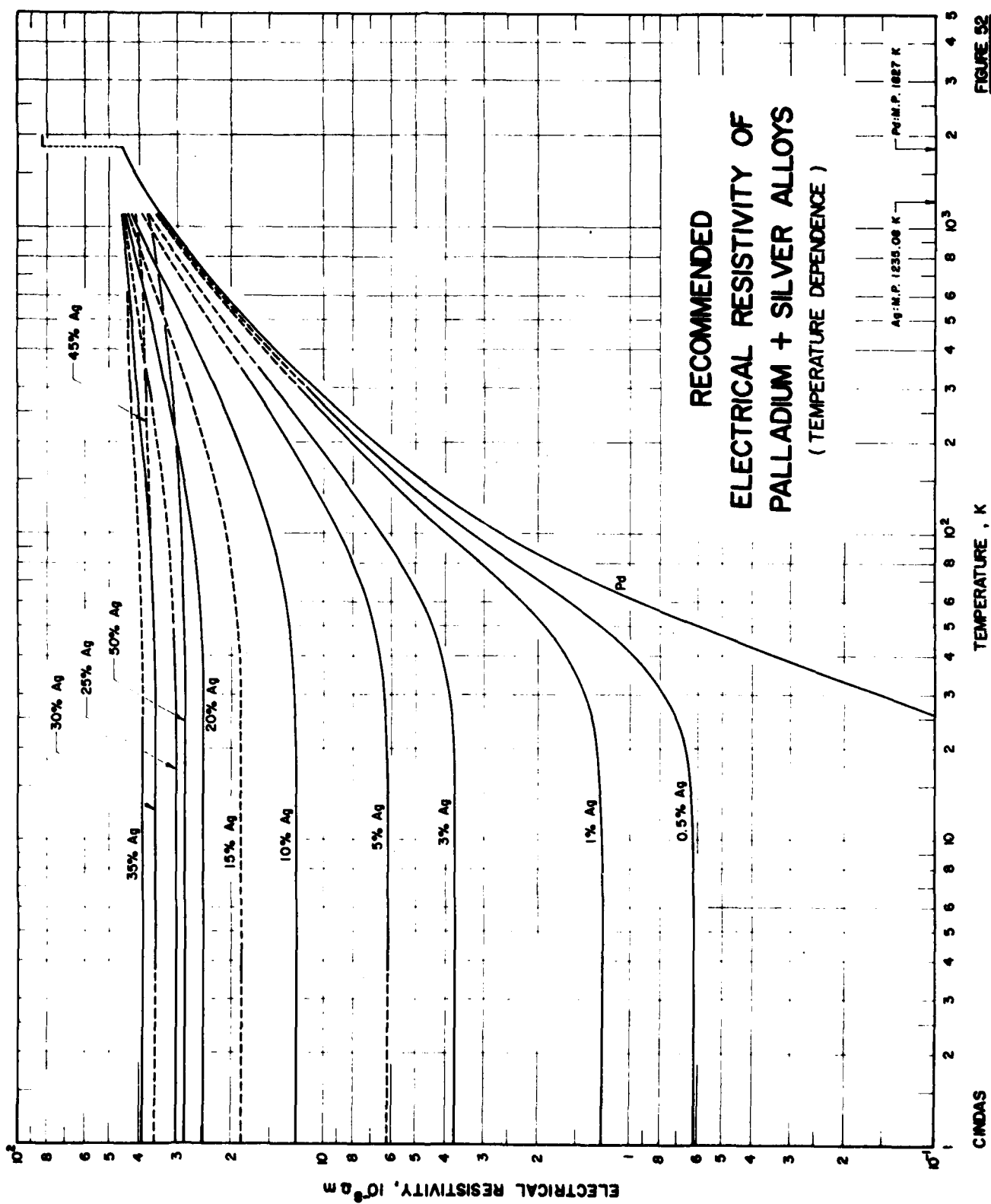


FIGURE 52



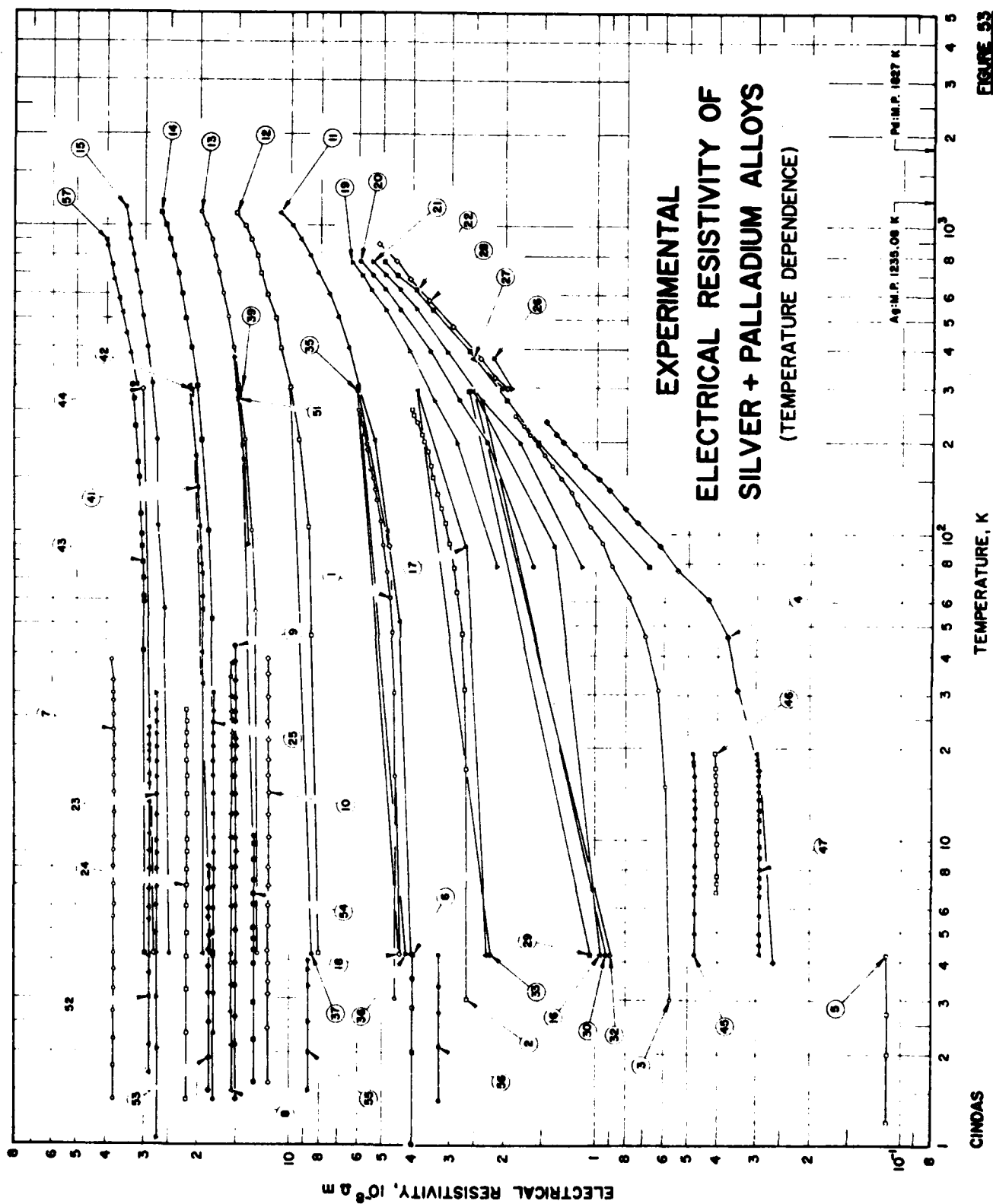


FIGURE 53

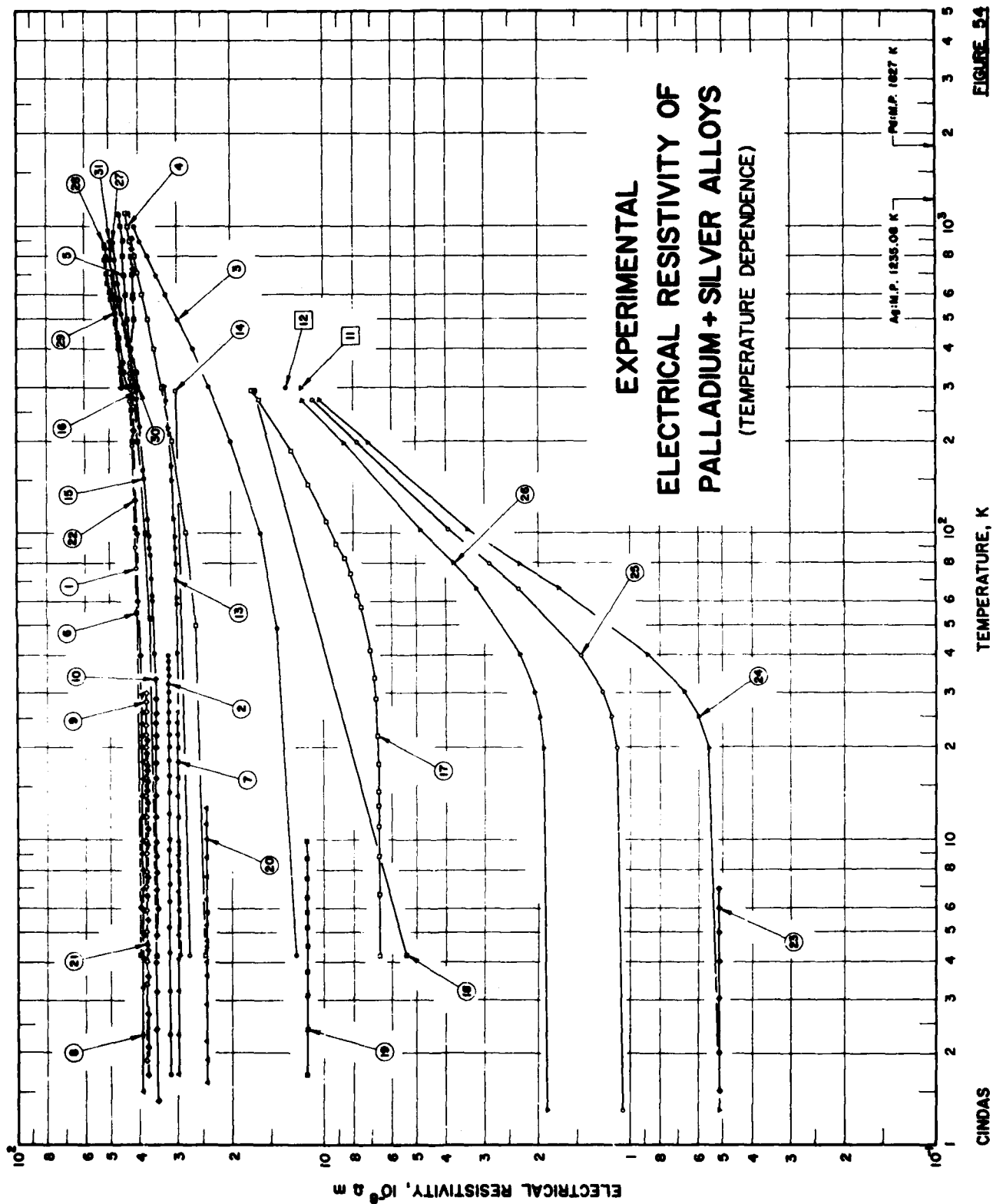


FIGURE 54

TABLE 65. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF SILVER + PALLADIUM ALLOYS (Temperature Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Ag	Composition (continued), Specifications, and Remarks
1	180	Schroeder, P. A., Wolf, R., and Woollam, J. A.	1965	A	3-250		9.88	Calculated composition (10.0 at/o Pd); $15 \pm 10$ ppm Fe impurity; Cominco 59 grade Ag ( $R_{973}/R_4 = 640$ ) with $3 \pm 3$ ppm Fe and sponge Pd from Johnson-Matthey and Co. Ltd. melted together in alumina crucible by induction heating, chill cast in ingot, drawn down to wire of 0.2 mm diam, and annealed in vacuum at 953 K; cross-sections determined from weight per unit length measurements and the densities of the pure components as given by the supplier and modified to allow for changes of lattice parameter on alloying; data read from figure.
2	180	Schroeder, P. A., et al.	1965	A	3-252		6.22	Calculated composition (6.30 at/o Pd); similar to the above specimen.
3	180	Schroeder, P. A., et al.	1965	A	3-243		1.26	Calculated composition (1.28 at/o Pd); similar to the above specimen.
4	180	Schroeder, P. A., et al.	1965	A	4-232		0.52	Calculated composition (0.53 at/o Pd); similar to the above specimen.
5	283	Bäcklund, N.	1968	A	1.2-4.2		0.207	Calculated composition (0.21 at/o Pd); prepared from spectroscopically pure materials from Johnson Matthey and Co., Ltd. by melting together in evacuated silica tube in high-frequency furnace; ingot slightly rolled, homogenized at 1073 K for about 2 h, then drawn to 0.2 mm wire; annealed at about 773 K prior to measurement; negative resistivity-temperature derivative indicated low temperature minimum; data read from figure.
6	283	Bäcklund, N.	1968	A	1.0-4.2		9.40	Calculated composition (9.52 at/o Pd); similar to the above specimen; resistivity-temperature derivative gave no indication of minimum within accuracy of measurements.
7	381	Chen, C. W., Edwards, L. R., and Legvold, S.	1968	A	1.4-38.2		50.3	Calculated composition (50 at/o Pd); chemical analysis showed 10 to 20 ppm Fe impurity; 99.999 pure Ag and 99.99 pure Pd levitation melted, swaged, drawn to wires of about 1 mm diam, cut to length of 5 cm, annealed in vacuum at 1273 K to recrystallize, and furnace-cooled; sample temperature controlled within $\pm 0.05$ K, relative error about $\pm 0.0004 \times 10^{-4}$ $\Omega$ m, $\pm 15$ uncertainty in resistivity due to slightly non-uniform diam; data read from figure.
8	381	Chen, C. W., et al.	1968	A	1.4-37.7	A	65.3	Calculated composition (35 at/o Pd); similar to the above specimen.
9	381	Chen, C. W., et al.	1968	A	1.4-42.4	A'	65.3	The above specimen reheated to 1273 K and quenched in ice water.
10	381	Chen, C. W., et al.	1968	A	1.6-38.6		75.2	Calculated composition (25 at/o Pd); similar to the sample of Ag + Pd data set 7.
11	378	Richter, T. and Pfleger, E.	1966	B	4.2-1103		9.9	Calculated composition (10 at/o Pd); 0.35 to 0.5 mm in diam and 50 to 100 cm long; supplied by Firma Degussa; prepared from 99.9 pure palladium and 99.99 pure silver; annealed in argon at 1073 K for 30 min and quenched in water; data read from figure.
12	378	Richter, T. and Pfleger, E.	1966	B	4.2-1102		19.8	Calculated composition (20 at/o Pd); similar to the above specimen.
13	378	Richter, T. and Pfleger, E.	1966	B	4.2-1106		28.7	Calculated composition (30 at/o Pd); similar to the above specimen.

TABLE 65. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF SILVER + PALLADIUM ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Ag Pd	Composition (continued), Specifications, and Remarks
14	378	Richter, T. and Pfleger, E.	1966	B	4.2-1103		39.7	Calculated composition (40 a/o Pd); similar to the above specimen.
15	378	Richter, T. and Pfleger, E.	1966	B	4.2-1105		49.7	Calculated composition (50 a/o Pd); similar to the above specimen.
16	244	Tainsh, R.J. and White, G.K.	1962		4.2-293		2.05	Calculated composition (2.08 a/o Pd); rod specimen 3 to 5 mm in diam and 6 cm long; annealed at 1213 K; data extracted from table.
17	244	Tainsh, R.J. and White, G.K.	1962		4.2-293		4.99	Calculated composition (5.06 a/o Pd); similar to the above specimen.
18	244	Tainsh, R.J. and White, G.K.	1962		4.2-293		9.78	Calculated composition (9.9 a/o Pd); similar to the above specimen.
19	226	Otter, F.A., Jr.	1956	A	78-767		4.07	Calculated composition (4.12 a/o Pd); 99.99% silver from Hardy and Harman and 99.99% palladium from Bishop Platinum Co. melted together under vacuum of less than one micron in aluminum crucibles with an induction furnace, superheated about 100 K above the melting point to insure solutions in a reasonable time, homogenized in vacuum for 24 h at 1173 K, ingot swaged or drawn to wires of about 0.030 in diam, and annealed about 1 h at 773 K before measuring in vacuum; data read from figure.
20	226	Otter, F.A., Jr.	1956	A	78-767		2.97	Calculated composition (3.01 a/o Pd); similar to the above specimen.
21	226	Otter, F.A., Jr.	1956	A	78-767		1.97	Calculated composition (2.00 a/o Pd); similar to the above specimen.
22	226	Otter, F.A., Jr.	1956	A	78-767		1.08	Calculated composition (1.10 a/o Pd); similar to the above specimen.
23	380	Edwards, L.R., Chen, C.W., and Legvold, S.	1970		1.1-30		50.34 49.66	Calculated composition (50 a/o Pd), analysis revealed 0.0050 to 0.0250 wt % Fe but authors concluded minimum in resistivity at low temperatures was not due to Fe content; data read from figure.
24	380	Edwards, L.R., et al.	1970		1.4-27		55.34 44.66	Calculated composition (45 a/o Pd); similar to the above specimen.
25	380	Edwards, L.R., et al.	1970		1.4-30		60.33 39.67	Calculated composition (40 a/o Pd); similar to the above specimen.
26	385	Pravoverov, N.L. and Tribunskaya, L.A.	1967	B	298-373	15	98.40 0.6	Composition determined by chemical analysis, microstructural analysis confirmed single phase; 10 g charge of 99.99 pure Ag and electron-beam-refined Pd (99.98 initial purity) melted in double evacuated quartz ampoules in high frequency furnace, homogenized in vacuum at 673 K for 60 h, drawn into wires 0.5 and 1 mm diam, and annealed at 623 K for 0.5 h; data extracted from table.
27	385	Pravoverov, N.L. and Tribunskaya, L.A.	1967	B	298-373	17	99.00 1.0	Similar to the above specimen.
28	386	Pravoverov, N.L. and Tribunskaya, L.A.	1967	B	298-673	16	99.20 0.8	Similar to the above specimen; measured in a vacuum of $6 \times 10^{-6}$ mm Hg.
29	376	Kemp, W.R.G., Klemens, P.G., Freedhar, A.K., and White, G.K.	1966	B	4.2-293		2.08	Wire specimen supplied by Johnson, Matthey and Co. Ltd.; wound on a mica former; measured in strained state; data extracted from table.

TABLE 65. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF SILVER + PALLADIUM ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Ag Pd	Composition (continued), Specifications, and Remarks
30	376	Kemp, W. R. G., Klemens, P. G., Sreedhar, A. K., and White, G. K.	1956	B	4.2, 293			The above specimen annealed at 883 K; data extracted from table; authors also gave temperature-dependent resistivity in figure and stated it was the same for all alloys with 2 to 30% Pd.
31*	376	Kemp, W. R. G., et al.	1956	B	4.2, 293		2.08	Rod specimen supplied by Johnson, Matthey and Co. Ltd.; measured in strained state; data extracted from table.
32	376	Kemp, W. R. G., et al.	1956	B	4.2, 293			The above specimen measured after annealing at 883 K.
33	376	Kemp, W. R. G., et al.	1956	B	4.2, 293		5.06	Wire specimen supplied by Johnson, Matthey and Co. Ltd.; wound on a mica former and annealed at 883 K; data extracted from table; authors also gave temperature-dependent resistivity in figure and stated it was the same for all alloys with 2 to 30% Pd.
34*	376	Kemp, W. R. G., et al.	1956	B	4.2, 293		5.06	Rod specimen supplied by Johnson, Matthey and Co. Ltd.; annealed at 883 K; data extracted from table.
35	376	Kemp, W. R. G., et al.	1956	B	4.2, 293		9.9	Wire specimen supplied by Johnson, Matthey and Co. Ltd.; wound on a mica former and annealed at 923 K; data extracted from table; authors also gave temperature-dependent resistivity in figure and stated it was the same for all alloys with 2 to 30% Pd.
36	376	Kemp, W. R. G., et al.	1956	B	4.2, 293		9.9	Rod specimen supplied by Johnson, Matthey and Co. Ltd.; annealed at 923 K; data extracted from table.
37	376	Kemp, W. R. G., et al.	1956	B	4.2, 293		20.08	Wire specimen supplied by Johnson, Matthey and Co. Ltd.; wound on a mica former and annealed at 923 K; data extracted from table; authors also gave temperature-dependent resistivity in figure and stated it was the same for all alloys with 2 to 30% Pd.
38*	376	Kemp, W. R. G., et al.	1956	B	4.2, 293		20.08	Rod specimen supplied by Johnson, Matthey and Co. Ltd.; annealed at 1073 K; data extracted from table.
39	376	Kemp, W. R. G., et al.	1956	B	4.2, 293		29.63	Wire specimen supplied by Johnson, Matthey and Co. Ltd.; wound on a mica former and annealed at 923 K; data extracted from table; authors also gave temperature-dependent resistivity and stated it was the same for all alloys with 2 to 30% Pd.
40*	376	Kemp, W. R. G., et al.	1956	B	4.2, 293		29.63	Rod specimen supplied by Johnson, Matthey and Co. Ltd.; annealed at 1073 K; data extracted from table.
41	376	Kemp, W. R. G., et al.	1956	B	4.2-300		40	Wire specimen supplied by Johnson, Matthey and Co. Ltd.; annealed at 1153 K, wound on a mica former, and again annealed at 773 K; values of temperature-dependent resistivity read from figure and added to tabulated residual resistivity to obtain total resistivity.
42	376	Kemp, W. R. G., et al.	1956	B	4.2, 293		40	Rod specimen supplied by Johnson, Matthey and Co. Ltd.; annealed at 1153 K; data extracted from table.
43	376	Kemp, W. R. G., et al.	1956	B	4.2-303		50	Wire specimen supplied by Johnson, Matthey and Co. Ltd.; annealed at 1153 K, wound on a mica former, and a gain annealed at 773 K; values of temperature-dependent resistivity read from figure and added to tabulated residual resistivity to obtain total resistivity.

\* Not shown in figure.

TABLE 65. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF SILVER + PALLADIUM ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Ag Pd	Composition (continued), Specifications, and Remarks
44	376	Kemp, W. R. G., Klemens, P. G., Sreedhar, A. K., and White, G. K.	1956	B	4.2-293		50	Rod specimen supplied by Johnson, Matthey and Co. Ltd.; annealed at 773 K; data extracted from table.
45	308	Barber, A. J. and Caplin, A. D.	1975	A	4.2-19.4	Sample 43	1.056	Calculated, nominal composition (1.07 a/o Pd); wire specimen 0.76 mm in diam; high purity Ag and Pd melted together in quartz ampoules, homogenized, and rolled and drawn into wire; annealed at 873 K for 6 to 10 h in a sealed capsule with up to 10 Torr oxygen or in a continuous flow of oxygen at 10 <sup>-1</sup> Torr in order to precipitate Fe; measured in helium environment; residual resistivity 0.472 x 10 <sup>-8</sup> Ωm; uncertainty in temperature less than ±200 mK below 20 K and ±100 mK below 10 K, less than 1% uncertainty in geometrical factor; data extracted from figure with error bars.
46	308	Barber, A. J. and Caplin, A. D.	1975	A	6.8-19.3	Sample 40	0.888	Calculated, nominal composition (0.90 a/o Pd); similar to the above specimen; 0.89 mm diam; residual resistivity 0.4000 x 10 <sup>-8</sup> Ωm.
47	308	Barber, A. J. and Caplin, A. D.	1975	A	4.2-19.4	Sample 42	0.651	Calculated, nominal composition (0.66 a/o Pd); similar to the above specimen; 0.65 mm diam; residual resistivity 0.2900 x 10 <sup>-8</sup> Ωm.
48*	308	Barber, A. J. and Caplin, A. D.	1975	A	4.3-19.8	Sample 57	0.0898	Calculated, nominal composition (0.091 a/o Pd); similar to the above specimen; 0.36 mm diam; residual resistivity 0.0422 x 10 <sup>-8</sup> Ωm.
49*	308	Barber, A. J. and Caplin, A. D.	1975	A	4.7-19.8	Sample 58	0.0898	Calculated, nominal composition (0.091 a/o Pd); similar to the above specimen; 0.36 mm diam; residual resistivity 0.0415 x 10 <sup>-8</sup> Ωm.
50*	308	Barber, A. J. and Caplin, A. D.	1975	A	2.1-19.5	Sample 90	0.00769	Calculated, nominal composition (0.0078 a/o Pd); similar to the above specimen; 0.25 mm diam; residual resistivity 0.00607 x 10 <sup>-8</sup> Ωm.
51	386	Devaz, J. and Fleming, J. A.	1982	B	91-373		80 20	Wire specimen 0.0762 mm (0.003 in) in diam and of the order of 50-100 cm long wound loosely on mica sheet; supplied by Johnson and Matthey; measurements taken in boiling water, air, melting ice, melting carbonic acid, boiling ethylene, and boiling oxygen; diameter measured to nearest 0.01 in; values reported are average of several measurements; no correction for thermal expansion; units of composition not stated, weight percent assumed.
52	379	Murani, A. P.	1974		1.7-23.0		50.3	Calculated composition (50 a/o Ag); preliminary results with no measurement information; data read from figure.
53	379	Murani, A. P.	1974		1.5-8.1		62.3	Calculated composition (62 a/o Ag); similar to the above specimen.
54	379	Murani, A. P.	1974		1.6-10.2		70.3	Calculated composition (70 a/o Ag); similar to the above specimen.
55	379	Murani, A. P.	1974		1.5-4.0		80.2	Calculated composition (80 a/o Ag); similar to the above specimen.
56	379	Murani, A. P.	1974		1.4-4.2		92.1	Calculated composition (92 a/o Ag); similar to the above specimen.
57	384	Arajs, S., Rao, K. V., Yao, Y. D., and Teoh, W.	1977	A	301-896		50.24	Calculated composition (49.9 a/o Ag); 99.99 pure Pd sponge from Engelhard Industries Inc. and 99.999 pure Ag from American Smelting and Refining Co. arc-melted to form ingot weighing about 10 g; ingot sealed in evacuated silica tube, homogenized at about 1300 K for 60 h, drawn to 1 mm <sup>2</sup> cross-section, cleaned, re-sealed in evacuated silica capsule; annealed at 1200 K for 1 h, and rapidly cooled in air without breaking the capsule; measurement temperatures produced by platinum wound vacuum furnace; data read from figure.

\* Not shown in figure.

TABLE 65. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF SILVER + PALLADIUM ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Ag Pd	Composition (continued), Specifications, and Remarks
58*	389	Pravoverov, N. L. and Savitskii, E. M.	1962	A	315-1020	3	85.2	Calculated composition (85 a/o Ag); 10 g charge melted under borax in induction furnace, cast, forged, rolled, and vacuum annealed for 120 h at 1273 K and 250 h at 1073 K in order to bring nearly to equilibrium; annealed as 1 mm diam wire, then drawn to 0.3 mm diam and again annealed for 1 h at 1173 K to remove work-hardening; sample specimen consisted of 250 mm of the wire wound into cylindrical coil of mean diam 3 mm and length 2.7 mm; coil welded to junctions of two thermocouples consisting of Pt and Pt-Rh electrodes; Pt electrodes formed part of the current-supplying circuit and Pt-Rh ones part of the current-removing circuit; recorded temperature was mean of values for the two thermocouples, with no significant temperature drop along the length of the coil; direction of the specimen current reversed and $\rho(T)$ resistivity recorded; data reported as $\rho(T)/\rho(298)$ , the ratio of the resistivity at the measurement temperature (T) to that at 298 K, without reporting $\rho(298)$ ; data read from figure.
59*	389	Pravoverov, N. L. and Savitskii, E. M.	1962	A	342-1006	4	60.3	Calculated composition (60 a/o Ag); similar to the above specimen.
60*	389	Pravoverov, N. L. and Savitskii, E. M.	1962	A	342-1149	5	55.3	Calculated composition (55 a/o Ag); similar to the above specimen.
61*	389	Pravoverov, N. L. and Savitskii, E. M.	1962	A	297-1173	6	50.3	Calculated composition (50 a/o Ag); similar to the above specimen; according to the authors, ordering takes place in this alloy upon prolonged annealing at 1073 to 1273 K.

\* Not shown in figure.







TABLE 66. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF SILVER + PALLADIUM ALLOYS (Temperature Dependence) (continued)

T	$\rho$	T	$\rho$	T	$\rho$	T	$\rho(T)$ $\rho(298)$	T	$\rho(T)$ $\rho(298)$
DATA SET 48*		DATA SET 50*		DATA SET 52		DATA SET 55		DATA SET 58*	
4.34	0.04221	2.11	0.006070	1.7	28.8455	1.5	8.68269	315	1.0128
4.67	0.04222	2.11	0.006070	2.1	28.8452	2.0	8.68275	347	1.0230
4.84	0.04222	2.36	0.006071	3.0	28.8449	2.5	8.68281	373	1.0335
5.26	0.04223	2.36	0.006071	3.6	28.8444	3.3	8.68298	410	1.0479
5.47	0.04223	2.57	0.006071	4.1	28.8436	3.7	8.68307	452	1.0576
5.88	0.04225	2.76	0.006071	5.4	28.8418	4.0	8.68317	475	1.0715
6.18	0.04225	2.96	0.006072	5.9	28.8411	DATA SET 56		522	1.0815
6.55	0.04227	3.18	0.006072	6.8	28.8397	1.4	3.25270	536	1.0894
6.83	0.04229	3.38	0.006073	7.9	28.8380	2.1	3.25270	536	1.0894
7.35	0.04232	3.59	0.006074	9.0	28.8353	2.7	3.25270	536	1.0894
7.57	0.04234	3.78	0.006074	10.3	28.8341	3.3	3.25275	536	1.0894
8.53	0.04240	3.99	0.006076	13.1	28.8298	4.2	3.25279	536	1.0894
9.08	0.04246	4.24	0.006077	15.0	28.8274	DATA SET 57		536	1.0894
10.30	0.04266	4.29	0.006078	16.2	28.8261	301	31.97	536	1.0894
11.14	0.04282	5.11	0.006087	17.9	28.8253	312	32.19*	536	1.0894
12.08	0.04306	6.11	0.006106	19.1	28.8251	347	32.69	536	1.0894
13.37	0.04343	6.58	0.006121	20.1	28.8253	382	33.29	536	1.0894
14.26	0.04383	7.15	0.006138	21.6	28.8263	444	34.21	536	1.0894
15.45	0.04443	7.85	0.006168	23.0	28.8281	520	35.40	536	1.0894
16.41	0.04508	8.00	0.006175	DATA SET 53		557	35.91*	536	1.0894
17.54	0.04593	8.81	0.006231	1.5	18.4114	576	36.21	536	1.0894
18.58	0.04686	9.37	0.006284	2.5	18.4120	608	36.61*	536	1.0894
19.77	0.04797	10.28	0.006366	3.1	18.4135	632	36.97*	536	1.0894
DATA SET 49*		10.89	0.006411	3.8	18.4169	662	37.37	536	1.0894
4.70	0.04152	10.87	0.006437	4.3	18.4191	691	37.76*	536	1.0894
4.99	0.04152	12.13	0.006635	4.8	18.4217	714	38.01*	536	1.0894
5.27	0.04153	13.00	0.006820	5.5	18.4251	742	38.36	536	1.0894
5.52	0.04153	14.76	0.007287	5.9	18.4291	762	38.61*	536	1.0894
5.94	0.04155	15.24	0.007476	6.8	18.4371	777	38.75*	536	1.0894
6.27	0.04156	15.78	0.007652	8.1	18.4525	791	38.94*	536	1.0894
6.64	0.04157	16.22	0.007911	DATA SET 54		801	39.06*	536	1.0894
7.02	0.04159	16.82	0.008038	1.6	13.1317	814	39.19*	536	1.0894
7.41	0.04162	17.91	0.008225	2.2	13.1325	851	39.53	536	1.0894
7.71	0.04164	18.37	0.008554	2.9	13.1338	875	39.74*	536	1.0894
8.61	0.04170	19.54	0.008998	3.9	13.1358	884	39.85*	536	1.0894
9.23	0.04177	DATA SET 51		4.3	13.1372	896	39.94	536	1.0894
10.28	0.04195	91	13.797	4.7	13.1381	DATA SET 59*		536	1.0894
11.17	0.04211	173	14.256	5.1	13.1396	342	1.0182	536	1.0894
12.16	0.04234	183	14.482	6.1	13.1443	393	1.0355	536	1.0894
13.34	0.04269	274	14.965	6.6	13.1476	420	1.0457	536	1.0894
14.32	0.04311	283	14.984	7.3	13.1526	459	1.0536	536	1.0894
15.45	0.04368	373	15.409	8.5	13.1636	487	1.0695	536	1.0894
16.44	0.04434	DATA SET 60*		9.6	13.1758	553	1.0875	536	1.0894
17.54	0.04516	342	1.0145	10.2	13.1838	575	1.0913	536	1.0894
18.62	0.04610	361	1.0219	DATA SET 61*		607	1.1038	536	1.0894
19.83	0.04723	391	1.0287	301	31.97	720	1.1377	536	1.0894
		431	1.0375	312	32.19*	736	1.1402	536	1.0894
		457	1.0488	347	32.69	787	1.1605	536	1.0894
		527	1.0636	382	33.29	861	1.1881	536	1.0894
		561	1.0778	444	34.21	882	1.1889	536	1.0894
				520	35.40	894	1.2222	536	1.0894
				557	35.91*	1006	1.2373	536	1.0894
				576	36.21	T	$\rho(T)$ $\rho(298)$	536	1.0894
				608	36.61*	342	1.0182	536	1.0894
				632	36.97*	393	1.0355	536	1.0894
				662	37.37	420	1.0457	536	1.0894
				691	37.76*	459	1.0536	536	1.0894
				714	38.01*	487	1.0695	536	1.0894
				742	38.36	553	1.0875	536	1.0894
				762	38.61*	575	1.0913	536	1.0894
				777	38.75*	607	1.1038	536	1.0894
				791	38.94*	720	1.1377	536	1.0894
				801	39.06*	736	1.1402	536	1.0894
				814	39.19*	787	1.1605	536	1.0894
				851	39.53	861	1.1881	536	1.0894
				875	39.74*	882	1.1889	536	1.0894
				884	39.85*	894	1.2222	536	1.0894
				896	39.94	1006	1.2373	536	1.0894

\* Not shown in figure.

TABLE 67. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF PALLADIUM + SILVER ALLOYS (Temperature Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Pd Ag	Composition (continued), Specifications, and Remarks
1	45	Coles, R. R. and Taylor, J. C.	1962	A	77-393		39.3	Calculated composition (39.0 a/o Ag), composition determined by chemical analysis, spectrographic analysis of a few samples showed no abnormal incidence of impurities; 1.524 mm (0.060 in.) wire specimen; supplied by Precious Metals Division of Mond Nickel Co. prepared from 99.95% pure metals by melting in argon atmosphere; annealed in vacuum at 1023 to 1173 K; $\rho_0 = 40.2 \times 10^{-8}$ $\Omega$ m, $d\rho/dT = 0.0037 \times 10^{-8}$ $\Omega$ m K <sup>-1</sup> at 293 K; $\pm 1\%$ uncertainty in cross-sectional area; data read from figure.
2	397	Sarachik, M. P.	1968	A	1.7-40		74.4 25.1	0.5 Fe; calculated composition (Pd <sub>3</sub> Ag+1 a/o Fe); $\pm 10\%$ error arises from sample geometry and composition; data read from figure.
3	378	Richter, T. and Pfüger, E.	1966	B	4.2-1102		89.9	Calculated composition (90 a/o Pd); 0.35 to 0.5 mm in diameter and 50 to 100 cm long; supplied by Firma Degussa; prepared from 99.9 pure palladium and 99.99 pure silver; annealed in argon at 1073 K for 30 min and quenched in water; data read from figure.
4	378	Richter, T. and Pfüger, E.	1966	B	4.2-1100		79.8	Calculated composition (80 a/o Pd); similar to the above specimen.
5	378	Richter, T. and Pfüger, E.	1966	B	4.2-1099		69.7	Calculated composition (70 a/o Pd); similar to the above specimen.
6	378	Richter, T. and Pfüger, E.	1966	B	4.2-1101		59.7	Calculated composition (60 a/o Pd); similar to the above specimen.
7	380	Edwards, L. R., Chen, C. W., and Legvold, S.	1970		1.7-26		25.26	Calculated composition (25 a/o Ag), 0.0050-0.0250 wt. % Fe present; authors concluded minimum in resistivity was not due to Fe; data read from figure.
8	380	Edwards, L. R., et al.	1970		1.5-30		40.33	Calculated composition (40 a/o Ag); similar to the above specimen.
9	380	Edwards, L. R., et al.	1970		1.9-30		43.34	Calculated composition (43 a/o Ag); similar to the above specimen.
10	380	Edwards, L. R., et al.	1970		1.4-34		45.34	Calculated composition (45 a/o Ag); similar to the above specimen.
11	284	Köster, W. and Hagmann, D.	1961		298.2		2.03	Calculated composition (2 a/o Ag); 1 mm wire specimen supplied by Firma Degussa; annealed at 1073 K for 1 h and furnace-cooled; data extracted from table.
12	284	Köster, W. and Hagmann, D.	1961		298.2		4.06	Calculated composition (4 a/o Ag); similar to the above specimen.
13	376	Kemp, W. R. G., Klemens, P. G., Sreedhar, A. K., and White, G. K.	1966	B	4.2-303		50 50	Wire specimen; supplied by Johnson, Matthey and Co.; annealed at 1153 and 773 K; data read from figure.
14	376	Kemp, W. R. G., et al.	1966		4.2, 293		50 50	Rod specimen; supplied by Johnson, Matthey and Co.; annealed at 1153 K; measured with aid of galvanometer amplifier; data extracted from table.
15	376	Kemp, W. R. G., et al.	1966	B	4.2-305		70 30	Wire specimen; supplied by Johnson, Matthey and Co.; annealed at 1153 and 773 K; data read from figure.
16	376	Kemp, W. R. G., et al.	1966		4.2, 293		70 30	Rod specimen; supplied by Johnson, Matthey and Co.; annealed at 1153 K; measured with aid of galvanometer amplifier; data extracted from table.

TABLE 67. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF PALLADIUM + SILVER ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Pd Ag	Composition (continued), Specifications, and Remarks
17	376	Kemp, W. R. G., Klemens, P. G., and Sreedhar, A. K., and White, G. K.	1956	B	4.3-293		95 5	Wire specimen; supplied by Johnson, Matthey and Co.; annealed at 1153 K and 773 K; data read from figure.
18	376	Kemp, W. R. G., et al.	1956		4.3-293		95 5	Rod specimen; supplied by Johnson, Matthey and Co.; annealed at 1153 K.
19	379	Murani, A. P.	1974		1.7-9.9		8.1	Calculated composition (8 a/o Ag); preliminary results with no measurement information; data read from figure.
20	379	Murani, A. P.	1974		1.6-12.7		19.2	Calculated composition (19 a/o Ag); similar to the above specimen.
21	379	Murani, A. P.	1974		1.7-21.1		33.3	Calculated composition (33 a/o Ag); similar to the above specimen.
22	175	Ahmad, H. M. and Greig, D.	1974		6-911		40.3	Calculated composition (40 a/o Ag); no measurement information; data read from figure.
23	382	Greig, D. and Rowlands, J. A.	1974	A	1.5-6.9		0.51	Calculated composition (0.5 a/o Ag). composition determined by weighing component metals; Pd with less than 0.001 wt. % metallic impurities from Englehard Industries and Ag from Johnson-Matthey Ltd. melted together in argon arc furnace which had tungsten electrode, copper hearth, and titanium getter; cooled into form of a 'button' which was turned over and remelted 5 times to ensure mixing, pressed, rolled, and swaged into 2 mm diam rods about 12 cm long, annealed for 24 h at 1123 K in a vacuum furnace, and cooled slowly; error in geometrical factor $\pm 0.1\%$ , temperature steady to $\pm 5$ mK and measured to $\pm 100$ mK, smallest resistivity measurable was 1 p $\Omega$ m; measured resistivity $\rho$ analyzed as $\rho = \rho_0 + \rho_T$ where the residual resistivity $\rho_0$ was determined by assuming the temperature dependent $\rho_T = \rho_0 \Delta T$ and making a best fit to the experimental data, $\rho_0$ was $0.512 \times 10^{-8} \Omega$ m; data reported in figure in the form of $\rho_T$ .
24	383	Greig, D. and Rowlands, J. A.	1974		1.3-273.0		0.51	Calculated composition (0.5 a/o Ag). composition determined by weighing components; Pd from Englehard Industries and Ag from Johnson-Matthey melted into a 'button' in argon arc furnace, homogenized by turning over and remelting 5 times, pressed, rolled, and swaged into rod 2 mm in diam and 12 cm long, annealed in vacuum for 24 h at 1123 K; geometrical factor known to $\pm 0.1\%$ , temperatures to $\pm 0.1$ K; measured in same run as two following specimens, temperature difference between specimens less than 0.001 K; within accuracy of measurements, the 1.3 K value may be taken as the residual resistivity; data extracted from table.
25	383	Greig, D. and Rowlands, J. A.	1974		1.3-273.0		1.0	Calculated composition (1 a/o Ag); similar to the above specimen.
26	383	Greig, D. and Rowlands, J. A.	1974		1.3-273.0		2.0	Calculated composition (2 a/o Ag); similar to the above specimen.

TABLE 67. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF PALLADIUM + SILVER ALLOYS (Temperature Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp. Range, K	Name and Specimen Designation	Composition (weight percent) Pd Ag	Composition (continued), Specifications, and Remarks
37	384	Arajs, S., Rao, K. V., Yao, Y. D. and Tech, W.	1977	A	301-891		44.64	Calculated composition (44.3 a/o Ag); 99.99 pure Pd sponge from Engelhard Industries Inc. and 99.999 pure Ag from American Smelting and Refining Co. arc-melted to form ingot weighing about 10 g; ingot sealed in evacuated silica tube, homogenized at about 1300 K for 60 h, drawn to 1 mm <sup>2</sup> cross-section, cleaned, re-sealed in evacuated silica capsule, annealed at 1200 K for 1 h, and rapidly cooled in air without breaking the capsule; measurement temperatures produced by platinum-wound vacuum furnace; data read from figure.
28	384	Arajs, S., et al.	1977	A	302-889		40.33	Calculated composition (40.0 a/o Ag); similar to the above specimen.
29	384	Arajs, S., et al.	1977	A	302-892		35.11	Calculated composition (34.8 a/o Ag); similar to the above specimen.
30	384	Arajs, S., et al.	1977	A	300-890		30.29	Calculated composition (30.0 a/o Ag); similar to the above specimen.
31	384	Arajs, S., et al.	1977	A	300-900		40.33	Calculated composition (40.0 a/o Ag); similar to the above specimen but cold-worked to show different temperature dependence of the resistivity.
33*	388	Pravoverov, N. L., and Savitskii, E. M.	1962	A	327-708	1	15.2	Calculated composition (15 a/o Ag); 10 g charge melted under borax in induction furnace, cast, forged, rolled, and vacuum annealed for 120 h at 1273 K and 250 h at 1073 K in order to bring nearly to equilibrium; annealed as 1 mm diam wire, then drawn down to 0.3 mm diam and again annealed for 1 h at 1173 K to remove work-hardening; sample specimen consisted of 250 mm of the wire wound into cylindrical coil of mean diam 3 mm and length 25 mm; coil welded to junctions of two thermocouples consisting of Pt and Pt-Rh; Pt electrodes formed part of the current-supplying circuit and Pt-Rh ones part of the current-removing circuit; recorded temperature was mean of values for the two thermocouples, with no significant temperature drop along the length of the coil; direction of the specimen current reversed and mean resistivity recorded; data reported as $\rho(T)/\rho(298)$ , the ratio of the resistivity at the measurement temperature (T) to that at 298 K, without reporting $\rho(298)$ ; data read from figure.
33*	388	Pravoverov, N. L., and Savitskii, E. M.	1962	A	354-1186	7	35.3	Calculated composition (35 a/o Ag); similar to the above specimen.
34*	388	Pravoverov, N. L., and Savitskii, E. M.	1962	A	293-875	2	40.3	Calculated composition (40 a/o Ag); similar to the above specimen.
35*	389	Pravoverov, N. L., and Savitskii, E. M.	1962	A	303-1024	8	45.3	Calculated composition (45 a/o Ag); similar to the above specimen.

\* Not shown in figure.





TABLE 66. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF PALLADIUM + SILVER ALLOYS (Temperature Dependence) (continued)

$$\frac{\rho(T)}{\rho(296)}$$

T

## DATA SET 34\*

283	1.0077
313	1.0075
356	1.0069
378	1.0212
410	1.0424
428	1.0578
466	1.0651
534	1.1091
546	1.1427
720	1.1804
743	1.2081
836	1.2326
878	1.2481

$$\frac{\rho(T)}{\rho(296)}$$

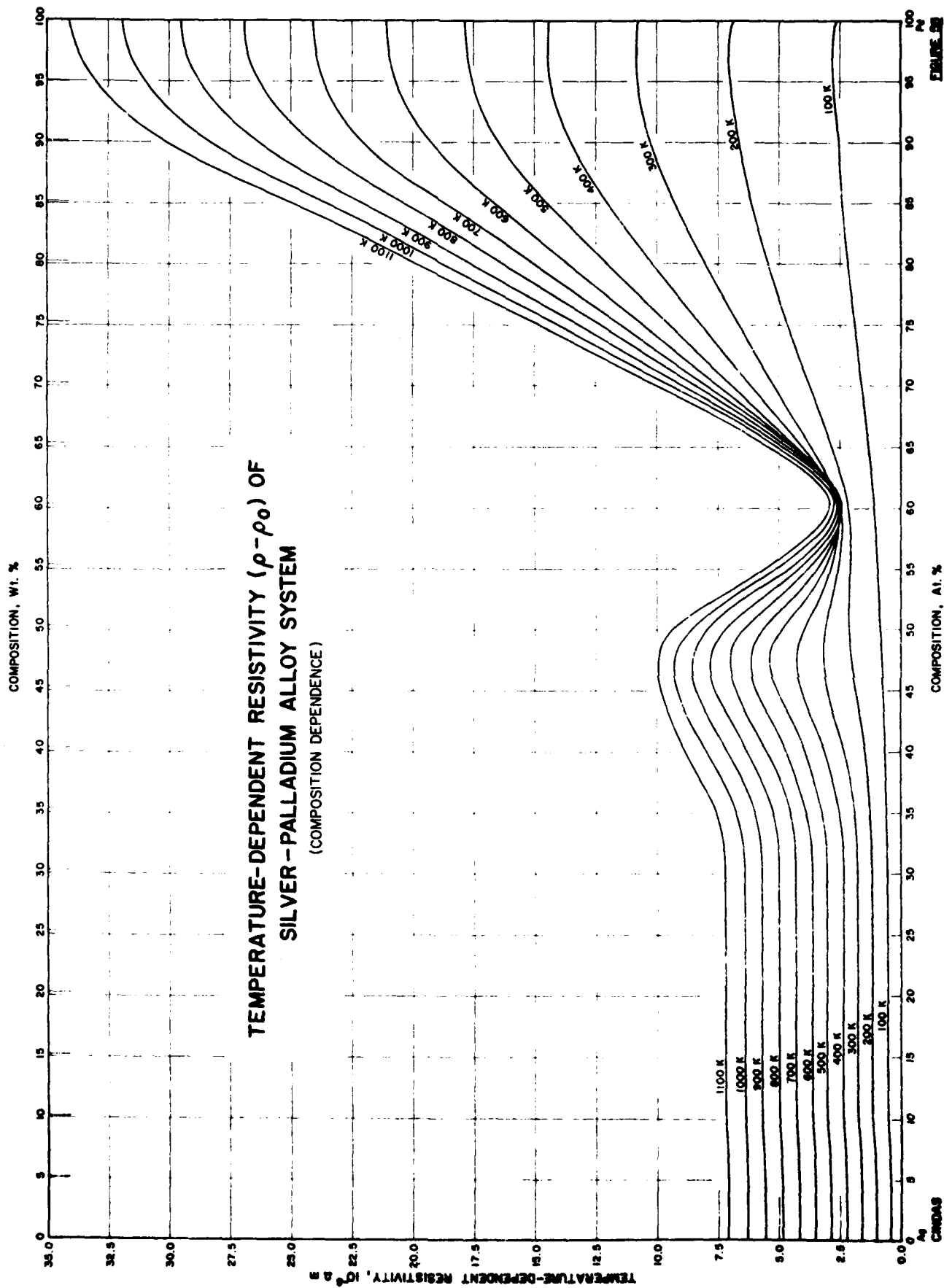
T

## DATA SET 35\*

303	1.0017
315	1.0010
335	1.0015
349	1.0063
390	1.0078
410	1.0085
452	1.0104
491	1.0157
518	1.0175
553	1.0195
575	1.0185
608	1.0187
616	1.0206
691	1.0221
775	1.0242
843	1.0223
912	1.0278
943	1.0282
992	1.0300
1094	1.0351

\* Not shown in figure.





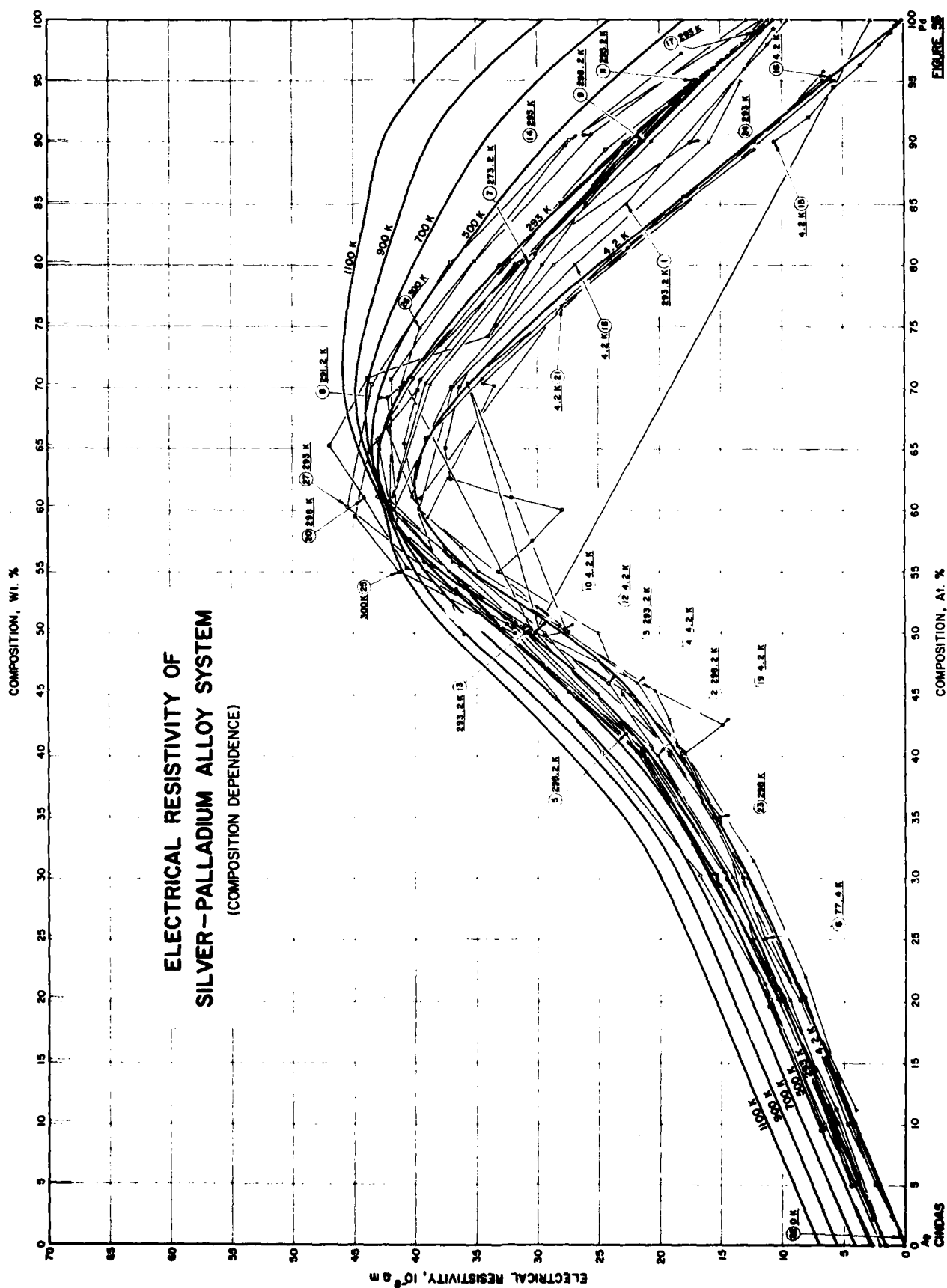


FIGURE 25

TABLE 69. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF SILVER-PALLADIUM ALLOY SYSTEM (Composition Dependence)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp., K	Composition Range (At. % of Pd)	Composition (weight percent), Specifications, and Remarks
1	312	Ricker, T.	1963	V	293.2	1-100	Compositions determined by chemical analysis; specimens prepared from 99.99 pure metals by arc-melting in inert gas; ingots homogenized and rolled to 0.1 mm thick; annealed at 1073 K for 1 h and air cooled; data extracted from table.
2	388	Seeman, H.J. and Rennelet, G.	1966	A	298.2	0-100	Prepared by melting in argon; rolled to 0.7 mm thick, annealed in vacuum at 1073 K for 30 min, rolled to 0.2 mm thick, and finally rolled to 0.1 mm thick after repeated annealing in vacuum; data extracted from table.
3	45	Coles, B. R. and Taylor, J. C.	1962	A	293.2	0-100	Compositions determined by chemical analysis, spectrographic analysis of selected samples showed no abnormal incidence of impurities; wire specimens 1.524 mm (0.060 in) in diam supplied by Precious Metals Division of Mond Nickel Co.; prepared from 99.95% pure metals by melting in argon; annealed in vacuum at 1023 to 1173 K; temperature coefficients of resistivity also reported; $\pm 1\frac{1}{2}$ uncertainty in cross-sectional area; temperature-dependent resistivities read from figure and added to residual resistivities of the following curve.
4	45	Coles, B. R. and Taylor, J. C.	1962	A	4.2	11.1-99.4	The above specimens; data read from figure.
5	373	Westerlund, T. W. and Nicholson, M. E.	1966	P	298	9.53-53.53	Compositions determined by gravimetric analysis; prepared from American Platinum Co. 99.99 pure silver and Englehard Industries Inc. 99.9 pure palladium by induction melting under vacuum in recrystallized alumina crucibles, homogenizing in vacuum of $5 \times 10^{-5}$ mm Hg at 1223 K for 1 week, cold-reducing, annealing in vacuum at 873 K for 2 h, and cold-rolling to 0.002 in thick; measured in annealed state in an oil bath maintained at $298 \pm 3$ K; specimen thickness determined using linear x-ray absorption; data read from figure.
6	370	Aarts, W. H. and Houston-MacMillan, A. S.	1957		77.4	5-100	Wire specimens of about 0.3 mm diam; palladium from Pure Metals Division of Mond Nickel Co.; measured in liquid nitrogen bath; standard deviations from the measured resistivities of the 5, 10, 20, 25, 30, 35, 40, 45, 50, and 100 a/o Pd samples were 0.01, 0.04, 0.03, 0.28, 0.04, 0.24, 1.13, 0.17, 1.9, and $0.01 \times 10^{-8} \Omega m$ , respectively; data extracted from table.
7	369	Geibel, W.	1911		273.2	0-100	Wire specimens of 0.1 mm diam; data extracted from table.
8	221	Svensson, B.	1932	B	291.2	0-100	Specimens 1 mm square and 42 mm long; prepared from Heraeus physically pure palladium and Hilger spectroscopically pure silver; annealed at 1273 to 1323 K for 10 to 12 h; data extracted from table.
9	286	Schulze, F. A.	1911		248.2	0-100	Wire specimens of 1 mm diam supplied by Firma Heraeus; data extracted from table.
10	376	Kemp, W. R. G., Klemens, P. G., Sreedhar, A. K., and White, G. K.	1956	B	4.2	2.11-95.06	Wire specimens supplied by Johnson, Matthey and Co.; 2.11 and 5.13 a/o Pd samples wound on mica former and annealed at 893 K; 10.02, 20.30, and 29.91 a/o Pd samples wound on mica former and annealed at 923 K; remaining samples annealed at 1153 K, wound on mica former, and reannealed at 773 K; data extracted from table.
11	376	Kemp, W. R. G., et al.	1956	B	293.2	2.11-95.06	The above specimens.
12	376	Kemp, W. R. G., et al.	1956	B	4.2	2.11-95.06	Rod specimens supplied by Johnson, Matthey and Co.; 2.11 and 5.13 a/o Pd samples annealed at 893 K; 10.02 a/o Pd sample annealed at 923 K; 20.30 and 29.91 a/o Pd samples annealed at 1073 K; remaining samples annealed at 1153 K; data extracted from table.
13	376	Kemp, W. R. G., et al.	1956	B	293.2	2.11-95.06	The above specimens.

TABLE 69. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF SILVER-PALLADIUM ALLOY SYSTEM (Composition Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp., K	Composition Range (At. % of Pd)	Composition (weight percent), Specifications, and Remarks
14	287	Rudolfskii, A. A.	1956		293	0-100	Manufactured in high-frequency furnace, drawn to 0.5 mm diam wire, annealed at 973 to 1073 K for 7 d; measurement temperature not stated explicitly, assumed to be 293 K; data read from figure; original work by Nemilov, V. A., Rudnitskii, A. A., and Vitusova, T. A. (Izv. Sektora platiny, 20, 225, 1946) but this document unavailable.
15	390	Greig, D. and Livesey, D.	1972	A	4.2	90.05-99.95	Foil specimens about 0.3 mm thick prepared by M. J. Walker, University of Leeds using Pd from Johnson, Matthey and Co. Ltd.; constituents melted in argon arc-furnace, formed into 'button' ingots, cold-rolled between plastic sheets, annealed at 1173 K for at least 16 h in sealed evacuated quartz tubes, and cooled at about 1 K min <sup>-1</sup> ; Hall coefficients at 4.2 K and room temperature also reported; residual resistivity obtained from resistance ratio by method which assumes Matthiessen's rule, with the fractional error in $\rho_0$ the same as the fractional deviation from the rule at room temperature; measured ratios of resistance at room temperature to resistance at 4.2 K were 1.99, 2.36, 2.87, 4.03, 6.42, 10.0, 16.4, 22.1, 46.7, and 81.5 for the 90.05, 92, 94.44, 96.3, 98, 99.01, 99.52, 99.75, 99.9 and 99.95 a/o Pd alloys respectively; data extracted from table.
16	32	Fletcher, R. and Greig, D.	1967	A	4.2	90.27, 95.22	Compositions determined by chemical analysis; prepared by International Nickel Ltd., annealed at 973 K for 24 h, outgassed 4 to 5 h at 773 K; residual resistivity values reported, assumed to be measured at 4.2 K; data extracted from table.
17	391	Zwingmann, G.	1963	V, B	293	94.02-99.50	Average purity of component metals 99.9; final wire specimens 0.8 mm in diam; components melted in DEGUSSIT AL23 or Zr23 crucible partly under Ar and partly under vacuum of 10 <sup>-3</sup> Torr, formed into round ingots of 9 mm diam, homogenized partly under Ar and partly under a vacuum of 10 <sup>-4</sup> Torr at 1173 to 1473 K, homogenizing times were 8 h for alloys with more than 5 a/o solute and 4-6 h for lesser concentrations, formed into wires with intermittent annealing, finally annealed in manner similar to the homogenization; change in resistivity upon alloying relative to pure Pd reported in figure; resistivity of pure Pd annealed at 923 K in high vacuum was $10.85 \times 10^{-6} \Omega \text{m}$ at 293 K; increase in resistivity per atomic percent Ag in Pd was $1.17 \times 10^{-4} \Omega \text{m}$ .
18	392	Simon, P. R. F.	1965		4.2	1.2-95.0	Spectrographically pure components; wire specimens 1 mm in diam and 100 mm long supplied by Johnson, Matthey, and Co. Ltd.; annealed under vacuum for 36 h at 955 K; data read from figure.

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ELECTRICAL RESISTIVITY OF TEN SELECTED BINARY ALLOY  
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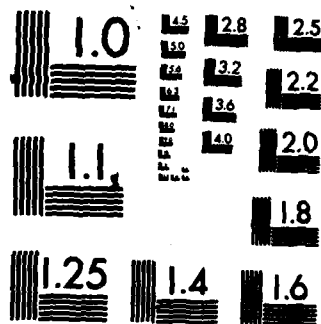
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TABLE 89. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF SILVER-PALLADIUM ALLOY SYSTEM (Composition Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp., K	Composition Range (At. % of Pd)	Composition (weight percent), Specifications, and Remarks
19	393	Gednault, A. M.	1974		4.2	1-95.8	Wire specimens from three sources: most were prepared by M. J. Walker, University of Leeds by melting, turning, and remelting the pure metal sponges in an arc furnace and then forming 0.5 mm diam wires by rolling and drawing through tungsten and diamond dies and finally annealing in vacuum; others were selected samples of Kemp, Klemens, Sreedhar, and White (Ag + Pd data sets 29-44 and Pd + Ag data sets 13-18); one sample was from earlier work by Gednault; these sources are denoted by [L], [K], and [G], respectively; compositions of all but the sample from [G] were determined by chemical analysis and trace impurities by mass spectrographic analysis by Fulmer Research Institute; the sources and wt. % impurities of the samples in the order in which they appear in the experimental data table are: [G], not determined; [L], 0.0020 Fe, 0.0005 Cu; [L], 0.0015 Cu, 0.0005 Fe; [L], 0.0020 Fe, 0.0010 Cu; [K], 0.0150 Cd, 0.0100 Cu, 0.0030 Fe, 0.0010 Pt, 0.0010 Zn; [K], 0.0150 Cd, 0.0100 Cu, 0.0100 Fe, 0.0010 Pt, 0.0010 Zn; [L], 0.0015 Cu, 0.0010 Fe; [L], 0.0030 Fe, 0.0015 Cu; [K], 0.1500 Fe, 0.0300 Cu, 0.0040 Pb, 0.0040 Zn, 0.0015 Ni; [L], 0.0005 Cu, 0.0003 Fe; [L], 0.0050 Cu, 0.0010 Zn; [K], 0.0050 Cu, 0.0030 Fe; [L], 0.0030 Ni, 0.0030 Pt, 0.0010 Au, 0.0010 Zn; [L], 0.0050 Cu, 0.0030 Fe; [L], 0.0050 Cu, 0.0050 Fe, 0.0010 Zn; [L], 0.0005 Cu, 0.0003 Fe; analysis of resistivity and thermopower of the 49.8 a/o Pd sample indicated a more likely composition of 51 a/o Pd, the 42.5 a/o Pd sample had an irregular large-grained appearance; data extracted from table.
20	394	Carson, A. W., Lewis, F. A., and Scharf, W. H.	1966		298	45.20-89.92	Wire specimens 0.122 and 0.274 mm in diam prepared by arc-melting under argon; fcc lattice parameters of the 50.74, 60.93, 70.78, 74.16, 80.91, and 89.92 a/o Pd alloys were 3.981, 3.962, 3.940, 3.937, 3.916, and 3.903 Å, respectively; data extracted from table.
21	395	Szafranski, A. W.	1973	A	4.2	60.9-95.0	Foil specimens 0.01 x 1 x 10 mm were cold-rolled and not annealed; data read from figure.
22	398	Barber, A. J. and Caplin, A. D.	1975	A	0	0.0078-1.07	Nominal compositions; wire specimens 0.25 mm to 0.89 mm in diam prepared by melting high purity Ag and Pd in quartz ampoules, homogenizing, rolling, and drawing; annealed at 873 K for 6-10 h in sealed capsule with up to 10 Torr oxygen or in a continuous flow of oxygen at 10 <sup>-1</sup> Torr in order to precipitate Fe; measured in helium environment; less than 1% uncertainty in geometrical factor; uncertainty in measurement temperatures less than ±100 mK; lowest measurement temperature was 1.5 K, residual resistivities obtained by extrapolation to 0 K were extracted from table.
23	349	Savitskii, E. M. and Pravoverov, N. L.	1961	A	298	5-95	20 g charges of 99.98 pure Pd and 99.99 pure Ag fused under molten borax in alumina crucibles in a high-frequency furnace, forged while hot, rolled on manual rolling mill without intermediate annealing, and finally annealed in evacuated quartz tubes at 1273 K for 124 h and at 1073 K for 200 h; loss in weight was not observed after fusion; microstructural analysis showed polyhedral crystals of solid solutions with well-defined grain boundaries for all compositions; authors concluded existence of ordered compounds at 50 and 80 a/o Pd upon annealing at 1273 to 1073 K; data read from figure.
24	396	Couper, A. and Metcalfe, A.	1966		293	0.6-99.2	Specific resistivity of filament specimens; measurement temperature not stated explicitly, assumed to be 293 K; data read from figure.

TABLE 69. MEASUREMENT INFORMATION ON THE ELECTRICAL RESISTIVITY OF SILVER-PALLADIUM ALLOY SYSTEM (Composition Dependence) (continued)

Data Set No.	Ref. No.	Author(s)	Year	Method Used	Temp., K	Composition Range (At. % of Pd)	Composition (weight percent), Specifications, and Remarks
25	384	Arajs, S., Rao, K. V., Yao, Y. D., and Teoh, W.	1977	A	~300	50.0-59.7	99.99 pure Pd sponge from Engelhard Industries Inc. and 99.998 pure Ag from American Smelting and Refining Co. arc-melted together to form ingot weighing about 10 g; ingots sealed in evacuated silica tubes, homogenized at about 1300 K for 60 h, drawn to 1 mm <sup>2</sup> cross-section, cleaned, re-sealed in evacuated silica capsules, annealed at 1200 K for 1 h, and rapidly cooled in air without breaking the capsules; data read from figure; pre-print kindly supplied by K. V. Rao in private communication.
26	397	Rao, K. V., Rapp, Ö, Johanneson, C., and Aström, H. U.	1977		300	49.9-97.2	Data not yet published at time of preparation of this report; authors will publish in "Electrical Resistivity of Fe Dissolved in Pd-Ag Matrix Alloys," Physica B, International Conference on Magnetism, 1976 Proceedings; the data were presented concomitantly in the pre-print of the article by Arajs et al. (see data set 25 of this table) which was supplied by K. V. Rao in private communication; no measurement information available; data read from figure.
27	398	Krüger, F. and Gehm, G.	1933		283	0-100	Homogenized in high vacuum for several hours between 1023 and 1073 K, quenched in water at room temperature; lattice constants of the 0, 10, 20, 30, 40, 50, 60, 70, 80, 90, and 100 at/o Pd alloys were 4.078, 4.052, 4.030, 4.011, 3.990, 3.970, 3.950, 3.933, 3.916, 3.899, and 3.884 Å respectively; measurement temperature was room temperature, assumed to be 293 K; data extracted from table.



TABLE 70. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF SILVER-PALLADIUM ALLOY SYSTEM (Composition Dependence)

[Composition, C, at/o Ag; Electrical Resistivity,  $\rho$ ,  $10^{-8}$  Qm]

DATA SET 1 ( $T = 293.2$ K)			DATA SET 3 (cont.) ( $T = 293.2$ K)			DATA SET 7 ( $T = 273.2$ K)			DATA SET 10 ( $T = 4.2$ K)			DATA SET 13 (cont.) ( $T = 293.2$ K)			DATA SET 17 (cont.) ( $T = 293$ K)		
C	$\rho$		C	$\rho$		C	$\rho$		C	$\rho$		C	$\rho$		C	$\rho$	
1	2.0		65.8	43.0		0	1.57		2.11	0.925		50.34	30.5		98.00	13.14	
4	3.2		70.7	40.1		10.08	6.06		5.13	2.21		70.29	40.9*		99.00	11.98	
10	5.5		80.5	32.3		20.16	10.44		10.02	4.39*		95.06	16.8		99.50	11.35	
20	10.0		89.4	22.4		30.22	15.41		20.30	8.45		DATA SET 14 ( $T = 293$ K)			DATA SET 18 ( $T = 4.2$ K)		
30	15.4		100	10.6		40.26	21.74		29.91	12.86		0	2.6		1.2	0.5	
40	21.3		DATA SET 4 ( $T = 4.2$ K)			50.27	32.79		40.33	19.16		09.4	6.9		9.9	4.7	
50	29.4		11.1	4.0		60.27	42.02		50.34	29.86		19.5	11.1		20.0	8.6*	
60	41.7		20.1	8.4		70.23	38.76		70.29	35.60		29.6	15.3		40.0	18.3	
70	37.0		30.5	13.1		80.17	30.67		95.06	6.61		40.5	21.5		43.0	19.3	
80	28.6		40.8	18.6		90.10	20.62		DATA SET 11 ( $T = 293.2$ K)			50.8	31.8		43.3	20.4	
85	22.7		45.9	21.9		100	10.56		2.11	2.54		60.7	42.0		50.0	27.7	
98	11.1		50.6	28.2		DATA SET 8 ( $T = 291.2$ K)			5.13	3.81		70.6	41.9		52.1	30.0	
100	9.5		55.8	36.9		0	1.6*		10.02	6.035*		80.3	35.1		53.0	34.8	
DATA SET 2 ( $T = 298.2$ K)			61.0	40.2		10.3	6.0		20.30	10.00		90.6	25.5		56.9	36.2	
0	1.66		65.8	39.1		21.6	10.8		29.91	14.55		100	12.8		60.0	39.6	
10	5.71		70.7	34.5		29.3	15.1		40.33	21.40		DATA SET 15 ( $T = 4.2$ K)			70.0	36.9	
20	9.44		80.5	24.1		40.0	21.4		50.34	33.00*		90.0	15.9		80.0	26.9	
30	14.1		89.4	12.2		50.5	31.9		70.29	40.90*		95.0	10.6		90.0	15.9	
40	20.2		DATA SET 5 ( $T = 298$ K)			58.5	41.3		95.06	17.20		DATA SET 19 ( $T = 4.2$ K)			95.0	13.3	
45	25.2		9.53	6.6		69.1	42.2		DATA SET 12 ( $T = 4.2$ K)			92	7.73		DATA SET 20 ( $T = 296$ K)		
50	30.6		21.43	11.4		70.3	40.8		2.11	0.89*		94.44	5.66		1	0.43	
55	37.2		32.81	17.3		85.1	28.0		5.13	2.20*		96.3	3.48		2.4	1.09	
60	42.0		41.80	22.8		94.4	17.8		10.02	4.15		98	1.95		9.8	4.13	
65	42.9		53.53	36.6		100	10.6*		20.30	8.25		99.01	0.95		21.9	8.2	
70	41.2		DATA SET 6 ( $T = 77.4$ K)			DATA SET 9 ( $T = 298.2$ K)			29.91	12.78*		99.75	0.50		31.4	12.4	
80	33.0		5	2.46		0	2.01		40.33	18.10		99.9	0.23*		40.2	18.0	
90	22.8		10	4.43		10.1	6.20		50.34	27.70		DATA SET 16 ( $T = 4.2$ K)			48.6	29.5	
100	11.1		20	8.61		20.2	10.56*		70.29	35.60*		90.27	12.18		49.4	14.9	
DATA SET 3 ( $T = 293.2$ K)			25	11.48		30.3	15.55		95.06	5.61		95.22	5.92		50.1	27.4	
0	1.6*		30	13.25		40.3	21.93		DATA SET 13 ( $T = 293.2$ K)			90.27	12.18		70.1	33.5	
11.1	5.6		35	15.60		60.3	42.02*		2.11	2.53*		95.22	5.92		70.3	34.4	
20.1	10.0		40	19.22		70.3	39.06		5.13	3.91*		DATA SET 17 ( $T = 293$ K)			81.4	22.6	
30.5	14.8		45	22.45		90.1	21.23		10.02	6.00*		94.02	17.63		91.7	10.3	
40.8	20.8		50	30.20		100	11.51		20.30	10.10*		96.01	15.50		95.8	6.5	
45.9	24.2		55.8	39.2		Not shown in figure.			40.33	21.10		DATA SET 20 ( $T = 296$ K)			45.20	37.42	
50.6	30.9		61.0	42.9								97.00	14.31				

\* Not shown in figure.

TABLE 70. EXPERIMENTAL DATA ON THE ELECTRICAL RESISTIVITY OF SILVER-PALLADIUM ALLOY SYSTEM (Composition Dependence) (continued)

DATA SET 20 (cont.) (T = 298 K)		DATA SET 23 (cont.) (T = 298 K)		DATA SET 27 (cont.) (T = 293 K)	
C	P	C	P	C	P
50.74	32.49	70	36.28	50.27	33.1*
55.25	40.60	75	33.33	60.27	45.5
60.93	44.14	80	29.53	70.23	43.5
65.24	46.96	85	26.00	80.17	37.0
70.78	43.81	90	21.53	90.10	27.3
74.16	33.96	95	16.53	100.0	11.6*
80.91	30.11				
89.93	22.56				
DATA SET 21 (T = 4.2 K)		DATA SET 24 (T = 293 K)			
60.9	39.6	0.6	2.00*		
76.7	28.0	10.1	6.14*		
85.6	18.0	20.3	10.24		
90.1	12.2	40.5	21.36*		
95.0	5.7	60.8	39.84		
		65.4	40.87		
		70.5	39.54		
		80.1	31.77		
		90.1	17.46		
		99.2	10.60		
DATA SET 22 (T = 0 K)		DATA SET 25 (T = 300 K)			
0.0078	0.00607	50.0	31.88		
0.091	0.0415*	55.0	41.28		
0.091	0.0423*	59.5	44.86		
0.66	0.2900	6.9	43.70		
0.90	0.4000*	7	39.73		
1.07	0.4720*				
DATA SET 23 (T = 298 K)		DATA SET 26 (T = 300 K)			
5	4.38	49.9	35.95		
10	5.93*	66.7	44.28		
15	8.00	74.8	39.54		
20	9.85	89.7	27.62		
25	12.40	97.2	18.12		
30	14.05*				
35	15.10				
40	18.03				
45	23.03				
50	24.95				
55	33.20				
57.5	30.35				
60	27.93				
61	32.10				
62.5	37.08				
65	37.43				
		DATA SET 27 (T = 293 K)			
		0.0	1.7*		
		10.08	6.6		
		20.16	11.0		
		30.22	16.7		
		40.26	24.7		

\* Not shown in figure.

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## 5. APPENDICES

### 5.1. Analysis of the Electrical Resistivity of Gold-Silver Alloy System

In deriving recommended values of  $\rho(c,T)$  for binary alloys from experimental results for various temperatures  $T$  and mole fractions  $c$  of one component, one faces the considerable problem of correlating the results in such a way as to smooth out, so far as is possible, the effects of the various errors that may influence the individual measurements. When the full range of  $T$  and  $c$  is to be dealt with, this cannot be done by getting a "best fit" (in some sense) of a smooth approximating function of  $T$  and  $c$  to the reported values of  $\rho$ , since it is not clear what manifold of approximating functions should be considered. Given, as in the case of the Au-Ag alloy system, values of  $\rho$  measured for a reasonably large number of values of  $T$  on samples with a reasonable number of values of  $c$ , cross-plotting is effective.

The precise tactics to be employed in cross-plotting may depend on the data available in a particular case. One may, for instance, begin with plots of measured  $\rho$  against  $T$  for each value of  $c$  for which sufficient data is available. Through each set of points one can draw a smooth curve, or  $T$ -plot, to represent  $\rho(c,T)$  for that value of  $c$ . The choice of such curves is always subject to arbitrariness, but an immediate check is available if one can assume that  $\rho(c,T)$  is a smoothly varying function of  $c$ : one can plot values of  $\rho$  read from these curves against  $c$ , for various test temperatures  $T_t$ . If each such set of points lies on a smooth curve, this  $c$ -plot can be taken to represent the variation of  $\rho(c,T)$  with  $c$  for the given  $T_t$ . If not, the original  $T$ -plots should be re-examined and revised until an acceptably smooth  $c$ -plot can be passed through the points corresponding to each  $T_t$ . When this has been accomplished, one can derive from the revised  $T$ -plots a set of  $c$ -plots for any desired set of temperatures  $T_1$ , and from these, by cross-plotting again, a set of  $T$ -plots for any desired set of  $c_j$ . Smoothing can be introduced at each step in repeated cross-plotting, but care must be taken that the end result of multiple smoothing remains consistent with the original data.

If one lacks assurance that  $\rho(c,T)$  is a smooth function of both  $c$  and  $T$ , the value of cross-plotting is correspondingly reduced. It may also be

noted that scattered values of  $\rho(c,T)$  are of little value during the process of cross-plotting, but may be useful in testing the final results.

Instead of cross-plotting  $\rho(c,T)$ , it is often better to work with

$$\rho_1(c,T) = \rho(c,T) - \rho_0(c), \quad (A1)$$

where  $\rho_0(c)$  is the residual resistivity of the alloy (or, usually, with adequate accuracy, its measured resistivity at 4K). This temperature-dependent part of  $\rho(c,T)$  changes more slowly with  $c$  than does the total  $\rho$ , and its isotherms may form a family more convenient for cross-plotting. In the case of the Au-Ag system it has proved very useful to go further in this direction, to work with the quantity

$$\Delta(c,T) = \rho_1(c,T) - \bar{\rho}_1(c,T), \quad (A2)$$

where  $\bar{\rho}_1$  is the atomic-fraction-weighted average of the intrinsic resistivities of the pure metals:

$$\bar{\rho}_1(c,T) = c\rho_1^{(Ag)}(T) + (1-c)\rho_1^{(Au)}(T) \quad (A3)$$

with  $c \equiv c_{Ag}$ . The motivation for this choice will be clear from figure A-1, where results of measurements of Giaque and Stout [297] (Au + Ag data sets 24-26, Ag + Au data sets 21 and 22, and Au-Ag data sets 16-20) on a series of alloys ( $c = 0.1$  to  $c = 0.9$ ) are presented by plotting  $\rho_1(c,T)/T$  against  $c$ , for seven temperatures ranging from 14 K to 298 K. Corresponding values for small  $c$ , derived from the results of Stewart and Huebener [301] (Au + Ag data sets 59 and 60 and Au-Ag data set 14), and for small  $(1-c)$ , derived from the results of Dugdale and Basinski [29] (Au-Ag data set 15) are also given for these temperatures. Comparison of the plotted values of  $\rho_1/T$  for each temperature with the straight line between the end points (the pure metals), which represents  $\bar{\rho}_1(c,T)/T$ , shows that  $\Delta/T$  is small and varies only slowly with  $c$  for  $0.2 \leq c \leq 0.8$ . By definition, it vanishes as  $c$  approaches 0 or 1 at any  $T$ , and (barring highly unlikely behavior of  $\rho$  at low temperatures) as  $T$  goes to 0 for any  $c$ . Figure A-1 also shows that  $\Delta/T$  (like  $\rho_1/T$ ,  $\rho_1^{(Ag)}/T$ , and  $\rho_1^{(Au)}/T$ ) changes only slowly with  $T$  at high temperatures, and even down to quite low temperatures.

The quantity  $\Delta$ , which is at most a few percent of  $\rho$ , is the small correction to the good approximation

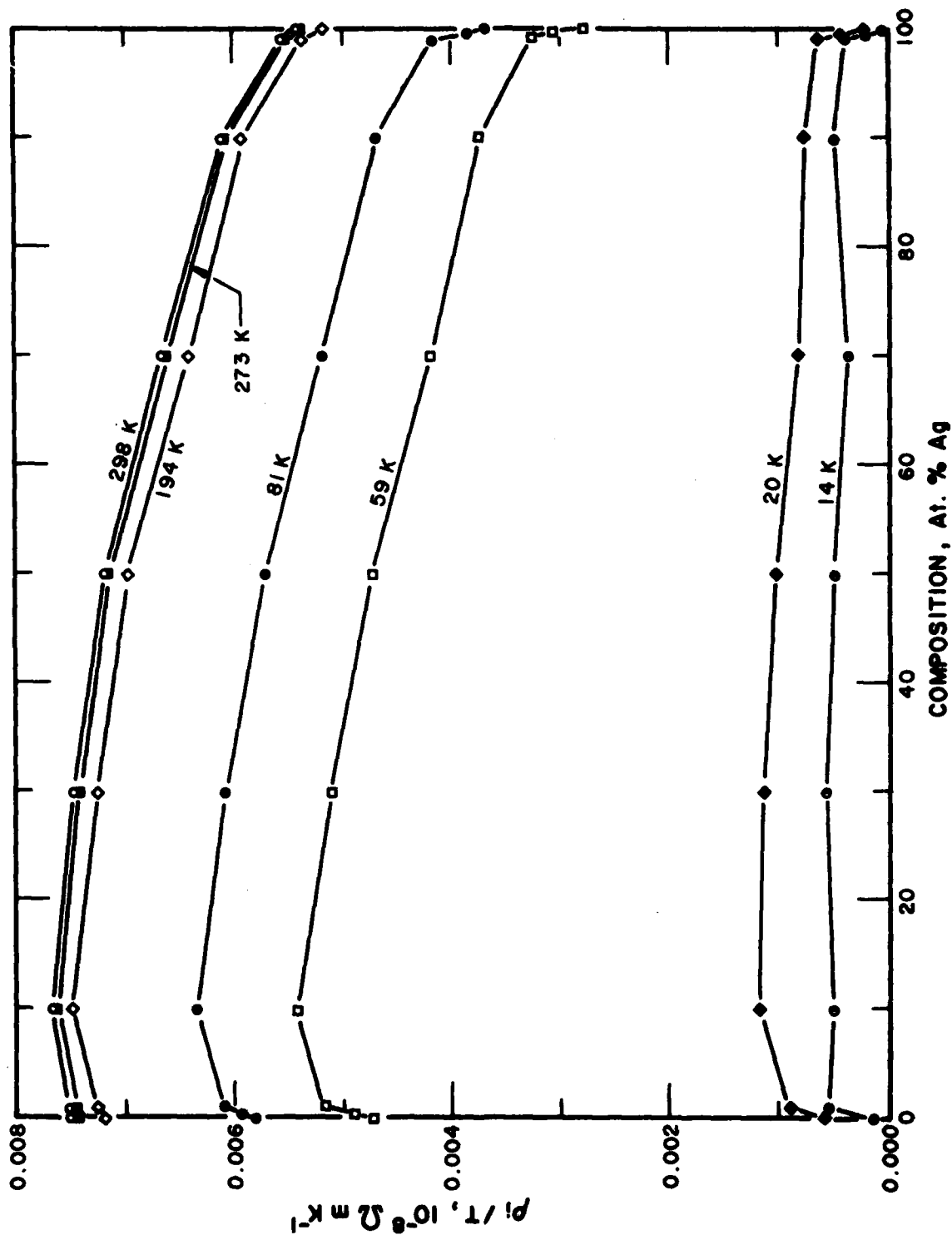


Figure A-1.  $\rho_l(c, T)/T$  as a Function of T. Pure Metal Values Recommended by CINDAS, Values for  $c = 0.25$  and  $1 \text{ At. \%}$  from data of Stewart and Huebner [30], for  $c = 10$  to  $90 \text{ At. \%}$  from data of Glauxue and Stout [29], and for  $c = 99$  and  $99.5 \text{ At. \%}$  from data of Dugdale and Basinski [29].

$$\rho(c,T) \simeq \rho_0(c) + \bar{\rho}_1(c,T). \quad (A4)$$

Irregularities and discrepancies in values of  $\rho(c,T)$  become very conspicuous in plots of  $\Delta$  or  $\Delta/T$ , and this increases the confidence with which one can reject some data as aberrant or unreliable. Cross-plotting of  $\Delta$  is convenient up to 300 K, and cross-plotting of  $\Delta/T$  down to 100 K; the range of overlap in  $T$  makes it useful to employ both types of plot in relating low- $T$  data to high- $T$  data.

As examples, figures A-2 and A-3 show values of  $\Delta/T$  derived from five sets of measurements, for  $c_{Ag} = 0.5$  and  $0.9$ , respectively. The indicated curves were derived by cross-plotting values of  $\Delta$ , starting from data on samples with  $c = 0.05, 0.1, 0.3, 0.5, 0.7, 0.75, 0.9, 0.95$ , and  $0.97$  for which the authors had measured  $\rho(c,4K)$  as well as  $\rho$  at higher temperatures. (As will be discussed later, availability of  $\rho(c,4K)$  eliminates some sources of error in the determination of  $\Delta$ .) The value of  $\rho(c,4K)$  was available for all data illustrated in figures A-2 and A-3 except that of Iyer and Asimow [302] (Au + Ag data sets 2-5, Ag + Au data sets 11-13, and Au-Ag data set 46), for which  $\rho_0(c)$  had to be deduced from a  $\rho_0$  versus  $c$  curve and the stated  $c$ . The correction for thermal expansion of the samples, made by Iyer and Asimow, has been removed in calculating  $\Delta/T$  in order to maintain uniformity in the neglect of this correction. Data of Grüneisen and Reddemann [128] (Au + Ag data sets 27 and 28 and Ag + Au data sets 23 and 24) is not represented in figure A-2, since two of the three values of  $\Delta/T$  derived from their measurements lie outside the range of the figure.

The results of cross-plotting  $\Delta$  to lower temperatures is illustrated, for the Ag-rich alloys, in figure A-4. The crosses represent values read from cross-plots of  $\Delta/T$ . Also shown are values of  $\Delta$  for  $c = 0.99$  and  $0.995$ , as deduced from the results of Dugdale and Basinski [29] (Au-Ag data set 15), who were primarily concerned with deviations from Matthiessen's Rule; their correction for thermal expansion was removed, but not their correction for change of volume on alloying, which they describe as small. These data are consistent with the idea (not necessarily correct) that  $\Delta$  increases monotonically with  $c$  at all  $T$ , and this idea has been used as a guide in extrapolating to the very small corrections at the lowest  $T$ . For small  $c$  a similar relation was found between values of  $\Delta$  derived from cross-

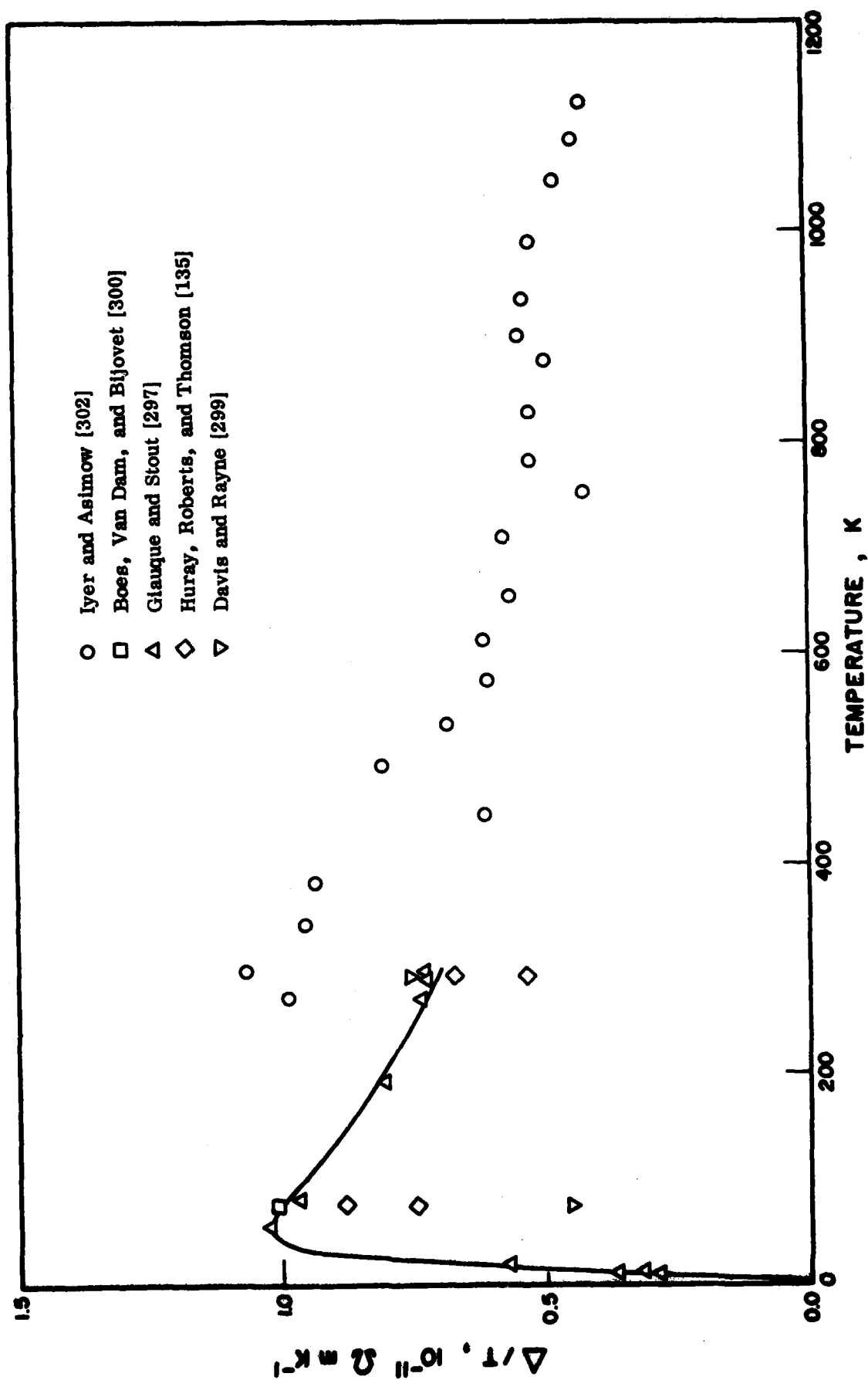


Figure A-2.  $\Delta/T$  as a Function of T for  $c_{\text{Ag}} = 50 \text{ At.}\%$ . Data Points and Estimated Curve at 300 K.



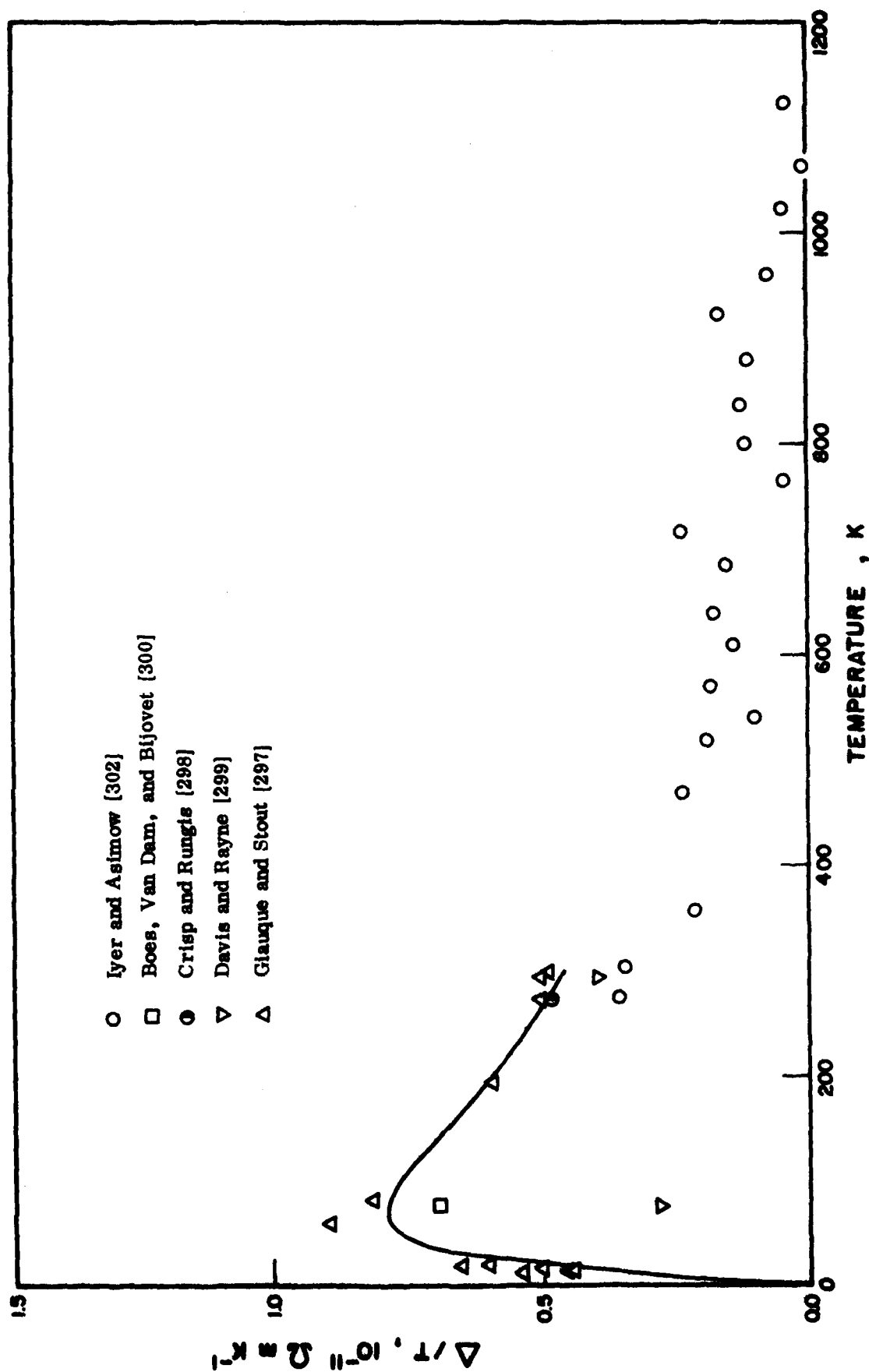


Figure A-3.  $\Delta/T$  as a Function of  $T$ , for  $c_{Ag} = 90 \text{ At.}\%$ . Data Points and Estimated Curve to 300 K.

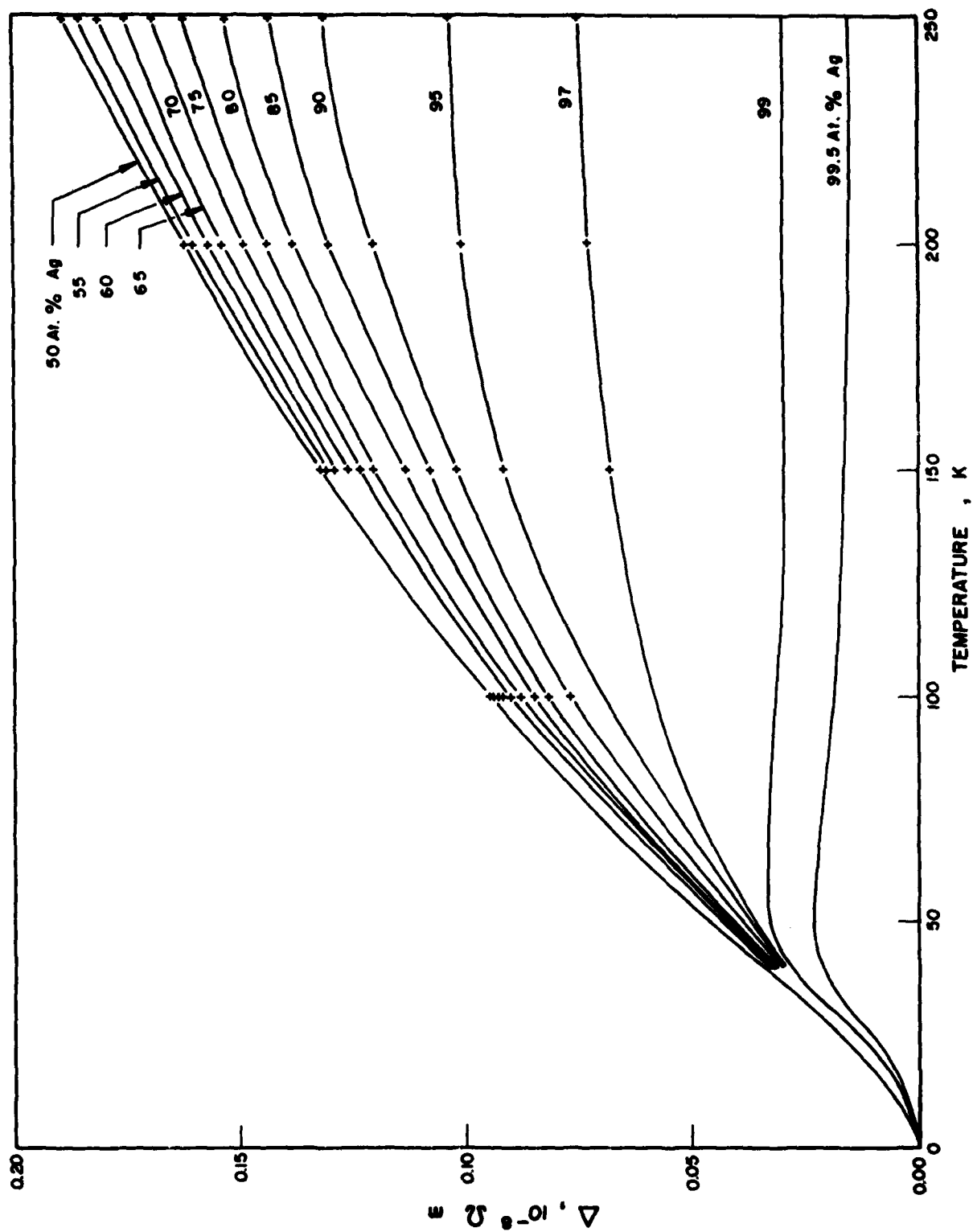


Figure A-4.  $\Delta$  as a Function of T, for Silver-Rich Alloys, as Derived from Cross Plots of  $\Delta/T$ . Values for  $c_{Ag} = 99$  and 99.5 At. % are Derived from Data of Dugdale and Basinski [29], as Described in Text.

plots of  $\Delta/T$  for  $c \geq 0.05$  and values derived from the results of Stewart and Huebener [301] (Au + Ag data sets 59 and 60 and Au-Ag data set 14) for  $c \leq 0.01$ . For interpolation to low values of  $c$  it is convenient to use cross-plots of  $\rho(c,T)/c$  against  $c$ .

For the solid above room temperature the only data useful in cross-plotting are those of Iyer and Asimow [302], for  $c = 0.05, 0.1, 0.3, 0.5, 0.7, 0.9$ , and  $0.95$ . As is illustrated in figures A-2 and A-3, values of  $\Delta$  derived from their data do not always join smoothly onto values derived from other data at room temperatures and below. This appears to be due in part to errors that (as concerns data obtained for a single sample at many temperatures) may be regarded as systematic, and that are absent from the low-T values of  $\Delta$ . The low-T values were computed as

$$\Delta(c,T) = \rho(c,T) - \rho(c,4K) - \bar{\rho}_1(c,T) \quad (A5)$$

with the first two terms on the right measured on the same sample. An error  $\delta c$  in control of the  $c$  for the sample will change the sum of these two terms by just  $(\partial \rho_1 / \partial c) \delta c$ . Errors in determination of the dimensions of the sample, said by Iyer and Asimow to be the major source of error in their results, will change the sum of these two terms by a fixed fraction of  $\rho_1$ , which we write as  $\rho_1 \delta l$ , with  $\delta l$  a measure of the overall effect of dimensional errors. Together, these errors change  $\Delta$  by

$$\delta \Delta = \frac{\partial \rho_1}{\partial c} \delta c + \rho_1 \delta l. \quad (A6)$$

Both terms on the right tend to be small because  $\rho_1$  is smaller than  $\rho$  and is, indeed, very small at low temperatures. In processing the data of Iyer and Asimow, on the other hand, it was necessary to use eq (A2) instead of eq (A5). The errors  $\delta c$  and  $\delta l$  then effect only the first term on the right of eq (A2), and there is no partially cancelling contribution from the second term; in addition, there is an independent error,  $-\delta \rho_0(c)$ , arising from any error  $\delta \rho_0(c)$  in the value assigned to  $\rho_0$  at the nominal  $c$ . Together, these sources of error will change  $\Delta$  by

$$\delta \Delta = \left\{ \frac{\partial \rho_0}{\partial c} + \frac{\partial \rho_1}{\partial c} \right\} \delta c + \{\rho_0 + \rho_1\} \delta l - \delta \rho_0(c). \quad (A7)$$

The difference between eqs (A6) and (A7) represents a systematic difference between the  $\Delta$ 's derived from the low-T data and the data of Iyer and Asimow, respectively.

Since the sources of error should be less important for the low-T results, it seems reasonable to use those results as reference points in reducing the effects of the errors  $\delta c$ ,  $\delta l$  and  $\delta \rho_0$  on the values of  $\Delta$  derived from the data of Iyer and Asimow. This can be done by adjusting the available parameters to make the corrected values  $\Delta - \delta \Delta$  for the high-T data continue smoothly, in the region 300-400 K, the trend established by cross-plotting the low-T data. The process involves no great arbitrariness because  $\delta \Delta$  for any sample is nearly a linear combination of two known functions of T: a constant, and  $\bar{\rho}_1(c, T)$ . For the Ag-Au alloys one has, with all the accuracy needed in computing this small correction,

$$\rho_1 \simeq \bar{\rho}_1, \quad (\text{A8})$$

$$\frac{\partial \rho_1}{\partial c} \simeq \frac{\partial \bar{\rho}_1}{\partial c} = \rho_1^{(\text{Ag})}(T) - \rho_1^{(\text{Au})}(T). \quad (\text{A9})$$

Further, though  $\bar{\rho}_1$  and  $(\partial \bar{\rho}_1 / \partial c)$  involve the metal resistivities in quite different ways, they change with T in nearly the same way because the T-dependences of the pure-metal resistivities are so similar. This is illustrated in table A-1.

Table A-1. Values of  $\rho(T)/\rho(300 \text{ K})$

T, K	$\frac{\rho_1^{(\text{Ag})}(T) - \rho_1^{(\text{Au})}(T)}{\rho_1^{(\text{Ag})}(300\text{K}) - \rho_1^{(\text{Au})}(300\text{K})}$	$\bar{\rho}_1(T) / \bar{\rho}_1(300 \text{ K})$				
		Mole Fraction of Silver				
		0.1	0.3	0.5	0.7	0.9
300	1	1	1	1	1	1
600	2.13	2.15	2.15	2.15	2.15	2.15
900	3.54	3.45	3.44	3.44	3.43	3.42
1200	5.43	5.01	4.98	4.95	4.92	4.89

Since  $\rho_1^{(\text{Ag})} - \rho_1^{(\text{Au})}$  is about a third as large as  $\rho_1$ , and only small corrections are involved, one can reasonably take the T-dependence of the second of the five terms on the right in eq (A7) to be the same as that of the fourth. Replacing  $(\partial \rho_1 / \partial c)$  by  $\kappa \rho_1(c)$ , we have

$$\delta\Delta \approx \left\{ \frac{\partial \rho_0}{\partial c} \delta c + \rho_0(c) \delta l - \delta \rho_0(c) \right\} + \bar{\rho}_1(T, c) \{ \kappa \delta c + \delta l \}, \quad (A10)$$

where the quantities in braces are independent of  $T$ . Adjustment of these two constants to achieve a smooth fit of high- $T$  to low- $T$  data does not, of course, make it possible to estimate the three errors  $\delta c$ ,  $\delta l$ ,  $\delta \rho_0$  for a particular high- $T$  sample.

In treating the Au-Ag alloys, the fit between high- $T$  and low- $T$   $\Delta$ 's was actually made on plots of  $\Delta/T$ , such as figures A-2 and A-3. Analysis of the low- $T$  data yielded plots that were nearly linear in  $T$  from 100 to 300 K, and appeared to fix  $\Delta$  at 300 K with an error not exceeding  $0.03 \times 10^{-8} \Omega m$  for any  $c$ . For each of the seven compositions mentioned above, smooth curves were then passed through the high- $T$  data points, with the intention of smoothing out the effect of random errors. This was done as carefully as possible by eye and French curve, without consideration of the relation of curves for different  $c$ , except in one case:  $c = 0.3$ . For this composition the three points derived from the data of Iyer and Asimow for  $293 \text{ K} \leq T \leq 350 \text{ K}$  were far out of line with the other points, and a curve passed through them could not be made to fit in with other curves, either before or after adding the correction  $-\delta\Delta$ . These three points were therefore ignored, and the smoothed curve was passed through the remaining points with knowledge of, but no detailed attention to, the trend of the corresponding curves for other compositions. Each smoothed high- $T$  curve was then matched to the corresponding smoothed low- $T$  curve by addition of a correction  $-\delta\Delta$  that brought it to the same value at 300 K and at 400 K. The  $T$ -plots formed by this union of the high- $T$  and low- $T$  parts were then tested by cross-plotting. The results were acceptable, and further study of the possibilities led to no change in the initial  $T$ -plots. From these  $T$ -plots a complete set of  $c$ -plots was derived, and from these in turn a complete set of  $T$ -plots.

Figure A-5 illustrates the effect of the correction  $-\delta\Delta$  on the high- $T$  data shown in figure A-3; it also shows the final  $T$ -plot for  $c = 0.9$ . Figure A-6 shows the final  $c$ -plots of  $\Delta/T$ , which are particularly useful for interpolation or extrapolation at high  $T$ .

It is now necessary to consider the uncertainty in the values of  $\rho(c, T)$  derived by adding the estimated  $\Delta(c, T)$  to the interpolative approxi-

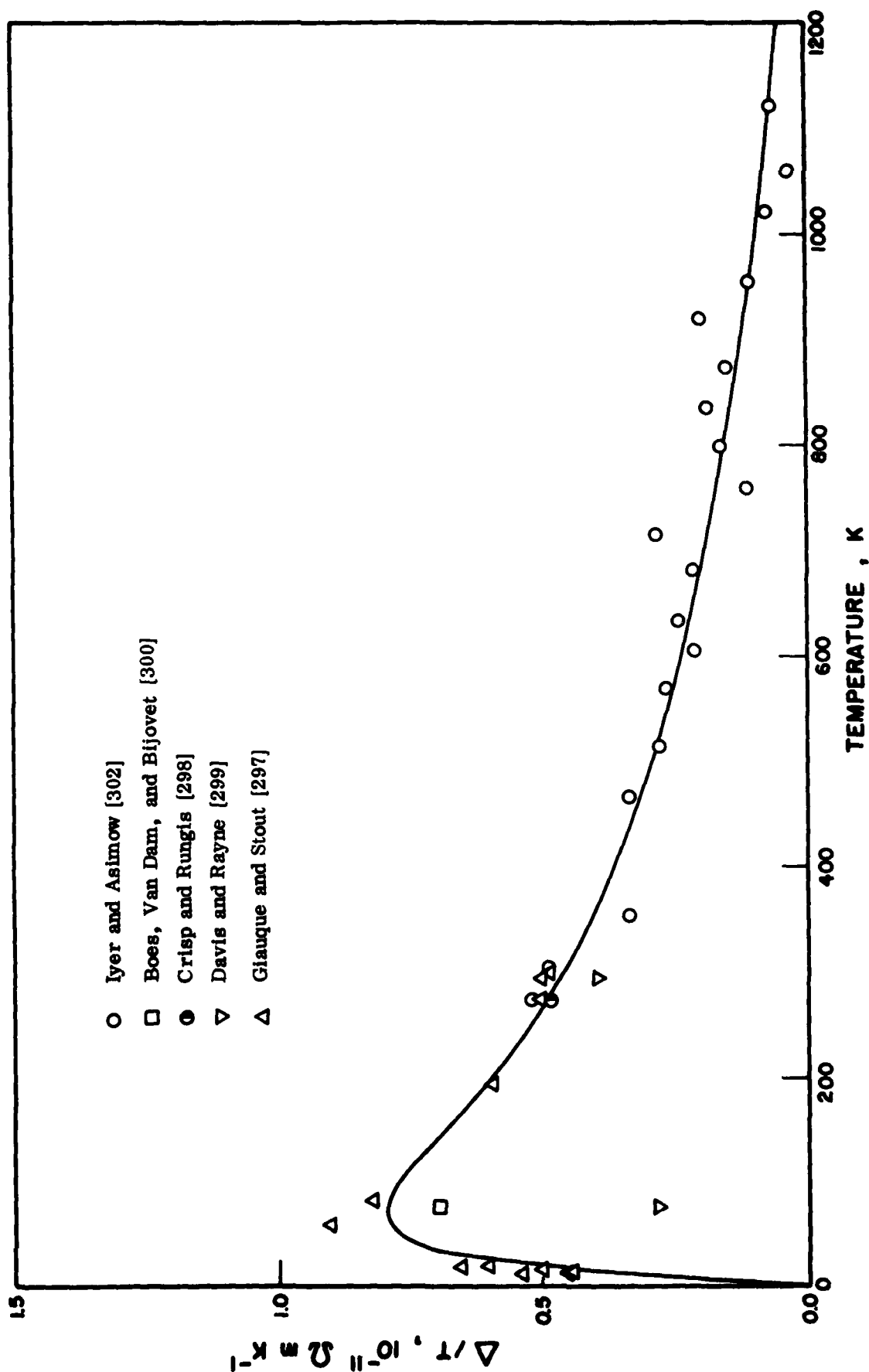


Figure A-5.  $\Delta/T$  as a Function of  $T$ , for  $c_{Ag} = 90$  At.%. Data Points of Iyer and Asimow are Adjusted as Described in Text, and Estimated Curve is Extended to 1200 K.

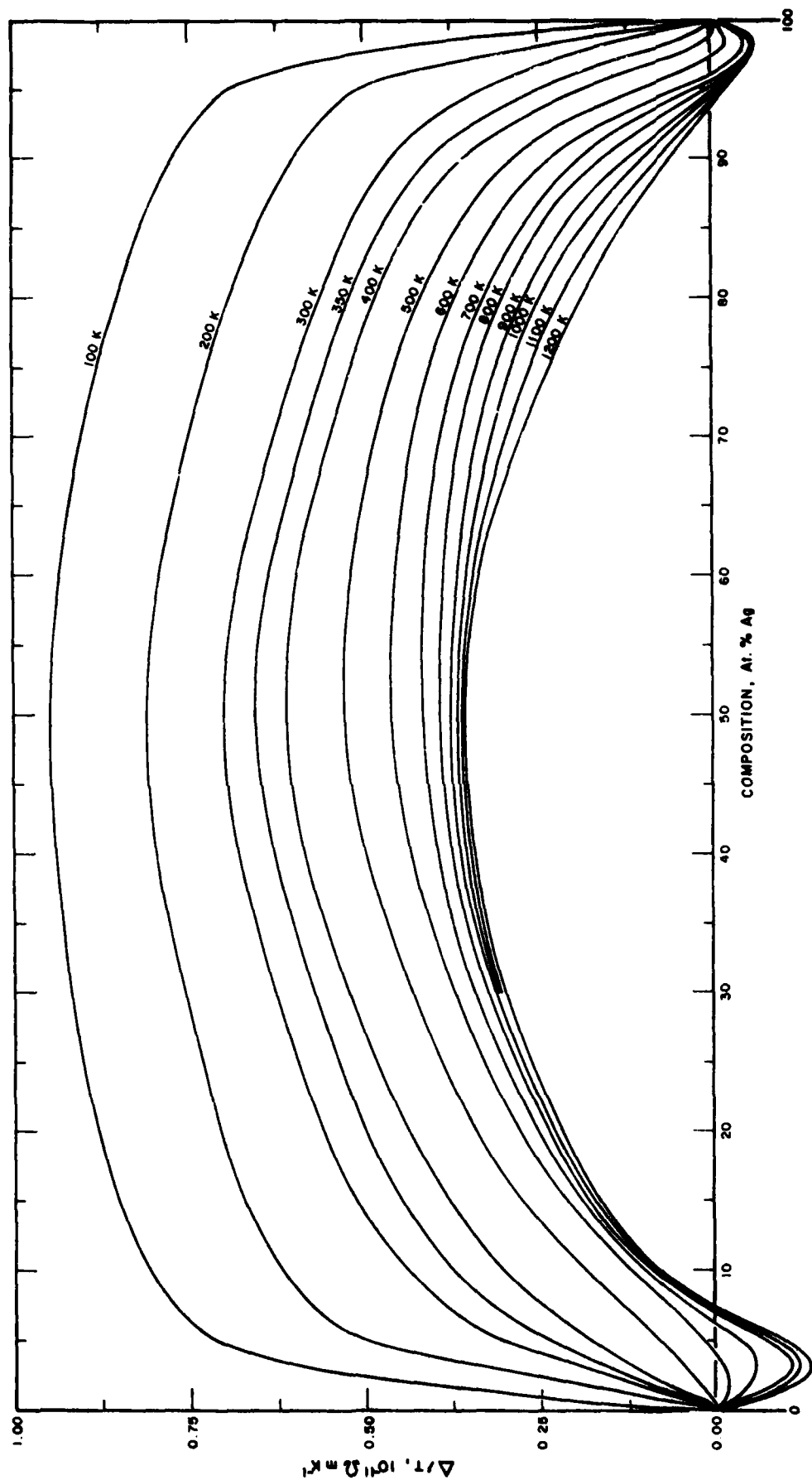


Figure A-6. Final Plot of  $\Delta/T$  as a Function of Composition for Fixed Temperatures.

mation given by eq (A4). This uncertainty arises, not just from the uncertainties involved in constructing plots of  $\Delta$ , but from the fact that one must use throughout, not the exact values of  $\rho_0(c)$  for alloys and of  $\rho_1(T)$  for the pure metals, but the best available approximations.

We consider first the uncertainties in the process of correcting the high-T data, of which a detailed study was made in the case of  $c = 0.5$ . It was found that alternative choices of the smoothed curve through the uncorrected data points, together with different ways in which the smoothed high-T curves could be fitted to the low-T curve without an evident discontinuity in the slope, could shift the inferred value of  $\Delta$  at 1000 K only by  $\pm 0.1 \cdot 10^{-8} \Omega m$ . Since independent adjustments of this magnitude in the T-plots for different  $c$  would make trouble with the c-plots,  $\pm 0.1 \cdot 10^{-8} \Omega m$  seems to be a reasonable, and even high, estimate of the uncertainty in  $\Delta$  at 1000 K due to uncertainties in the fitting process. To this error of fitting one must add the uncertainty of the low-temperature  $\Delta$  at 300 K. Estimates of the possible constructional errors in  $\Delta$ , intended to be safe maxima, are shown in table A-2. These apply to the construction of all the original T-plots, for  $c$  ranging from 0.05 to 0.95.

Table A-2. Estimated Maximum Error in Constructing T-Plots

T, K	Uncertainty in $\Delta$ , $10^{-8} \Omega m$
100	0.015
200	0.02
400	0.04
600	0.06
800	0.085
1000	0.11
1200	0.14

In this method of analysis, the primary reason for subtracting  $\rho_0(c)$  and  $\bar{\rho}_1(c, T)$  from the observed  $\rho(c, T)$  is to get a smaller or more slowly changing quantity ( $\Delta(c, T)$ , or  $\Delta/T$ , or  $\Delta/c$ ) to study. Then inevitable errors in the assumed  $\rho_0(c)$ ,  $\rho_1^{(Ag)}(T)$ , and  $\rho_1^{(Au)}(T)$ , if smoothly changing, will result only in smooth changes in the c-plots and T-plots that represent  $\Delta$ . Smoothness of the plots constructed from given data on  $\rho$  may thus provide some check on the smoothness, but none on the accuracy of the assumed  $\rho_0$  and  $\rho_1$ . To the extent that the analysis only smooths out the



contribution to  $\Delta$  of the experimental errors in  $\rho$ , recommended values of  $\rho(c,T)$  derived as

$$\rho(c,T) = \rho_0(c) + \bar{\rho}_1(c,T) + \Delta(c,T) \quad (A11)$$

will not contain errors due to the inaccuracies in  $\rho_0$  or  $\bar{\rho}_1$ , these being cancelled out by the difference between the "exact" and the derived values of  $\Delta$ . If, however, one constrains the derived  $\Delta$  to have special properties known to be possessed by the exact  $\Delta$ , this will no longer be the case. If, for instance, one imposes on the derived  $\Delta$ 's the condition that they vanish as  $c \rightarrow 0$ , as  $c \rightarrow 1$ , and as  $T \rightarrow 0$ , then errors in the assumed  $\rho_0(c)$  will necessarily appear as errors in the derived  $\rho(c,0K)$ , and errors in the assumed  $\rho_1^{(Ag)}$  and  $\rho_1^{(Au)}$  will appear as errors in the derived  $\rho(1,T)$  and  $\rho(0,T)$ , respectively. This is the best one can do unless the data under consideration justify a different extrapolation to the indicated limits — in which case one would in effect be using them to arrive at improved estimates of  $\rho_0$  and of  $\rho_1$ .

In the present case, to extrapolate the alloy data smoothly:  $c = 0, 1$  would yield values of the  $\Delta$ 's that are negative, and values of  $\rho(0,T)$  and  $\rho(1,T)$  that are unacceptably low. The plots of  $\Delta$  in figure A-6 have therefore been constructed subject to the condition that the  $\Delta$ 's vanish as  $c \rightarrow 0, 1$ . The possibility of removing the somewhat ill-defined minima near the ends of the curves, by reinterpretation of the data for  $c = 0.05, 0.1, 0.9$  and  $0.95$ , has been examined. The minima for small  $c$  can be removed by changing the corrections discussed above, while maintaining the smoothness of cross-plots, but this cannot be done with the minima for large  $c$ . The initial and most natural interpretation of the data is that presented in figure A-6.

An estimate of the constructional uncertainties remaining in the  $\Delta$ 's of figure A-6, after the full cross-plotting procedure, is given in table A-3; to these should be added any systematic errors not removed in the reconciling of high-T and low-T data.

Table A-3. Estimated Uncertainties in the Final Values of  $\Delta$ 

T, K	Uncertainty in $\Delta$ , $10^{-8} \Omega m$		
	Mole Fraction of Silver		
	0.01	0.03-0.97	0.99
100	0.005	0.015	0.01
200	0.005	0.02	0.01
400	0.02	0.03	0.02
600	0.025	0.04	0.02
800	0.03	0.06	0.03
1000	0.04	0.08	0.04
1200	0.05	0.10	0.05

The uncertainties in the values of  $\rho$  computed using eq (A11) have been estimated as follows. For the more concentrated alloys,  $c$  or  $1-c$  equal to or greater than  $0.05 \cdot 10^{-8} \Omega m$ , the estimated error has been taken to be the estimated error in  $\rho_0$  (in the first row of table A-4) plus twice the estimated constructional error. Inclusion of the first term reflects the doubt that the uncertainty in  $\rho_0$  has been much reduced in the cross-plotting, and the doubling of the second term is intended to allow for systematic errors not removed in that process. For the most dilute alloys, the estimated error has been taken to be the estimated constructional error in  $\Delta$ , plus the uncertainty in  $\rho_0$ , plus the uncertainty in the  $\rho_1$  of the pure solvent metal (first or last column of table A-4). The estimated uncertainties are shown in table A-4, both in absolute terms, and as a percentage of the estimated  $\rho$ .

The low estimates of errors for the concentrated alloys,  $0.3 \leq c \leq 0.7$ , reflect some optimism that the correlation of high-T and low-T data have reduced (by about 50%) the errors estimated by Iyer and Asimow for the high temperature data; the percentage error in  $\rho$  due to the uncertainty in  $\Delta$  also tends to be small because  $\rho$  is itself so large. The percentage uncertainty is decidedly larger for the more dilute ( $c = 0.01-0.03$ ) alloys, for which observations are entirely lacking for  $T > 373$  K, and interpolations of  $\Delta$  are relatively insecure.

Table A-4. Estimated Uncertainties in the Recommended Values for the Electrical Resistivity of Gold-Silver Alloy System.

T, K	Uncertainty in $\rho$ , $10^{-8} \Omega m$ and % (in parentheses)														
	Mole Fraction of Silver														
	0.00	0.01	0.03	0.05	0.10	0.30	0.50	0.70	0.90	0.95	0.97	0.99	1.00		
0	0	0.01	0.03	0.05	0.05	0.05	0.05	0.05	0.05	0.05	0.03	0.01	0		
50	0.004 (2.0)	0.019 (3.6)	0.039 (3.5)	0.058 (3.5)	0.058 (1.9)	0.058 (0.8)	0.058 (0.6)	0.058 (0.7)	0.058 (1.7)	0.058 (3.1)	0.049 (4.2)	0.019 (3.9)	0.004 (4.0)		
100	0.008 (1.3)	0.028 (2.9)	0.053 (3.4)	0.065 (3.0)	0.065 (1.9)	0.065 (0.8)	0.065 (0.7)	0.065 (0.8)	0.065 (1.7)	0.065 (3.0)	0.066 (4.4)	0.026 (3.3)	0.006 (1.5)		
200	0.015 (1.0)	0.035 (2.0)	0.055 (2.3)	0.07 (2.4)	0.07 (1.6)	0.07 (0.8)	0.07 (0.7)	0.07 (0.8)	0.07 (1.6)	0.07 (2.5)	0.08 (3.7)	0.035 (2.5)	0.010 (1.0)		
300	0.011 (0.5)	0.051 (2.0)	0.066 (2.1)	0.08 (2.1)	0.08 (1.6)	0.08 (0.9)	0.08 (0.7)	0.08 (0.8)	0.08 (1.6)	0.08 (2.3)	0.09 (3.3)	0.05 (2.5)	0.010 (0.6)		
400	0.03 (1.0)	0.08 (2.4)	0.09 (2.3)	0.09 (2.0)	0.09 (1.5)	0.09 (0.9)	0.09 (0.7)	0.09 (0.8)	0.09 (1.6)	0.09 (2.2)	0.11 (3.3)	0.07 (2.7)	0.02 (0.9)		
600	0.10 (2.1)	0.16 (3.1)	0.17 (3.0)	0.10 (1.6)	0.10 (1.3)	0.10 (0.9)	0.10 (0.7)	0.10 (0.8)	0.10 (1.3)	0.10 (1.9)	0.16 (3.5)	0.10 (2.6)	0.05 (1.4)		
800	0.15 (2.2)	0.22 (3.2)	0.24 (3.2)	0.11 (1.4)	0.11 (1.2)	0.11 (0.8)	0.11 (0.7)	0.11 (0.8)	0.11 (1.3)	0.11 (1.7)	0.22 (3.7)	0.14 (2.7)	0.07 (1.4)		
1000	0.22 (2.5)	0.31 (3.4)	0.33 (3.5)	0.13 (1.3)	0.13 (1.1)	0.13 (0.8)	0.13 (0.8)	0.13 (0.9)	0.13 (1.3)	0.13 (1.6)	0.29 (3.9)	0.19 (2.9)	0.10 (1.6)		
1200	0.33 (2.9)	0.44 (3.9)	0.46 (3.9)	0.15 (1.2)	0.15 (1.1)	0.15 (0.8)	0.15 (0.8)	0.15 (0.9)	0.15 (1.3)	0.15 (1.5)	0.36 (4.0)	0.24 (2.9)	0.13 (1.6)		

## 5.2. Methods for the Measurement of Electrical Resistivity.

At the Center for Information and Numerical Data Analysis and Synthesis (CINDAS) of Purdue University, the experimental methods for the measurement of electrical resistivity have been classified into various categories according to a similar scheme used by CINDAS for the classification of methods for the measurement of thermal conductivity [399, pp. 13a-25a]. This classification scheme of CINDAS is presented below. Note that the letters in parentheses following the respective methods are the code letter used in the "Method Used" column of the Table of Measurement Information for indicating the experimental methods used by the various authors.

### Methods for the Measurement of Electrical Resistivity

#### A. Steady-State Methods

1. Voltmeter and ammeter direct reading method (V) [400, p. 159; 401, pp. 244-5]
2. Direct-current potentiometer method (A) [19, pp. 151-8]
  - a. 4-probe potentiometer method
3. Direct-current bridge methods (B) [19, pp. 144-51]
  - a. Kelvin double bridge method
  - b. Mueller bridge method
  - c. Wheatstone bridge method
4. Van der Pauw method (P) [402, 403]
5. Galvanometer amplifier method (G) [12, pp. 159-62]

#### B. Non-stead-state Methods

1. Periodic current method
  - a. Direct connection to sample
    - (1) Alternating-current potentiometer method (C) [19, pp. 161-2]
    - (2) Alternating-current bridge method (D) [19, p. 162]
  - b. No connection to sample
    - (1) Mutual inductance method (M) [404]
    - (2) Self-inductance method (S) [405]
    - (3) Rotating field method (R) [406]
2. Non-periodic current method
  - a. Direct connection to sample
    - (1) Transient (subsecond) method (T) [407]
  - b. No connection to sample
    - (1) Eddy current decay method (E) [408; 19, p. 103]

### 5.3. Conversion Factors for the Units of Electrical Resistivity

The recommended values and experimental data for the electrical resistivity of alloys tabulated in this work are in the units:  $10^{-8} \Omega \text{ m}$ . Conversion factors for the units of electrical resistivity, which may be used to convert the values given in ( $10^{-8} \Omega \text{ m}$ ) to values in other units, are given in table A-5.

Table A-5. Conversion Factors for the Units of Electrical Resistivity

Units to Convert to	Multiply the Value Given in ( $10^{-8} \Omega \text{ m}$ ) by
ohm-meter ( $\Omega \text{ m}$ )	$1 \times 10^{-8}$
ohm-centimeter ( $\Omega \text{ cm}$ )	$1 \times 10^{-6}$
ohm-inch ( $\Omega \text{ in.}$ )	$3.937 \times 10^{-7}$
ohm-foot ( $\Omega \text{ ft}$ )	$3.281 \times 10^{-8}$
microohm-centimeter ( $\mu\Omega \text{ cm}$ )	1
abohm-centimeter ( $\text{ab}\Omega \text{ cm}$ )	$1 \times 10^3$
statohm-centimeter ( $\text{stat}\Omega \text{ cm}$ )	$1.113 \times 10^{-10}$
emu (= $\text{ab}\Omega \text{ cm}$ )	$1 \times 10^3$
esu (= $\text{stat}\Omega \text{ cm}$ )	$1.113 \times 10^{-10}$
ohm-circular mil per foot ( $\Omega \text{ cmil ft}^{-1}$ )	6.015

Example:  $1.000 \times 10^{-8} \Omega \text{ m} = 3.937 \times 10^{-7} \Omega \text{ in.}$

## 6. REFERENCES

1. Ho, C.Y., Ackerman, M.W., Wu, K.Y., Oh, S.G., and Havill, T.N., "Thermal Conductivity of Ten Selected Binary Alloy Systems," J. Phys. Chem. Ref. Data, 7(3), 959-1177, 1978.
2. Matthiessen, A., "Electrical Resistivity of Alloys," Ann. Physik, 110, 190-221, 1860.
3. Matthiessen, A. and Vogt, C., "The Influence of Temperature on the Electrical Conductivity of Alloys," Ann. Physik, 122, 19-78, 1864.
4. Sondheimer, E.H., "The Theory of the Transport Phenomena in Metals," Proc. Roy. Soc. London, A203(1072), 75-98, 1950.
5. Sommerfeld, A. and Bethe, H., "Electron Theory of Metals," in Handbuch der Physik, Vol. 24/2, Springer-Verlag, Berlin, 333-622, 1933.
6. Mott, N.F. and Jones, H., The Theory of the Properties of Metals and Alloys, Clarendon Press, Oxford, England, 1936; Dover Publications, New York, 326 pp., 1958.
7. Wilson, A.H., The Theory of Metals, Cambridge University Press, 1936; 2nd Edition, 346 pp., 1953.
8. Bardeen, J., "Electrical Conductivity of Metals," J. Appl. Phys., 11, 88-111, 1940.
9. Seitz, F., The Modern Theory of Solids, McGraw-Hill Book Co., New York, 698 pp., 1940.
10. Mott, N.F., "Recent Advances in the Electron Theory of Metals," Progr. Metal Phys., 3, 76-114, 1952.
11. Peierls, R.E., Quantum Theory of Solids, Oxford University Press, 229 pp., 1955.
12. MacDonald, D.K.C., "Electrical Conductivity of Metals and Alloys at Low Temperatures," in Handbuch der Physik, Vol. 14, Springer-Verlag, Berlin, 137-97, 1956.
13. Gerritsen, A.N., "Metallic Conductivity, Experimental Part," in Handbuch der Physik, Vol. 19, Springer-Verlag, Berlin, 137-226, 1956.

14. Jones, H., "Theory of Electrical and Thermal Conductivity in Metals," in Handbuch der Physik, Vol. 19, Springer-Verlag, Berlin, 227-315, 1956.
15. Dekker, A.J., Solid State Physics, Prentice-Hall, Inc., Englewood Cliffs, N.J., 540 pp., 1957.
16. Ziman, J.M., Electrons and Phonons, Oxford University Press, 554 pp., 1960.
17. Olsen, J.L., Electron Transport in Metals, Interscience Publishers, New York, 121 pp., 1962.
18. Rosenberg, H.M., Low Temperature Solid State Physics, Oxford University Press, 420 pp., 1963.
19. Meaden, G.T., Electrical Resistance of Metals, Plenum Press, New York, 218 pp., 1965.
20. Blatt, F.J., Physics of Electronic Conduction in Solids, McGraw-Hill Book Co., New York, 446 pp., 1968.
21. Allen, P.B. and Butler, W.H., "Electrical Conduction in Metals," Phys. Today, 31(12), 44-9, 1978.
22. Bloch, F., "On the Quantum Mechanics of Electrons in a Crystalline Lattice," Z. Physik, 52, 555-600, 1928.
23. Bloch, F., "The Electrical Resistance Law at Low Temperatures," Z. Physik, 59, 208-14, 1930.
24. Gruneisen, E., "The Dependence of the Electrical Resistance of Pure Metals on the Temperature," Ann. Physik, 16, 530-40, 1933.
25. Meissner, W., "Galvanic and Thermomagnetic Effects," Handbuch der Experimentelle Physik, Vol. 11, Part 2, 311-94, 1935.
26. White, G.K., Experimental Techniques in Low Temperature Physics, Clarendon Press, Oxford, 328 pp., 1959.
27. Howie, A., "The Electrical Resistivity of Stacking Faults," Phil. Mag., Ser. 8, 5(51), 251-71, 1960.
28. Sondheimer, E.H. and Wilson, A.H., "The Theory of the Magneto-Resistance Effects in Metals," Proc. Roy. Soc. London, A190(1023), 435-55, 1947.
29. Dugdale, J.S. and Basinski, Z.S., "Matthiessen's Rule and Anisotropic Relaxation Times," Phys. Rev., 157(3), 552-60, 1967.

30. Ziman, J.M., "Approximate Calculation of the Anisotropy of the Relaxation Time of the Conduction Electrons in the Noble Metals," *Phys. Rev.*, 121(5), 1320-4, 1961.
31. Campbell, I.A., Fert, A., and Pomeroy, A.R., "Evidence for Two Current Conduction Iron," *Phil. Mag.*, Ser. 8, 15(137), 977-83, 1967.
32. Fletcher, R. and Grieg, D., "The Lattice Thermal Conductivity of Some Palladium and Platinum Alloys," *Phil. Mag.*, 16(140), 303-15, 1967.
33. Bass, J., "Deviations from Matthiessen's Rule," *Adv. Phys.*, 21(91), 431-604, 1972.
34. Cimberle, M.R., Bobel, G., and Rizzuto, C., "Deviations from Matthiessen's Rule at Low Temperatures: An Experimental Comparison Between Various Metallic Alloy Systems," *Adv. Phys.*, 23(4), 639-71, 1974.
35. Nordheim, L., "The Electron Theory of Metals," *Ann. Physik*, 9(5), 607-78, 1931.
36. Norbury, A.L., "The Electrical Resistivity of Dilute Metallic Solid Solutions," *Trans. Faraday Soc.*, 16, 570-96, 1921.
37. Linde, J.O., "Electrical Properties of Dilute Mixed Crystal Alloys. I. Gold Alloys," *Ann. Physik*, 10, 52-70, 1931.
38. Linde, J.O., "Electrical Properties of Dilute Solid Solution Alloys. II. Resistivity of Silver Alloys," *Ann. Physik*, 14, 353-66, 1932.
39. Linde, J.O., "Electrical Properties of Dilute Mixed Crystal Alloys. III. Resistance of Copper- and Gold Alloys. Regularity of Increase of Resistance," *Ann. Physik*, 15, 219-48, 1932.
40. Mott, N.F., "The Electrical Resistance of Dilute Solid Solutions," *Proc. Camb. Phil. Soc.*, 32, 281-90, 1936.
41. Damask, A.C., "Some Resistivity Effects of Short-Range Order in Alpha-Brass," *J. Appl. Phys.*, 27(6), 610-6, 1956.
42. Damask, A.C., "Residual Resistivity vs. Short-Range Order in 75% Copper-25% Gold Alloy," *J. Phys. Chem. Solids*, 1(1/2), 23-6, 1956.
43. Korevaar, B.M., "The Influence of Lattice Defects on the Electrical Resistivity of a Gold-Copper Alloy - 7 at.% Copper," *Acta Metallurgica*, 6(9), 572-9, 1958.



44. Mott, N.F., "A Discussion of the Transition Metals on the Basis of Quantum Mechanics," Proc. Phys. Soc. (London), 47, 571-88, 1935.
45. Coles, B.R. and Taylor, J.C., "The Electrical Resistivities of the Palladium-Silver Alloys," Proc. Roy. Soc., A267(1328), 139-45, 1962.
46. Dugdale, J.S. and Grénault, A.M., "The Low Temperature Transport Properties of the Palladium-Silver Alloy Series," Phil. Mag., Ser. 8, 13(123), 503-13, 1966.
47. Vuillemin, J.J. and Priestley, M.G., "De Haas-Van Alphen Effect and Fermi Surface in Palladium," Phys. Rev. Letters, 14(9), 307-9, 1965.
48. Mott, N.F. and Stevens, K.W.H., "The Band Structure of the Transition Metals," Phil. Mag., Ser. 8, 2(20), 1364-86, 1957.
49. Coles, B.R., "Spin-Disorder Effects in the Electrical Resistivities of Metals and Alloys," Adv. Phys., 7(8), 40-71, 1958.
50. Matula, R.A., "Electrical Resistivity of Copper, Gold, Palladium, and Silver," J. Phys. Chem. Ref. Data, 8(4), 1147-298, 1979.
51. Chi, T.C., "Electrical Resistivity of Alkaline Earth Elements," J. Phys. Chem. Ref. Data, 8(2), 439-97, 1979.
52. Chu, T.K. and Ho, C.Y., "Electrical Resistivity of Chromium, Cobalt, Iron, and Nickel," Purdue University, CINDAS Report to NBS, in preparation.
53. Chu, T.K. and Ho, C.Y., "Electrical Resistivity of Molybdenum, Tungsten, and Zinc," Purdue University, CINDAS Report to NBS, in preparation.
54. Chi, T.C. and Ho, C.Y., "Electrical Resistivity of Aluminum, Manganese, Vanadium, and Zirconium," Purdue University, CINDAS Report to NBS, in preparation.
55. Smith, A.W., "The Thermal Conductivities of Alloys," Ohio State Univ., Eng. Expt. Sta. Bull., 31, 5-61, 1925.
56. Griffiths, E. and Schofield, F.H., "The Thermal and Electrical Conductivity of Some Aluminum Alloys and Bronzes," J. Inst. Metals, 39, 337-74, 1928.
57. Gaudig, W. and Warlimont, H., "Direct Observations of the Short-Range Order and of a Stable Superlattice Phase in Alpha Copper-Aluminum Alloys," Z. Metallk., 60(5), 488-98, 1969.

58. Panin, V.Ye., Zenkova, E.K., and Fadin, V.P., "Study of the Ordering Effect in the Alloy Cu-Al. I. Homogeneous Solid Solutions," *Fiz. Met. Metall.*, 13(1), 86-92, 1962; Engl. transl.: *Phys. Met. Metall.*, 13(1), 76-81, 1962.
59. Panin, V.Ye., Fadin, V.P., and Solov'yev, L.A., "Order in Cu-Al Alloys. II. Alloys Close to the Solubility Boundary," *Fiz. Met. Metall.*, 13(2), 219-24, 1962; Engl. transl.: *Phys. Met. Metall.*, 13(2), 59-62, 1962.
60. Hibbard, W.R., Jr., "The Effect of Cold Work on the Electrical Resistivity of Copper Solid Solution Alloys," *Acta Metall.*, 7(8), 565-74, 1959.
61. Wechsler, M.S. and Kernohan, R.H., "Neutron-Irradiation Effects on Copper Aluminum Alloys," *Phys. and Chem. of Solids*, 7(4), 307-26, 1958.
62. Gulyaev, A.P. and Trusova, E.F., "Mechanism of the Property Change in Solid Solutions," *Zh. Tekhn. Fiz.*, 20(1), 66-78, 1950.
63. Linde, J.O., "The Effect of Cold-Work on the Electrical Resistivity of Alloys and the Law of Recovery," *Appl. Sci. Res., Sec. B*, 4(1-2), 73-86, 1954.
64. Smith, C.S. and Palmer, E.W., "Thermal and Electrical Conductivities of Copper Alloys," *Am. Inst. Mining Metall. Eng., Tech. Publ. No. 648*, 19 pp., 1935.
65. Pecijare, O. and Janssen, S., "Electricity - A Study on the Resistance of  $V_2$  Phase of Alloys Cu-Al," *Compt. Rend.*, 245, 1228-30, 1957.
66. Janssen, S. and Pecijare, O., "Electricity - A Study on the Resistance of Alloys Cu-Al," *Compt. Rend.*, 244, 2017-20, 1957.
67. Köster, W. and Rothenbacher, P., "Conductivity and Hall Constant. 35. Influence of the Valency of B-Metals on the Change in Electrical Magnitude by Short Term Ordering in Ternary Alpha Mixed Crystals of Copper and Silver Alloys," *Z. Metallk.*, 58(2), 93-8, 1967.
68. Hishiyama, Y., "Ordinary Transport Properties of the  $\gamma$ -Phase Alloys," *J. Phys. Soc. Jpn.*, 22(2), 449-56, 1967.
69. Sinha, T.H.D. and Prasad, R.S., "Thermopower and Resistivity of Cu-Al Alloys," *Indian J. Pure Appl. Phys.*, 9(4), 246-8, 1971.

70. Weinberg, I., "Phonon-Drag Thermopower in Copper-Aluminum and Copper-Silicon Alloys," Phys. Rev., 139(3A), A838-43, 1965.
71. Chu, T.K. and Lipschultz, F.P., "A Study of Cyclic Fatigue Damage in Copper-Aluminum Alloys by Thermal Conductivity Measurements," J. Appl. Phys., 43(6), 2505-10, 1972.
72. Chu, T.K. and Lipschultz, F.P., "A Study of Cyclic Fatigue Damage in Copper-Aluminum Alloys by Thermal Conductivity Measurements," AFOSR-TR-72-0409, 21 pp., 1972.
73. Salter, J.A.M. and Charsley, P., "The Effect of Grain Size on the Lattice Thermal Conductivity of Copper Aluminum Alloys," Phys. Status Solidi, 21, 357-68, 1967.
74. Charsley, P. and Salter, J.A.M., "The Lattice Thermal Conductivities of Annealed Copper-Aluminum Alloys," Phys. Status Solidi, 10(2), 575-83, 1965.
75. Kusunoki, M. and Suzuki, H., "Lattice Thermal Conductivity of Deformed Copper Aluminum Alloy Crystals at Low Temperatures," J. Phys. Soc. Japan, 26(4), 932-8, 1969.
76. Lindenfeld, P. and Pennebaker, W.C., "Lattice Conductivity of Copper Alloys," Phys. Rev., 127(6), 1881-9, 1962.
77. Kapoor, A., Rowlands, J.A., and Woods, S.B., "Lattice Thermal Conductivity of Cold-Worked Noble-Metal Alloys Between 0.5 and 4 K," Phys. Rev., B9(4), 1223-9, 1974.
78. Mitchell, M.A., Klemens, P.G., and Reynolds, C.A., "Lattice Thermal Conductivity of Plastically Deformed Copper Plus 10 Atomic Percent Aluminum Specimens in the Temperature Range 1-4 K," Phys. Rev., B3(4), 1119-30, 1971.
79. Mannchen, W., "Heat Conductivity, Electrical Conductivity and the Lorenz Number for a Few Light-Metal Alloys," Z. Metallkde., 23, 193-6, 1931.
80. Grard, C. and Villey, J., "On the Thermal Conductivity of Some Light Alloys," Compt. Rend., 185, 856-8, 1927.
81. Elflein, M., "Thermal and Electrical Conductivity of Cast Al Alloys with Especial Attention to Self-Aging Alloys," Forsch. Metallk. Röntgen., 23, 63 pp., 1937.

82. Turnbull, D., Rosenbaum, H.S., and Treafitis, H.N., "Kinetics of Clustering in Some Aluminum Alloys," *Acta Metall.*, 8(5), 277-95, 1960.
83. Caplin, A.D. and Rizzuto, C., "Systematics of Matthiessen's Rule Breakdown at Low Temperatures," *Aust. J. Phys.*, 24, 309-16, 1971.
84. Bollenrath, F. and Hank, V.Z., "Electrical Resistivity of Aluminum + Copper Alloy," *Z. Metallkunde*, 39, 106-8, 1948.
85. Robinson, A.T. and Dorn, J.E., "Effect of Alloying Elements on the Electrical Resistivity of Aluminum Alloys," *Trans. AIME, J. Metals*, 457-60, 1951.
86. Caplin, A.D. and Rizzuto, C., "Breakdown of Matthiessen's Rule in Aluminum Alloys," *J. Phys. C*, 3(6), L117-20, 1970.
87. Korol'kov, A.M. and Shashkov, D.P., "Electrical Resistivity of Some Alloys in the Liquid State," *Russ. Met. Fuels*, 1, 49-54, 1962.
88. Gniewek, J.J. and Wasik, C.A., "Electrical Resistivity and Hall Coefficient of  $\text{CuAl}_2$ ," *J. Appl. Phys.*, 42(5), 2151, 1971.
89. Watanabe, K. and Katsui, A., "Temperature Dependencies of Resistivity and Hardness of  $\text{CuAl}_2$ ," *J. Jap. Inst. Metals*, 32(8), 808, 1968.
90. Leaver, A.D.W. and Charsley, P., "Low Temperature Lattice Thermal Conductivity of Deformed Copper Alloys," *J. Phys.*, 1F(1), 28-37, 1971.
91. Eretnov, K.I. and Lyubimov, A.P., "Viscosity and Electrical Resistance of Binary Melts with Electron Compounds in the Solid State," *Ukr. Fiz. Zh.*, 12(2), 216-23, 1967.
92. Sidorova, T.S., Panin, V.Ye., and Bol'shanina, M.A., "Investigation of the Nature of Low-Temperature Transformations in Deformed Cu-Al Alloys," *Fiz. Met. Metall.*, 14(5), 750-6, 1962; Engl. transl.: *Phys. Met. Metall.*, 14(5), 97-103, 1962.
93. Scala, E. and Robertson, W.D., "Electrical Resistivity of Liquid Metals and of Dilute Liquid Metallic Solutions," *Trans. Amer. Inst. Mining Eng.*, 197, 1141-7, 1953.
94. Vozdvizhenskii, V.M., "Microheterogeneities in Metal Alloys," *Phys. Met. Metall. (USSR)*, 11(2), 151-3, 1961.
95. Aliev, N.A., "Relationship of the Thermal Conductivity of Copper-Aluminum Alloys to Their Structure," *Trudy. Inst. Fiz. i Mat. Akad. Nauk, Azerb. SSR, Ser. Fiz.*, 6, 62-8, 1953.

96. Bradley, J.M. and Stringer, J., "Hall Effect Measurements in Aluminum Alloys," J. Phys. F, 4, 839-47, 1974.
97. Köster, W. and Rave, H.P., "Conductivity and Hall Constant. XIII.  $\alpha$ -Copper-Aluminum Alloys," Z. Metall., 52(3), 158-67, 1961.
98. Koester, W. and Rave, H.P., "General Remarks on Mixed Crystal Alloys of Copper and Silver with B-Metals," Z. Metallk, 55(12), 750-62, 1964.
99. Idase, R., Heierberg, R., and Walkenhorst, W., "Thermal Conductivity of Aluminum and Aluminum-Magnesium Alloys," Aluminum, 22, 631-5, 1940.
100. Cordier, H. and Detert, K., "Investigation of the Dissociation Process at Low Temperatures in an Aluminum + 10 wt.% Mg Alloy," Z. Metallkd, 52(5), 321-8, 1961.
101. Seth, R.S. and Woods, S.B., "Electrical Resistivity and Deviations from Matthiessen's Rule in Dilute Alloys of Aluminum, Cadmium, Silver, and Magnesium," Phys. Rev. B (Solid State), 2(8), 2961-72, 1970.
102. Clark, A.F., Childs, G.E., and Wallace, G.H., "Low Temperature Electrical Resistivity of Some Engineering Alloys," Cry. Eng. Conf. (UCLA, Los Angeles, CA), 1-15, 1969.
103. Clark, A.F., Childs, G.E., and Wallace, G.H., "Electrical Resistivity of Some Engineering Alloys at Low Temperatures," Cryogenics, 10(4), 295-305, 1970.
104. Clark, A.F. and Tryon, P.V., "Material Variability as Measured by Low Temperature Electrical Resistivity," Cryogenics, 12(6), 451-61, 1972.
105. Staebler, J., "Electrical and Thermal Conductivity and the Number of Wiedemann Franz of Light Metals and Magnesium Alloys," Tech Hochschule (of Breslau), Wroclaw, Poland, Ph.D. Thesis, 35 pp., 1929.
106. Powell, R.W., Hickman, M.J., and Tye, R.P., "The Thermal and Electrical Conductivity of Magnesium and Some Magnesium Alloys," Metallurgia, 70(420), 159-63, 1964.
107. Hedgcock, F.T. and Muir, W.B., "Influence of Lattice Scattering on Matthiessen's Rule in Dilute Binary Magnesium Alloys," Phys. Rev., 136(2A), A561-8, 1964.

108. Doyama, M., Koehler, J.S., Lwin, Y.N., Ryan, E.A., and Shaw, D.G., "Annealing Study of Dilute Aluminum Alloys Electron Irradiated at Liquid-Nitrogen Temperature," *Phys. Rev. B, Ser. 3*, 3(4), 1069-81, 1971.
109. Materials in Design Engineering, "Properties of Materials: Nonferrous Metals," *Materials in Design Engineering*, 50, 98-141, 1959.
110. Täubert, P., Thom, F., and Gammert, U., "The Lattice Heat Conductivity of Aluminum Alloys During Age-Hardening," *Cryogenics*, 13(3), 147-9, 1973.
111. Romanova, O.V. and Persion, Z.V., "Effect of Some Alloying Elements on Aluminum Electric Conductivity," *Ukr. Fiz. Zh.*, 18(6), 1030-2, 1973.
112. Rapp, O. and Fogelholm, R., "Resistivity of Dilute Alloy AlMg and AlMn Between 0 and 100°C a Generalized Approach," *Solid State Commun.*, 15, 1291-4, 1974.
113. Blatt, F.J., "Thermoelectric Power of Magnesium and Magnesium Alloys," *Helv. Phys. Acta*, 41(617), 693-700, 1968.
114. Collings, E.W., Hedgcock, F.T., Muir, W.B., and Muto, Y., "Magnetic and Electrical Properties of Some Ternary Mg-Mn-Al Alloys at Low Temperatures," *Philos. Mag.*, 10(103), 159-67, 1964.
115. Kikuchi, R., "The Thermal and Electrical Conductivities of Magnesium Alloys," *Kinzoku No Kenkyu*, 9(6), 239-43, 1932.
116. Matsuda, T. and Sato, T., "Hall Coefficient of Aluminum Rich Alloys," *J. Phys. Soc. Jpn.*, 21(8), 1494-503, 1966.
117. Salkovitz, E.I., Schindler, A.I., and Kammer, E.W., "Transport Properties of Dilute Binary Magnesium Alloys," *Phys. Rev.*, 105(3), 887-96, 1957.
118. Steeb, S. and Wörner, S., "Atomic Distribution and Some Physical Properties of Molten Aluminum-Magnesium Alloys," *Z. Metallkd.*, 56(11), 771-5, 1965.
119. Hirabayashi, M., "Electrical Resistivity and Superstructure of CuAu<sub>3</sub>," *J. Phys. Soc. Jpn.*, 14(3), 262-73, 1959.
120. Sato, H., "Electrical Resistivity of Au<sub>3</sub>Cu at Low Temperatures," *Phys. Rev.*, 106(4), 674-5, 1957.
121. Yonemitsu, K. and Sato, T., "Variation of Hall Coefficient of Some Non-Ferromagnetic Superlattice Alloys," *J. Phys. Soc. Jpn.*, 13(9), 998-1004, 1958.

122. Johansson, C.H. and Linde, J.O., "Radiographic and Electric Investigation of Cu-Au Systems," Ann. Phys. (Leipzig), 25(1), 1-49, 1936.
123. Borelius, G., Johansson, C.H., and Linde, J.O., "The Lattice Structure-Transformation in Metallic Mixed Crystals," Ann. Phys. (Leipzig), 86, 291-304, 1928.
124. Passaglia, E. and Love, W.F., "Electrical Resistance of Copper-Gold Alloys at Low Temperature," Phys. Rev., 98(4), 1006-10, 1955.
125. Tainsh, R.J. and White, G.K., "Lattice Thermal Conductivity of Copper-and-Silver Alloys at Low Temperatures," Phys. Chem. Solids, 23(9), 1329-35, 1962.
126. Linde, J.O., "Electric Properties of Dilute Solid Solution Alloys. III. Resistance of Cu-Au Alloys: Regularity of Resistance Increases," Ann. Phys. (Leipzig), 15, 219-48, 1932.
127. Roll, A. and Motz, H., "The Electrical Resistivity of Metallic Melts. III. The Electrical Resistivity of Alloy Melts of the Mixed Crystal Systems Silver-Gold and Copper-Gold as Well as the Eutectic Systems Silver-Copper, Tin-Zinc and Aluminum-Zinc," Z. Metallkd., 48(9), 495-502, 1957.
128. Grüneisen, E. and Reddemann, H., "Electronic and Lattice Conduction of Heat in Metals," Ann. Phys. (Leipzig), 20, 843-77, 1934.
129. Kemp, W.R.G., Klemens, P.G., and Tainsh, R.J., "Lattice Thermal Conductivity of Some Copper Alloys," Aust. J. Phys., 10, 454-61, 1957.
130. Sedström, E., "Peltier-Heat, Together with Thermal and Electrical Conductivities of Several Metallic Solid Solutions," Ann. Phys. (Leipzig), 59, 134-44, 1919.
131. Röhl, H., "The Elastic Properties of Mixed-Crystals Au-Cu and Au-Pd and the Alloys Cu<sub>3</sub>Pt, Cu<sub>3</sub>Pd, and CuPd," Ann. Phys. (Leipzig), 18(5), 155-68, 1933.
132. Pospisil, V., "Investigation of the Au-Cu System by Measuring Its Resistivity at Low Temperatures," Ann. Phys. (Leipzig), 18(5), 497-514, 1933.
133. Jones, F.W. and Sykes, C., "Atomic Rearrangement Process in the Copper Gold Alloy Cu<sub>3</sub>Au II," Proc. R. Soc. London, A166, 376-90, 1938.

134. Davis, T.H. and Rayne, J.A., "Specific Heat and Residual Resistivity of Binary and Ternary Noble-Metal Alloys," *Phys. Rev.*, B6(8), 2931-42, 1972.
135. Huray, P.G., Roberts, L.D., and Thomson, J.O., "Study of the Cu-Au and Ag-Au Alloy Systems as a Function of Composition and Order Through the Use of the Mössbauer Effect for  $^{197}\text{Au}$ ," *Phys. Rev.*, B4(7), 2147-61, 1971.
136. Ascoli, A., Bergamini, P., and Queirolo, G.T., "Impurity-Vacancy Binding Energy Obtained from Equilibrium Resistivity of Cu-Doped Au," *Scripta Metall.*, 6, 641-6, 1972.
137. Damon, D.H., Mathur, M.P., and Klemens, P.G., "Phonon-Assisted Impurity Scattering in Gold Alloys," *Phys. Rev.*, 176(3), 876-85, 1968.
138. Sedström, E., "On the Knowledge of Gold-Copper Alloys," *Ann. Phys. (Leipzig)*, 75, 549-55, 1924.
139. Eitel-McCullough, Inc., "Metallurgical Research and Development for Ceramic Electron Devices," Quarterly Report No. 5, AD 428 529, 38-9, 1963.
140. Broniewski, W. and Wesolowski, K., "On the Structure of Gold-Copper Alloys," *C. R. Hebd. Seances Acad. Sci.*, 198, 370-2, 1934.
141. Lengeler, B., Schilling, W., and Wenzl, H., "Deviations from Matthiessen's Rule and Longitudinal Magnetoresistance in Copper," *J. Low Temp. Phys.*, 2(1), 59-86, 1970.
142. Köster, W. and Schüle, W., "Conductivity and Hall Coefficient. IV. Copper-Nickel Alloys," *Z. Metallkd.*, 48, 592-4, 1957.
143. Schüle, W. and Kehrler, H.P., "Studies in Copper-Nickel System with Regard to Vacancies and Precipitation by Means of Electrical Resistivity," *Z. Metallkd.*, 52(3), 168-79, 1961.
144. Hedman, L.E. and Mattuck, R.D., "Effect of Heat Treatment and Plastic Deformation on the Paramagnetic Susceptibility of Copper-Nickel Alloys," *J. Phys. Chem. Solids*, 23, 955-61, 1962.
145. Robbins, C.G., Claus, H., and Beck, P.A., "Magnetism in Ni-Cu Alloys," *Phys. Rev. Lett.*, 22(24), 1307-10, 1969.
146. Mozer, B., Keating, D.T., and Moss, S.C., "Neutron Measurement of Clustering in the Alloy CuNi," *Phys. Rev.*, 175(3), 868-76, 1968.



147. Hicks, T.J., Rainford, D., Kouvel, J.S., Low, G.G., and Comly, J.B., "Giant Moments in Nickel-Copper alloys Near the Critical Composition," Phys. Rev. Lett., 22(11), 531-4, 1969.
148. Kussmann, A. and Wollenberger, H., "Influence of Thermal Treatment and Deformation on the Paramagnetic Behavior of Nickel-Copper Alloys," Z. Metallkd., 54(9), 521-4, 1963.
149. Furukawa, G.T., Reilly, M.L., and Saba, W.G., "Electrical Resistances of Wires of Low Temperature Coefficient of Resistance Useful in Calorimetry, 10-380 K," Rev. Sci. Instrum., 35(1), 113-4, 1964.
150. Rode, V.E., Blyushke, A., Finkel'berg, S.A., and Leichenko, A.I., "Temperature Dependence of the Susceptibility of Ni-Cu Alloys," Sov. Phys.-JETP, 35(3), 568-70, 1972.
151. Ahmad, H.M. and Greig, D., "Electrical Resistivity and Thermopower of Nickel-Copper Alloys," J. Phys. (Paris) Colloq., 4, 223-6, 1974.
152. Ahern, S.A., Martin, M.J.C., and Sucksmith, W., "The Spontaneous Magnetization of Nickel + Copper Alloys," Proc. R. Soc. London, 248A, 145-52, 1958.
153. Ahern, S.A. and Sucksmith, W., "Saturation Magnetization in Copper-Nickel Alloys," Proc. Phys. Soc., London, 69B(10), 1050-2, 1956.
154. Jackson, P.J. and Saunders, N.H., "Electrical and Thermal Conductivity of Nickel-Copper Alloys in the Neighborhood of the Curie-Point," Phys. Lett., 28A(1), 19-20, 1968.
155. Alder, M., University of Zurich, Ph.D. Thesis, 1916; as reported by Néel, L., "Magnetic Properties of the Metallic State and Energy of Interaction Between Magnetic Atoms," Ann. Phys. (Paris), 5, 232-79, 1936.
156. Meyer, A.J.P. and Wolff, C., "The Magnetic Properties of a Series of Nickel-Copper Alloys," C. R. Hebd. Seances Acad. Sci., 246(4), 567-9, 1958.
157. Torkar, K. and Goetz, H., "Magnetic Analysis of Copper-Nickel Sinter Bodies," Z. Metallkd., 45(5), 371-7, 1955.
158. Yao, Y.D., "Electrical Resistivity of Nickel-Rich Nickel-Copper Alloys Between 78 and 700 K," Annual Rept. Inst. Phys., Acad. Sinica, 8, 11-7, 1978.
159. Bagguley, D.M.S. and Heath, M., "Ferromagnetic Resonance in a Series of Alloys. I. Binary Alloys of Nickel with Copper, Tungsten, Cobalt, and Chromium," Proc. Phys. Soc., London, 90, Part 4, no. 570, 1029-45, 1967.

160. Standley, K.J. and Reich, K.H., "Ferromagnetic Resonance in Nickel and in Some of Its Alloys," *Proc. Phys. Soc., London*, 68B(10), 713-22, 1955.
161. Bloch, D. and Pauthenet, R., "Magnetic Properties Under Pressure of Some Transition Metal Alloys," *J. Appl. Phys.*, 36(3), 1229-30, 1965.
162. Belov, K.P. and Goriaga, A.N., "Influence of the Structural Characteristics of Ferromagnets on the Temperature Dependence of the Spontaneous Magnetization," *Bull. Acad. Sci. USSR, Phys. Ser. (Engl. Transl.)*, 21, 1043-50, 1957.
163. Van Elst, H.C., Lubach, B., and Van den Berg, G.J., "The Magnetization of Some Nickel Alloys in Magnetic Fields up to 15 kOe Between 0 and 300 K," *Physica*, 28(12), 1297-317, 1962.
164. Svensson, B., "Ferromagnetic Resistivity Increase of Copper-Nickel Alloys," *Ann. Phys. (Leipzig)*, Ser. 5, 25(5), 263-71, 1936.
165. Weiss, R.J., "The Origin of the Invar Effects," *Army Material Research Agency MRL Rept. 133*, 15 pp., 1963. [AD 415 635]
166. Yamasawa, K. and Murakami, K., "Experimental Study of Basic Characteristics of Temperature-Sensitive Magnetic Materials," *Tohoku Univ. Technol. Rept.*, 38(1), 225-36, 1973.
167. Wheeler, M.A., "The Paramagnetic Susceptibility of Copper-Nickel and Zinc-Nickel Alloys," *Phys. Rev.*, 56, 1137-45, 1939.
168. Marian, V., "The Ferromagnetic Curie Points and Absolute Saturation of Several Nickel Alloys," *Ann. Phys. (Paris)*, 7, 459-527, 1937.
169. Sousa, J.B., Chaves, M.R., Pinheiro, M.F., and Pinto, R.S., "Electrical Resistivity in Ferromagnetic Nickel-Copper Alloys Near the Curie Point," *J. Low Temp. Phys.*, 18(1-2), 125-37, 1975.
170. Skoskiewicz, T. and Baranowski, B., "Low Temperature Minimum of the Electrical Resistance in Ni-Cu and Ni-Cu-H Alloys," *Solid State Commun.*, 7, 647-9, 1969.
171. Crangle, J. and Butcher, P.J.L., "Observation of Resistance Minima in Cu-Ni Alloys Near the Critical Concentration for Ferromagnetism," *Phys. Lett.*, 32A(2), 80-1, 1970.

172. Houghton, R.W., Sarachik, M.P., and Kouvel, J.S., "Electrical Resistivity and Magnetoresistance of Paramagnetic Nickel-Copper Giant-Moment Alloys," *Solid State Commun.*, 8(12), 943-5, 1970.
173. Houghton, R.W., Sarachik, M.P., and Kouvel, J.S., "Anomalous Electrical Resistivity and the Existence of Giant Magnetic Moments in Ni-Cu Alloys," *Phys. Rev. Lett.*, 25(4), 238-9, 1970.
174. Kondorskii, E.I., Galkina, O.S., and Chernikova, L.A., "Electrical Resistance Maximum for Ferromagnets at Their Curie Points at Low Temperatures," *Sov. Phys.-JETP*, 11(2), 464-6, 1960.
175. Ahmad, H.M. and Greig, D., "High Temperature Resistance Minima in Concentrated Pd-Ag and Ni-Cu Alloys," *Phys. Rev. Lett.*, 32(15), 833-5, 1974.
176. Eagen, C.F., "Resistivity and Magneto-Resistance of Dilute Solutions of Cr in Cu-Ni Alloys," Iowa State Univ., Ph.D. Thesis, 70 pp., 1972.
177. Eagen, C.F. and Legvold, S., "Resistivity and Magnetoresistance of Dilute Solutions of Cr in Cu-Ni Alloys," *Phys. Rev.*, B6(5), 1830-8, 1972.
178. Legvold, S., Peterson, D.T., Burgardt, P., Hofer, R.J., Lundell, B., and Vyrostek, T.A., "Residual Resistivity and Aging-Clustering Effects of Cu-Rich Cu-Ni Alloys," *Phys. Rev.*, B9(5), 2386-9, 1974.
179. Domenicali, C.A. and Christenson, E.L., "Effects of Transition Metal Solutes on the Electrical Resistivity of Copper and Gold Between 4 and 1200 Degrees K," *J. Appl. Phys.*, 32(11), 2450-6, 1961.
180. Schroeder, P.A., Wolf, R., and Woollam, J.A., "Thermopowers and Resistivities of Silver-Palladium and Copper-Nickel Alloys," *Phys. Rev.*, 138(1A), A105-11, 1965.
181. Bulow, C.L., "Properties of Copper Alloys at Cryogenic Temperatures," *Electrical Design News*, 9(6), 70-8, 1964.
182. Jowitz, A.E., "Cryogenics, Environment, Phenomena, Application," *Electro-Technol.*, 68(3), 121-33, 1961.
183. Chang, H., "Electrical and Thermoelectric Properties of Some Metallic Thermoelectric Materials," *Energy Conversion*, 10(2), 65-72, 1970.
184. Pollock, D.D. and Finch, D.I., "Effects of Alloying Elements on the Electrical Properties of Manganin-Type Alloys," *Trans. Amer. Inst. Min. Met. Eng.*, 206, 203-10, 1956.

185. Mikryukov, V.E., "Thermal and Electrical Properties of Copper Alloys," Vestnik Moskov Univ. Ser. Mat. Mekh. Astron. Fiz. Khim., 11(2), 53-70, 1956.
186. Kierspe, W., "Influence of Transition Elements on Thermal Conductivity of Copper," Z. Metallkd., 58(12), 895-902, 1967.
187. Dutta-Roy, S.K. and Subrahmanyam, A.V., "Hall Effect, Magnetic Resistivity, and Magnetic Susceptibility of Nickel and Nickel-Copper Alloys," Phys. Rev., 177(3), 1133-8, 1969.
188. Lecordier, J.C. and Colombani, A., "Electric Properties of Thin Films of a Copper-Nickel Alloy," C. R. Hebd. Seances Acad. Sci., Ser. A/B, 266(23), 1426-9, 1968.
189. Sager, G.F., "Investigation of the Thermal Conductivity of the System Copper-Nickel," Eng. Sci. Series, Rensselaer Polytech. Inst. Bull., 27, 3-48, 1930.
190. Barratt, T., "Thermal and Electrical Conductivities of Some of the Rarer Metals and Alloys," Proc. Phys. Soc., London, 26, 347-71, 1914.
191. Grüneisen, E. and Goens, E., "Investigations of Metal Crystals. V. Electric and Thermal Conductivity of Single Crystal and Polycrystalline Metals of the Cubic System," Z. Phys., 44, 615-42, 1927.
192. Jaeger, W. and Diesselhorst, H., "Heat Conduction, Electric Conduction, Heat Capacity, and Thermoelectrical EMF of Several Metals," Wiss. Abh. Phys.-Tech. Reichsanst., 3, 269-425, 1900.
193. Zimmerman, J.E., "Heat Conduction in Alloys and Semi-Conductors at Low Temperatures," Carnegie Institute of Technology, Ph.D. Thesis, 1951.
194. Estermann, I. and Zimmerman, J.E., "Effect of Lattice Defects on the Thermal Conductivity of Certain Alloys at Low Temperatures," Bull. Intern. Inst. Refrig. Annexe 1, 35-42, 1952.
195. Berman, R., "The Thermal Conductivity of Some Alloys at Low Temperatures," Philos. Mag., 42, 642-50, 1951.
196. Ellis, W.C., Morgan, F.L., and Sager, G.F., "The Thermal Conductivities of Copper, Nickel, and Some Alloys of Nickel," Eng. Sci. Series, Rensselaer Polytech. Inst. Bull., No. 21, 23 pp., 1928.

197. Grüneisen, E., "Thermal Conductivity of Metals and Their Relation to the Electrical Conductivity," *Ann. Phys. (Leipzig)*, Ser. 4, 3, 43-74, 1900.
198. Mikryukov, V.E., "Thermal and Electrical Properties of Copper Alloys," *Vestnik Moskov Univ. Ser. Mat. Mekh. Astron. Fiz. Khim.*, 12(2), 85-93, 1957.
199. Hulm, J.K., "The Thermal Conductivity of Copper-Nickel Alloy at Low Temperatures," *Proc. Phys. Soc., London*, Ser. B, 64, 207-11, 1951.
200. Telkes, M., "Solar Thermoelectric Generators," *J. Appl. Phys.*, 25(6), 765-77, 1954.
201. Lerner, E. and Daunt, J.G., "Magnetoresistance Effect in Advance and Evanohm Wires at Low Temperatures in Fields up to 50 kG," *Rev. Sci. Instrum.*, 35(8), 1069-70, 1964.
202. Harvey, A.R., Legvold, S., and Peterson, D.T., "Resistivity and Magnetoresistance of Dilute Solutions of Mn in Cu-Ni Alloys," *Phys. Rev.*, B4(11), 4003-8, 1971.
203. Los, G.J. and Gerritsen, A.N., "Resistance and Magneto-Resistance of Dilute Alloys of Copper and Gold with Nickel at Low Temperatures," *Physica*, 23, 633-40, 1957.
204. Kondorskii, E.I., Galkina, O.S., and Chernikova, L.A., "Nature of Electrical Resistivity of the Ferromagnetic Metals at Low Temperatures," *J. Appl. Phys.*, 29(3), 243-6, 1958.
205. Allison, F.E. and Pugh, E.M., "Temperature Dependence of the Hall Coefficient in Some Copper Nickel Alloys," *Phys. Rev.*, 102, 1281-7, 1956.
206. Galkina, O.S. and Chernikova, L.A., "The Relation Between the Temperature Dependence of Electrical Resistance at Low Temperatures and the Galvanomagnetic Effect in Strong Magnetic Fields," *Sov. Phys.-JETP*, 11(1), 1-3, 1960.
207. Farrell, T. and Greig, D., "The Electrical Resistivity of Nickel and Its Alloys," *Proc. Phys. Soc. J. Phys. C*, 1, 1359-69, 1968.
208. Kondorskii, E.I., Galkina, O.S., and Chernikova, L.A., "Low-Temperature Resistivity of Nickel Alloys and Its Variation in Magnetic Fields," *Acad. of Sci., USSR, Bull., Phys. Ser.*, 21, 1109-15, 1958.

209. Greig, D. and Harrison, J.P., "The Low Temperature Electrical Transport Properties of Nickel and Dilute Nickel-Copper Alloys," *Philos. Mag.*, 12, 71-9, 1965.
210. Farrell, T. and Greig, D., "The Thermal Conductivity of Nickel and Its Alloys," *J. Phys. C, Ser. 2*, 2, 1465-73, 1969.
211. Yarbrough, D.W., Williams, R.K., and Graves, R.S., "Transport Properties of Concentrated Ag-Pd and Cu-Ni Alloys from 300-1000 K," Paper presented at the 16th Int. Therm. Cond. Conf., Chicago, IL, Nov. 7-9, 1979.
212. Greig, D. and Rowlands, J.A., "Electrical Resistivity of Dilute Nickel Alloys," *Low Temp. Phys.-LT13 (Proc. 13th Int. Conf. Low Temp. Phys.)*, 4, 233-5, 1974.
213. Lihl, F. and Wildhack, H., "Recovery of Lattice Defects in Polycrystalline Copper, Nickel, and Copper-Nickel Alloys After Tensile Deformation at 78 K," *Z. Metallkd.*, 62(2), 143-7, 1971.
214. Aoyama, S. and Ito, T., "Thermal Conductivity of Some Cu-Ni Alloys between 80 K and Room Temperature," *Nippon Kinzoku Gakkaishi*, 4, 3-7, 1940.
215. Erdmann, J.C. and Jahoda, J.A., "Temperature Dependence of the Lattice Thermal Conductivity of Copper-Nickel Alloys at Low Temperatures," in Thermal Conductivity: Proc. 7th Thermal Cond. Conf., NBS Spec. Publ. 302, 259-70, 1968.
216. Erdmann, J.C. and Jahoda, J.A., "Thermal Conductivity of Copper-Nickel Alloys at 4.2 K," Boeing Sci. Res. Lab. Rept. D1-82-0333, 22 pp., 1964. [AD 600 457]
217. Gartner, H., Zrudsky, D.R., and Legvold, S., "Low Temperature Electrical Resistivity of Dilute Solutions of Fe in Cu-Ni Alloys," *Solid State Commun.*, 8, 913-8, 1970.
218. Linz, R.J., Bowley, A.C., Klaffky, R.W., and Damon, D.H., "The Electrical and Thermal Conductivities of Cu-4, 10, and 20 at.% Ni from 1 to 25 K," *IEEE Trans. Magn.*, MAG-11(2), 305-8, 1975.
219. Kaul, S.N., "Anisotropy in Low Field Transverse Magnetoresistivity of Nickel-Copper Alloys at Room Temperature," *Indian J. Phys.*, 49, 143-54, 1975.

220. Van Elst, H.C. and Gorter, C.J., "The Ferromagnetic Anisotropy of the Specific Resistance of Some Nickel Alloys," Appl. Sci. Res., Ser. B, 4(1-2), 87-90, 1954.
221. Svensson, B., "Magnetic Susceptibility and Electrical Resistivity of Solid Solutions of Pd-Ag and Pd-Cu," Ann. Physik, Ser. 5, 14, 699-711, 1932.
222. Johansson, C.H. and Linde, J.O., "Lattice Structure and Electrical Conductivity of the Solid Solutions of Au-Cu, Pd-Cu, and Pt-Cu," Ann. Physik, 82(4), 449-78, 1927.
223. Pott, F.P., "Investigation of the Wiedemann-Franz-Lorenz Law for Ordered and Disordered Phases of a Range of Copper-Palladium Alloys," Z. Naturforsch., 13a(3), 215-21, 1958.
224. Köster, W. and Lang, W., "Several Physical Properties for Superstructures with Dislocations," Z. Metallkd., 49(9), 443-9, 1958.
225. Jaumot, F.E. and Sawatzky, A., "Order-Disorder and Cold-Work Phenomena in Cu-Pd Alloys," Acta Metall., 4, 118-44, 1956.
226. Otter, F.A., Jr., "Thermoelectric Power and Electrical Resistivity of Dilute Alloys of Mn, Pd, and Pt, in Cu, Ag, and Au," J. Appl. Phys., 27(3), 197-200, 1956.
227. Bäcklund, N., "Electrical Resistance of Some Noble-Metal Alloys at Liquid-Helium Temperatures," Phys. Chem. Solids, 7(1), 94-5, 1958.
228. Yonemitsu, K. and Sato, T., "Variation of Hall Coefficient of Some Non-ferromagnetic Superlattice Alloys," J. Phys. Soc. Jpn., 13(9), 998-1004, 1958.
229. Ugodnikova, L.A., Volkenshtein, N.V., and Tsiovkin, Yu.N., "Thermo-emf and Resistivity of Copper-Palladium Solid Solutions," Fiz. Met. Metalloved. Akad. Nauk SSSR Ural Filial, 31(3), 543-7, 1971.
230. Rennollet, G. and Seemann, H.J., "Influence of Temperature on the Hall Constant and the Electrical Conductivity of Copper-Palladium Alloys with Random and Ordered Atomic Distribution," Z. Angew. Phys., 28(3), 148-58, 1969.
231. Taylor, R., "Transformations in the Copper-Palladium Alloys," J. Inst. Metals, 54, 255-72, 1934.

232. Kim, M.J. and Flanagan, W.F., "The Effect of Plastic Deformation on the Resistivity and Hall Effect of Copper-Palladium and Gold-Palladium Alloys," *Acta Metall.*, 15, 735-45, 1967.
233. Buynova, L.N., Syutkina, V.I., Shashkov, O.D., and Yakovleva, E.S., "Influence of Domain Size on the Properties of Copper-Palladium Alloys," *Fiz. Met. Metalloved.*, 33(6), 1195-206, 1972; Engl. transl.: *Phys. Met. Metallogr.*, USSR, 33(6), 67-77, 1972.
234. Holgersson, S. and Sedström, E., "Experimental Measurement of the Lattice Structure of Several Metal Alloys," *Ann. Physik*, 75(18), 143-62, 1924.
235. Sedström, E., "Several Physical Properties of Metallic Alloys," University of Stockholm, Ph.D. Thesis, 1924.
236. Mallory Metallurgical Company, "Electrical Contacts," Mallory, P.R. and Co., Inc., Bull. No. 3-25, 71 pp., 1964.
237. Massalski, T.B. and Kittl, J.E., "The Low Temperature Solid Solubility Limits of the  $\alpha$  and  $\beta$  Phases in the Cu-Zn System," *J. Aust. Inst. Met.*, 8(1), 91-7, 1963.
238. Shinoda, G. and Amano, Y., "The Eutectoid Transformation of the  $\beta'$  Phase in Cu-Zn Alloys," *Trans. Jap. Inst. Met.*, 1(1), 54-7, 1960.
239. Argent, B.B. and Lee, K.T., "The Electrical Resistivities and Lattice Parameters of Copper-Gold-Zinc Alloys," *Br. J. Appl. Phys.*, 15(12), 1523-8, 1964.
240. Henry, W.G. and Schroeder, P.A., "The Low-Temperature Resistivities and Thermopowers of Alpha-Phase Copper-Zinc Alloys," *Can. J. Phys.*, 41(7), 1076-93, 1963.
241. Fairbank, H.A., "The Electrical Resistivity of Copper-Zinc and Copper-Tin Alloys at Low Temperature," *Phys. Rev.*, 66(9/10), 274-81, 1944.
242. Smith, C.S., "Thermal Conductivity of Copper Alloys. I. Cu-Zn Alloys," *Trans. AIME*, 89, 84-106, 1930.
243. Takano, K., "Hall Coefficient of Beta-Phase Alloys," *J. Phys. Soc. Jpn.*, 26(2), 362-70, 1969.



244. Tainsh, R.J. and White, G.K., "Lattice Thermal Conductivity of Copper- and Silver-Alloys at Low Temperatures," *Phys. Chem. Solids*, 23(9), 1329-35, 1962.
245. Crisp, R.S. and Henry, W.G., "The Temperature Dependence of the Characteristic Thermopowers of Zinc, Gallium, Germanium, and Arsenic in Copper and Silver," *Philos. Mag.*, 11(112), 841-51, 1965.
246. Kemp, W.R.G., Klemens, P.G., Tainsh, R.J., and White, G.K., "The Electrical and Thermal Conductivities of Some Brasses at Low Temperatures," *Acta Metall.*, 5, 303-9, 1957.
247. Kemp, W.R.G., Klemens, P.G., and Tainsh, R.J., "Lattice Thermal Conductivity of Copper Alloys: Effect of Plastic Deformation and Annealing," *Philos. Mag.*, 4(43), 845-57, 1959.
248. Eucken, A. and Neumann, O., "Zur Kenntnis des Wiedemann-Franz'schen Gesetzes. I," *Z. Physik Chem.*, 111, 431-46, 1924.
249. Olsen, T., "Lattice Thermal Conductivity in Copper Alloys," *J. Phys. Chem. Solids*, 12(2), 167-74, 1960.
250. Gordon, J.E. and Amstutz, L.I., "A Brass Thermometry for Use in Determining Temperature Below 1 K," *Cryogenics*, 329-32, 1965.
251. Srivastava, B.N., Chatterjee, S., and Sen, S.K., "Thermal and Electrical Conductivities of Alloys at Low Temperatures," *Indian J. Phys.*, 43, 213-22, 1969.
252. Maxwell, E., "Conductivity Loss Measurements at K-Band," MIT Radiation Laboratory, Report 854, 26 pp., 1946. [AD 24 458]
253. Domenicali, C.A. and Otter, F.A., "Thermoelectric Power and Electron Scattering in Metal Alloys," *Phys. Rev.*, 95(5), 1134-42, 1954.
254. Crisp, R.S., Henry, W.G., and Schröder, P.A., "The Thermopowers and Resistivities of the Primary Solid Solutions of Zinc, Gallium, Germanium, and Arsenic in Copper," *Philos. Mag.*, 10(106), 553-77, 1964.
255. Muto, Y. and Noto, K., "Magnetoresistivity Behavior of Some Dilute Cu-Fe, Cu-Mn, and Cu-Zn Alloys at Liquid Helium Temperatures," *Can. J. Phys.*, 42, 15-25, 1964.

256. Fusfeld, H.I., "Effect of Cold Work and Anneal on Resistivity of Alpha Brass," J. Appl. Phys., 24(8), 1062-3, 1953.
257. Ayers, J.D. and Massalski, T.B., "A Kinetic Study of the  $\beta$  to  $\alpha$  Massive Transformation in a Cu-Zn Alloy," Metall. Trans., 3(12), 3185-90, 1972.
258. Broom, T., "The Effect of Temperature of Deformation on the Electrical Resistivity of Cold-Worked Metals and Alloys," Proc. Phys. Soc. London, 65, 871-81, 1952.
259. Sato, T. and Noguchi, S., "Note on Some Properties of Gamma Brass," J. Phys. Soc. Jpn., 12(4), 335-9, 1957.
260. Martin, M.C., "Effect of Quenching on the Electrical Resistivity of a  $\beta$ -Brass Single Crystal," J. Appl. Phys., 34(6), 1835-6, 1963.
261. Aoyama, S. and Ito, T., "On the Thermal and Electrical Conductivities of Metals and Alloys at Low Temperatures. III. On Cu-Zn Alloys," Nippon Kinzoku Gakkai-Si, 4, 37-40, 1940.
262. Köster, W. and Schüle, W., "Conductivity and Hall-Constant. III.  $\alpha$ -Copper-Zinc Alloy," Z. Metall., 48(11), 588-91, 1957.
263. Presnyakov, A.A., Dautova, L.I., and Klychnikov, Yu.F., "Anomalies in the Electrical Resistivity of Brasses and Aluminum Bronzes," Fiz. Met. Metall., 10(5), 676-9, 1961; Engl. Transl.: Phys. Met. Metall. (USSR), 10(5), 41-5, 1961.
264. Hansen, M., Constitution of Binary Alloys, McGraw-Hill Book Co., New York, NY, 1305 pp., 1958.
265. Nagasawa, A., Matsuo, Y., and Kakinoki, J., "Ordered Alloys of Gold-Palladium System. I. Electron Diffraction Study of Evaporated Au<sub>3</sub>Pd Films," J. Phys. Soc. Jpn., 20(10), 1881-5, 1965.
266. Copeland, W.D. and Nicholson, M.E., "X-Ray Evidence of Short-Range Order in Au-Pd System," Acta Metall., 12, 321-2, 1964.
267. Iveronova, V.I. and Katsnel'son, A.A., "Existence of Short-Range Order in Gold-Palladium Alloys," Sov. Phys. Crystallogr. (Engl. Transl.), 9(4), 467-8, 1965.
268. Iveronova, V.I. and Katsnel'son, A.A., "Short-Range Order in a Gold-Palladium Alloy of Equiatomic Composition," Sov. Phys. Crystallogr. (Engl. Transl.), 11(4), 504-7, 1967.

269. Lin, W., Spruiell, J.E., and Williams, R.O., "Short-Range Order in a Gold-40.0 Atomic Percent Palladium Alloy," *J. Appl. Crystallogr.*, 3, 297-305, 1970.
270. Devi, U., Rao, C.N., and Rao, K.K., "Effect of Temperature on the Lattice Parameter of a 58.98 at.% Gold-41.02 at.% Palladium Alloy," *Acta Metall.*, 13(1), 44-5, 1965.
271. Köster, W. and Halpern, T., "Conductivity and Hall Constant. XXI. Gold-Palladium Alloys," *Z. Metallkd.*, 52, 821-5, 1961.
272. Logie, H.J., Jackson, J., Anderson, J.C., and Nabarro, F.R.N., "Effect of Plastic Deformation on Resistivity of Gold-Palladium Alloys," *Acta Metall.*, 9(8), 707-13, 1961.
273. Kim, M.J. and Flanagan, W.F., "The Recovery Kinetics of Deformed Copper-Palladium and Gold-Palladium Alloys," *Acta Metall.*, 15(5), 753-64, 1967.
274. Haas, H. and Lücke, K., "On the Short-Range Order Formation in Gold-Palladium Alloys," *Scr. Metall.*, 6(8), 715-20, 1972.
275. Lücke, K., Haas, H., and Schulze, H.A., "Equilibrium Values of Short Range Order in Gold-Silver and Gold-Palladium Alloys," *J. Phys. Chem. Solids*, 37, 979-87, 1976.
276. Rowland, T., Cusack, N.E., and Ross, R.G., "The Resistivity and Thermo-electric Power of the Palladium-Gold Alloy System," *J. Phys. F*, 4, 2189-202, 1974.
277. Kim, M.J., University of Washington, Ph.D. Thesis, 1966.
278. Kim, M.J. and Flanagan, W.F., "An Approximate Density of States Curve and Its Relation to the Measured Electrical Resistivity of Gold-Palladium Alloys," *Acta Metall.*, 15(5), 747-52, 1967.
279. Hau, N.H., "The Transport Properties of Palladium-Gold Alloys," Ohio University, Ph.D. Thesis, 108 pp., 1966.
280. Conybeare, J.G.G., "The Resistance of Palladium and Palladium-Gold Alloys," *Proc. Phys. Soc. London*, 49, 29-37, 1937.
281. Laubitz, M.J. and Van der Meer, M.P., "High Temperature Transport Properties of Some Platineal Alloys," in Thermal Conductivity - Proc. Seventh Conf., NBS Spec. Publ. 302, 325-30, 1968.

282. Ugodnikova, D.A., Beilin, B.M., and Tsiovkin, Yu.Yu.N., "Observation of Palladium Clusters in the Alloy  $Au_{0.6}Pd_{0.4}$ ," JETP Lett. (Engl. Transl.), 19(7), 234-6, 1974.
283. Bäcklund, N., "Electrical Resistance of Some Noble-Metal Alloys at Liquid-Helium Temperatures," Phys. Chem. Solids, 7(1), 94-5, 1958.
284. Köster, W. and Hagmann, D., "Conductivity and Hall Constant, XVI. The Solid Solutions of Palladium with Neighboring Elements in the Periodic System," Z. Metallkd., 52, 721-7, 1961.
285. Geibel, W., "Several Electrical and Mechanical Properties of Noble Metal Alloys. I," Z. Anorg. Chem., 69, 38-46, 1911.
286. Schulze, F.A., "The Thermal Conductivity of Several Series of Precious Metal Alloys," Physik. Z., 12, 1028-31, 1911.
287. Rudnitskii, A.A., "Thermoelectric Properties of the Noble Metals and Their Alloys," USAEC Rept. AEC-TR-3724, 233 pp., 1956.
288. Borelius, G., "Conductivity of Electricity in Alloys Containing Mixed Crystals," Ann. Physik, 4, 77(10), 109-37, 1925.
289. Elliott, R.P., Constitution of Binary Alloys, First Supplement, McGraw-Hill Book Co., New York, NY, 877 pp., 1965.
290. Grum-Grzhmailo, N.V., "Electrical Resistance and Hall Effect of Gold-Silver Alloys," J. Inorg. Chem., USSR, 1(9), 118-22, 1956.
291. Schulze, H.A. and Lücke, K., "Short-Range Order Formation in Dilute Alloys Due to Quenched-in Vacancies," J. Appl. Phys., 39, 4860-2, 1968.
292. Schulze, A. and Lücke, K., "The Influence of Vacancies on Short Range Order Formation in Au-Ag Alloys," Acta Met., 20, 529-42, 1972.
293. Lücke, K. and Haas, H., "The Resistivity Change During Short-Range Order Formation in a Au-15 At.% Ag Alloy," Scr. Met., 7, 781-6, 1973.
294. Cowley, J.M., "An Approximate Theory of Order in Alloys," Phys. Rev., 77(5), 669-75, 1950.
295. Giardina, M.D., Schüle, W., Frank, W., and Seeger, A., "Ordering Properties and Interpretation of Radiation Damage in Au-15 At.% Ag Alloys," Rad. Eff., 12, 277-80, 1972.

296. Schüle, W. and Crestoni, G., "Short-Range Order in Gold-Silver Alloys," *Z. Metallkd.*, 66(12), 728-33, 1975.
297. Giauque, W.F. and Stout, J.W., "Low Temperature Resistance of Gold-Silver Alloys," *J. Am. Chem. Soc.*, 60(1), 388-93, 1938.
298. Crisp, R.S. and Rungis, J., "Thermoelectric Power and Thermal Conductivity in the Silver-Gold Alloy System from 3-300°K," *Phil. Mag.*, 22(176), 217-36, 1970.
299. Davis, T.H. and Rayne, J.A., "Specific Heat and Residual Resistivity of Binary and Ternary Noble-Metal Alloys," *Phys. Rev.*, B6(8), 2931-42, 1972.
300. Boes, J., VanDam, A.J., and Bijovet, J., "Kondo Effect in the Electrical Resistivity of Silver-Gold Alloys with Ytterbium," *Phys. Lett.*, 28A(2), 101-2, 1968.
301. Stewart, R.G. and Huebener, R.P., "Deviations from Matthiessen's Rule in Dilute Gold and Platinum Alloys," *Phys. Rev.*, B1(8), 3323-38, 1970.
302. Iyer, V.K. and Asimow, R.M., "The Electrical Resistivity of Gold-Silver Alloys," *J. Less-Common Metals*, 13(1), 18-23, 1967.
303. Van der Sijde, B., "The Resistivity Changes Caused by Ordering in Some Quenched AuAg Alloys," *Physica*, 29(5), 559-61, 1963.
304. Dewar, J. and Fleming, J.A., "The Electrical Resistance of Metals and Alloys at Temperatures Approaching the Absolute Zero," *Phil. Mag.*, 5, 36, 271-99, 1893.
305. Dawson, H.I., "Recovery of Point Defects in Cold Worked Au, Ag, and Cu," *Acta Met.*, 13, 453-64, 1965.
306. Clay, J., "On the Change with Temperature of the Electrical Resistance of Alloys at Very Low Temperatures," *Univ. Leiden, Comm. Phys. Lab. No. 107d*, 30-51, 1908.
307. Weinberg, I., "Thermoelectric Power in Silver-Gold and Silver-Germanium Alloys," *Phys. Rev.*, 157(3), 564-9, 1967.
308. Barber, A.J. and Caplin, A.D., "The Low Temperature Electrical Resistivity of High Purity Ag and Ag-Based Alloys," *J. Phys. F: Metal Phys.*, 5, 679-96, 1975.

309. Sparks, L.L. and Hust, J.G., "Thermoelectric Voltage of Silver-28 Atomic Percent Gold Thermocouple Wire, SRM 733, versus Common Thermocouple Materials between Liquid Helium and Ice Fixed Points," NBS Spec. Publ. 260-34, 37 pp., 1972.
310. Van Baarle, C., Gorter, F.W., and Winsemius, P., "Thermal Conductivity and Thermopower of Silver and Silver-Base Alloys at Low Temperatures. III. Alloys with Gold, Antimony, and Palladium," *Physica*, 35(2), 223-40, 1967.
311. Howe, R.A. and Enderby, J.E., "The Thermoelectric Power of Liquid Silver-Gold," *Phil. Mag.*, 16(149), 467-76, 1967.
312. Ricker, T., "Magnetic Properties of Silver Alloyed with Gold, Palladium, Cadmium, and Aluminum," *Z. Metallk.*, 54(12), 718-24, 1963.
313. Beckmann, B., University of Uppsala, Ph.D. Thesis, 1911.
314. Broniewski, W. and Wesolowski, K., "Silver-Gold Alloys as Typical of Continuous Solid Solutions," *Compt. Rend.*, 194, 2047-9, 1932.
315. Shimizu, Y., "The Magnetic Susceptibility of Several Systems of Binary Alloys," *Tohoku Univ., Science Reports*, 21, 826-50, 1932.
316. Linde, J.O., "An Experimental Study of the Resistivity-Concentration Dependence of Alloys," *Helv. Phys. Acta*, 41, 1007-15, 1968.
317. Strouhal, V. and Barus, C., "Electrical Resistivity of Gold-Silver Alloy," *Abh. d. K. Böhm. Ges. d. Wiss.*, 6, 12, 1883.
318. Clay, J., "On the Change with Temperature of the Electrical Resistance of Alloys at Very Low Temperatures," *Jahrb. d. Radioakt.*, 8, 399, 1911.
319. Auer, H., Riedl, E., and Seemann, H.J., "Magnetic, Electric, and Spectrographic Investigations upon Gold-Silver Alloys," *Z. Physik*, 92(5-6), 291-302, 1934.
320. Bridgman, P.W., "Compressibilities and Pressure Coefficients of Resistance of Certain Elements, Compounds, and Alloys," *Proc. Am. Acad. Arts Sci.*, 68(2), 27-93, 1933.
321. Shunk, F.A., Constitution of Binary Alloys, Second Supplement, McGraw-Hill Book Co., New York, NY, 720 pp., 1969.

322. Jones, F.W. and Pumphrey, W.I., "Free Energy and Metastable States in the Iron-Nickel and Iron-Manganese Systems," J. Iron Steel Inst. London, 163, 121-31, 1949.
323. Machlin, E.S. and Cohen, M., "Burst Phenomenon in the Martensitic Transformation," Trans. AIME, 191, 746-54, 1951.
324. Machlin, E.S. and Cohen, M., "Isothermal Mode of Martensitic Transformation," Trans. AIME, 194, 489-500, 1952.
325. Fert, A. and Campbell, I.A., "Electrical Resistivity of Ferromagnetic Nickel and Iron Based Alloys," J. Phys. F, 6(5), 849-71, 1976.
326. Farrell, T. and Greig, D., "The Electrical Resistivity of Nickel and Its Alloys," J. Phys. C, Ser. 2, 1, 1359-69, 1968.
327. Schwerer, F.C. and Conroy, J.W., "Two-Current Conduction in Nickel-Base Alloys," J. Phys. F, 1(6), 877-92, 1971.
328. Schwerer, F.C. and Cuddy, L.J., "Spin Disorder Scattering in Iron-Base Alloys," J. Appl. Phys., 41(3), 1419-20, 1970.
329. Schwerer, F.C. and Cuddy, L.J., "Spin Disorder Scattering in Iron- and Nickel-Base Alloys," Phys. Rev. B, 2(6), 1575-86, 1970.
330. Shirakawa, Y., "On the Longitudinal Magnetoresistance Effect at Various Temperatures in Iron Nickel Alloys," Tohoku Imp. Univ., Sci. Repts., 27, 485-531, 1939.
331. Ascher, E.A., "Effet Hall, Aimantation Spontane'e at Temperature," Helv. Phys. Acta, 28(7), 667-93, 1955.
332. Ingersoll, L.R., et al., "Some Physical Properties of Nickel-Iron Alloys," Phys. Rev., 16, 126-32, 1920.
333. Reed, R.P., Clark, A.F., and Schramm, R.E., "Defect Annealing (4 to 295 K) after Martensitic Phase Transformation in an Iron-29 Nickel Alloy," Scr. Metall., 5(6), 485-8, 1971.
334. Livingston, H. and Mukherjee, K., "Martensitic Pretransformation Phenomena in the Gold 47.5 at.% Cadmium and the Iron-29.7 at.% Nickel Alloys," J. Appl. Phys., 43(12), 4944-50, 1972.
335. Armstrong, B.E. and Fletcher, R., "Resistivity and Thermopower of a Series of Nickel-Iron Invar Alloys," Can. J. Phys., 50(3), 244-50, 1972.

336. Gautier, F. and Loegel, B., "Electrical Resistivity of Concentrated Nickel Alloys," *Solid State Commun.*, 11(9), 1205-8, 1972.
337. Mikhailova, G.N., "Thermal and Electrical Conductivity of Certain Technical Materials in the Temperature Range 0.4-1.5 K," *Zh. Tekh. Fiz.*, 41(4), 800-3, 1971; Engl. transl.: *Sov. Phys.-Tech. Phys.*, 16(4), 626-8, 1971.
338. Larikov, L.N., Usov, Yu.V., and Boychuk, I.N., "Temperature Dependences of the Electrical Resistivity of Iron-Nickel Alloys," *Dopov. Akad. Nauk Ukr. SSR, Fiz. Mat. Tekh. Nauk*, A(5), 464-6, 1977.
339. Kondorskii, E.I. and Sedov, V.L., "The Change of Atomic Magnetic Moments in Ferromagnetic Metals Under Hydrostatic Compression," *Zh. Eksp. Teor. Fiz.*, 38, 773-9, 1960; Engl. transl.: *Sov. Phys.-JETP*, 11(3), 561-5, 1960.
340. Jellinghaus, W. and de Andrés, M.P., "Hall Effect and Conductivity of Iron-Nickel Alloys," *Ann. Phys.*, 5(7), 187-99, 1960.
341. Kalinin, V.M., Danilov, M.A., Komarova, L.K., and Tseytlin, A.M., "Influence of Titanium on the Physical Properties of Iron-Nickel Invars," *Fiz. Met. Metalloved.*, 36(2), 310-5, 1973; Engl. transl.: *Phys. Met. Metallogr.*, 36(2), 75-80, 1973.
342. Window, B., "Invar Anomalies," *J. Appl. Phys.*, 44(6), 2853-65, 1973.
343. Somura, T., "Magneto Volume Contribution to the Temperature Variation of the Magneto-Resistivity in Iron-Nickel Invar Alloys," *J. Phys. Soc. Jpn.*, 42(3), 826-32, 1977.
344. Tanji, Y., Moriya, H., and Nakagawa, Y., "Anomalous Concentration Dependence of Thermoelectric Power of Iron-Nickel (fcc) Alloys at High Temperatures," *J. Phys. Soc. Jpn.*, 45(4), 1244-8, 1978.
345. Källback, O., "Determination of the Order-Disorder Transition Point in 25% Iron - 75% Nickel," *Arkiv. Matematik Astron. Fys.*, 34B(17), 1-6, 1947.
346. Moore, J.P., Kollie, T.G., Graves, R.S., and McElroy, D.L., "Thermal Transport Properties of Ordered and Disordered 75% Nickel - 25% Iron," *J. Appl. Phys.*, 42(8), 3114-20, 1971.
347. Wakelin, R.J. and Yates, E.L., "A Study of the Order-Disorder Transformation in Iron-Nickel Alloys in the 25 Percent Iron Region," *Proc. Phys. Soc. London*, B66, Part 3, 221-40, 1953.



348. Szentirmay, Zs., "The Influence of Magnetic Annealing on the Transport Properties of Some Iron-Nickel Alloys," *Acta Physica et Chimia Debricina*, 20, 25-33, 1976.
349. Ono, Y. and Yagi, T., "Electrical Resistivity of Molten Iron-Nickel and Iron-Cobalt Alloys," *Trans. Iron Steel Inst. Jpn.*, 12(4), 314-6, 1972.
350. Baum, B.A., Tyagunov, G.V., Gel'd, P.V., and Khasin, G.A., "Viscosity and Electrical Resistance of Iron-Nickel Melts," *Izv. Vyssh. Uchebn. Zaved., Chern. Metall.*, 14(10), 5-8, 1971.
351. Epin, V.N., Baum, B.A., and Tyagunov, G.V., "Electrical Resistance of Iron-Nickel Alloys in Temperature Range 1200-1700°C," *Izv. Vyssh. Uchebn. Zaved., Fiz.*, 20(7), 145-6, 1977; Engl. transl.: *Sov. Phys. J.*, 20(7), 968-9, 1977.
352. Filippov, E.S. and Krestonikov, A.N., "Structural Transitions in Liquid Fe-Ni Alloys," *Russ. Metall.*, 4, 112-4, 1969.
353. Burgess, C.F. and Aston, J., "The Electrical Resistance of Iron Alloys," *Trans. Am. Electrochem. Soc.*, 20, 205, 1911.
354. Honda, K., "On the Thermal and Electrical Conductivities of Nickel Steels," *Sci. Rep. Tohoku Imp. Univ.*, 7, 59-66, 1918.
355. Bridgman, P.W., "The Effect of Pressure on the Resistance of Three Series of Alloys," *Proc. Amer. Acad. Arts Sci.*, 63, 329-45, 1928.
356. Bäcklund, N.G., "An Experimental Investigation of the Electrical and Thermal Conductivity of Iron and Some Dilute Iron Alloys at Temperatures Above 100°K," *J. Phys. Chem. Solids*, 20(1/2), 1-16, 1961.
357. Soffer, S., Dreesen, J.A., and Pugh, E.M., "Hall Effects, Resistivity, and Thermopower in Iron and Iron-Nickel Systems for  $x$  Equals 0 to 0.2," *Phys. Rev.*, 140(2A), 668-75, 1965.
358. Cadeville, M.C. and Loegel, B., "Transport Properties in Concentrated Nickel-Iron Alloys at Low Temperatures," *J. Phys. F*, 3(7), L115-9, 1973.
359. Berger, L. and Rivier, D., "Electrical and Thermal Resistivity of Pure Nickel and of an Iron-Nickel Alloy in a Magnetic Field at Low Temperatures," *Helv. Phys. Acta*, 35(7-8), 715-32, 1962.

360. Kohlhaas, R. and Kierspe, W., "The Thermal Conductivity of Pure Iron and Some Ferritic and Austenitic Steels Between the Temperature of Liquid Air and Room Temperature," Arch. Eisenhuettenwes., 36(4), 301-9, 1965.
361. Griffiths, E., Powell, R.W., and Hickman, M.J., "Second Report of Subcommittee 'A', Thermal Treatment. 'Part 3. The Physical Properties of a Series of Steels I,'" Iron Steel Inst. (London) Spec. Rept., 24, 215-51, 1939.
362. Bungardt, K. and Spyra, W., "Thermal Conductivity of Alloyed and Plain Steels and Alloys at Temperatures Between 20 and 700°," Arch. Eisenhuettenwes., 36(4), 257-67, 1965.
363. Wotruba, K., "The connection Between the Coercive Force and the Initial Susceptibility of Plastically Deformed Ferromagnetics," Chek. Fiz. Zh., 7(5), 568-76, 1957.
364. "Alloys, Review of Properties," Wilbur B. Driver Co., Product Guide, 1966.
365. Smolin, M.D., "Temperature Dependence of the Resistivity of Metals in the Two-Zone Approximation," Fiz. Met. Metalloved., 21(4), 630-2, 1966; Engl. transl.: Phys. Met. Metallogr., 21, 148-9, 1966.
366. Matsumoto, G., Satoh, T., and Iida, S., "Dynamic Properties of Permalloy Thin Films," Phys. Soc. Jpn., 21(2), 231-7, 1966.
367. Broniewski, W. and Smolinski, J., "The Structure of Iron-Nickel Alloys," Compt. Rend., Acad. Sci. (Paris), 196, 1793-6, 1933.
368. Savitskii, E.M. and Pravoverov, N.L., "Kurnakov Phases in the Palladium-Silver System," Russ. J. Inorg. Chem., 6(2), 253-4, 1961.
369. Pravoverov, N.L. and Savitskii, E.M., "Effect of Alloying Additions on the Electrical Properties of the Tensometric Palladium-Silver Alloy PdS-35," Russ. J. Inorg. Chem., 7(6), 687-91, 1962.
370. Aarts, W.H. and Houston-MacMillan, A.S., "Anomalous Behavior of Ag-Pd on Plastic Deformation," Acta Metall., 5, 525-7, 1957.
371. Rao, K.K., "Electrical Resistivity Recovery in Cold-Worked 60 percent Silver - 40 percent Palladium Alloy," Acta Metall., 10(9), 900, 1962.

372. Chen, W.K. and Nicholson, M.E., "The Influence of Annealing on the Electrical Properties of Cold-Worked Silver-Palladium Alloys," *Acta Metall.*, 12(6), 687-96, 1964.
373. Westerlund, R.W. and Nicholson, M.E., "Effect of Plastic Deformation on the Resistivity and Hall Constant of Silver-Palladium Alloys," *Acta Metall.*, 14(5), 569-74, 1966.
374. Chen, W.K. and Nicholson, M.E., "Reply to Resistivity Recovery Stages in a Silver-41.8 Weight Percent Palladium Alloy," *J. Appl. Phys.*, 39(7), 3496-7, 1968.
375. Rao, C.N. and Rao, K.K., "Effect of Temperature on the Lattice Parameters of Silver-Palladium Alloys," *Can. J. Phys.*, 42(7), 1336-42, 1964.
376. Kemp, W.R.G., Klemens, P.G., Sreedhar, A.K., and White, G.K., "The Thermal and Electrical Conductivity of Silver-Palladium and Silver-Cadmium Alloys at Low Temperatures," *Proc. R. Soc.*, A233, 480-93, 1956.
377. Mott, N.F., "Electrons in Transition Metals," *Adv. Phys.*, 13(51), 325-422, 1964.
378. Ricker, T. and Pflüger, E., "Temperature Dependence of Resistivity and Other Electrical Properties of Palladium Silver and Palladium Rhodium Alloys," *Z. Metallkd.*, 57(1), 39-45, 1966.
379. Murani, A.P., "Localized Enhancement Effects in Pd-Ag Alloys," *Phys. Rev. Letters*, 33(2), 91-4, 1974.
380. Edwards, L.R., Chen, C.W., and Legvold, S., "Resistivity Minima in Pd-Ag and Pd-Au Alloys," *Solid State Commun.*, 8(17), 1403-5, 1970.
381. Chen, C.W., Edwards, L.R., and Legvold, S., "Resistance Anomaly in Palladium-Silver Alloys Containing Rare Earth Impurities," *Phys. Status Solidi*, 26(2), 611-9, 1968.
382. Greig, D. and Rowlands, J.A., "The  $T^2$  Electrical Resistivity in Nickel and Palladium Alloys," *J. Phys. F*, 4, 232-45, 1974.
383. Greig, D. and Rowlands, J.A., "Resistance Minima and Deviations from Matthiessen's Rule in Dilute Palladium Alloys," *J. Phys. F*, 4, 536-49, 1974.
384. Araj, S., Rao, K.V., Yao, Y.D., Teoh, W., "Electrical Resistivity of Palladium-Silver Alloys at High Temperatures," Private Communication from Rao, K.V., 1977.

385. Pravoverov, N.L. and Tribunskaya, I.A., "Properties of Silver Alloys Containing Small Amounts of Palladium and Cadmium," *Russ. Metall.*, 1, 75-8, 1967.
386. Dewar, J. and Fleming, J.A., "On the Electrical Resistance of Pure Metals, Alloys, and Non-Metals at the Boiling Point of Oxygen," *Philos. Mag.*, 34, 326-37, 1892.
387. Sarachik, M.D., "Resistivity of the 4d Transition Series Ru-Rh-Pd-Ag Containing Fe," *J. Appl. Phys.*, 39(2), 699-701, 1968.
388. Seemann, H.J. and Rennollet, G., "Studies of the Silver-Palladium Mixed Crystal Series by Means of Hall Effect and Electrical Conductivity," *Z. Phys.*, 196(5), 486-94, 1966.
389. Geibel, W., "On Several Electrical and Mechanical Properties of Noble Metal Alloys. II," *Z. Anorg. Chem.*, 70, 240-54, 1911.
390. Greig, D. and Livesey, D., "The Hall Coefficient of Dilute Palladium and Platinum Alloys," *J. Phys. F*, 2, 699-708, 1972.
391. Zwingmann, G., "Several Electrical and Mechanical Properties of Binary Palladium Alloys. I. Palladium Alloys With Elements of the Second Large Period," *Z. Metallkd.*, 54(5), 286-92, 1963.
392. Simon, P.R.F., "Resistivity and Thermoelectric Power of Silver-Palladium Alloys," in *Proc. 9th Int. Conf. on Low Temp. Phys.*, Part B, Plenum Press, New York, 1045-9, 1965.
393. Guénault, A.M., "Low-Temperature Thermoelectric Power of Palladium-Silver Alloys," *Philos. Mag.*, 30(3), 641-9, 1974.
394. Carson, A.W., Lewis, F.A., and Schurter, W.H., "Relationships Between the Hydrogen Content and Electrical Resistance of Palladium + Silver Alloys," *Trans. Faraday Soc.*, 63, 1447-52, 1966.
395. Szafranski, A.W., "Electrical Resistance of the Palladium-Silver-Hydrogen Alloys From 4 to 300 K," *Phys. Status Solidi*, 19A(2), 459-66, 1973.
396. Couper, A. and Metcalfe, A., "Electrical Resistivity of Silver-Palladium Alloys," *J. Phys. Chem.*, 70(6), 1850-3, 1966.
397. Rao, K.V., Rapp, Ö., Johanneson, C., and Aström, H.U., "Electrical Resistivity of Fe Dissolved in Pd-Ag Matrix Alloys," *Physica B, Proc. 1976 Int. Conf. Magnetism*, 1977.

398. Krüger, F. and Gehm, G., "Electrical Resistivity of Silver-Palladium Alloys," *Ann. Phys.*, 16(5), 190-202, 1933.
399. Touloukian, Y.S., Powell, R.W., Ho, C.Y., and Klemens, P.G., Thermal Conductivity - Metallic Elements and Alloys, Vol. 1 of Thermophysical Properties of Matter - The TPRC Data Series, IFI/Plenum Data Corp., New York, 1595 pp., 1970.
400. Laws, F.A., Electrical Measurements, 2nd Edition, McGraw-Hill Book Co., Inc., New York, 739 pp., 1938.
401. Harris, F.K., Electrical Measurements, John Wiley and Sons, Inc., New York, 784 pp., 1952.
402. Van der Pauw, L.J., "A Method of Measuring Specific Resistivity and Hall Effect of Discs of Arbitrary Shape," *Philips Res. Rep.*, 13, 1-9, 1958.
403. Van der Pauw, L.J., "A Method of Measuring the Resistivity and Hall Coefficient on Lamellae of Arbitrary Shape," *Philips Tech. Rev.*, 20(8), 220-4, 1958-9.
404. Chambers, R.G. and Park, J.G., "Measurement of Electrical Resistivity by a Mutual Inductance Method," *Brit. J. Appl. Phys.*, 12, 507-10, 1961.
405. Zimmerman, J.E., "Measurement of Electrical Resistivity of Bulk Metals," *Rev. Sci. Instrum.*, 32(4), 402-5, 1961.
406. Radenac, A., Lacoste, M., and Roux, C., "Apparatus Meant for the Measurement of the Electrical Resistivity of Metals and Alloys by the Method of the Rotating Field Up to About 2000 K," *Rev. Int. Hautes Temper. Refract.*, 7(4), 389-96, 1970.
407. Cezairliyan, A. and McClure, J.L., "Thermophysical Measurements on Iron Above 1500 K, Using a Transient (Subsecond) Technique," *J. Res. Nat. Bur. Stand.*, 78A(1), 1-4, 1974.
408. Bean, C.P., DeBlois, R.W., and Nesbitt, L.B., "Eddy-Current Methods for Measuring the Resistivity of Metals," *J. Appl. Phys.*, 30, 1976-80, 1959.